



# Seeing Electrochemistry Live at the Atomic Scale



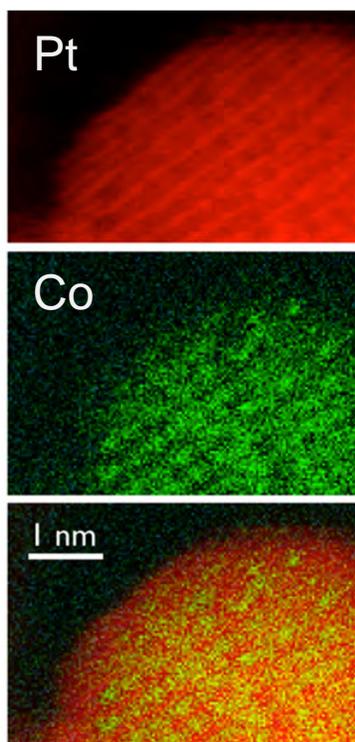
David A. Muller, Cornell University



*Ex Situ*

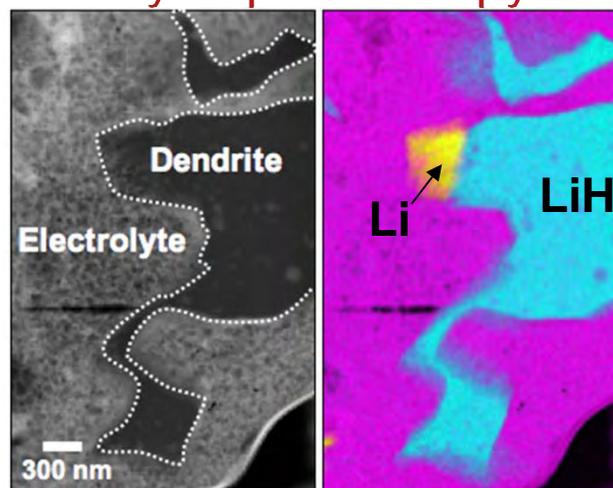


*In Situ/Operando*

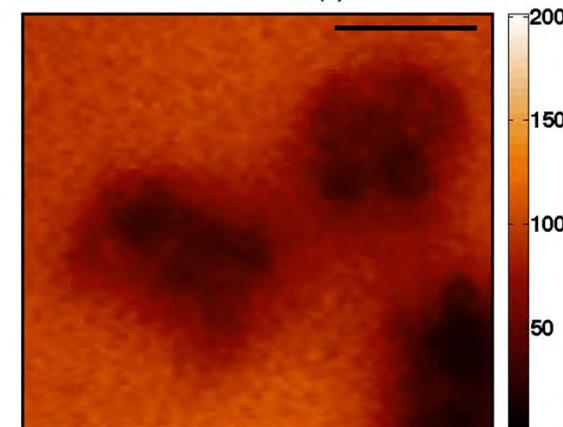
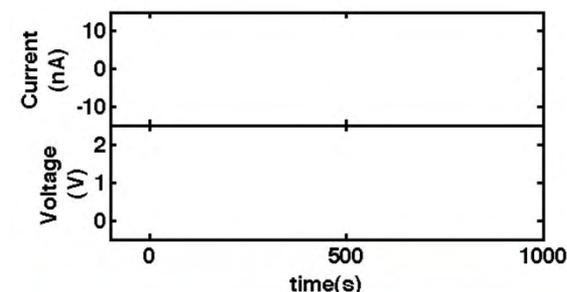


*Atomic Resolution*

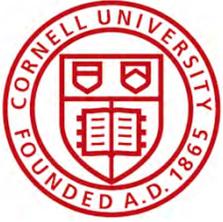
**Cryo-spectroscopy**



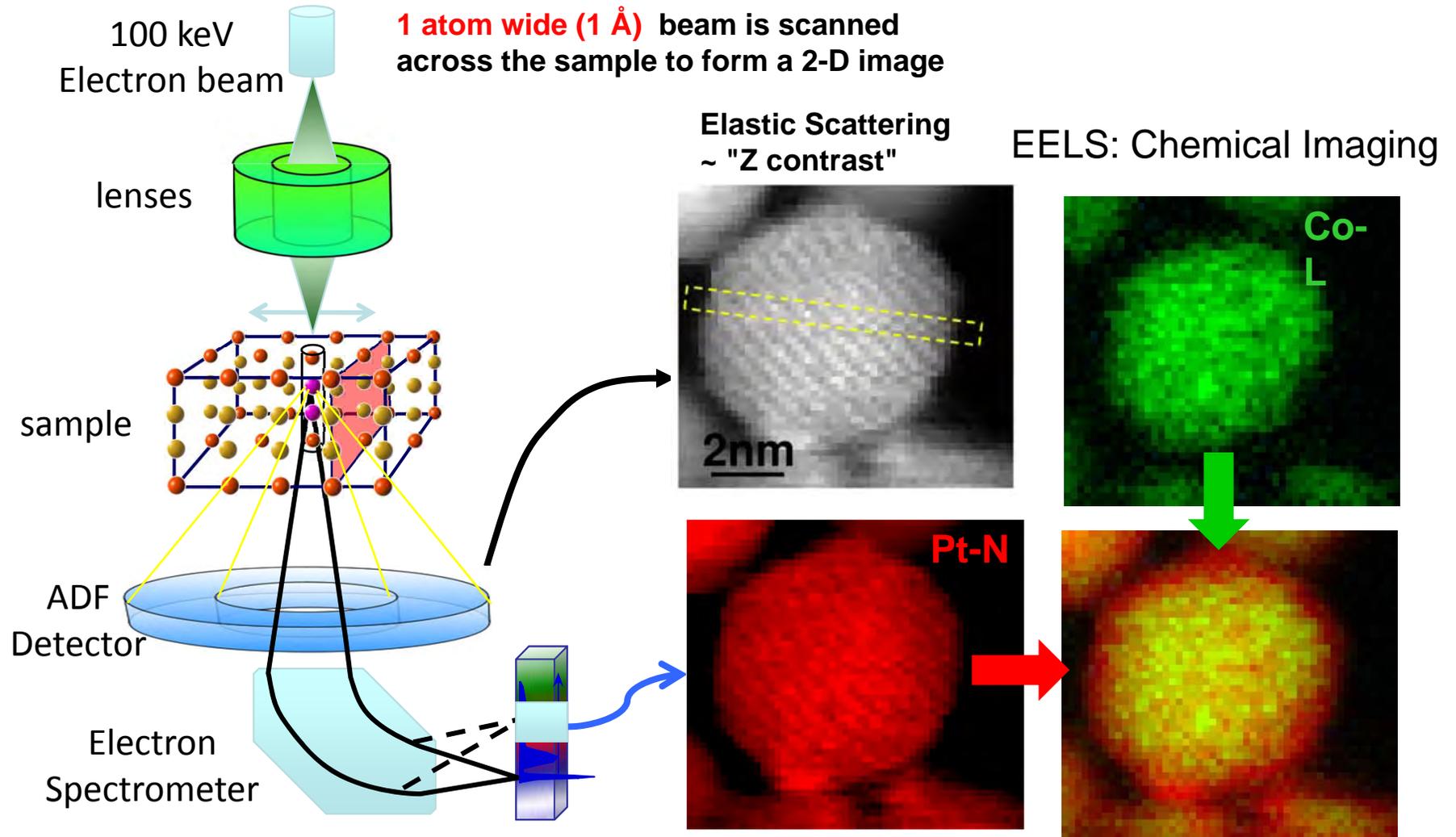
*Tracking chemistry and bonding at liquid/solid interfaces*



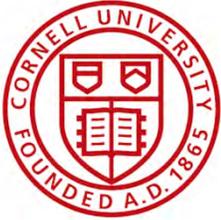
*Electrochemistry in liquids*



# Scanning Transmission Electron Microscopy

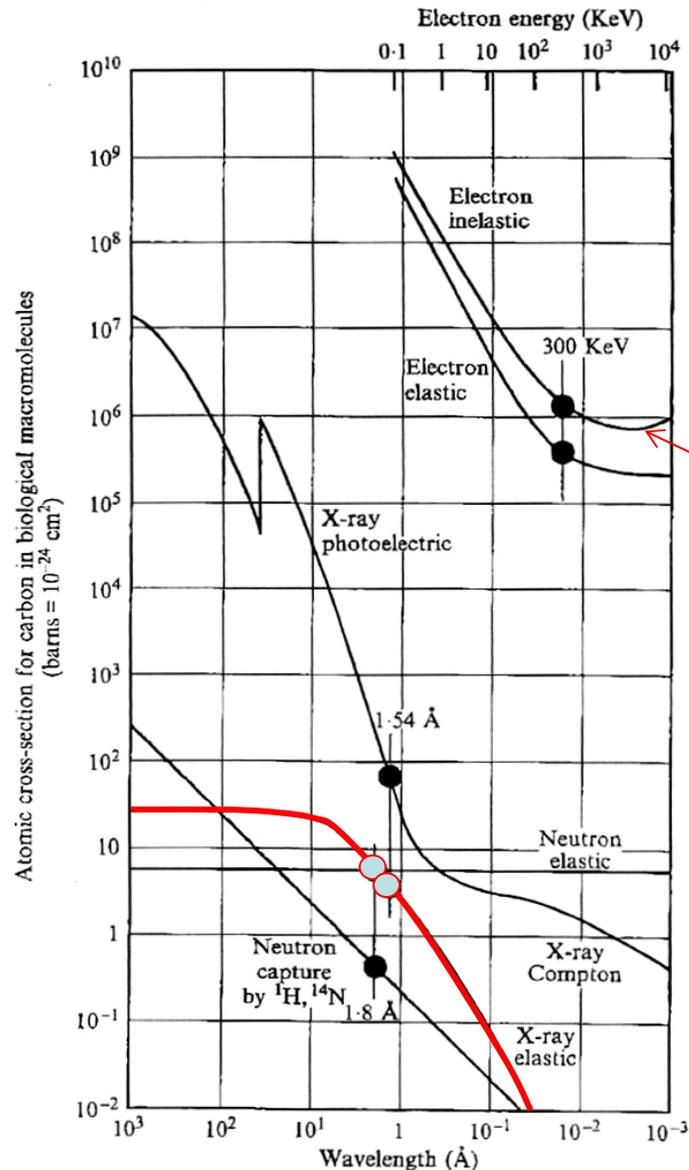


Muller, Kourkoutis, Murfitt, Song, Hwang, Silcox, Dellby, Krivanek, *Science* **319**, 1073 (2008).



# How Bad is Radiation Damage?

R. Henderson, Quarterly Reviews of Biophysics 28 (1995) 171-193.



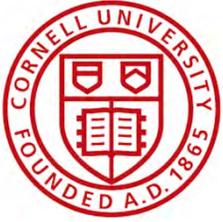
It's not the cross-section, but

How many damaging events per useful imaging event?

Least Damage:

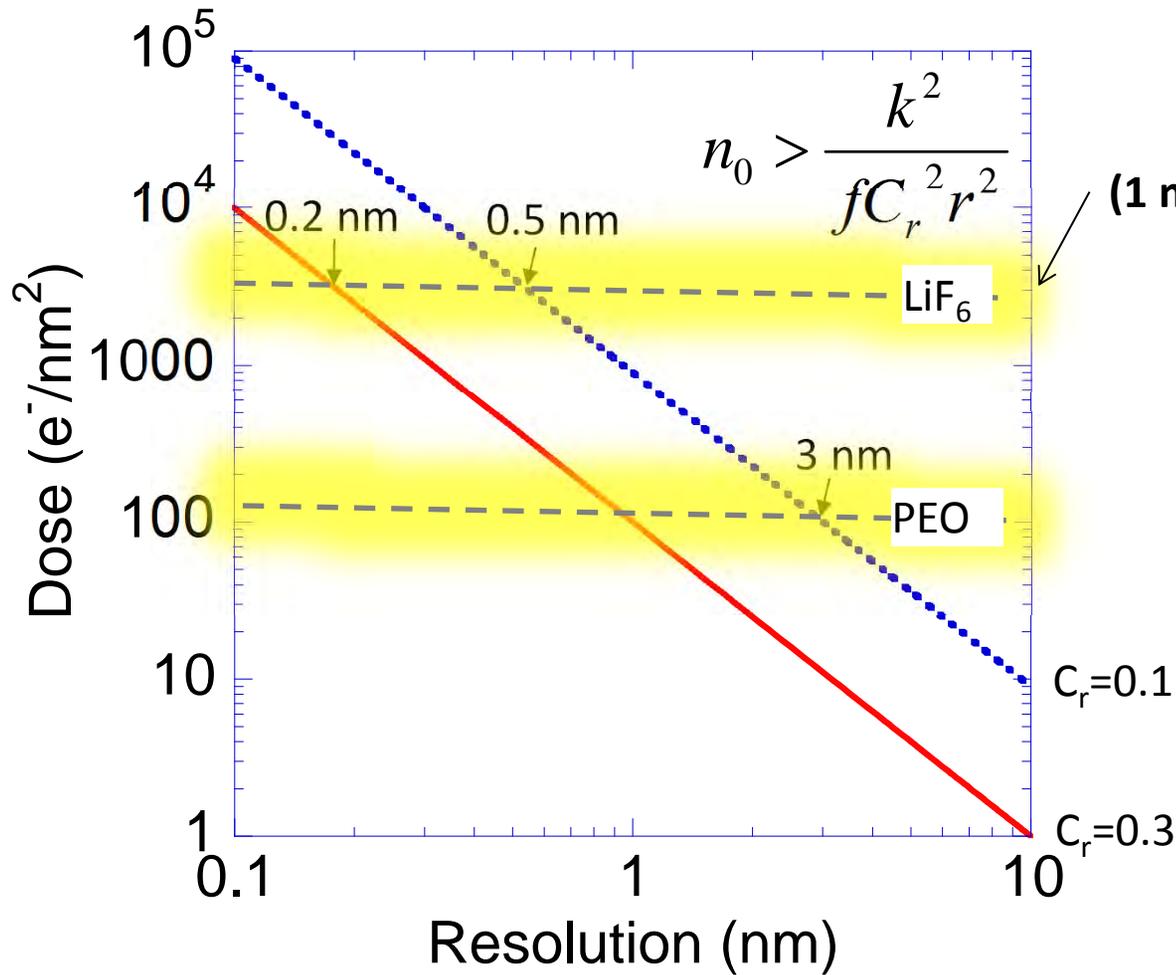
Elastic imaging - Electrons wins

Inelastic imaging - Soft X-rays win



# Dose Required for 2D-Imaging

Saxberg & Saxton, Ultramicroscopy 6, 85–90 (1981)



Resolution  $\sim 1/(\text{Dose})^{1/2}$

Resolution  $\sim 1/(\text{Contrast})$

$n_0$  : dose

$k$  : S/N = 3

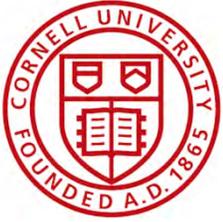
$f$  : fraction contributing to background = 1

$C_r$  : contrast = 0.1-0.3

$r$  : resolution

Maybe can do 1 atomic-resolution image in water, but not organic electrolyte

For 10-100 images, expect 1-10's of nm resolution

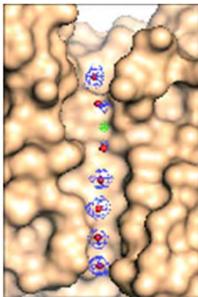
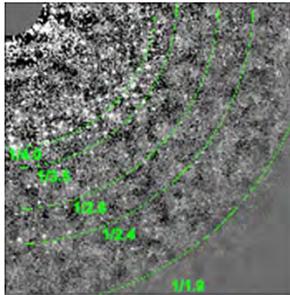


# 3D Cryo - Electron Microscopy

Biological structures are inherently 3D and require 3D techniques

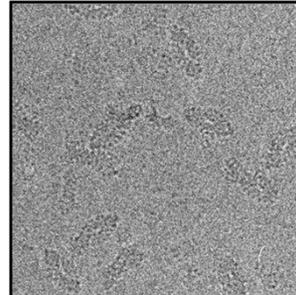
## Electron crystallography

2-D crystals of membrane proteins  
in their native environment  
near-atomic resolution



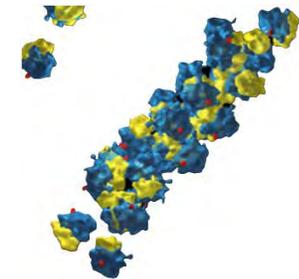
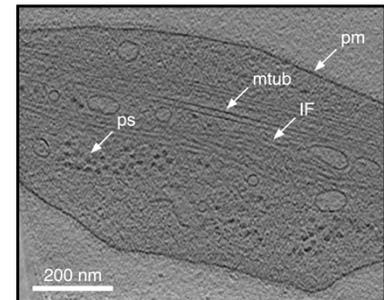
## Single-particle analysis

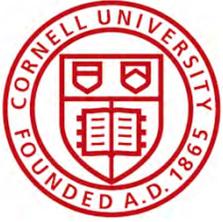
Purified molecules in  
solution ~200-10 MDa  
~2.2 Å resolution



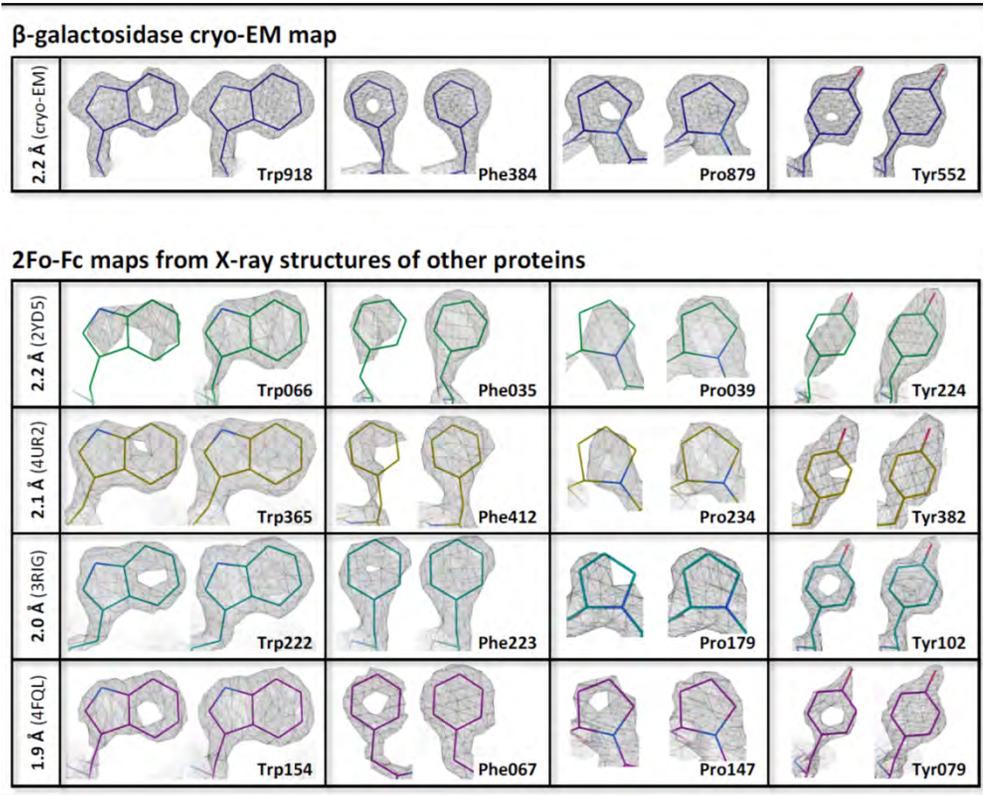
## Electron tomography

Pleiomorphic samples,  
e.g., cells and organelles  
~ 2-4 nm resolution

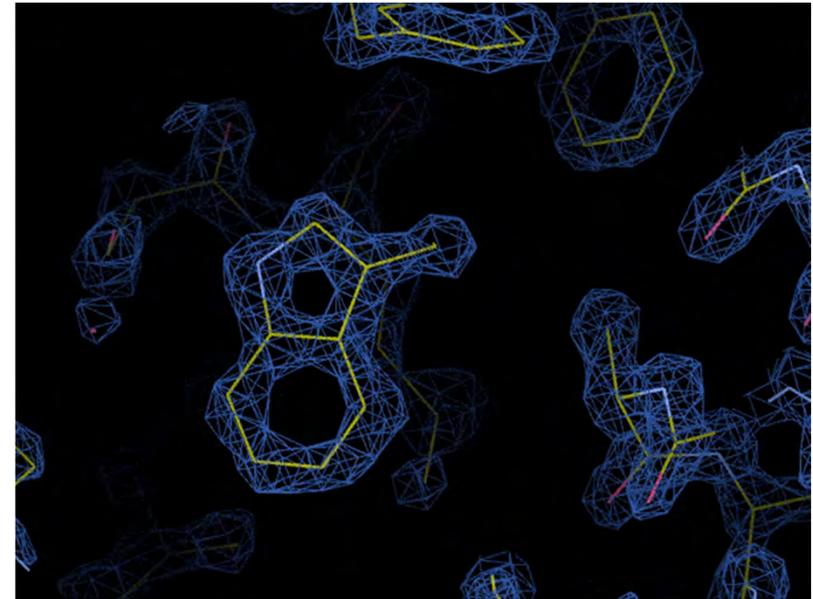




# Single-Particle CryoEM vs X-ray

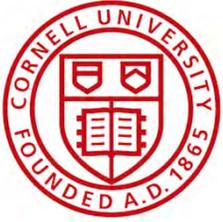


Bartesaghi, Science **348** (2015)



1.55 Å Cryo EM, Takayuki Kato, (2019)  
24 hrs data collection, 120,000 particles

Single-Particle Methods should work for molecular and bio-Echem (Freeze at bias)  
-needs similar molecules to work (so not so good for nanoparticle catalysts)



# Imaging vs Diffraction/SPEM

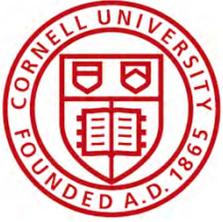


Ensemble - 1:1 mapping

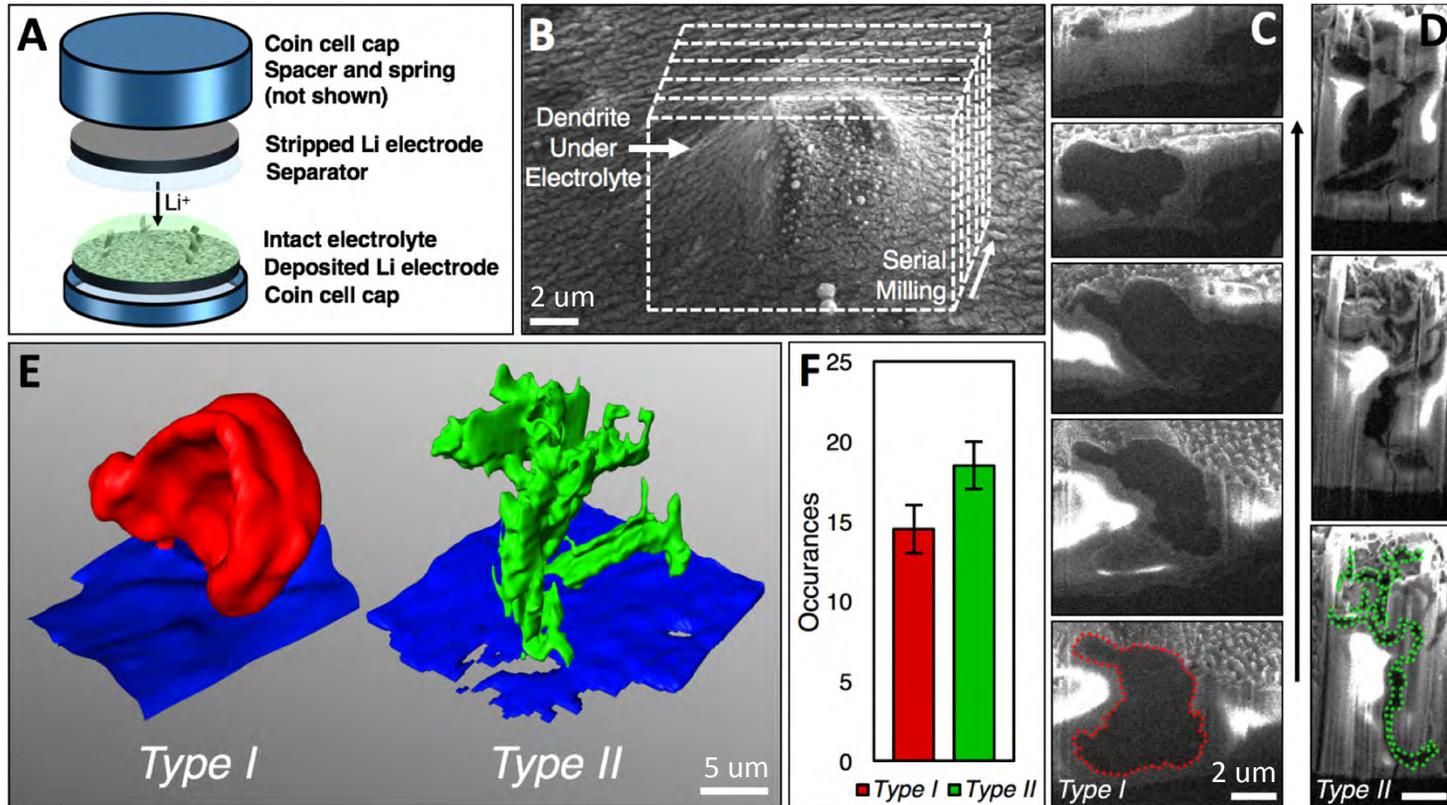


Ensemble Average  
Many:1 mapping

**Catalysts, like people  
are inhomegenous**

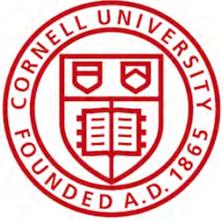


# Dendrite morphology from cryo-FIB



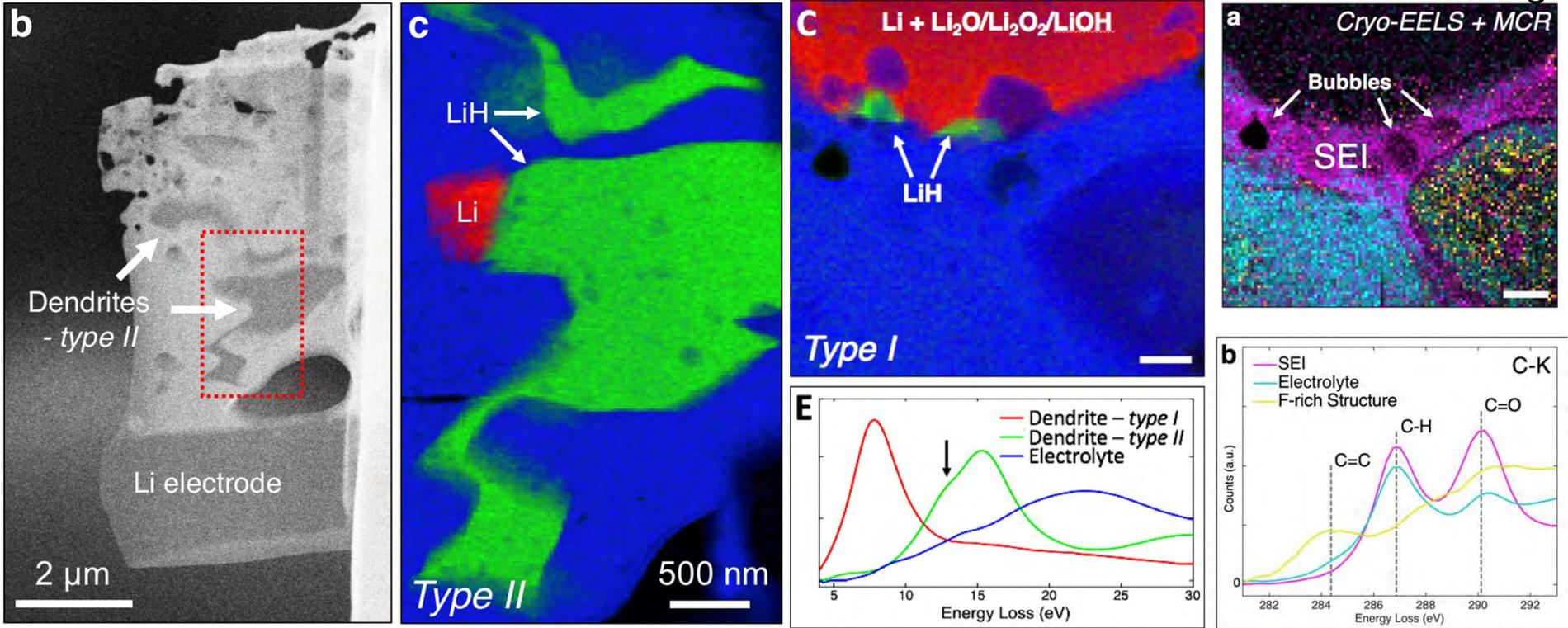
Serial cross-sectional cryo-FIB/SEM revealed two distinct dendrite morphologies  
Narrow connection sites for Type II might give rise to “dead” Li

Zachman and Kourkoutis, *Nature*, (2018)



# Two structurally and chemically distinct dendrite structures

*Spatial variation of carbon bonding*

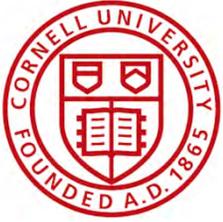


Type I dendrite:

- lithium metal, partially oxidized at its core
- transitioning to peroxide and hydroxides near the surface

Type II dendrite:

- uniform lithium hydride



# Electron Microscopy for Catalyst Characterization

*New Detector Records every electron that passes through the sample, at every probe position!*

Scanning beam

pixel array detector

**EMPAD:**  
High Speed: 0.86 ms/frame  
Single Electron Sensitivity  
 **$e^-$  Dynamic Range > 1,000,000:1**

*Microsc. & Microanal.* **22**, 237 (2016)

Bilayer  $\text{MoS}_2$

5 Å

GUINNESS WORLD RECORDS

Nature **559**, 343 (2018)

### Sub-nm Resolution, pm-Precision Strain Mapping

$\text{Pt}_3\text{Co}$

2 nm

20 mrad

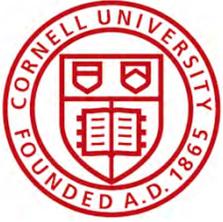
$1 \text{ \AA}^{-1}$

### Strain profile

$a/a_0$        $b/b_0$

1.08  
1.04  
1.00

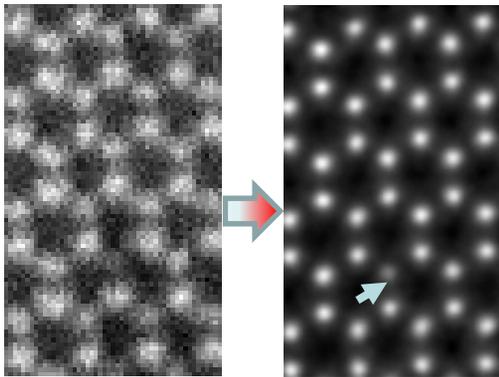
strain maps of core-shell nanoparticles with **1,000X less dose** than EELS or AC-STEM atom fitting & any orientation.



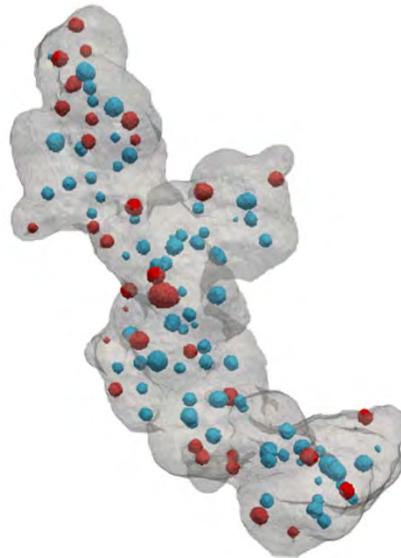
# Electron Microscopy for Catalyst Characterization

1. Structure of Catalytic Sites    2. Catalysts/Support Interactions    3. Catalyst Aging Mechanisms

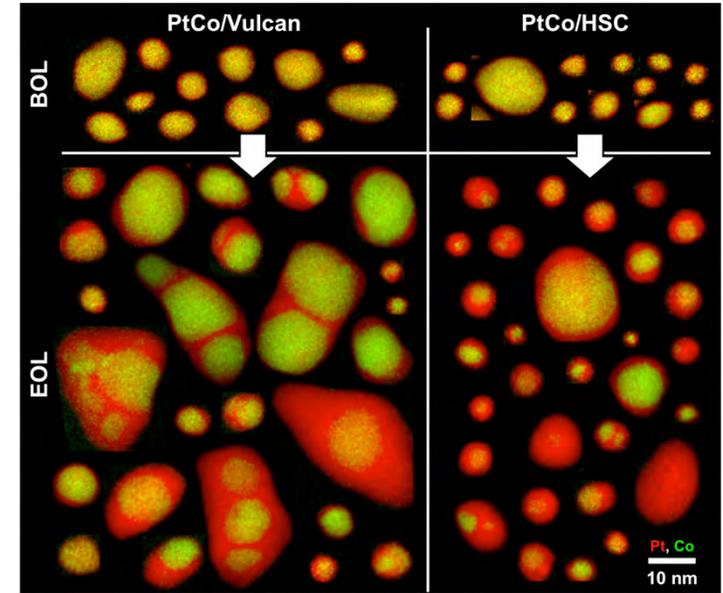
Improved resolution means improved detection of local atomic structure of catalytic sites



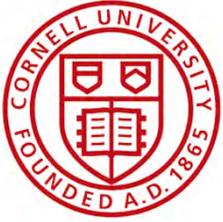
Sulfur monovacancy in MoS<sub>2</sub>



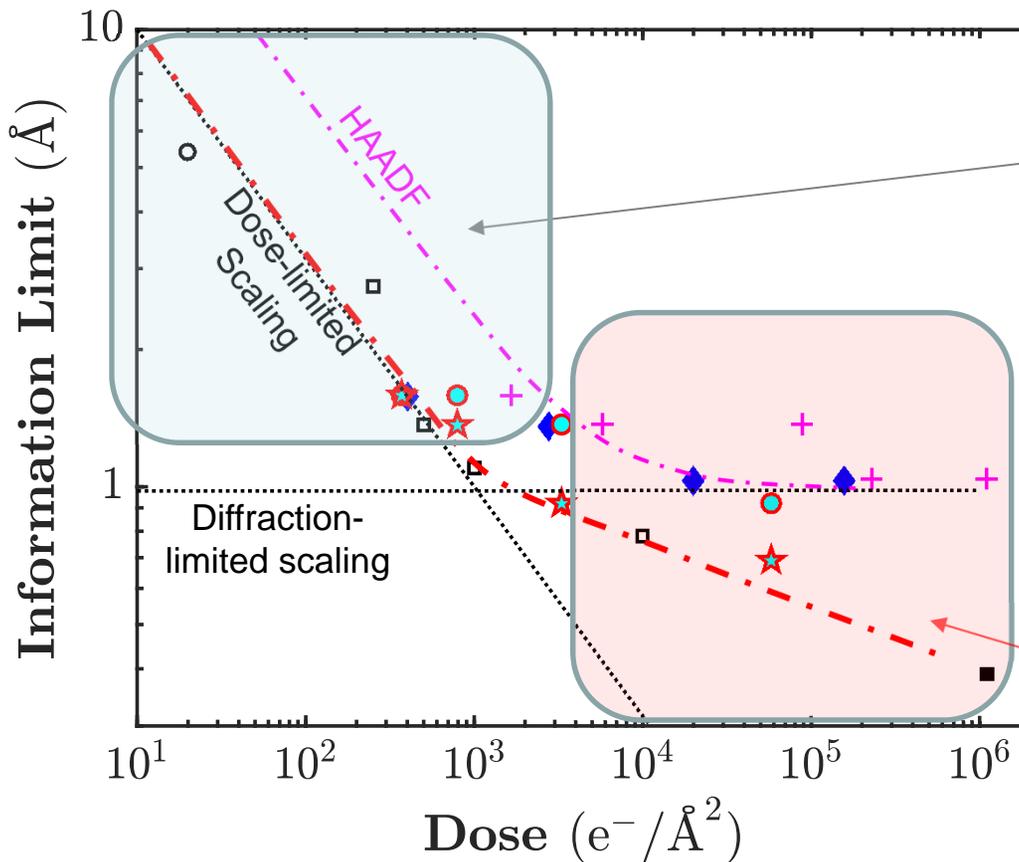
Pt catalysts dispersed in High-surface-area C



Capabilities include in-situ and electrochemical liquid cell microscopy



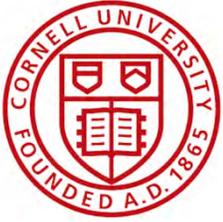
# Dose vs. Resolution: 2 Regimes



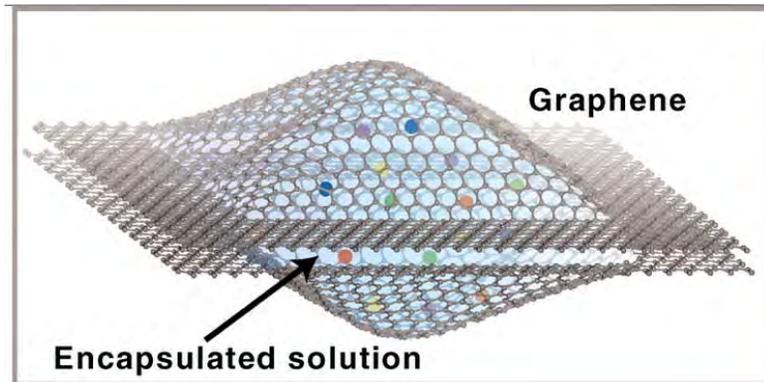
- Low Dose:**
- Poisson Uncertainty  $\gg$  Probe Size
  - $d \propto 1/\sqrt{N}$
  - Ptychography vs ADF:  
double info limit for the same dose

- High Dose:**
- Probe size set by diffraction limit
  - ADF resolution saturates
  - Ptychography resolution continues to improve with dose (2-3 x).

Imaging methods enabled by new detectors require  $\sim 10x$  less dose

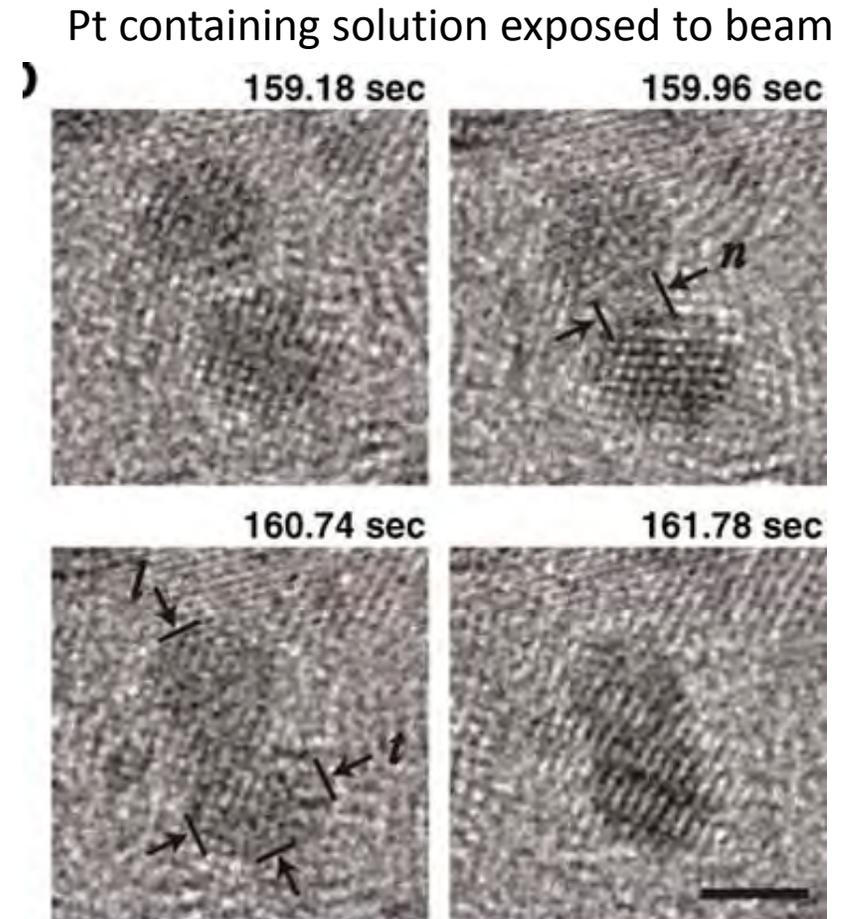


# Best resolution = graphene liquid cell

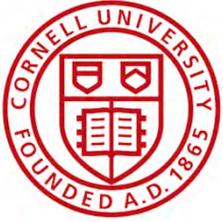


Thin liquid + thin encapsulating sheets  
= atomic resolution Pt growth and  
coalescence

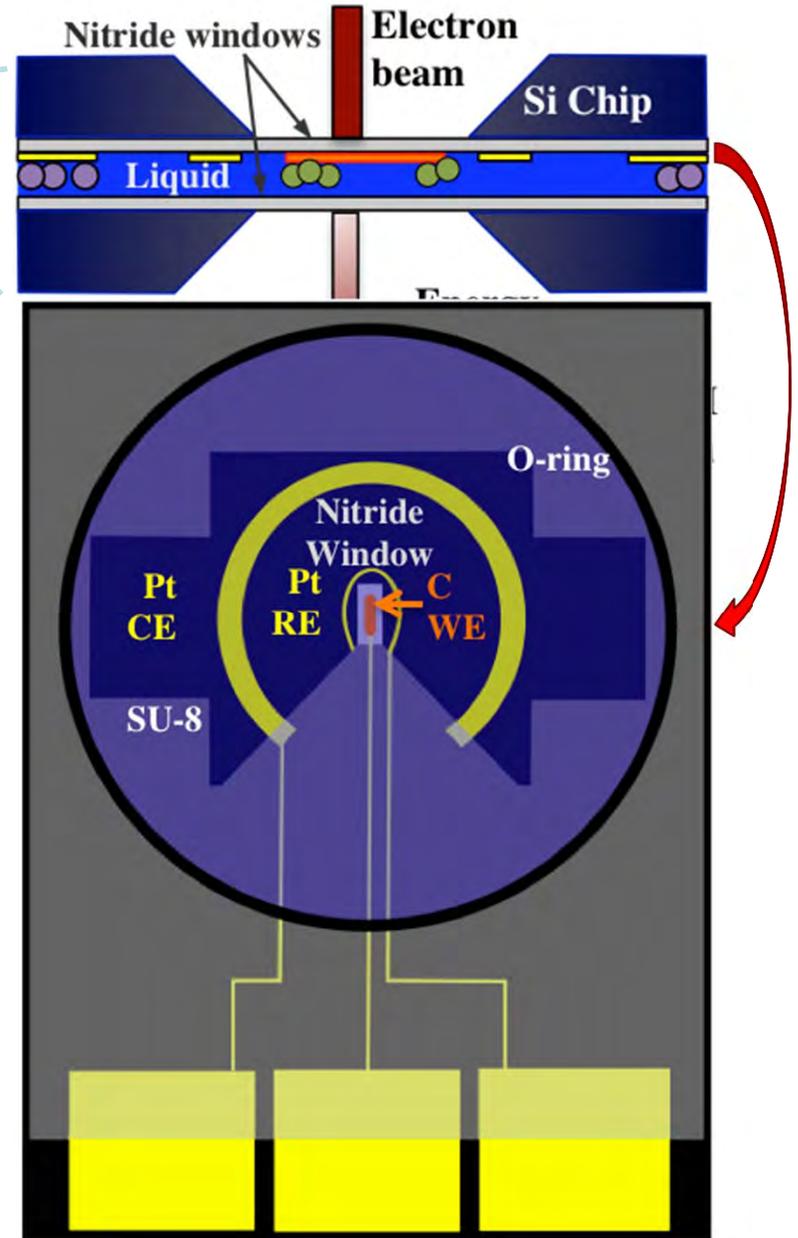
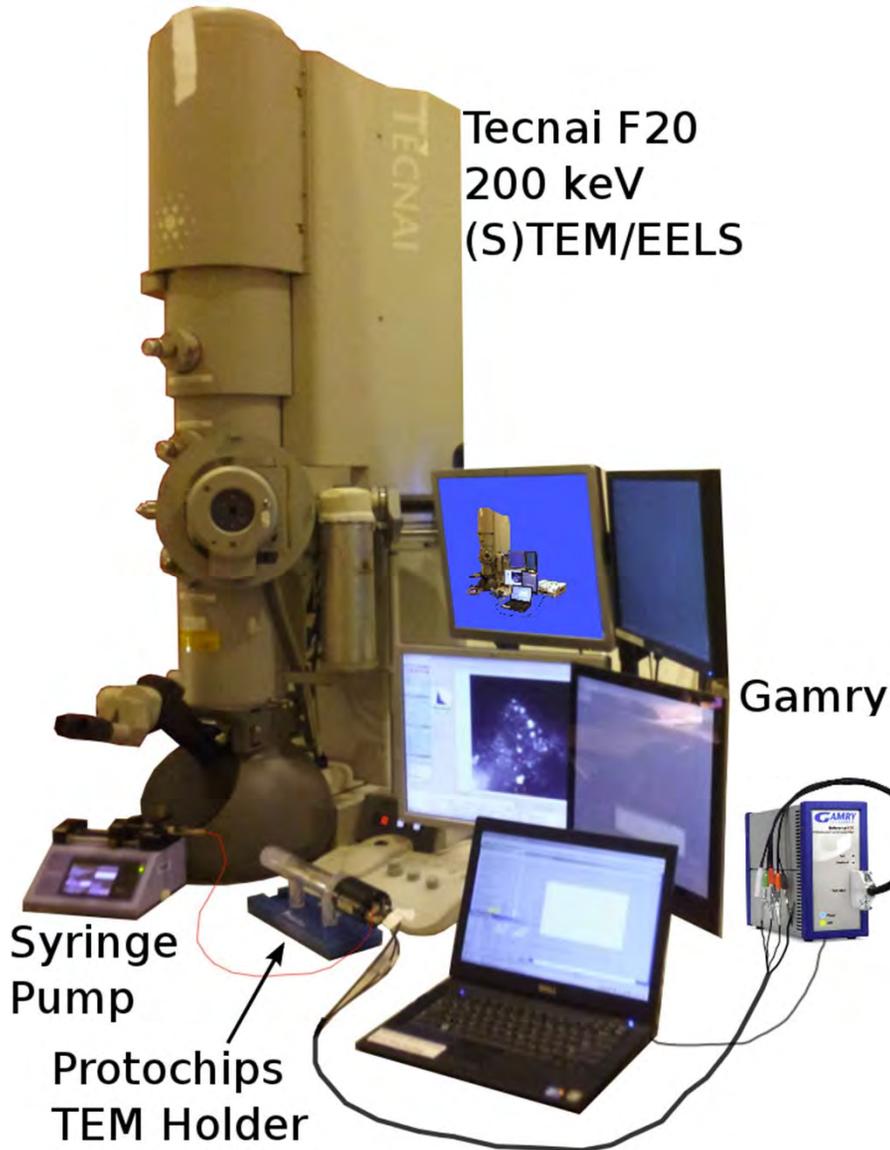
**Beam induced phenomenon!**  
Electron beam can make a  
very reducing environment.



Yuk et al. Science **336** (2012)



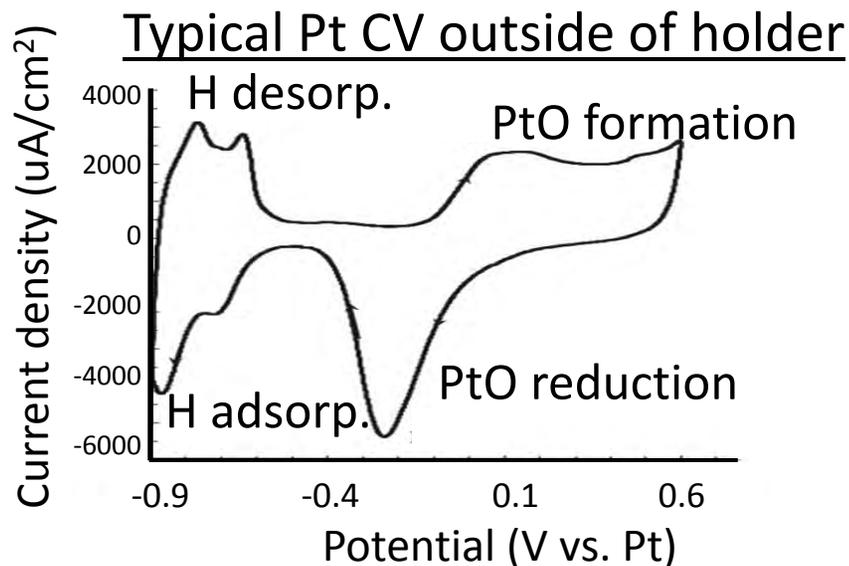
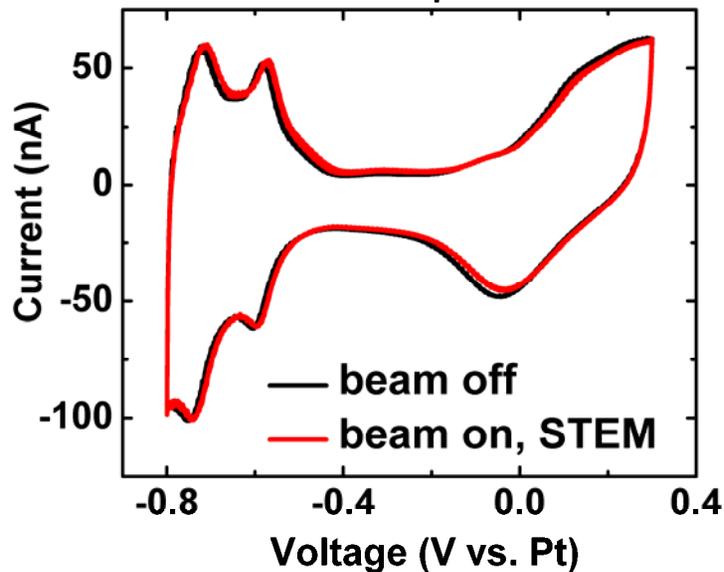
# TEM electrochemical cell





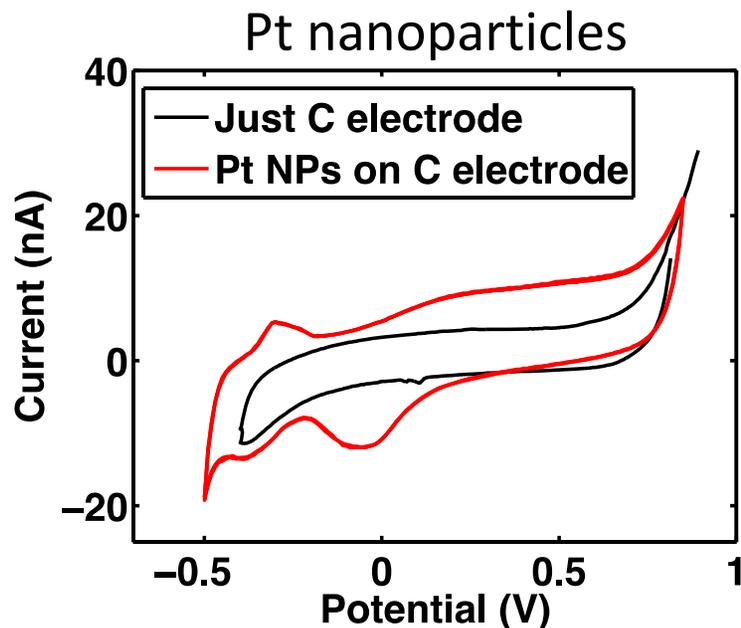
# Electrochemistry in the TEM

CV of Pt film deposited on chip

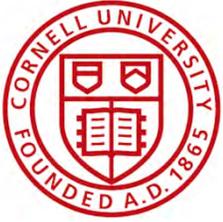


Adapted from M.G. Hosseini, et al. *Electrochimica Acta*, Volume 70, 30 (2012), 1–9

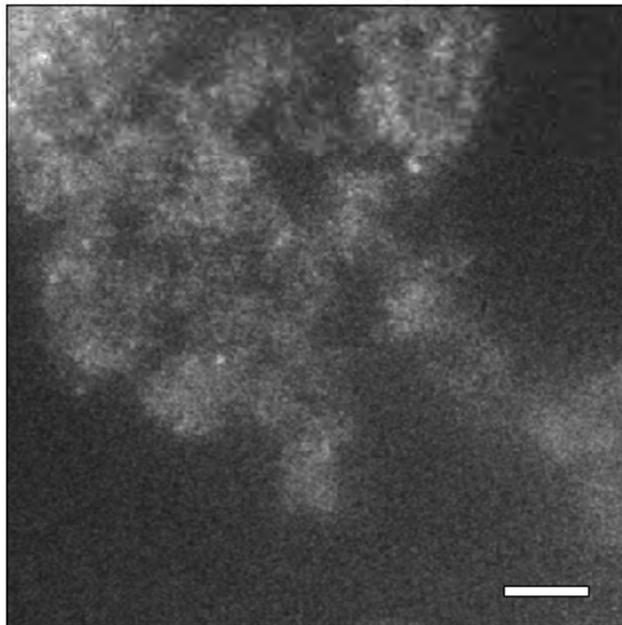
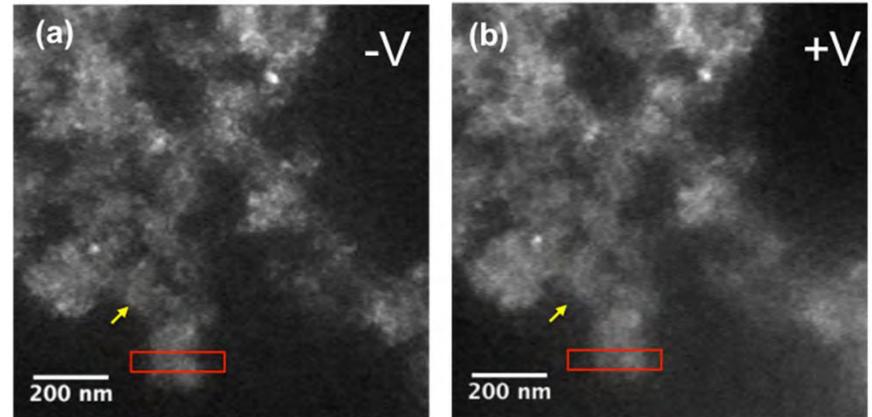
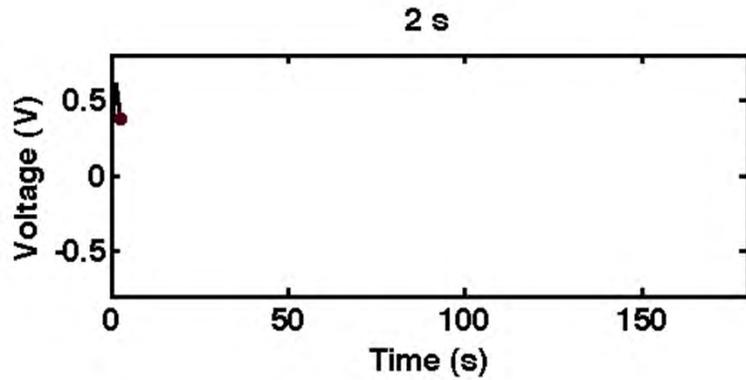
Similar behavior to a standard microelectrode!



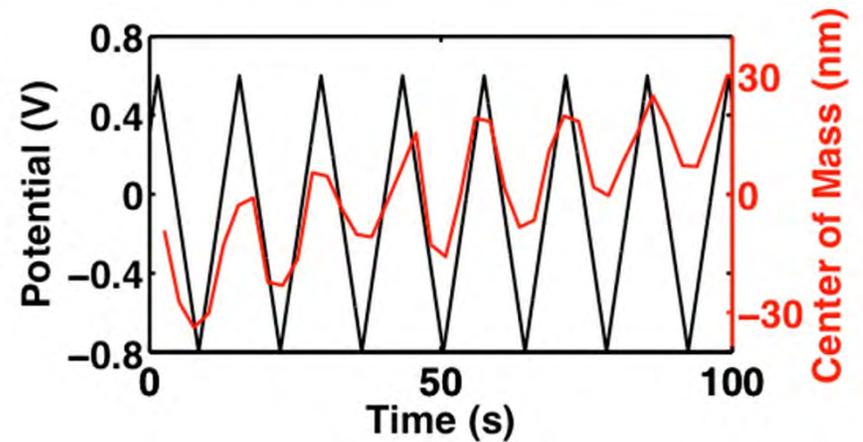
Holtz et. al. *NanoLetters*. 14(3) 1453 2014.



# Carbon support bending during potential cycling



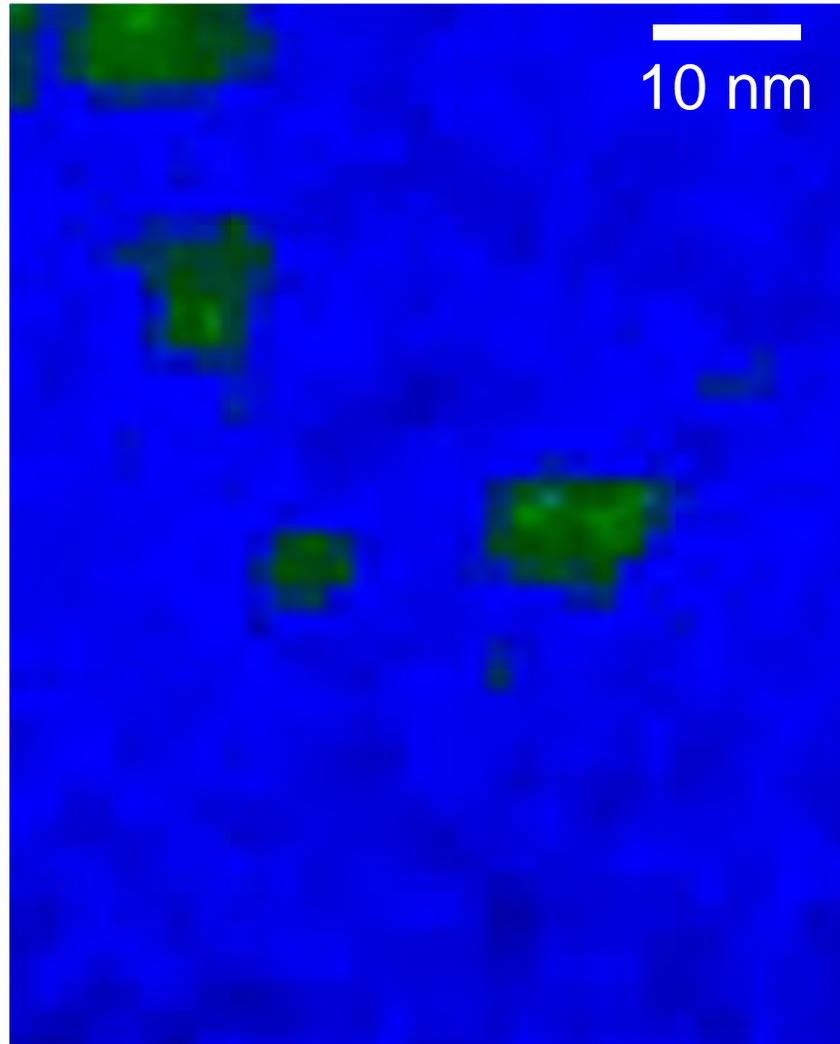
200 nm



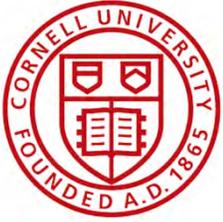
200 mV/s

0.1 M H<sub>2</sub>SO<sub>4</sub>

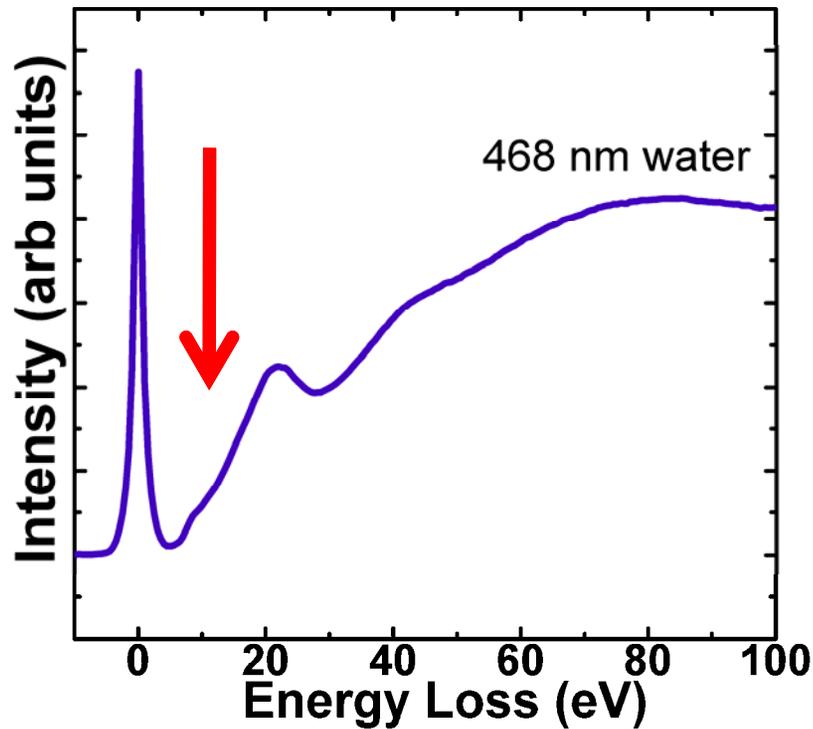
# Watching Coalescence



Cycling +0.5 V to +1.2 V at 500 mV/s for 220 cycles

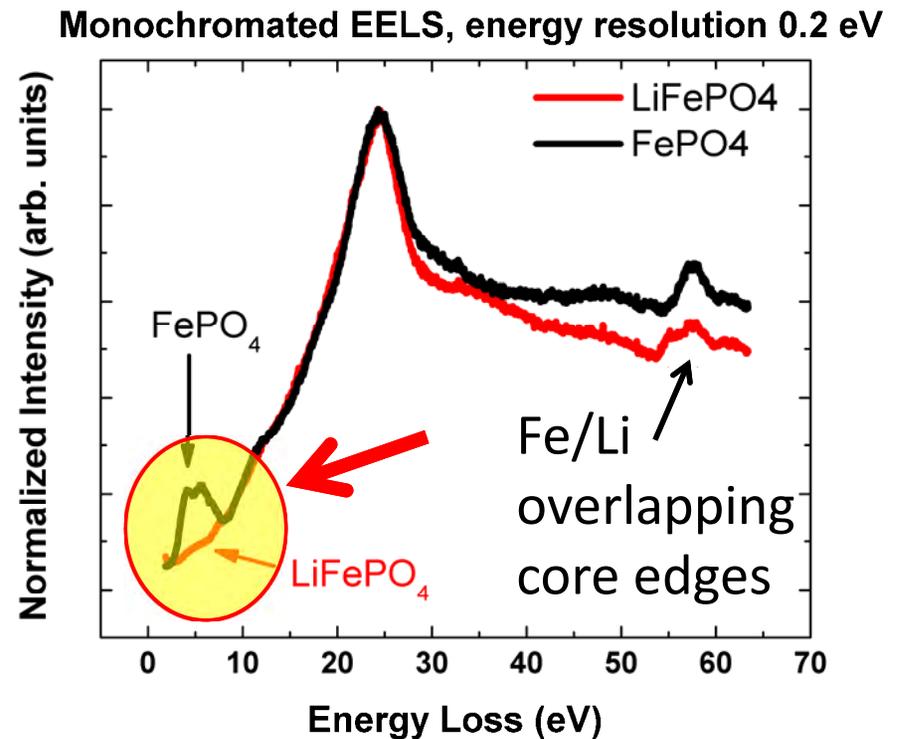


# Valence EELS for Batteries



This is interesting for  
Li-ion battery applications!  
Want Rapid Spectroscopy

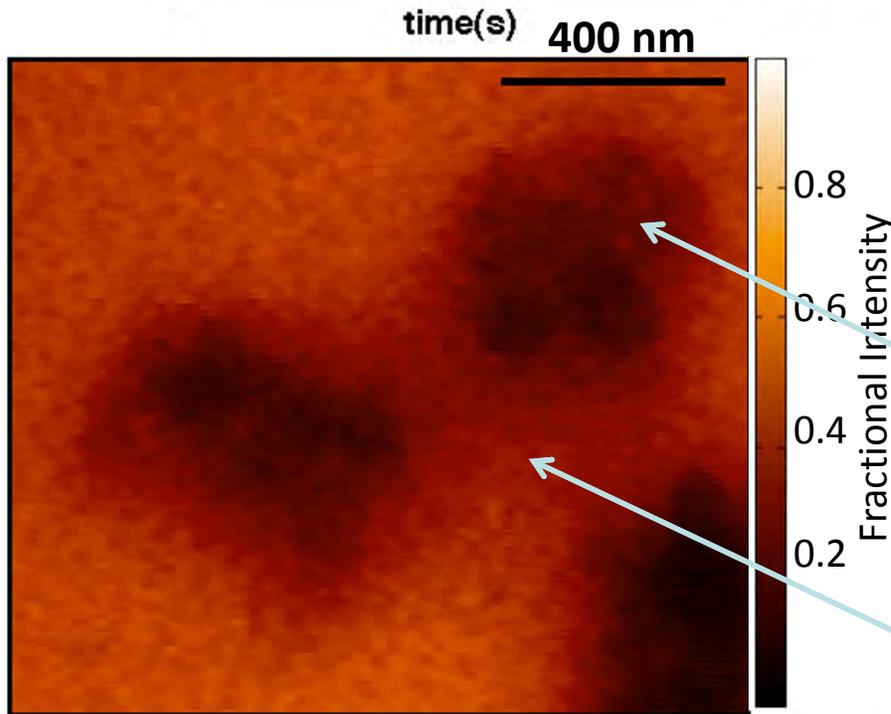
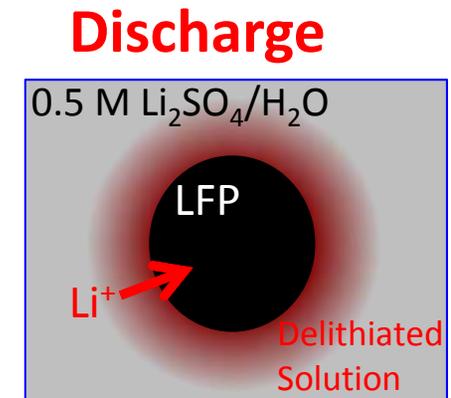
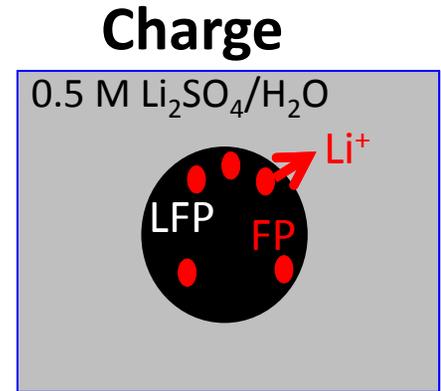
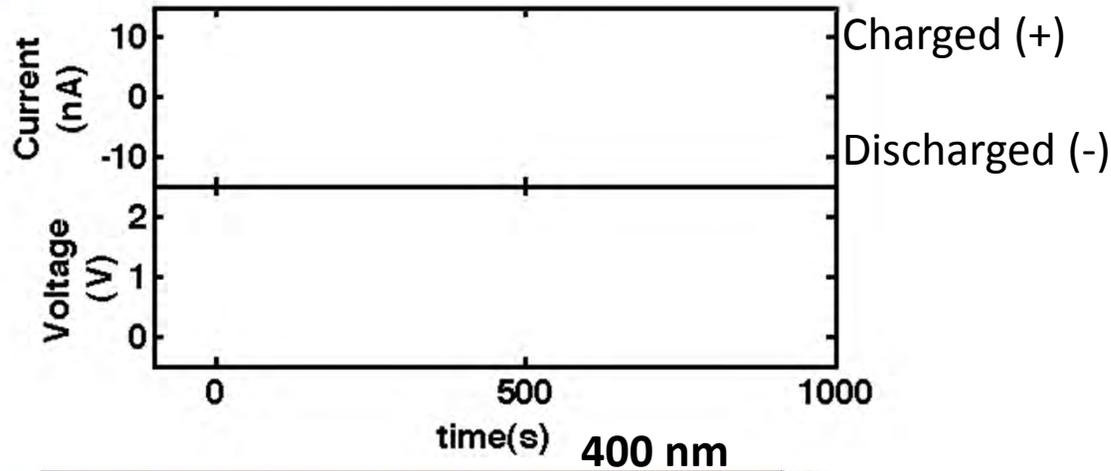
- Before Plasmon  
(in the optical gap!)  
monochromator helps



Holtz et. al. NanoLetters. 14(3) 1453 2014.



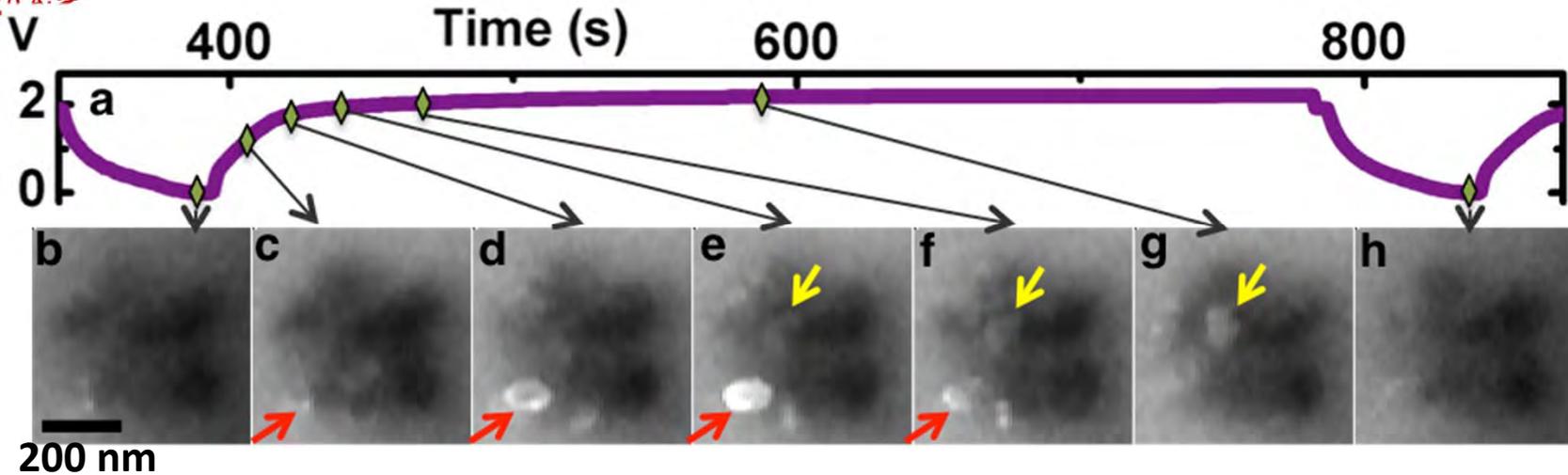
# We can observe the lithiation *in situ*!



5 eV EFTEM image!



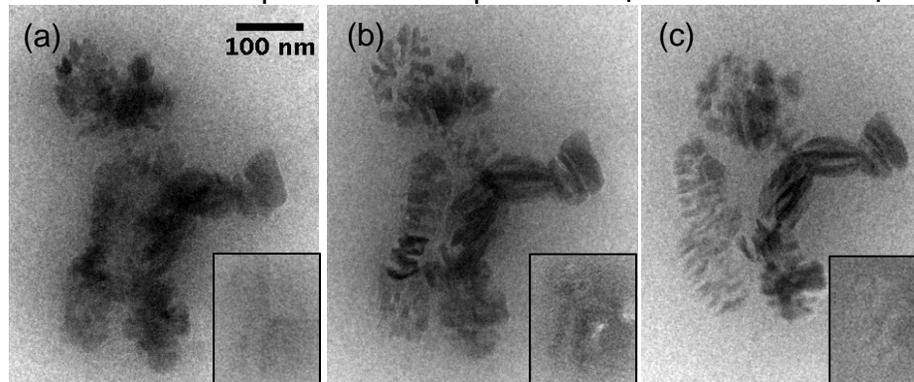
# Mechanism of Lithiation

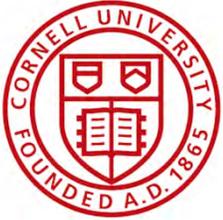


2 dominant modes of lithiation on same aggregate

Fracturing:

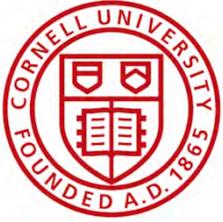
3 cycles + Charge 1 cycle + Discharge  
 $\text{LiFePO}_4 \rightarrow \text{FePO}_4 \rightarrow \text{LiFePO}_4$





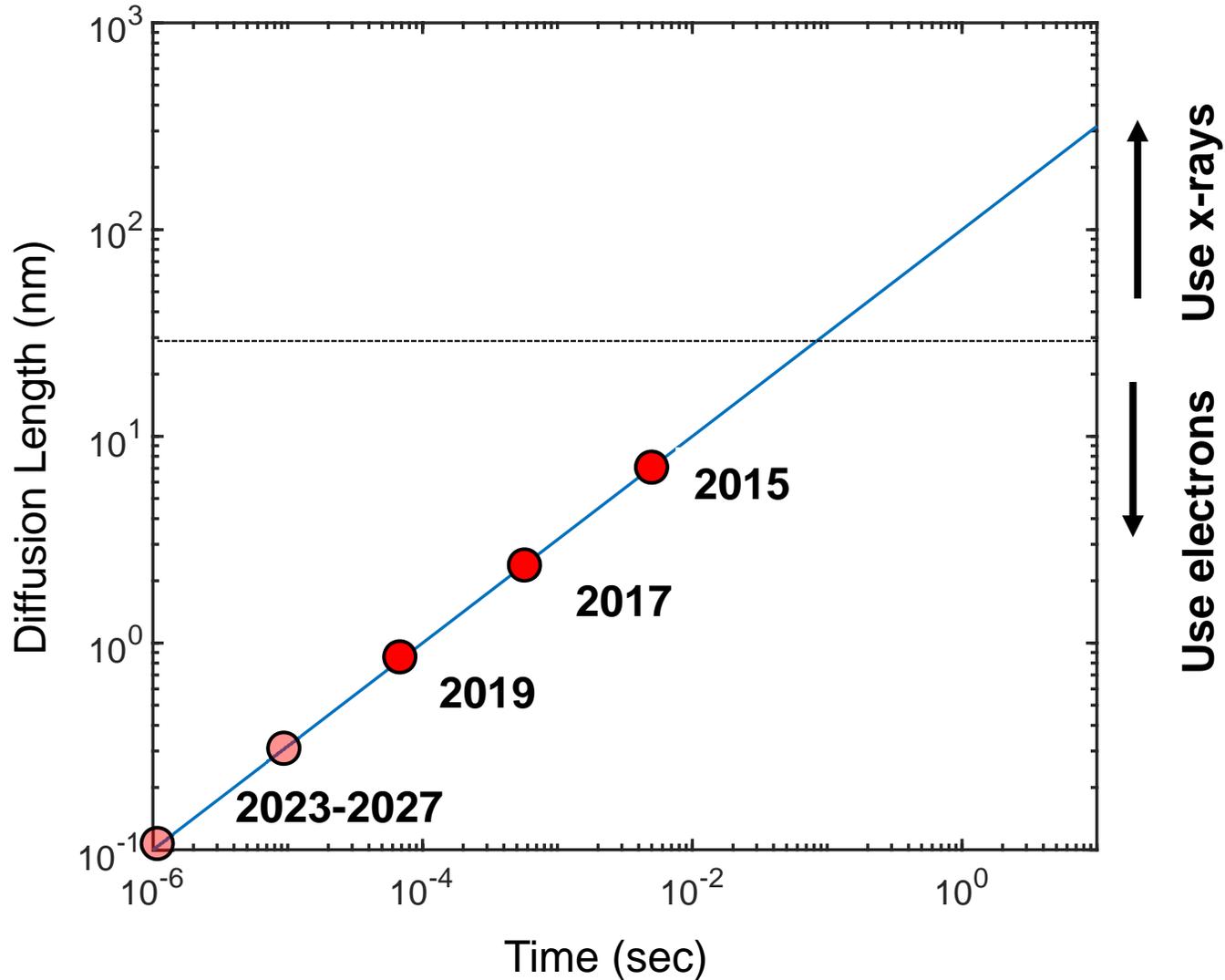
## Time and Length Scales for EChem

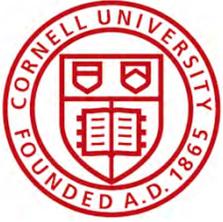
- *Transition States:* picoseconds & Angstroms
  - Today: need light activation
    - Diffraction from crystals, but not imaging
    - XFEL's are designed for fs, not ps!
  - Opportunities for ERL & next gen light sources
  - Ultrafast electron diffraction and imaging (2<sup>nd</sup> harder)
- *Diffusion of ions:*  $l = \sqrt{Dt}$ 
  - Spans ns (liquids) to seconds
  - But lengths scale as well
  - Opportunities for TEM & Synchrotrons



# Detector Speed to Image Diffusion

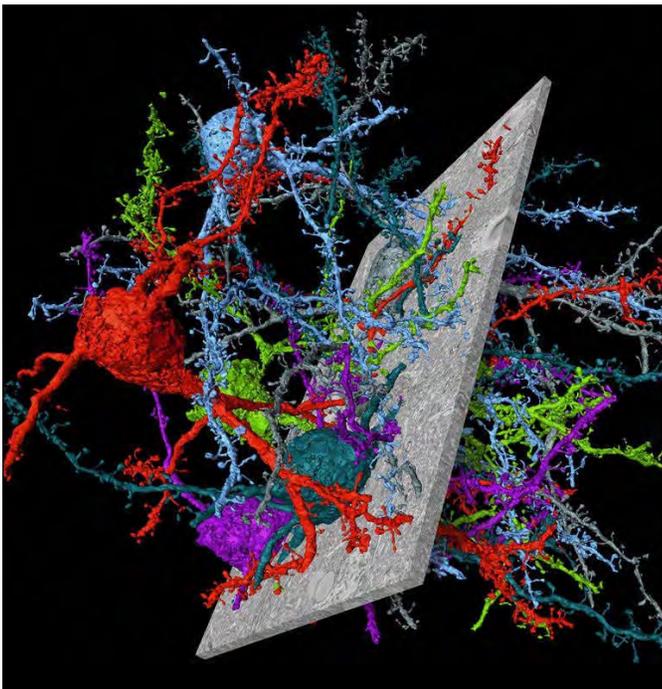
For  $D \sim 10^{-14} \text{ m}^2/\text{s}$  - typical for Li in  $\text{CoO}_2$





# The Connectome: A multi-scale problem

Map the brain's interconnectivity of the  
 $\sim 10^{10}$  nerves and  $\sim 10^{14}$  synapses



MPI Frankfurt

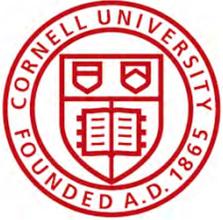
2018: EM maps  $0.4 \times 0.4 \times 1 \text{ mm}^3$  volume at  $\sim 30 \text{ nm}$  resolution in 300-500 days ( $6 \times 10^{12}$  voxels)

Today: EM maps  $1 \times 1 \times 1 \text{ mm}^3$  volume at  $\sim 4 \times 4 \times 40 \text{ nm}$  resolution in 90 days ( $1.5 \times 10^{15}$  voxels)

Goal: Map  $1 \text{ mm}^3$  (as  $7 \times 7 \text{ mm}^2 \times 20 \text{ microns}$ ) at  $30 \text{ nm}$  resolution in  $< 1$  day. ( $36 \times 10^{12}$  voxels)

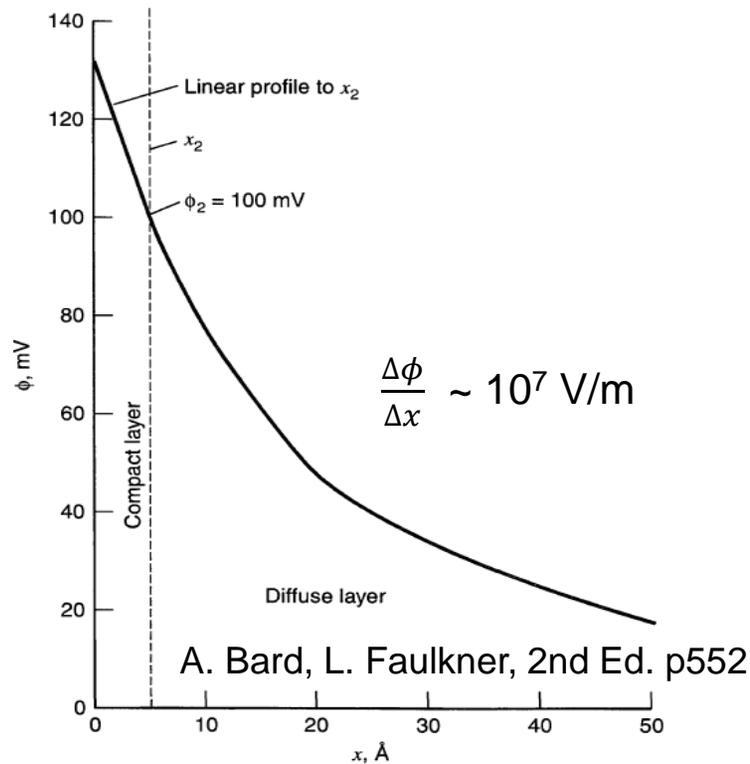
Do-able with a  $1 \text{ MHz}$  frame rate detector and out-of-focus ptychography at a beam line.  
( $100 \text{ nm}$  sampling, 50 projections = 7 hours  
&  $4\text{-}16 \times 10^{15}$  Bytes)

Next: Find and trace all connections!

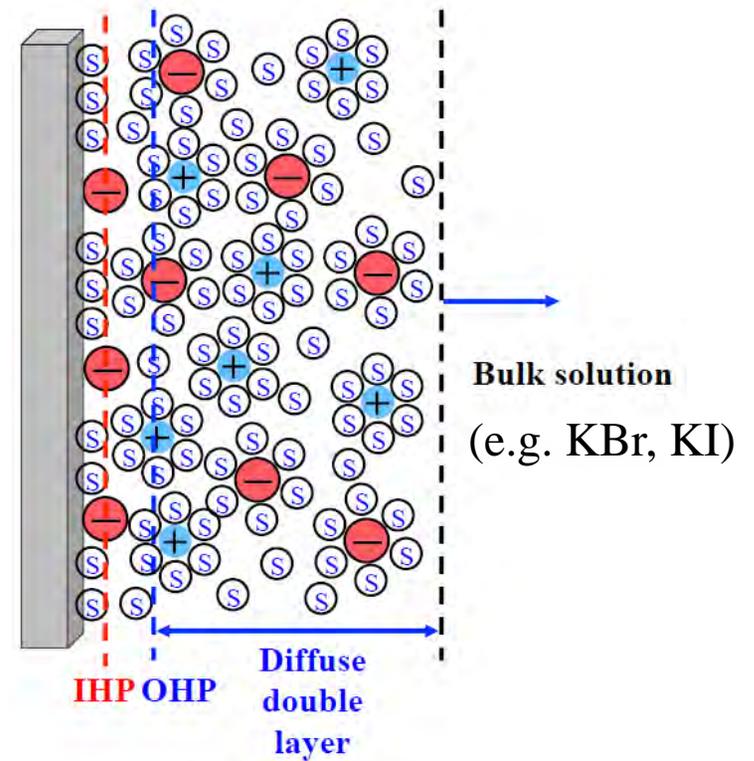


# Can we resolve the Electrochemical Double Layer?

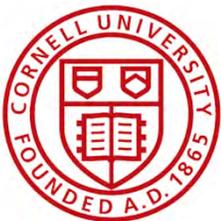
## Gouy-Chapman-Stern (GCS) Model Potential Profile of 0.01 M Solution(aq)



## Anion Adsorption on Electrode Surface

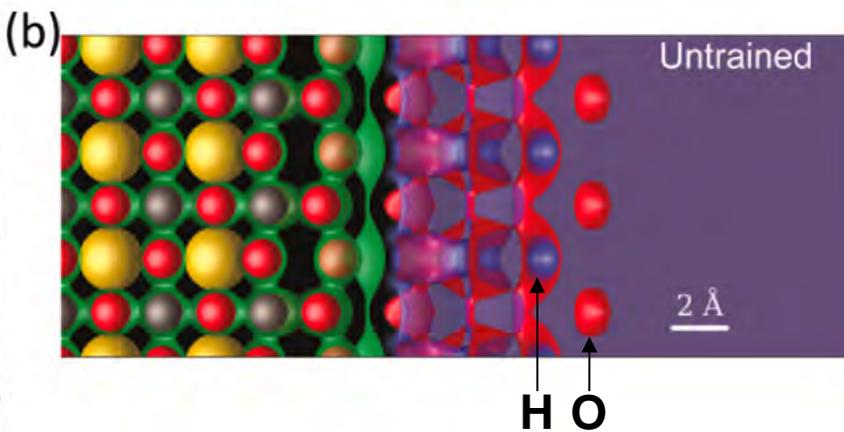
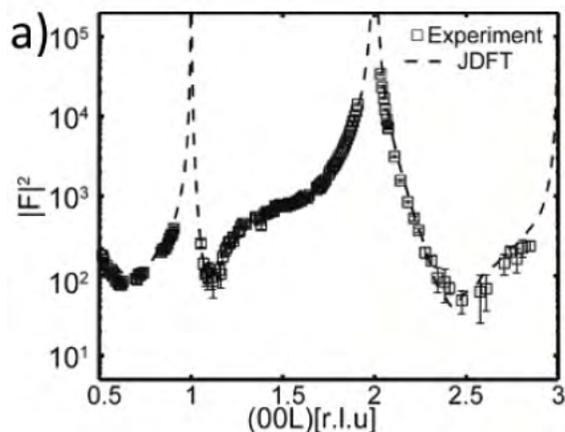


- With heavy ions in solution, screening charge detectable by TEM
- Electric field measurable as deflection of the beam (DPC)
- At nanometer resolution, nanoparticles are as easy to study as crystals

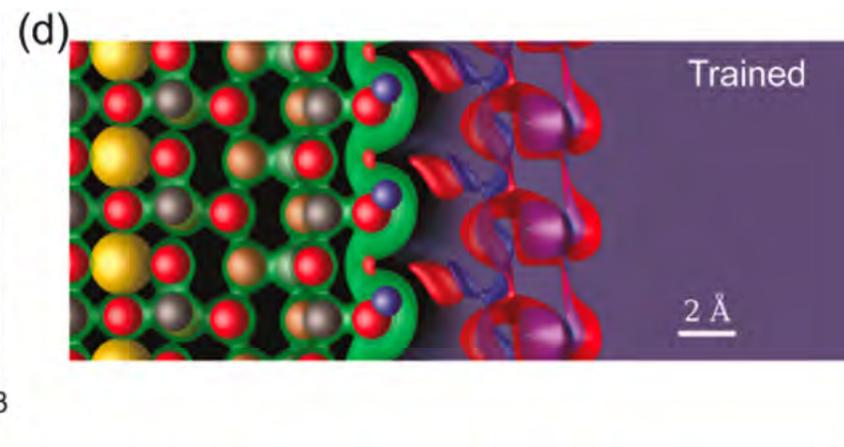
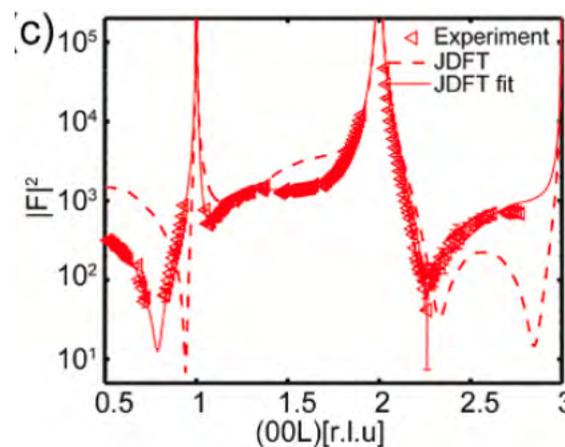


# Seeing the Double Layer in XRD?

X-Ray diffraction of Single-Crystal SrTiO<sub>3</sub> in water & JDFT prediction



flat



+defects

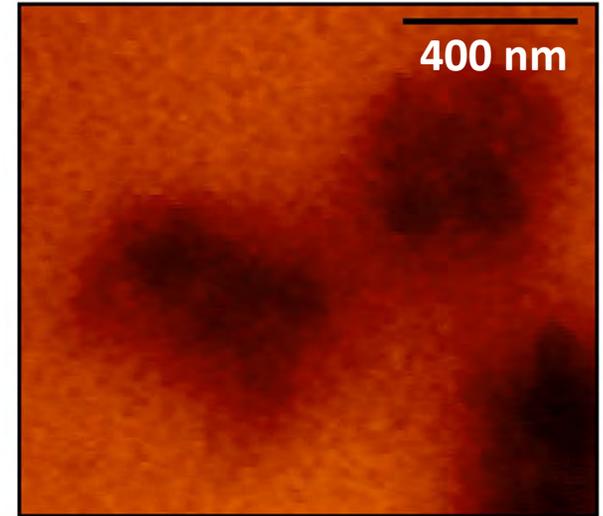
Plaza et al, JACS (2016) V138, 7816

With high-dynamic range detectors, density profile of the electrolyte should be accessible

# Outlook

## Electrochemistry in the TEM

- Radiation damage limits resolution (1-10 nm)
- CryoEM ~ 10-100x dose increase
- Single-Particle EM – atomic resolution by averaging
- EMPAD detector for dose-efficient imaging
- Multi-scale imaging enabled by fast detectors
- Embrace, do not fear big data.



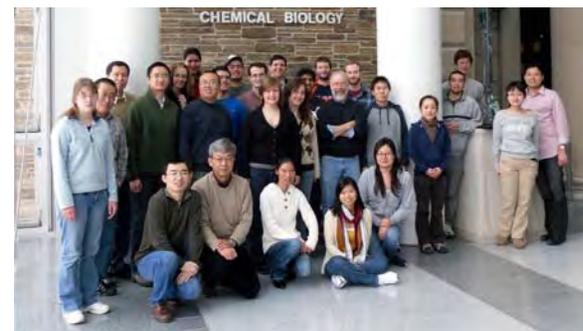


**David Muller**

Megan Holtz  
Elliot Padgett  
Robert Hovden  
Paul Cueva  
Kayla Nguyen  
Barnaby Levin  
Huolin Xin

**Lena Kourkoutis**

Michael Zachman  
Berit Goodge  
  
**Tomas Arias**  
Deniz Gunceler  
Shankar Sundararaman  
Katie Schwartz



**Abruña Group**

Yao Yang      Deli Wang  
Yingchao Yu      Johary Rivera  
Nikky Ritzert      Eric Rus

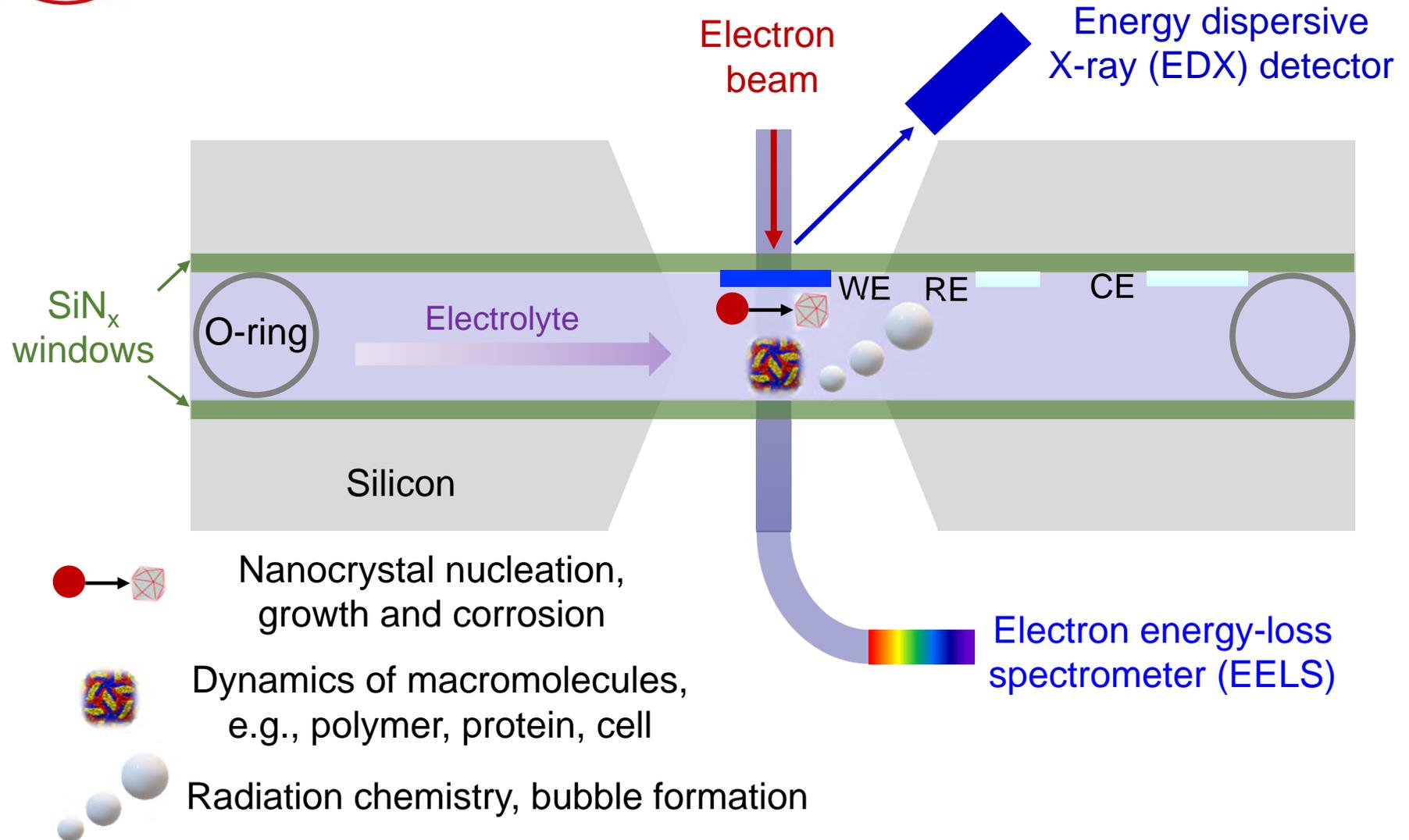


**Sol Gruner**

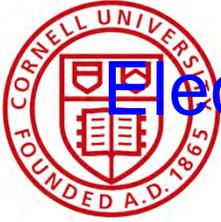
Mark Tate      Hugh Philips  
Kate Shanks      Praful Purohit



# In Situ TEM to Tackle Dynamic (Electro)Chemical Reactions at the solid-liquid interface

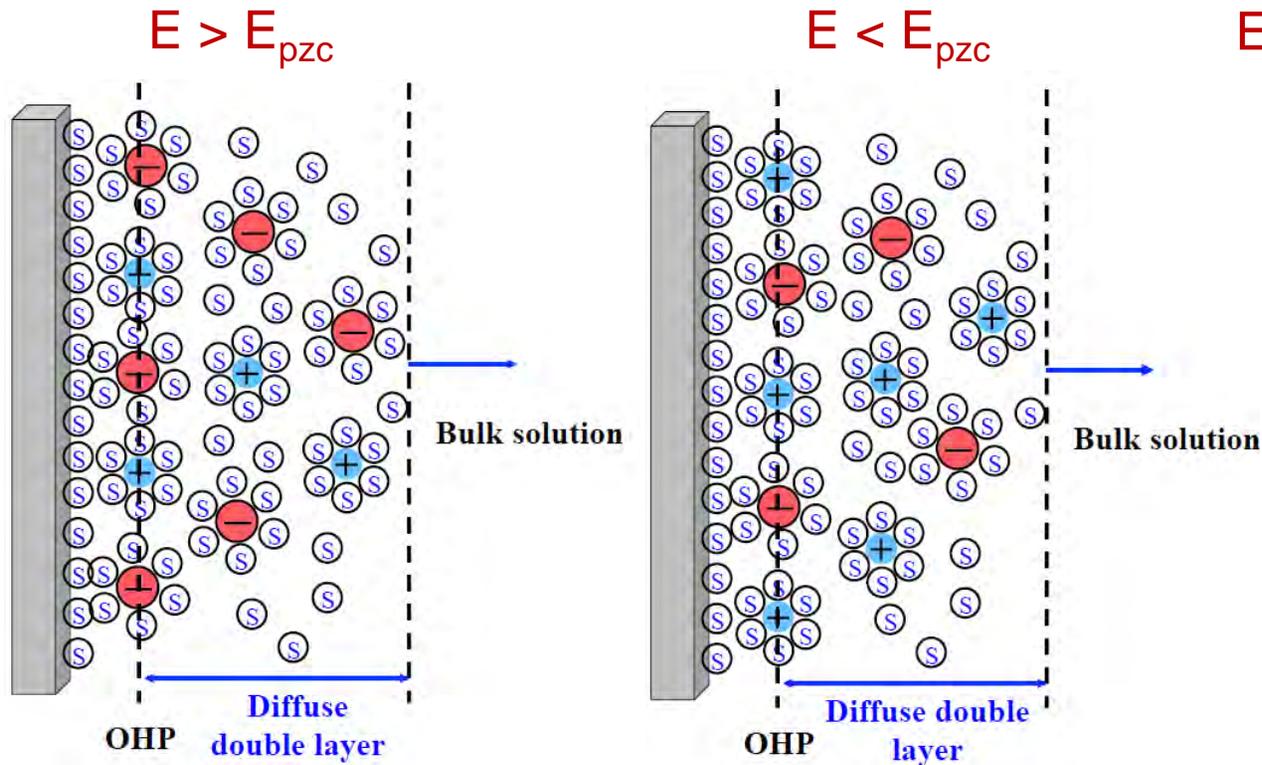


WE, CE, RE: three-electrode system  
(working, counter and reference electrodes)



# Electrochemical Double Layer at Various Conditions

Potential of zero charge ( $E_{PZC}$ )  
the potential at which the electrode  
will be electrically neutral, relative to the solution

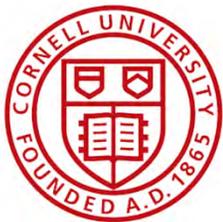


$E_{pzc}$  of Hg in 1M solution

Electrolyte	$E_{PZC} / V$
NaF	-0.47
NaCl	-0.56
KBr	-0.65
KI	-0.82

D. Grahame, *Chem. Rev.*  
1947, 41, 441

The electrochemical double layer depends on  
applied potentials, types and concentration of electrolyte



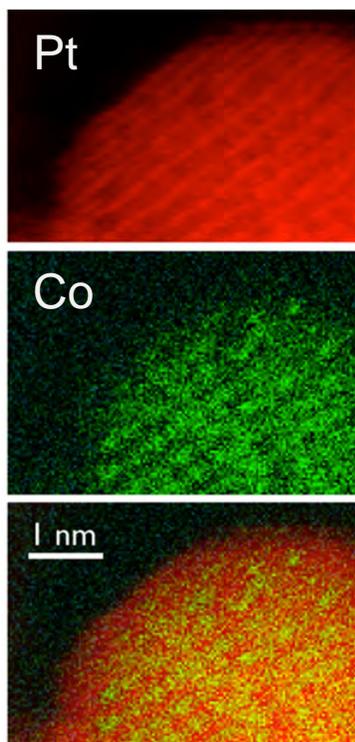
# Seeing Electrochemistry Live at the Atomic Scale

*David A. Muller, Cornell University*

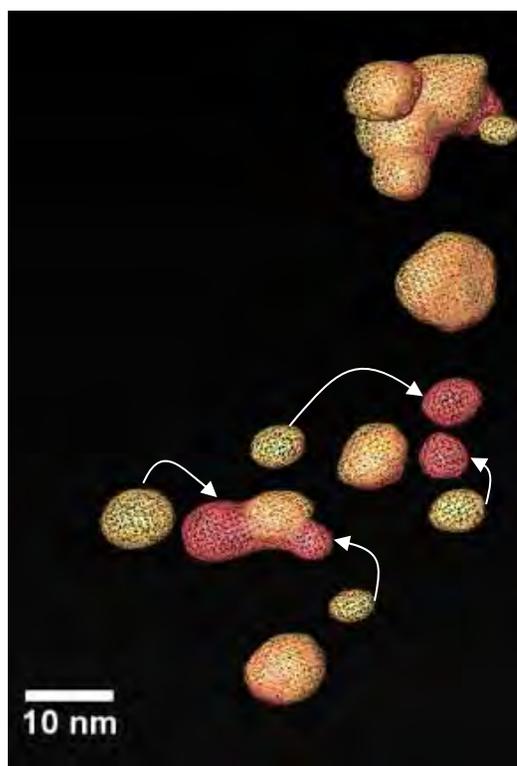
*Ex Situ*



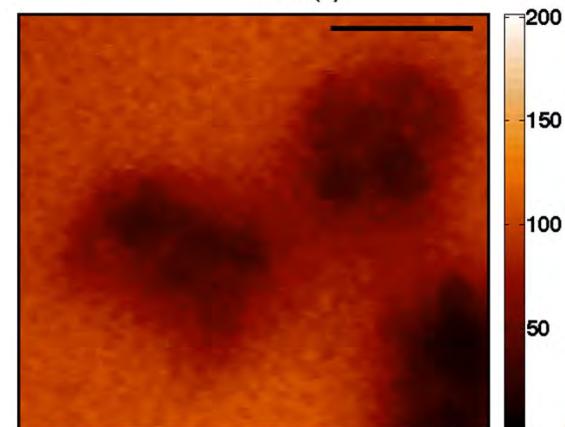
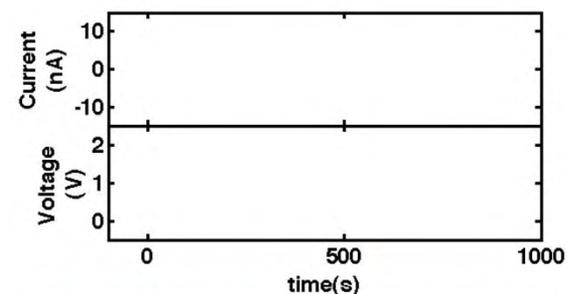
*In Situ/Operando*



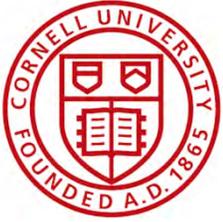
*Atomic Resolution*



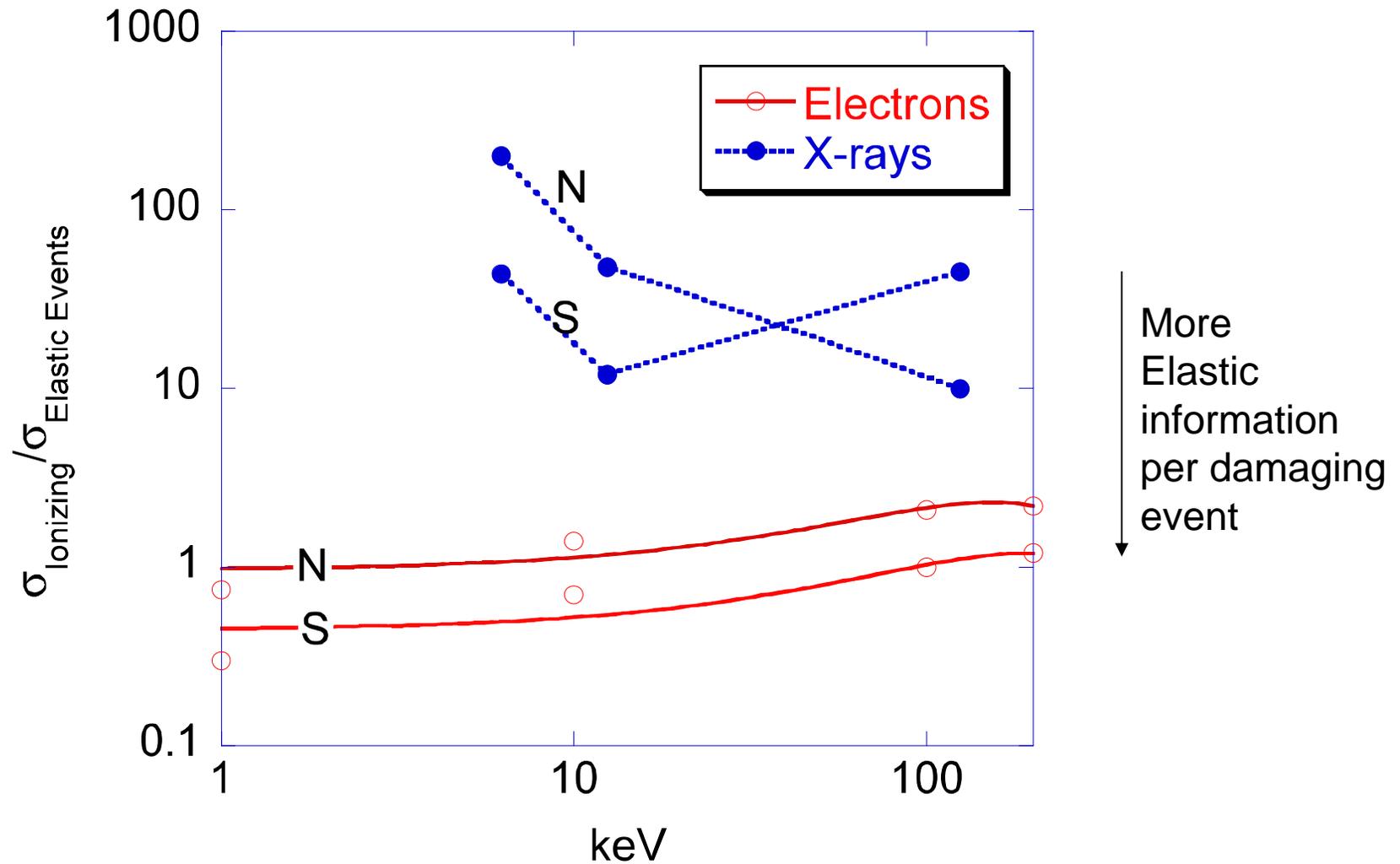
3D



*Electrochemistry in liquids*

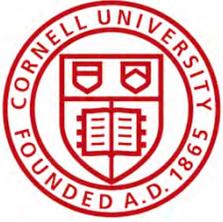


# Radiation Damage as a Fundamental limit



Data from Breedlove and Trammell, Science 170 (1970) 1310-1313

For electrons  $\sigma_i / \sigma_e \sim \ln(E)$

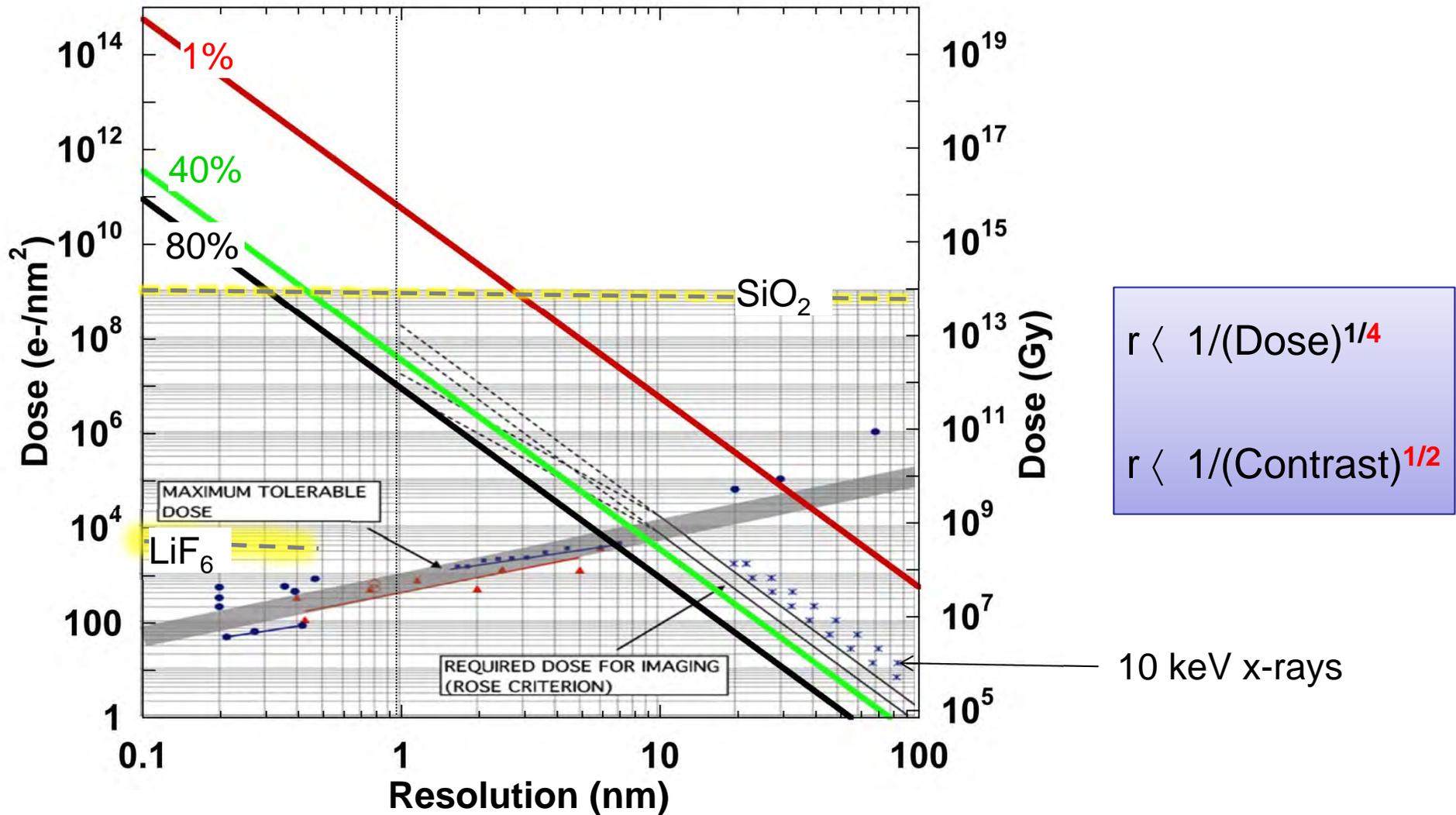


# Dose Required for 3-D Reconstructions is high

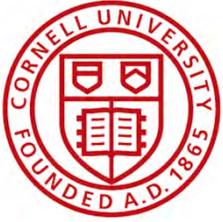
B. F. McEwen et al, Journal of Structural Biology **138** 47–57 (2002)

Saxberg & Saxton, Ultramicroscopy **6**, 85–90 (1981)

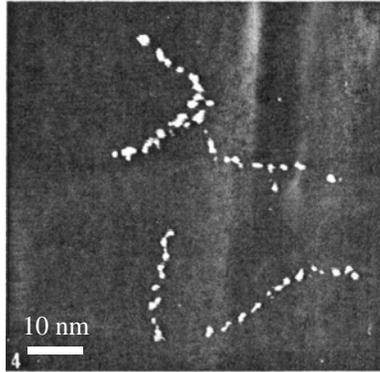
M.R. Howells et al, J. Electron Spec. Rel. Phenom. **170** 4 (2009)



Dose-limited resolution for oxides is ~0.2-2 nm and ~3-10 nm for batteries

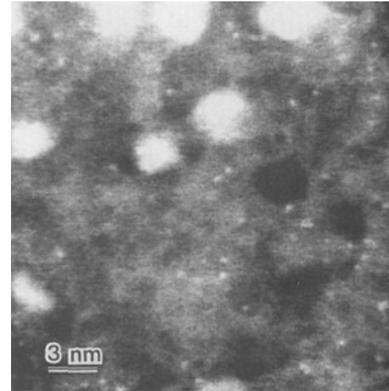


# Single Atom Imaging: Catalysis and Dopants



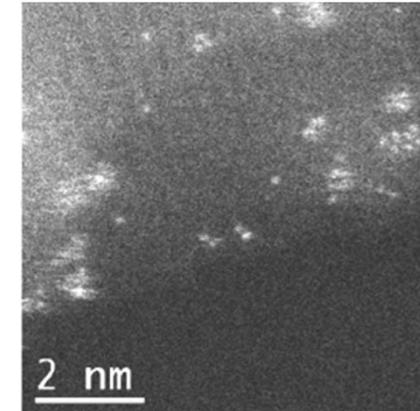
Th on a-C

Crewe *et al.*, *Science* **168**, 1338 (1970)



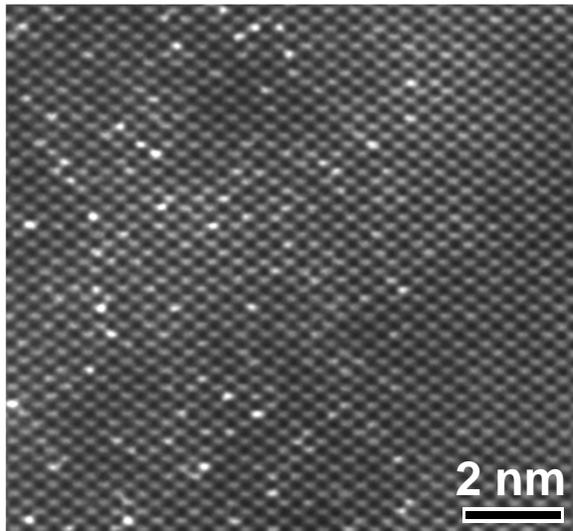
U on a-C

Treacy & Rice, *J. Microscopy* **156**, 211 (1989)



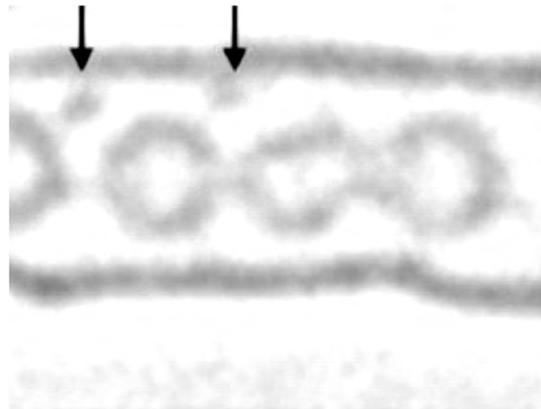
Pt on a-Al<sub>2</sub>O<sub>3</sub>

Blom *et al.*, *Microsc. Microanal.* **12**, 483, 2006



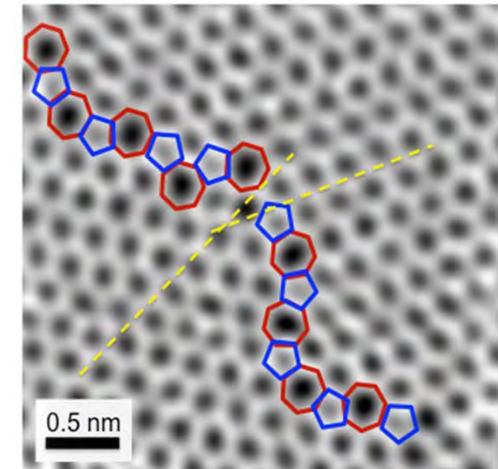
Sb in Si

Voyles *et al.*, *Nature* **416** 826 (2002)



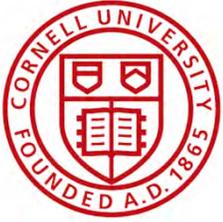
K-doped C<sub>60</sub> peapods

Guan *et al.*, *PRL* **94**, 045502 (2005)

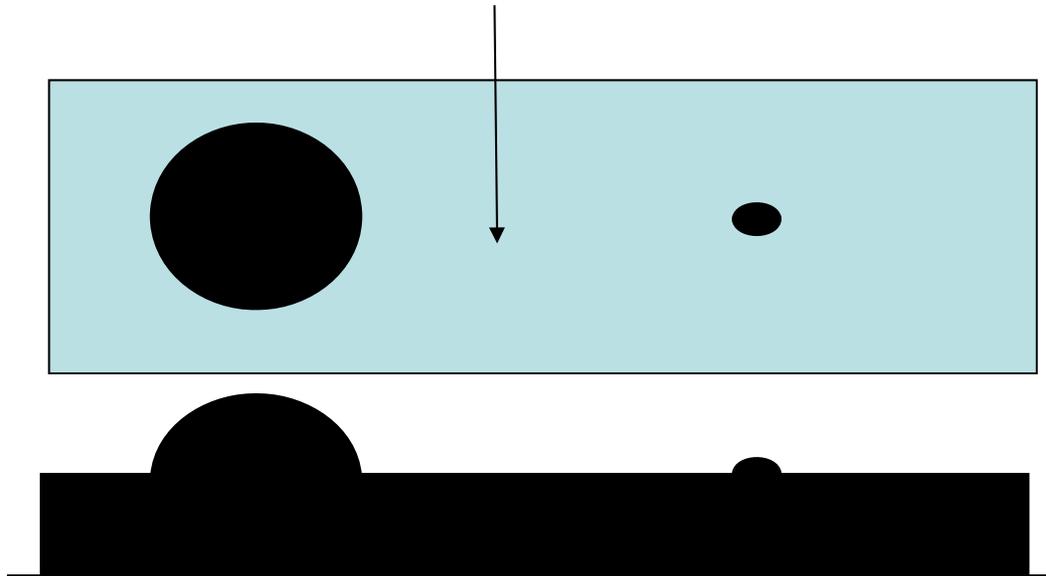


Graphene Grain Boundaries

Huang *et al.*, *Nature* **469**, 389 (2011)

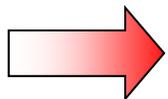


# *High Resolution= Thin Sections*



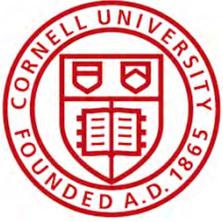
Small features have low contrast (and for a fixed dose we trade 2D resolution for contrast)

**Resolution < Sample Thickness**



Need to make thin samples (true for x-rays as well as electrons)

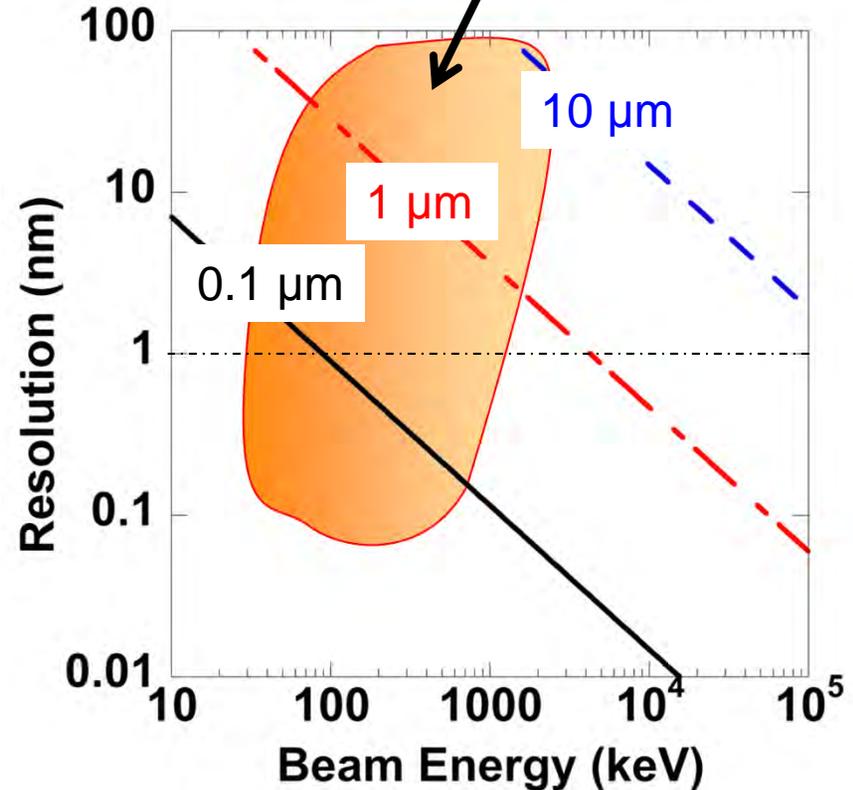
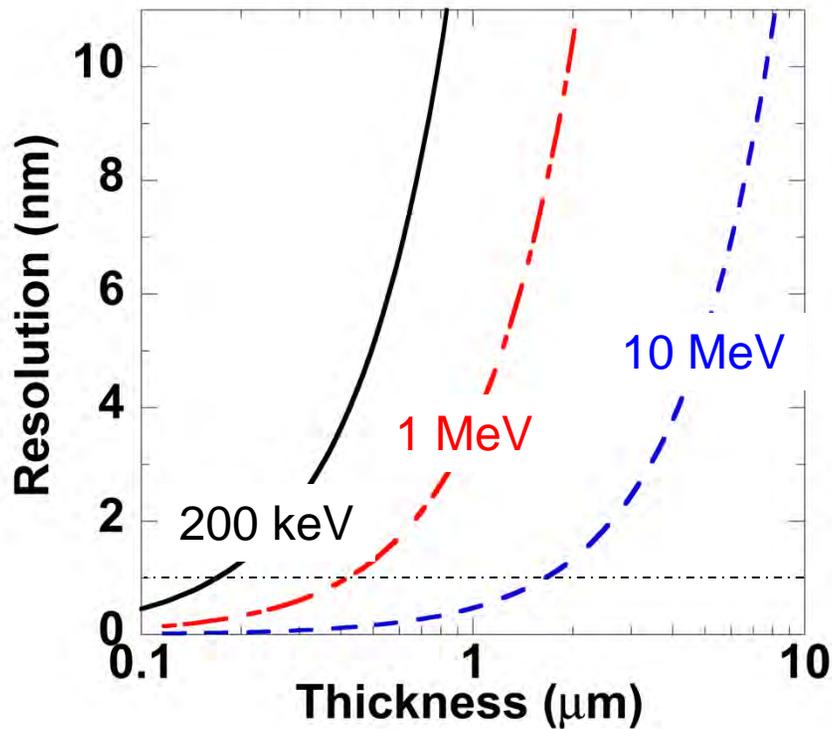
(unless we have a fluorescence detection method & there is only 1 object)



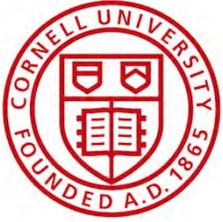
# Beam Spreading (in Water)

Beam Spreading  $b \propto \sqrt{\rho} \frac{Z}{E} t^{3/2}$

Electron's  
Home court advantage



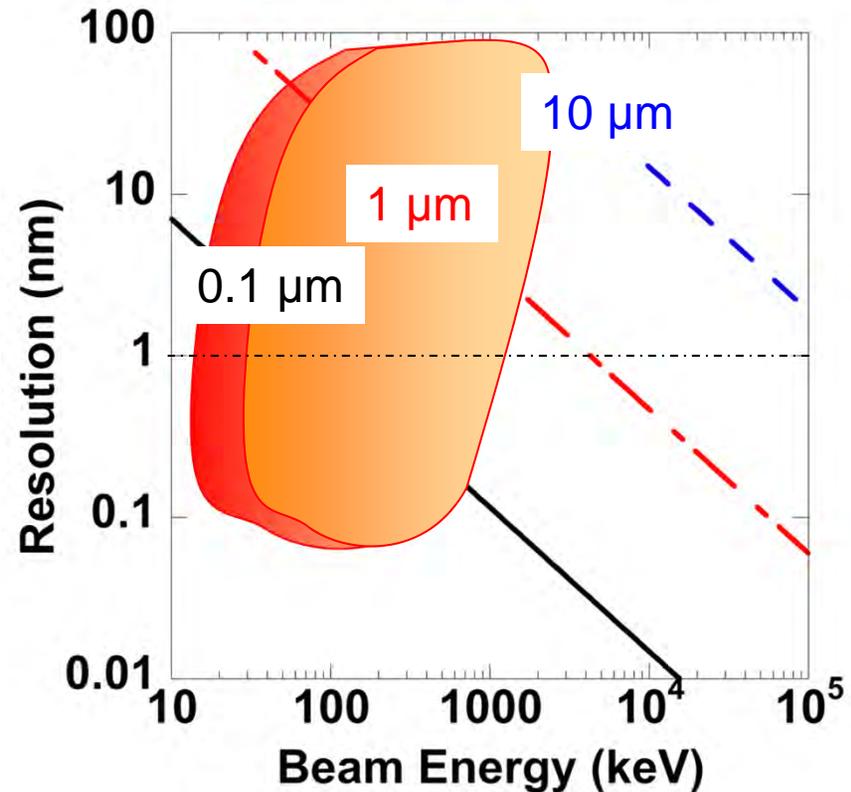
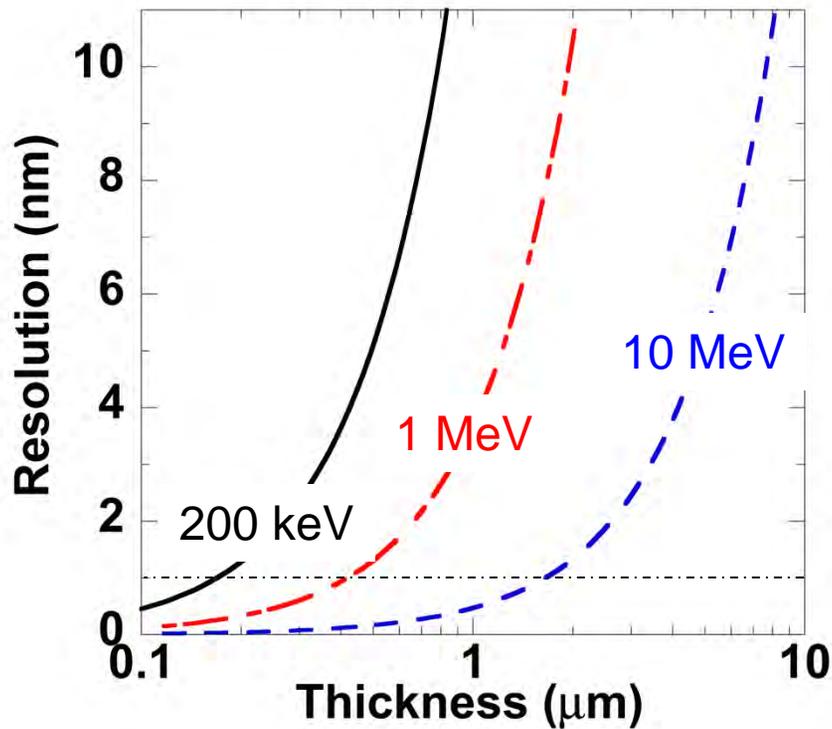
By changing the collection angle, can trade off a linear improvement in resolution (~ x 2) for an exponential drop in signal



# Beam Spreading (in Water)

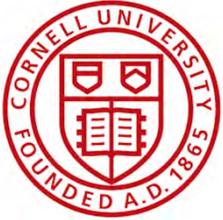
$$\text{Beam Spreading } b \propto \sqrt{\rho} \frac{Z}{E} t^{3/2}$$

Electron's  
Home court advantage



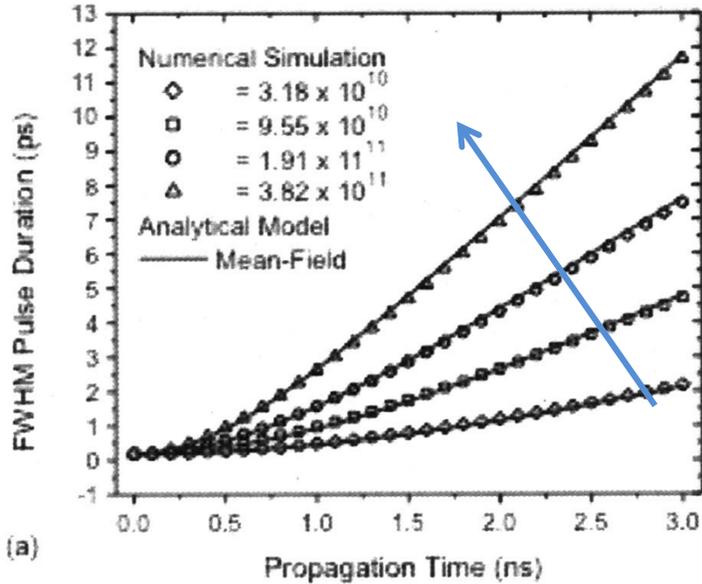
By changing the collection angle, can trade off a linear improvement in resolution ( $\sim \times 2$ ) for an exponential drop in signal

*Appl. Phys. Lett.* **88** 243116 (2006)



# Coulomb Blur Limits Resolution

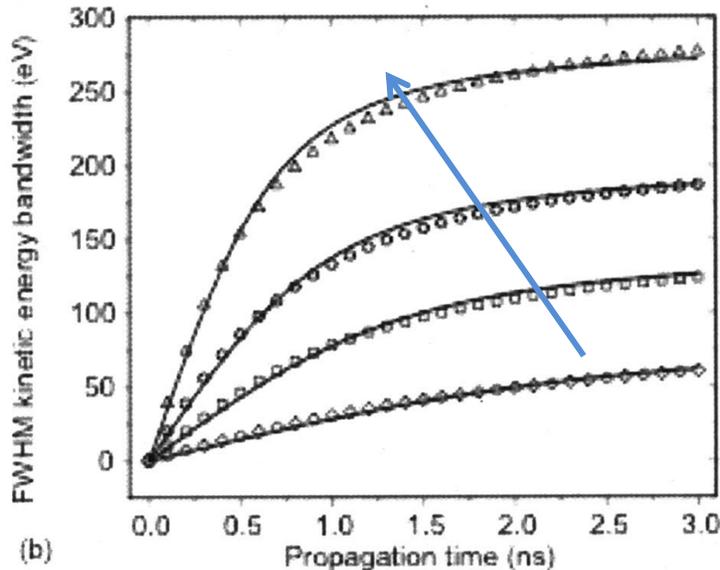
King et al, Journal Of Applied Physics **97**, 111101 (2005)



Typical e<sup>-</sup> transit times are 5-20 ns

Increasing current density degrades pulse duration

( $10^{21}$  e/m<sup>2</sup> for atomic resolution)



Increasing current density degrades energy spread

Chromatic blur limits spatial resolution

Soln: Either - Reversible single e<sup>-</sup> pump probe (atomic resolution, fs times)

- Single shot

( **0.3 nm in 100 ns, 10 nm in 10 ns** )



# *What Electrons do well*

## *Resolution Limits*

- 0.05 nm, 10 meV (0.1 nm & 50 meV together)
- Higher brightness Sources
- Radiation Damage limits the imaging of UV-sensitive materials (~10x better for electrons for elastic imaging)

## *Tomography*

- 0.2 - 3 nm resolution at 0.1-1  $\mu$ m thick
- 10 nm for 10-20 microns @ MeV
- Atomic resolution looks possible in inorganics

## *EELS & EDX Imaging (not time-resolved)*

- 2D Chemical information at the atomic scale
- 3D at the 1-nm scale

## *Magnetic & Electric Field Imaging ( $F=qE + qv \times B$ )*

- 0.3 – 3nm (holography, Lorentz)



# *What Electrons do Badly*

## *Spectroscopy in thick samples ( $> 1 \mu\text{m}$ )*

- XRF and IXS (especially for Li and other light elements)

## *Imaging degrades rapidly beyond a few microns*

- XRT at 1-10 nm in a few microns or thicker

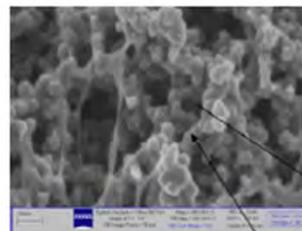
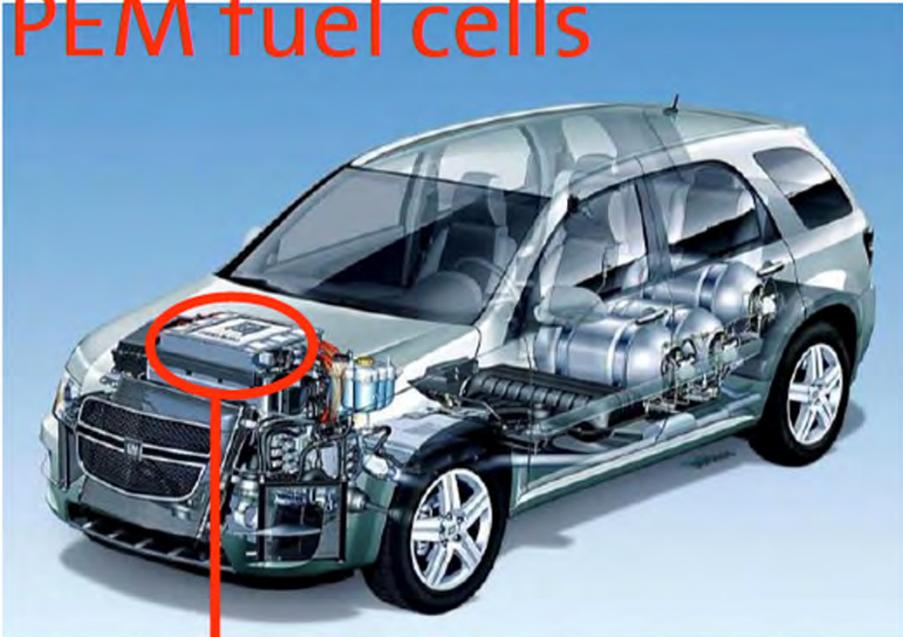
## *Time-resolved studies*

- Electrons are not bosons, and are not neutral (Coulomb Blur)
- Pump-Probe – 1 electron at a time
  - slow, but atomic resolution at fs-ps
- Single-Shot - 0.3 nm at 1  $\mu\text{s}$ 
  - 10 nm at 1 ns
  - Faster than this and x-rays will win.

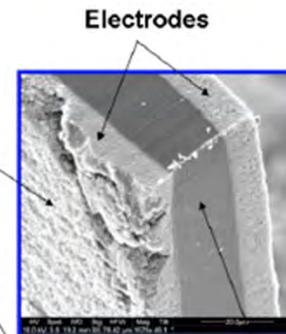
## *Spin Detection*

- Electrons are not neutral (charge dominates)
- EMCD is not as efficient as XMCD – vortex beams may get to 0.1 nm

# PEM fuel cells



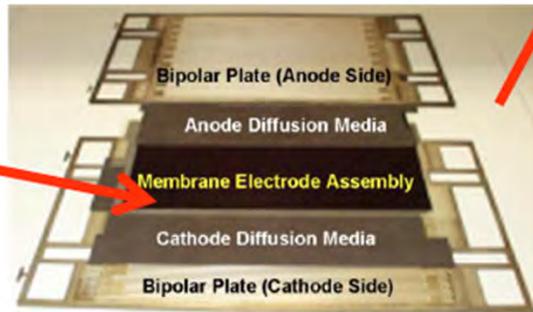
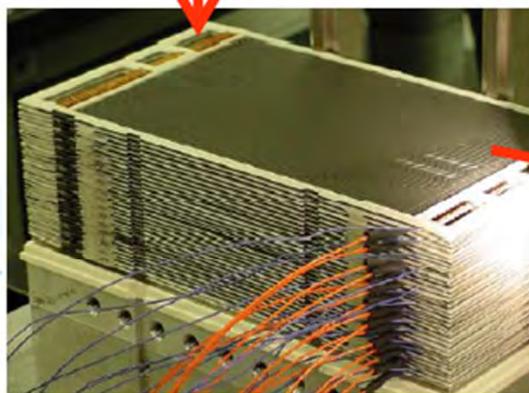
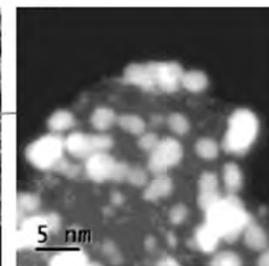
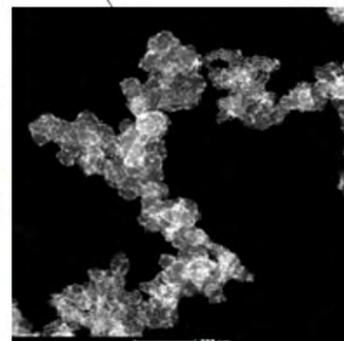
Top down electrode image (FESEM). Porous structure comprised of high structure carbon blacks bound together with a perfluorosulfonic acid polymer (ionomer). The carbon blacks are decorated with 2 nm Pt particles.



Electrodes

Membrane electrode assembly (MEA) cross section (SEM)

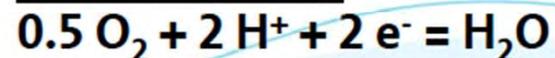
Membrane



**Anode (HOR):** (CATALYST REQUIRED)



**Cathode (ORR):** (CATALYST REQUIRED)

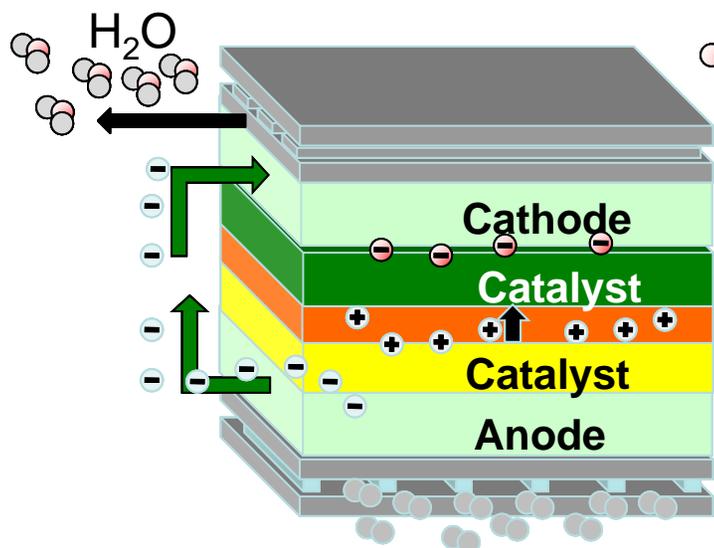


**PEM Fuel Cell Overall reaction:**



THE WORLD'S BEST VEHICLES

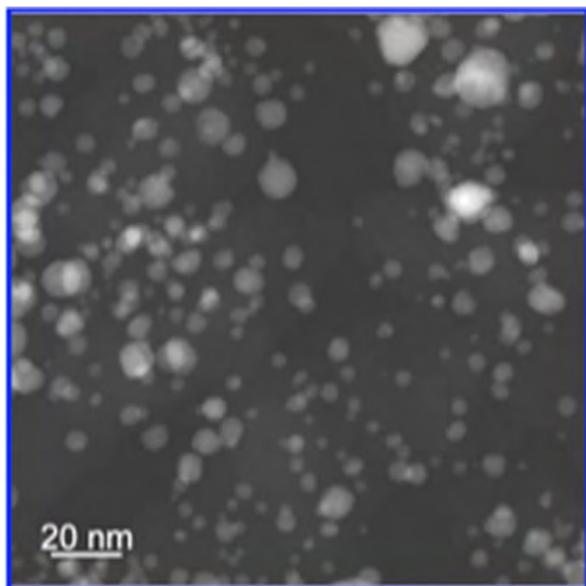
# Pt-Co nanoparticles for electrocatalysis



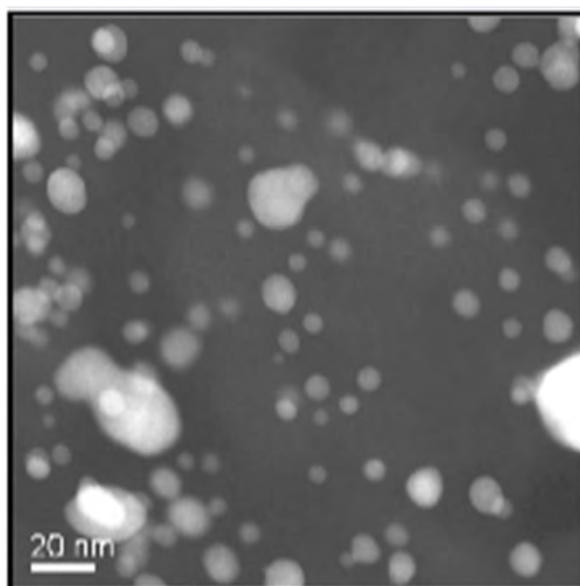
Oxygen Reduction Reaction:  
 $\frac{1}{2} O_2 + 2e^- \rightarrow O^{2-}$   
Proton Exchange Membrane



Starting Sample

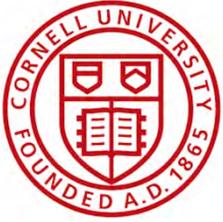


After 30k Voltage Cycles

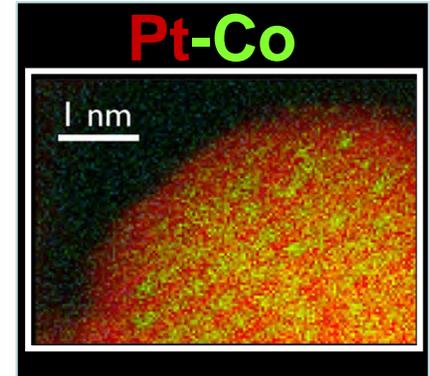
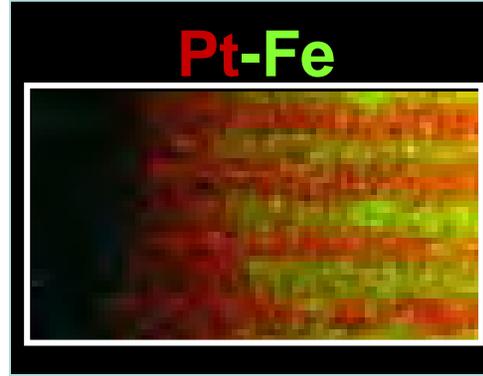
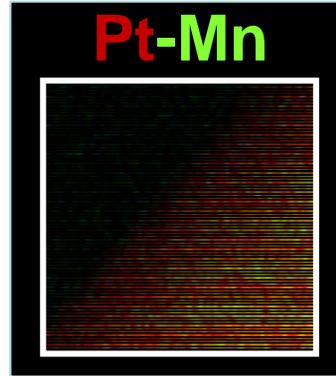
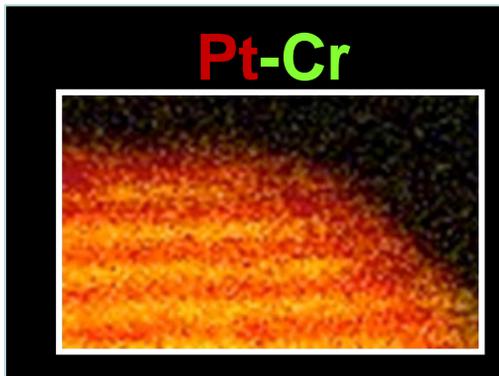
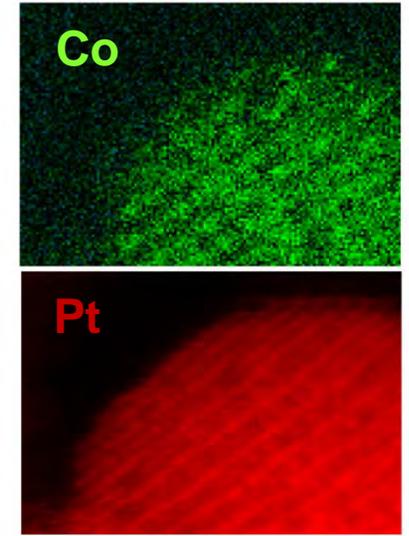
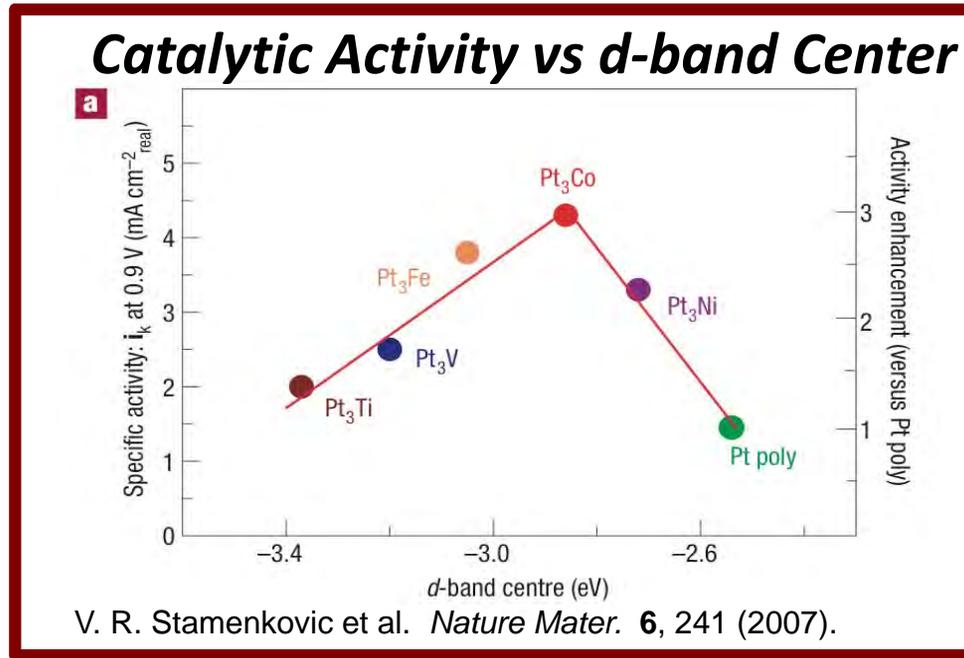


- Loss of catalytic surface area
- 37% drop in specific activity

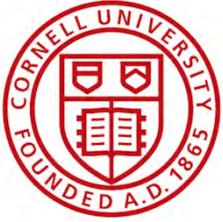
What microscopic mechanism is responsible?



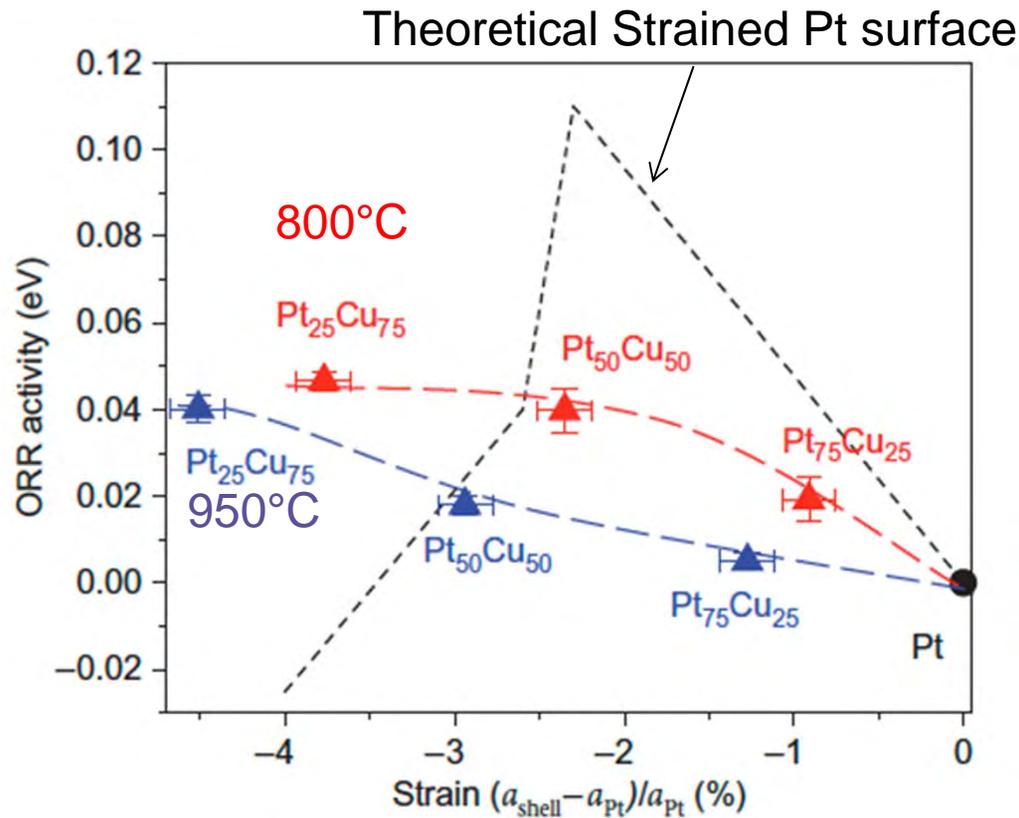
# Band Engineering of Catalysts?



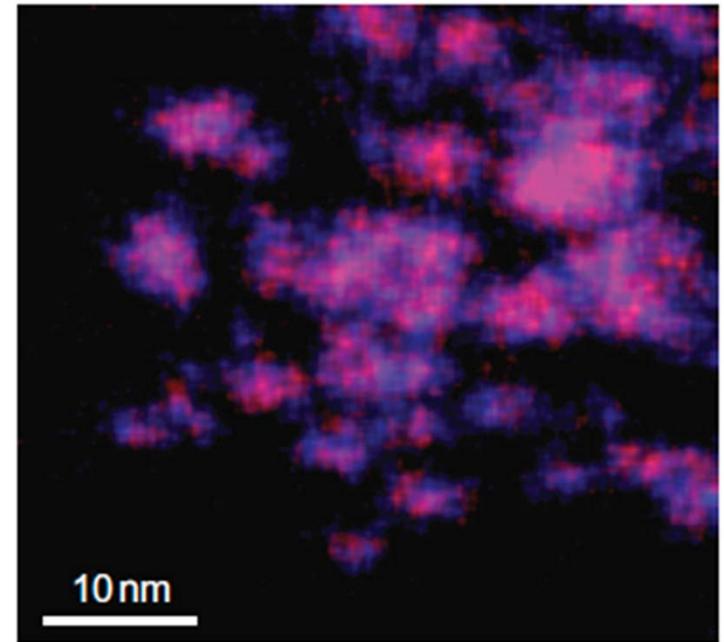
~ 3 monolayers of Pt segregation after a few cycles



# ORR Activity should track applied strain (*d*-band shifts with strain)

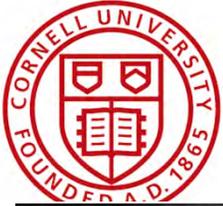


EDX map of Pt/PtCu<sub>3</sub>

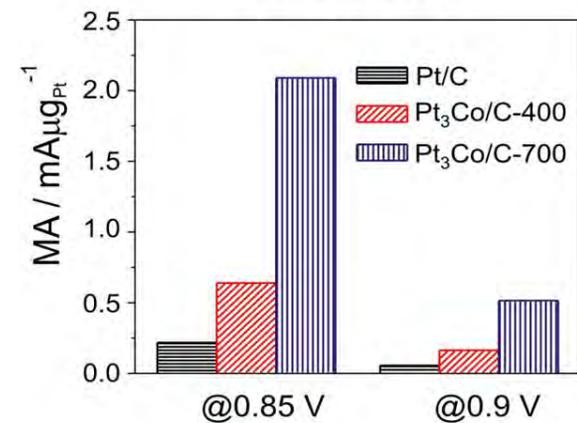
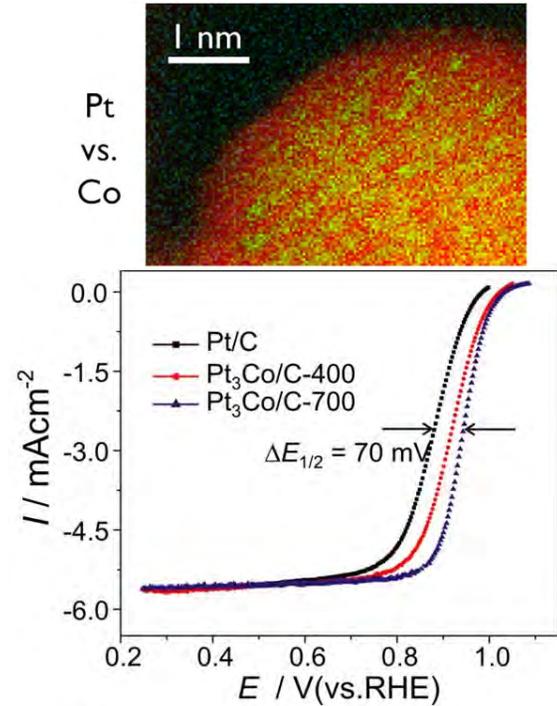
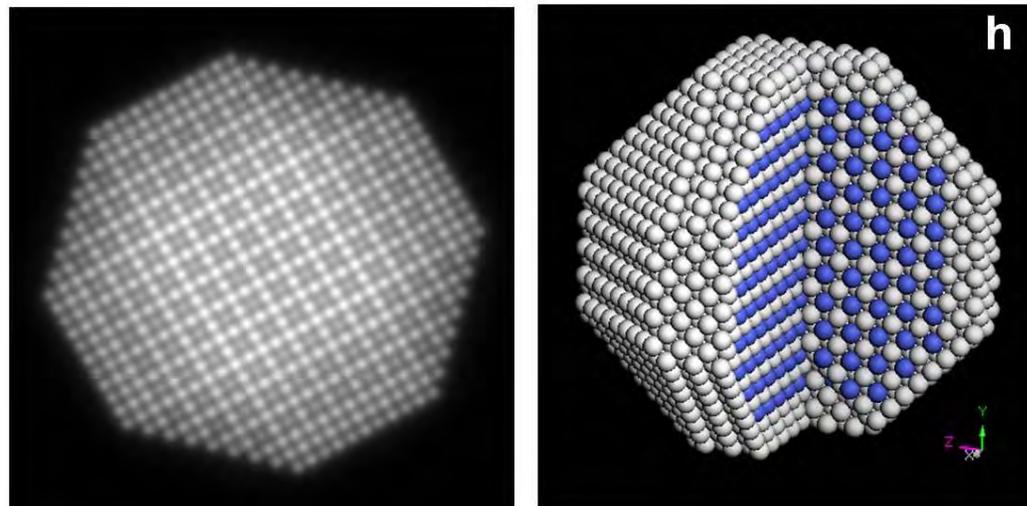
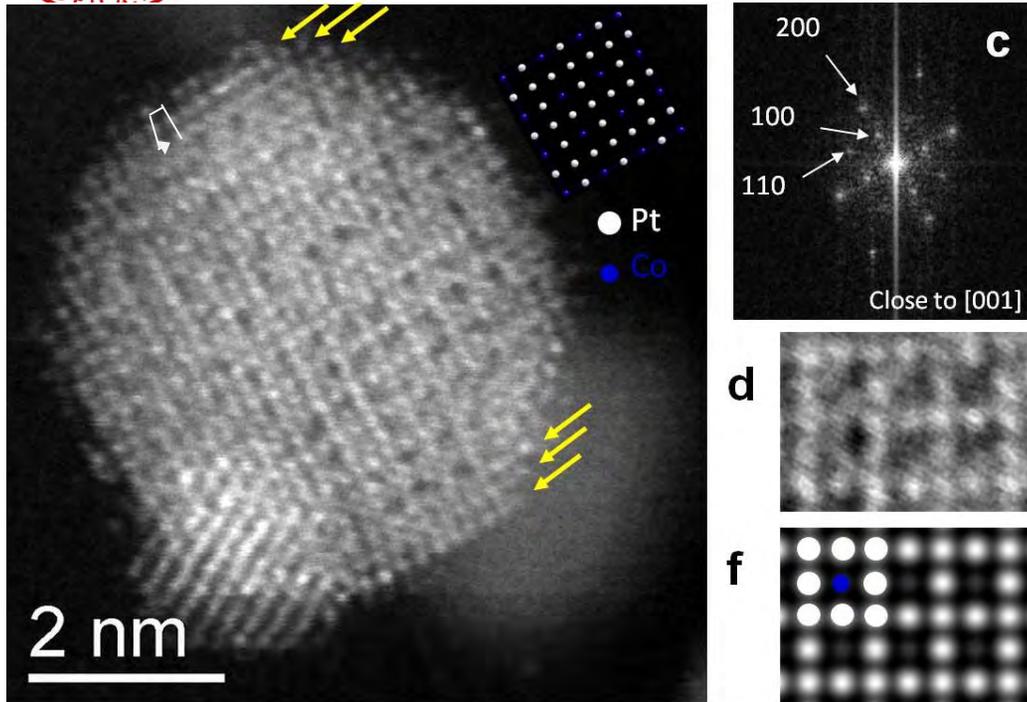


P. Strasser et al. *Nat. Chem.* **2**, 454 (2010).

Why?



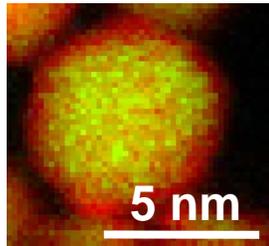
# Pt<sub>3</sub>Co ORR Catalyst Ordered Intermetallic vs. Disordered Alloy



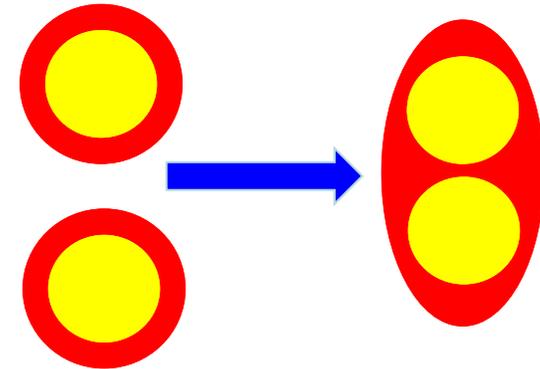
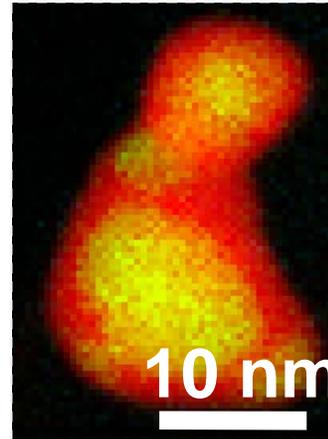
Wang et al, Nature Materials, 12, 81 (2013)  
Also Pt<sub>3</sub>Fe<sub>2</sub> Prabhudev, ACS Nano (2013)

# Determining the Coarsening Mechanism

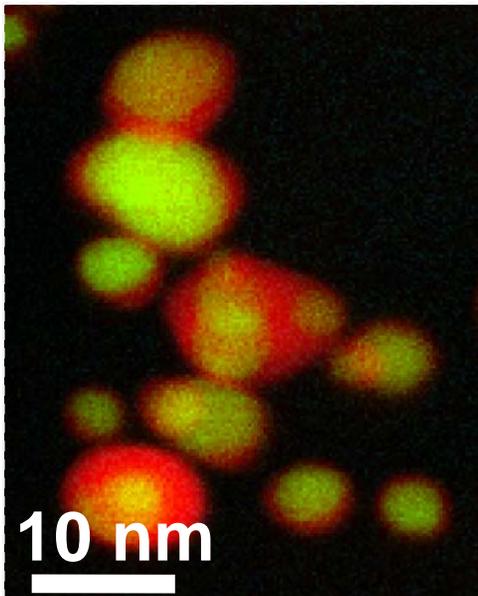
**Starting Sample**



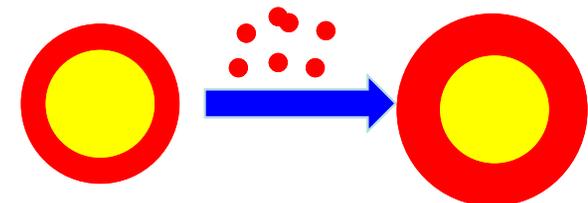
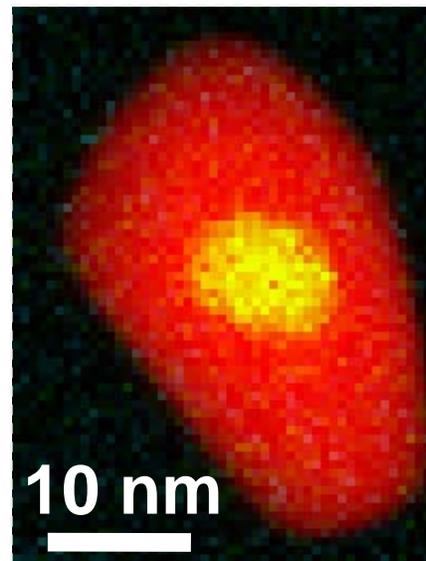
**Coalescence**



**End of Lifetime**



**Ostwald Ripening**  
*thru dissolution (Pt,Co)  
and redeposition (Pt)*

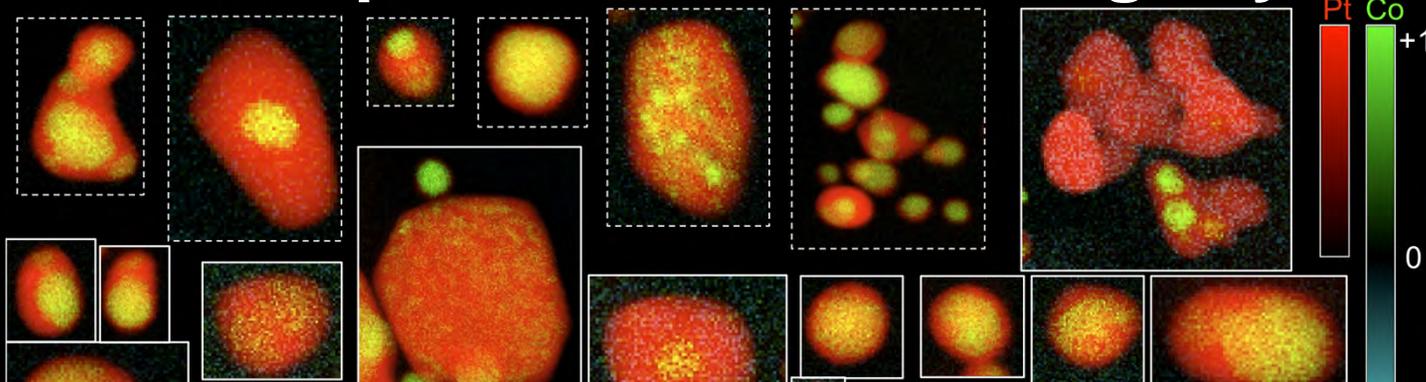
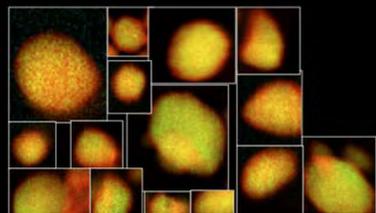


● Pt

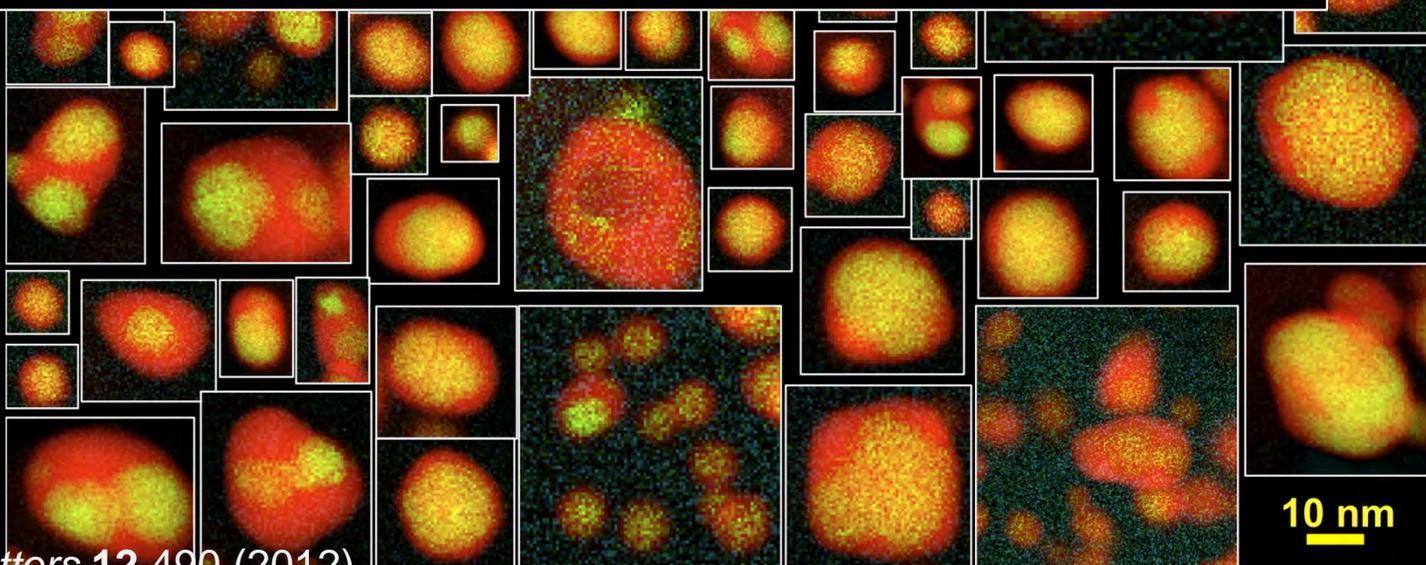
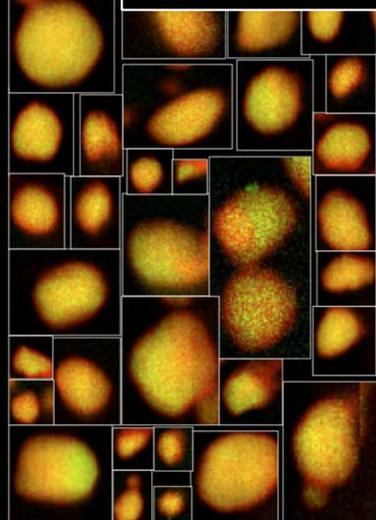
● Pt-Co Alloy Core

Pt-Co Before  
Voltage  
Cycling

Pt-Co particles 30 000 voltage cycles



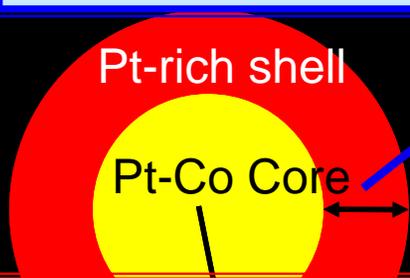
*Sampling hundreds of particles allows statistics on facet dependent segregation and coarsening mechanisms during voltage cycling.*



# Statistical Comparison: Pre- and Post-Cycling

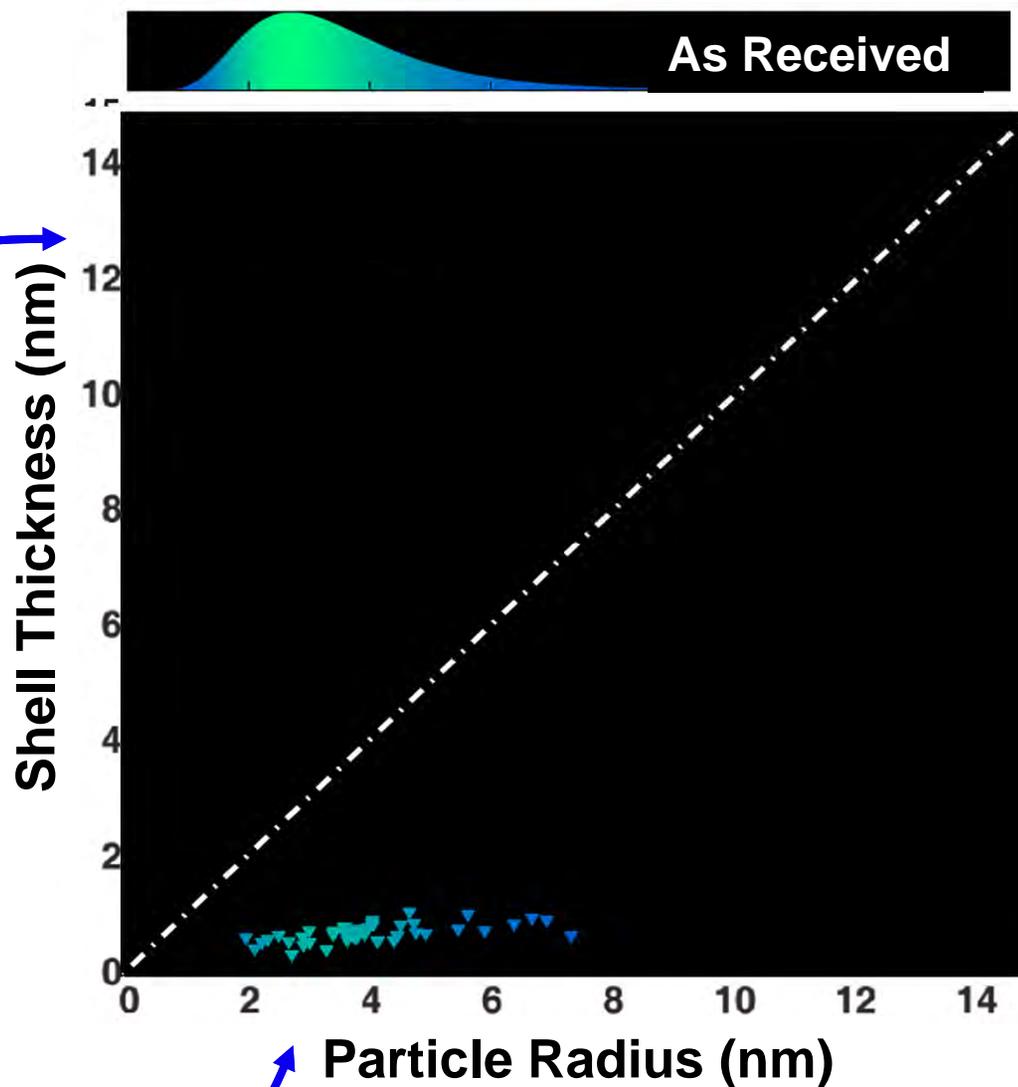
## **As Received:**

~ 2.5 monolayer  
( $5.7 \pm 0.3 \text{ \AA}$ ) Pt shell

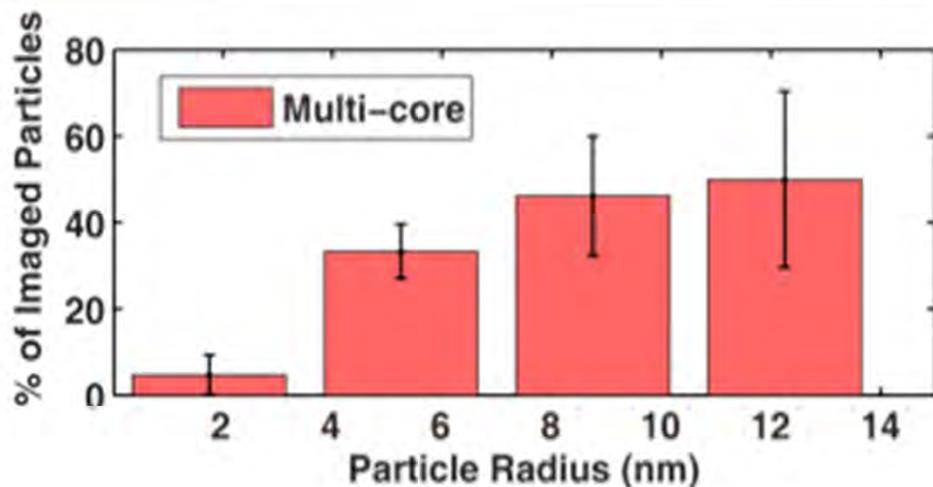


## **Voltage Cycled:**

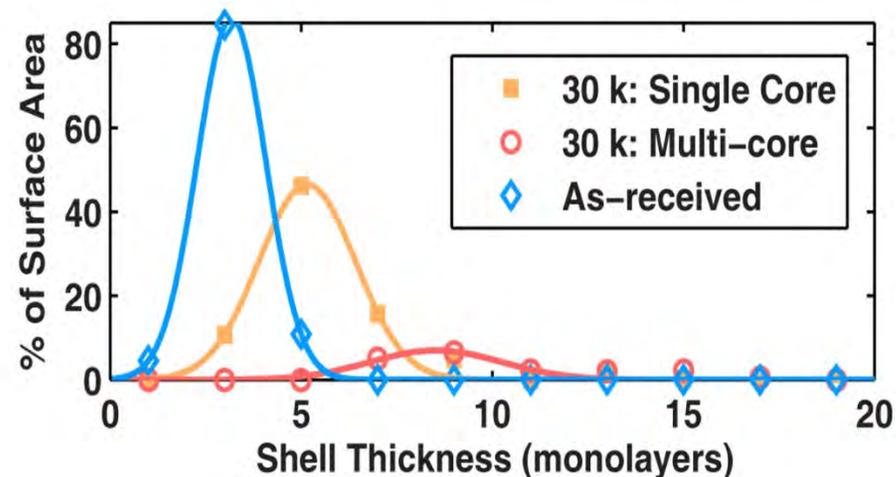
- Strong dependence of shell thickness on particle size.
- Large Particles approaching total depletion of Co.



# Statistics on Particle Coarsening Mechanisms



About half the large particles are multi-core



More Pt re-deposition onto multi-core particles

***Ostwald ripening***

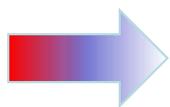
***only:***

Adds 2 monolayers of

Pt to shell

***Coalesced:***

- Adds 5 monolayers of Pt to shell



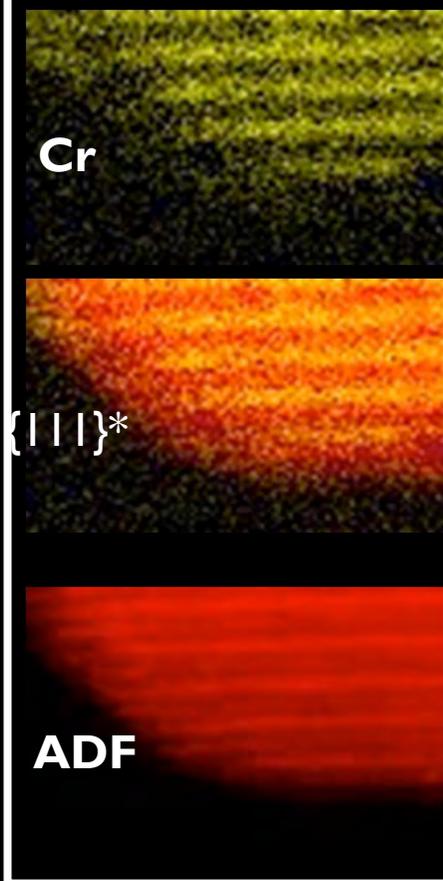
Ostwald Ripening is faster on highly-curved coalesced particles



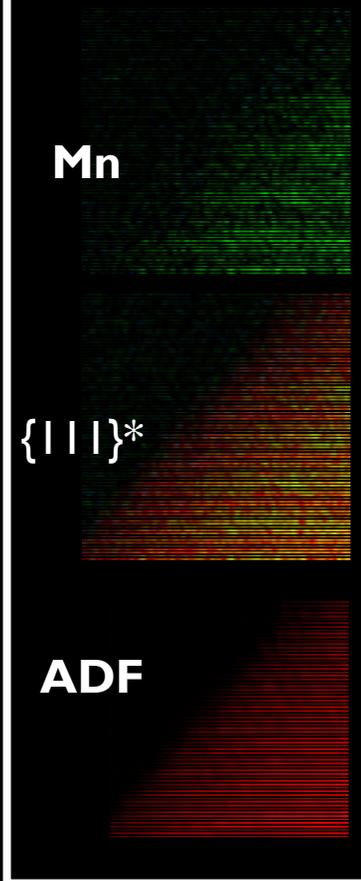
# Atomic Resolution imaging of segregation



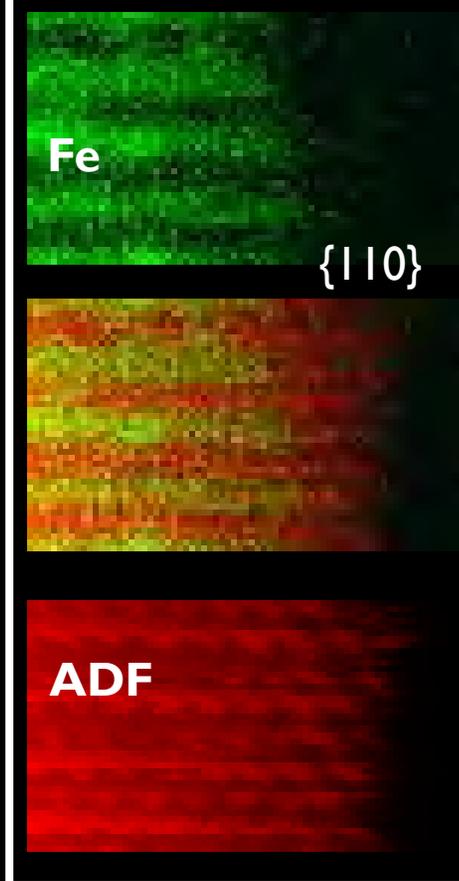
## Pt-Cr



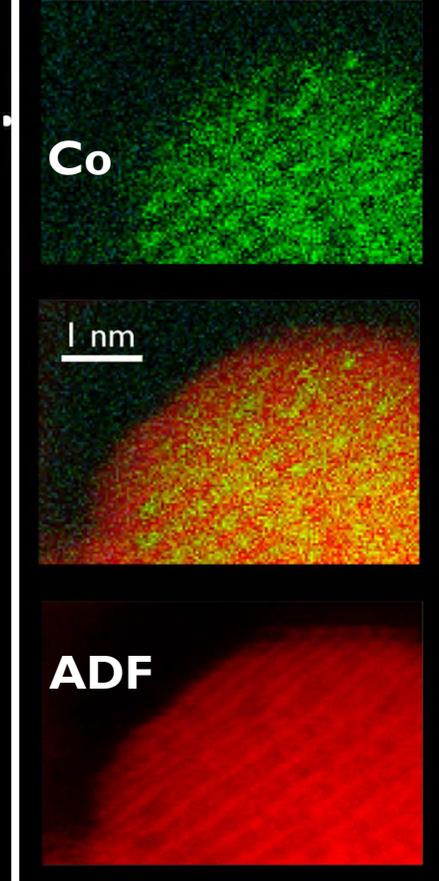
## Pt- Mn



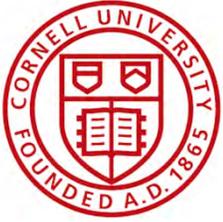
## Pt-Fe



## Pt-Co

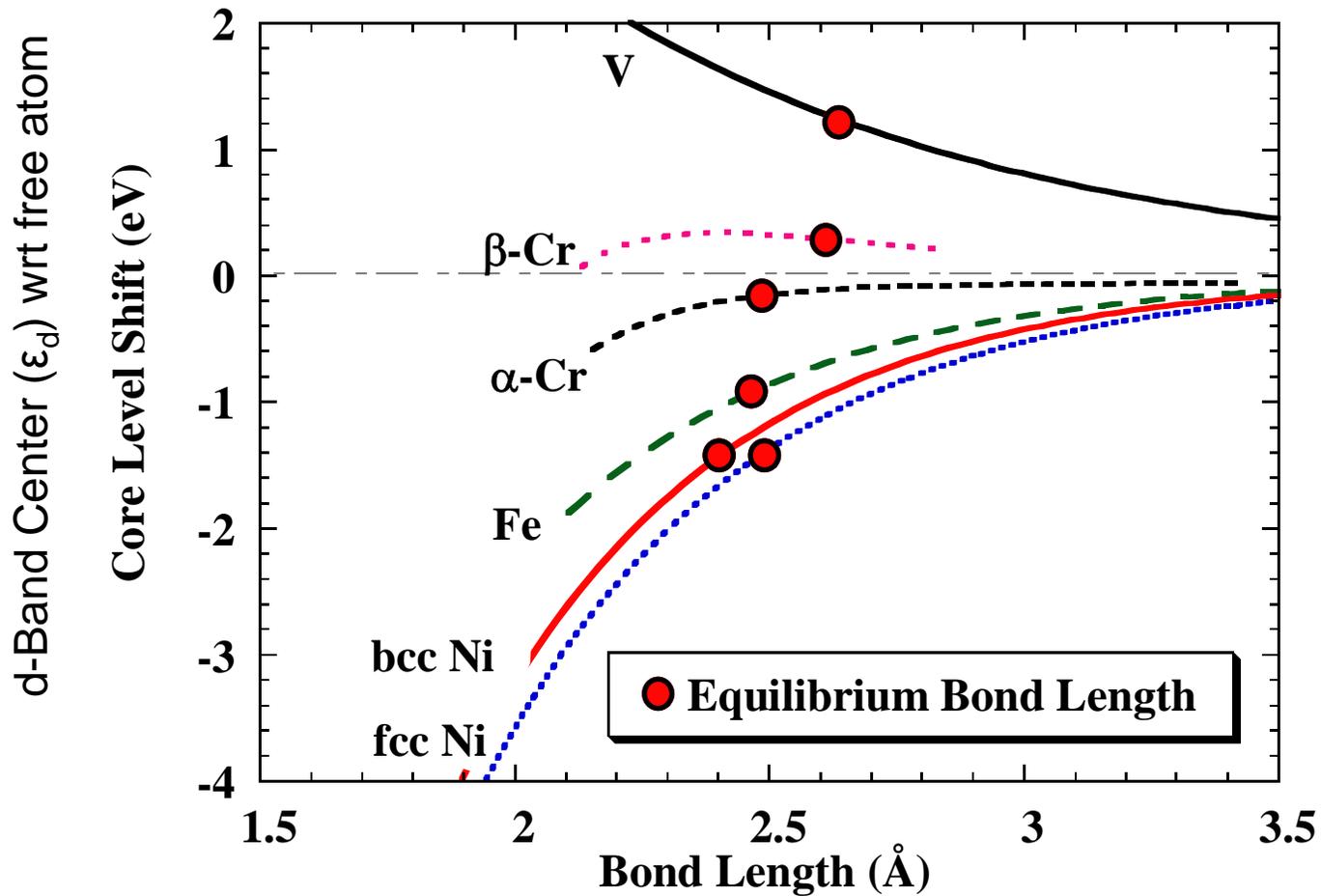


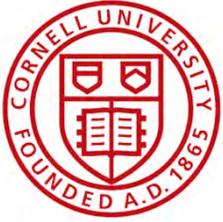
~ 3 monolayers of Pt segregation after a few cycles



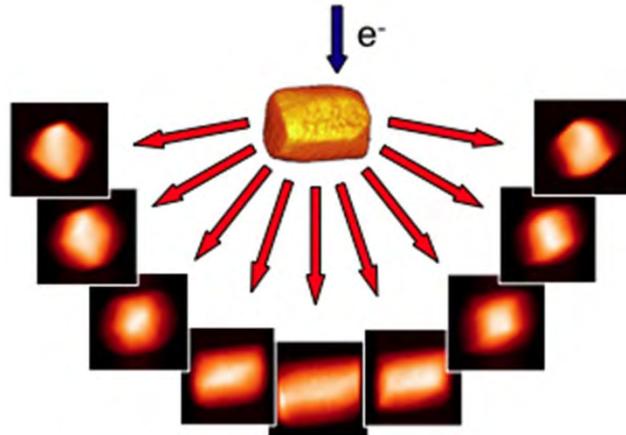
# Core Level Shifts in Metals (CLS tracks d-Band center)

We expect  $\delta\epsilon_d \propto \frac{(N_d - 5)}{10} \sqrt{z} \exp(-\beta r)$

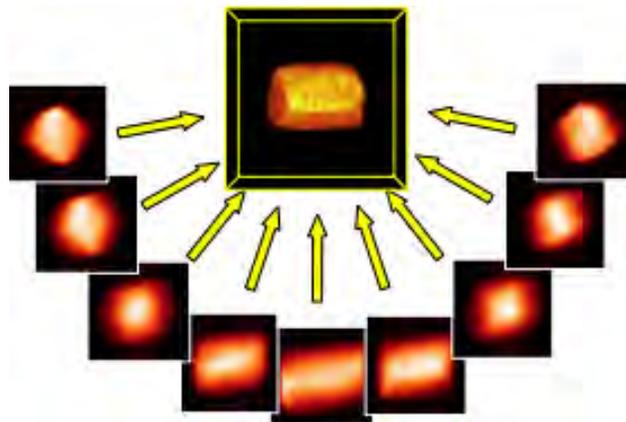




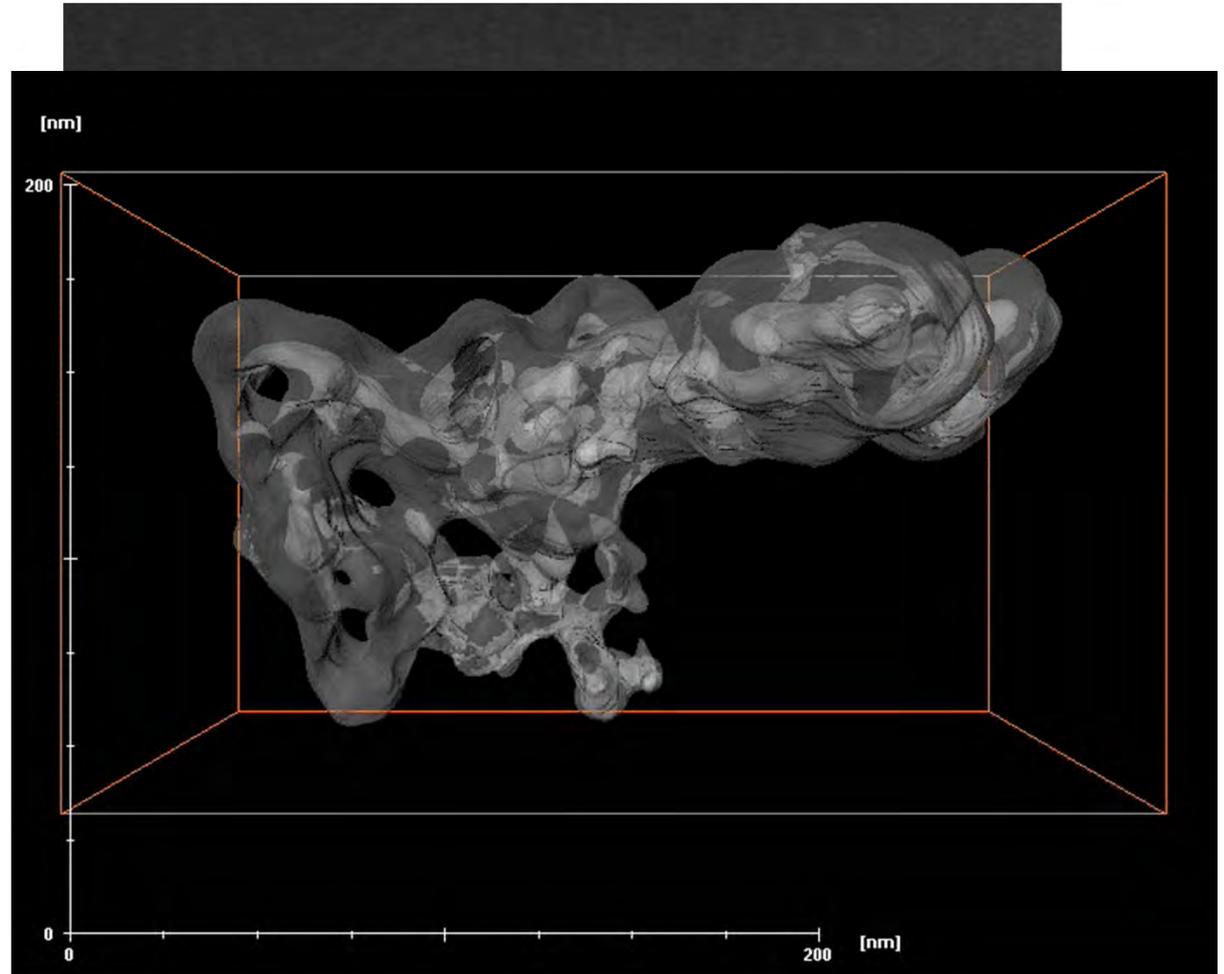
# 3D Reconstruction by Electron Tomography



Tilt from -70 to +70 degree



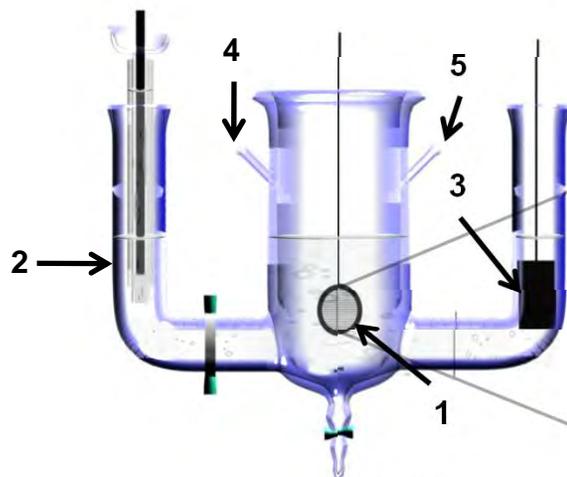
Reconstruction by back projection



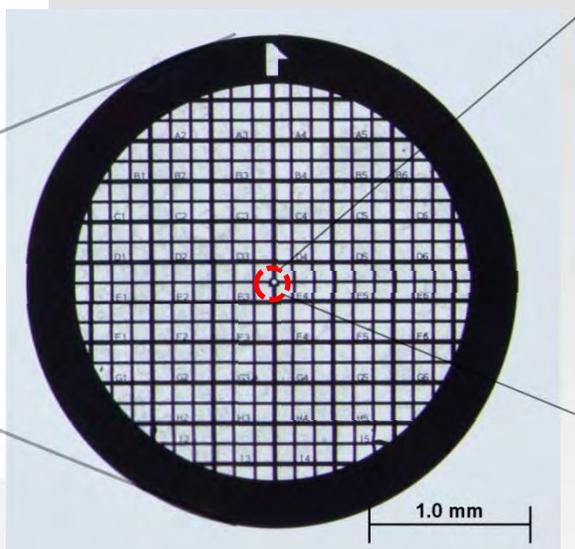
<http://youtu.be/DVsJqhd0oJU>

Z. Y. Liu et al, *Journal of The Electrochemical Society*, **155** B979 (2008).

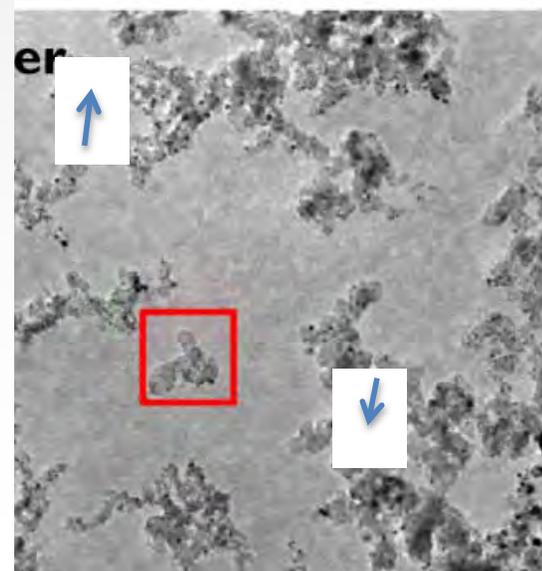
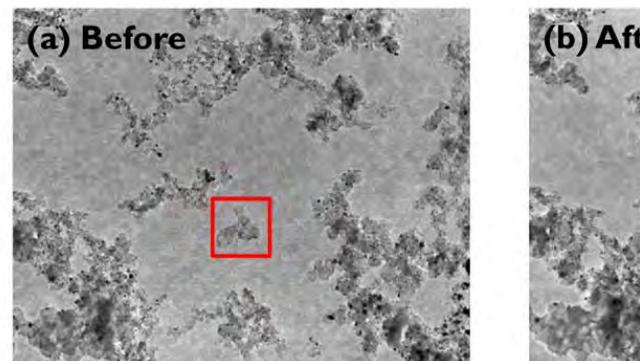
# E-Chem on a Grid: Observe Coarsening during Voltage Cycling of $Pt_3Co$



1. Gold index grid (working electrode)
2. Reversible Hydrogen Electrode (reference electrode)
3. Platinum foil (counter electrode)
4. Argon gas inlet
5. Argon gas outlet



$Pt_3Co$  on Vulcan

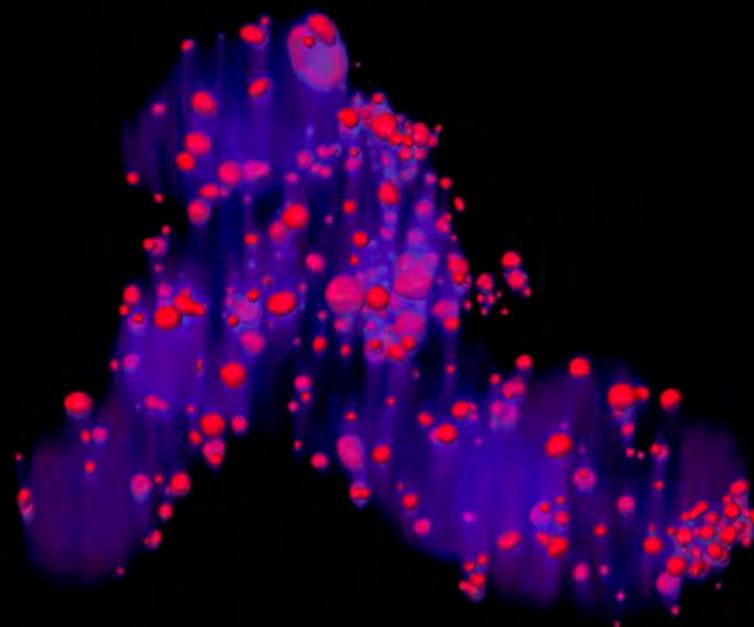
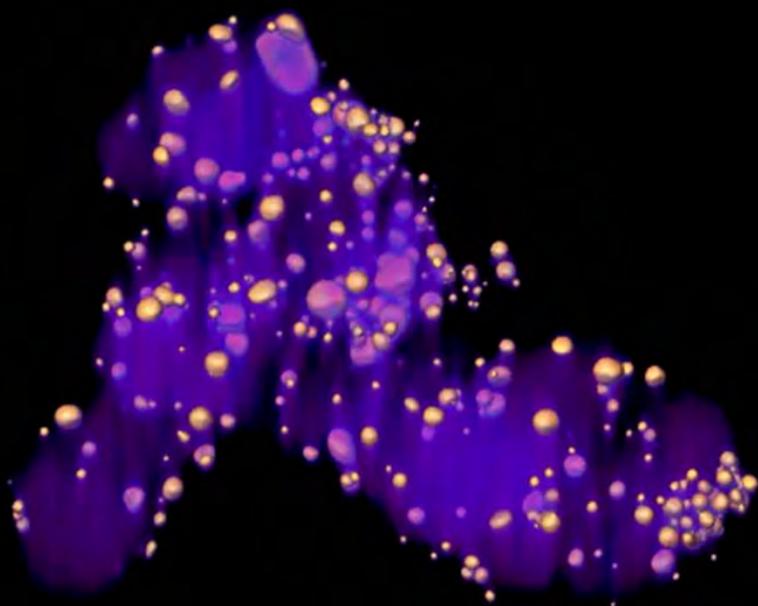


**Enables 3-D imaging for the first time of the same region before and after**

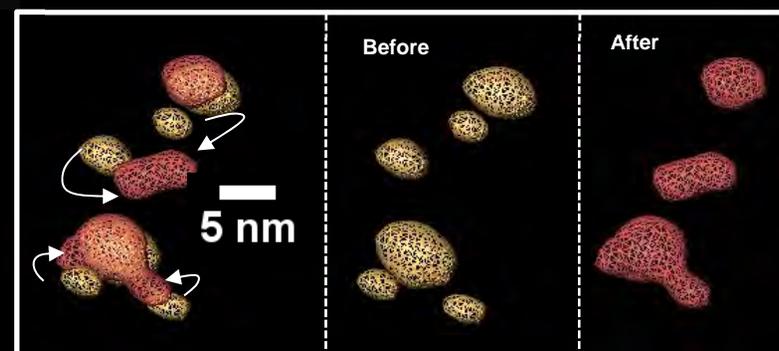
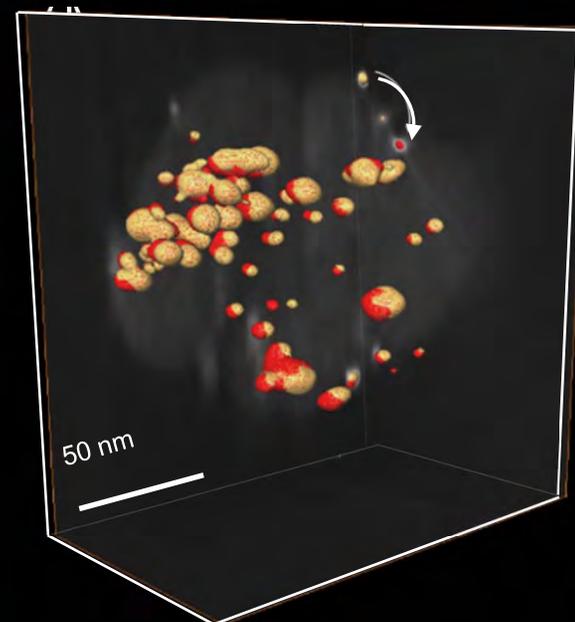
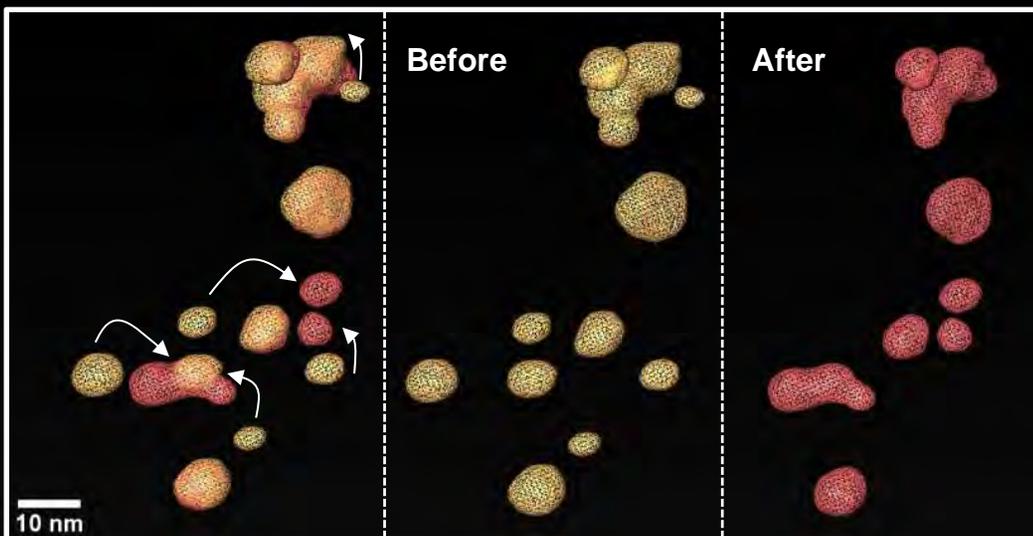
**Before**

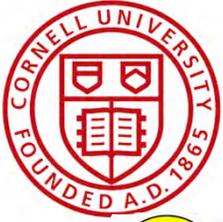
**vs.**

**After**



# Tracking Particle Coalescence in 3D

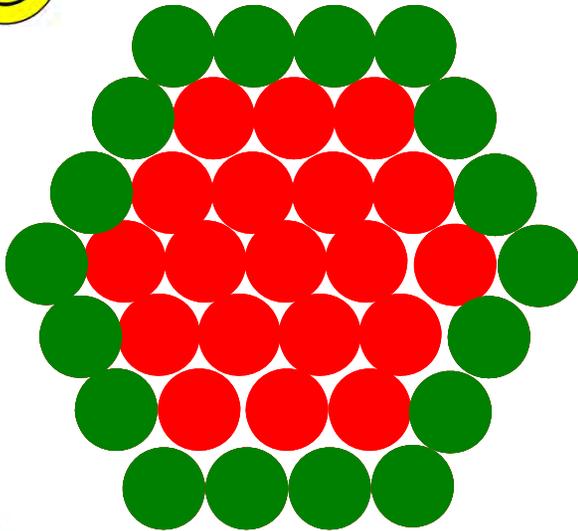




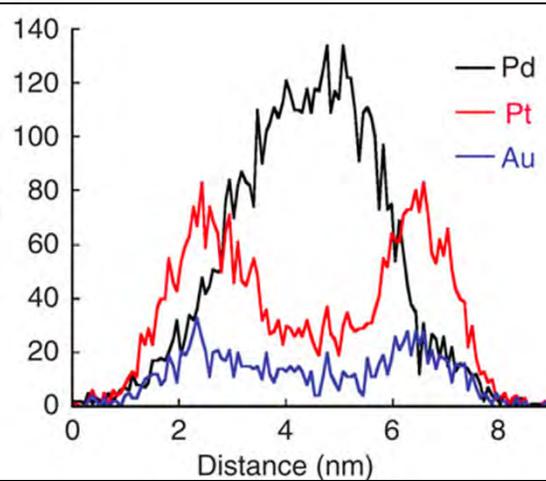
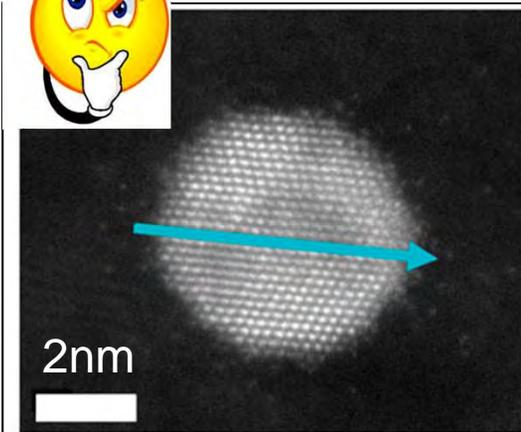
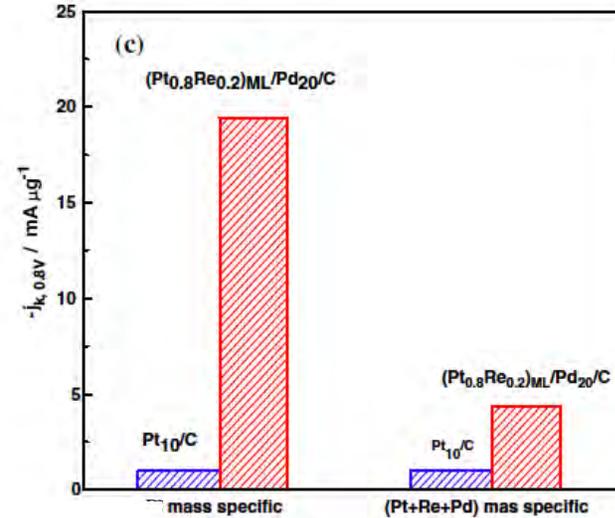
# Pt Monolayer Reduces Cost of Fuel Cell



**Pd@Pt**

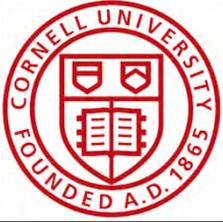


Pt Mass Activity Increases x10



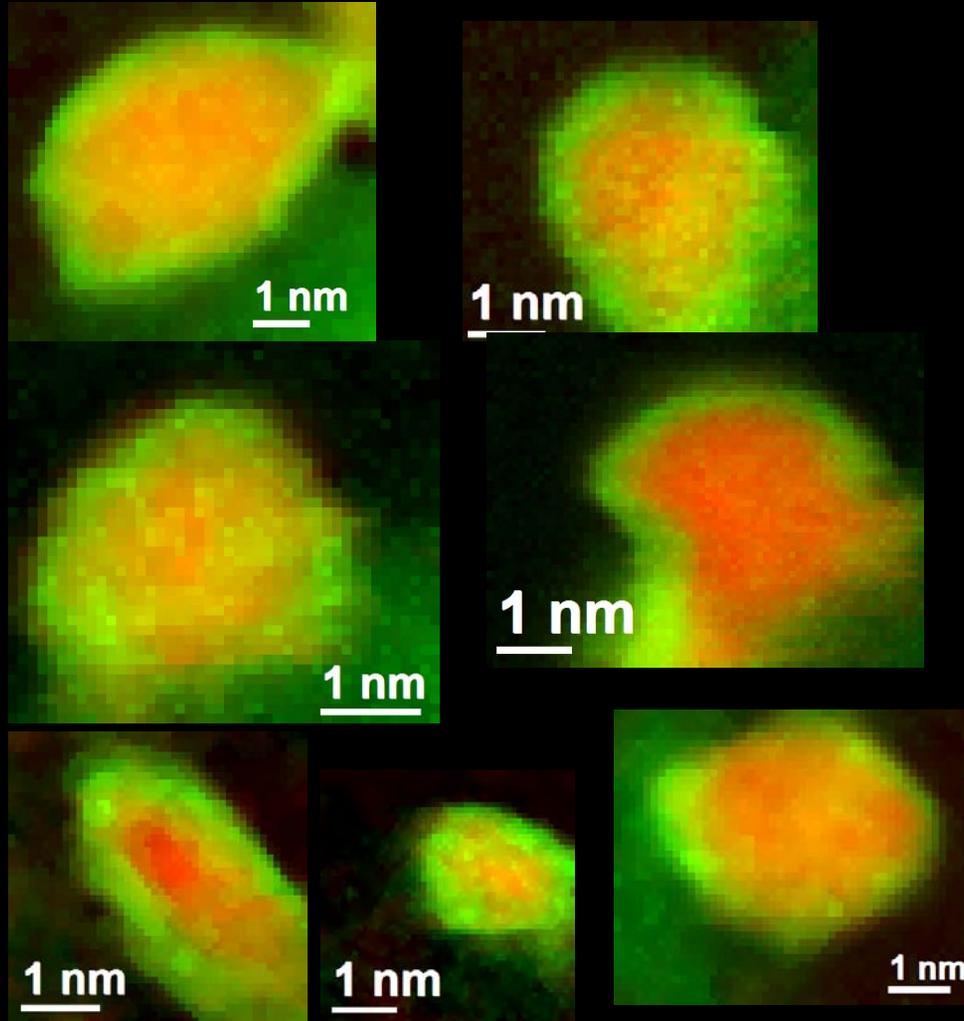
- Statistics -> Uniformity
- Durability -> Practical

RR. Adzic, et al @ BNL  
 Top Catal., 2007, 46:249–262  
 JPCB, 2005, 109, 22701-22704  
 Nature Comm., 2012, 3,1115

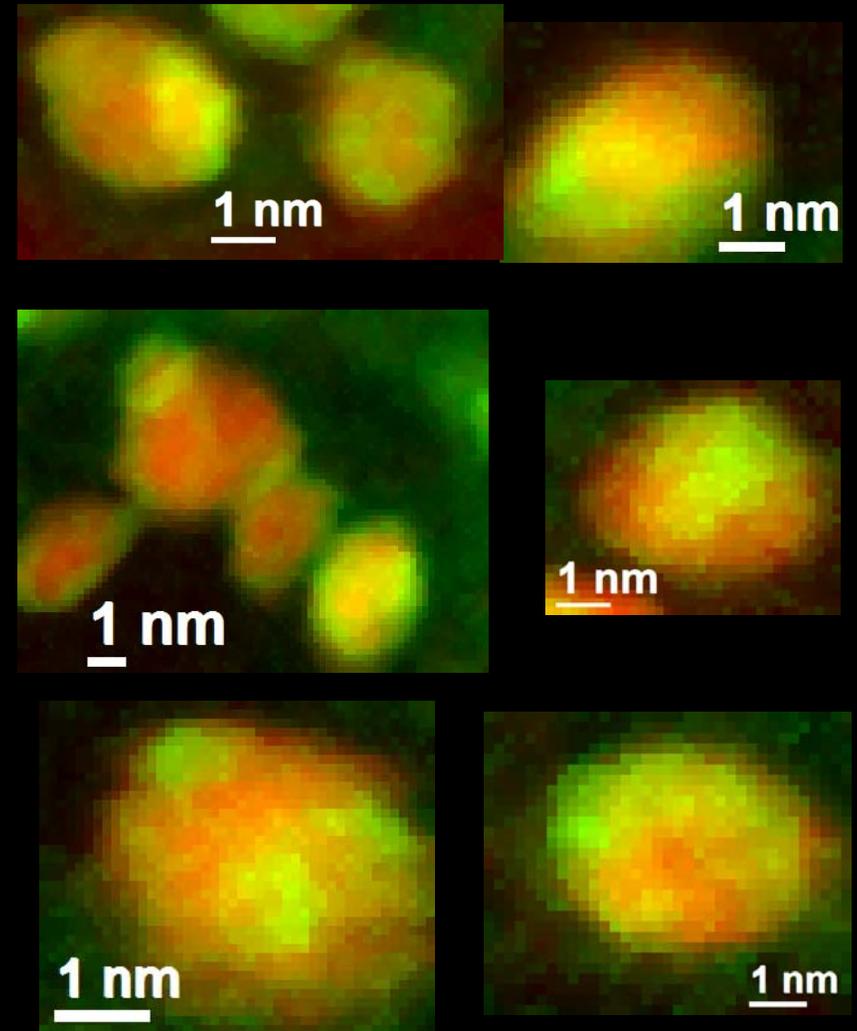


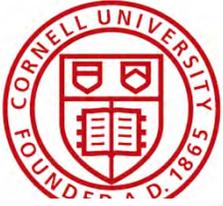
# As Prepared Pd@Pt: Not Uniform

## Core-Shell

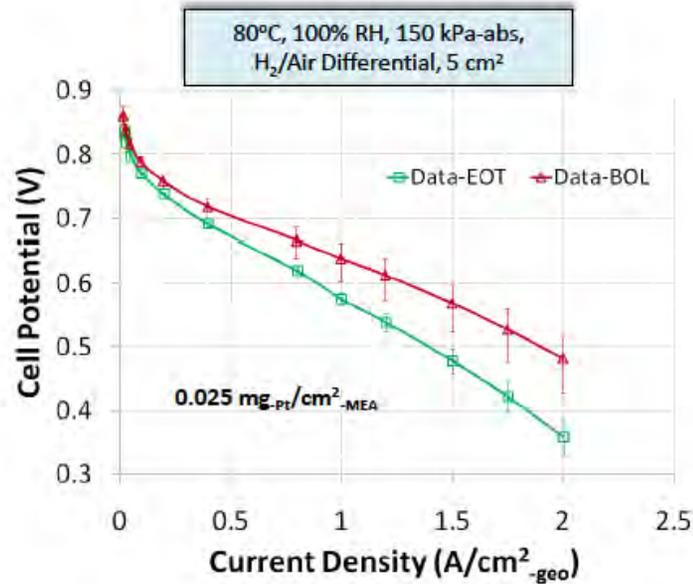


## Not Core-Shell





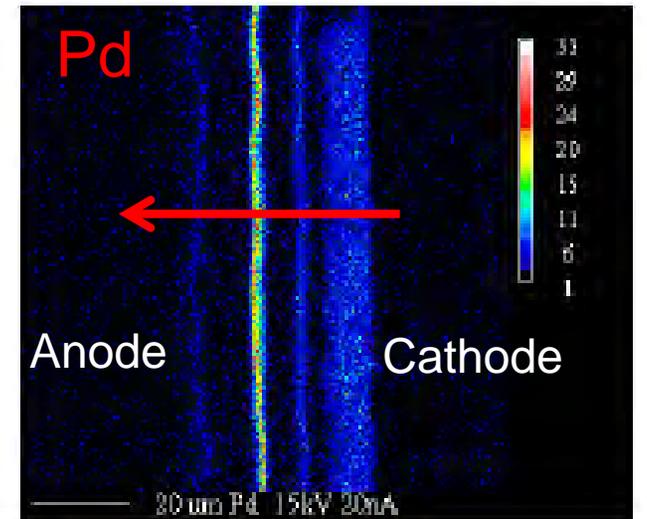
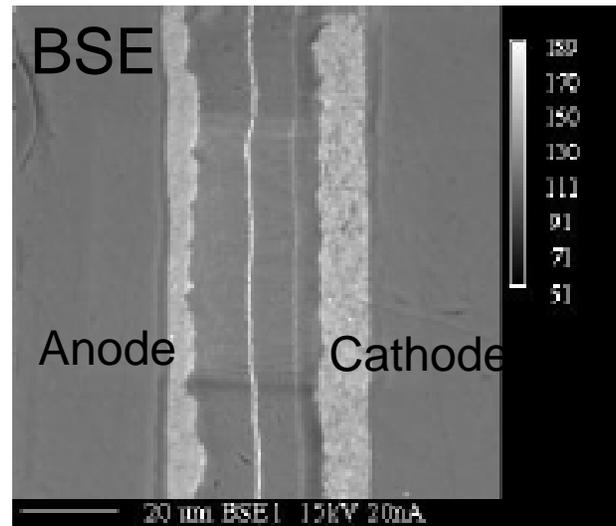
# Fuel Cell Testing



Samples	BOL	EOT
Pt Surface Area (m <sup>2</sup> <sub>-Pt</sub> /g <sub>-Pt</sub> )	88	45
Mass Activity (A/mg <sub>-Pt-eq</sub> )	0.298	0.217

**Pd migration to anode!!!**

EPMA mapping of fuel cell slice (EOT)





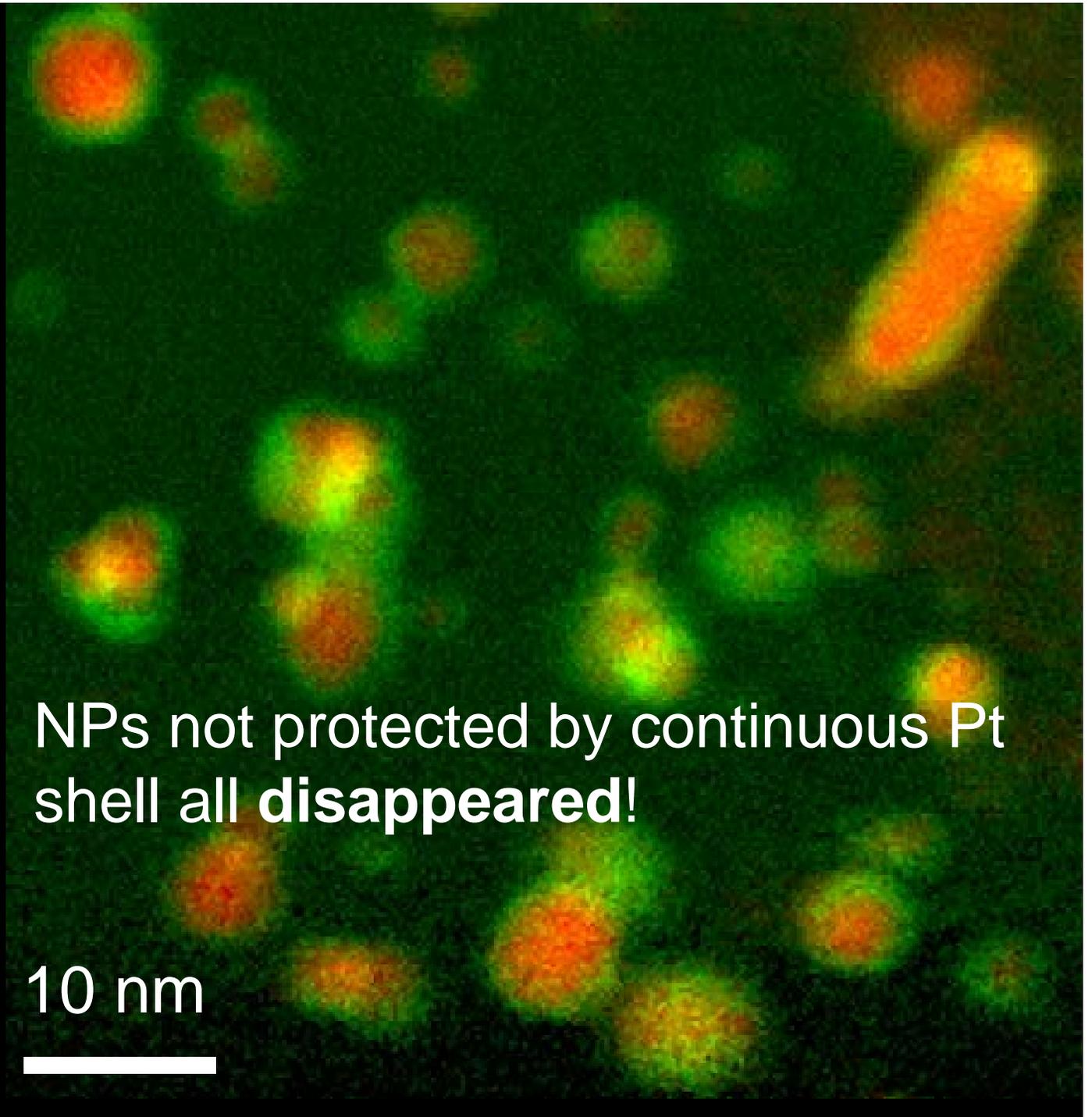
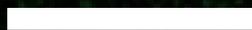
# Cathode Pd@Pt After Aging

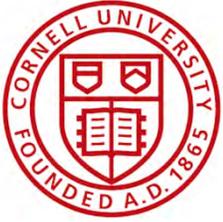
- ✓ New Spectrometer (Enfinium)
- ✓ Dual EELS
- ✓ Larger energy range
- ✓ Faster readout speed

Improved  
~ 300 by 300

NPs not protected by continuous Pt shell all **disappeared!**

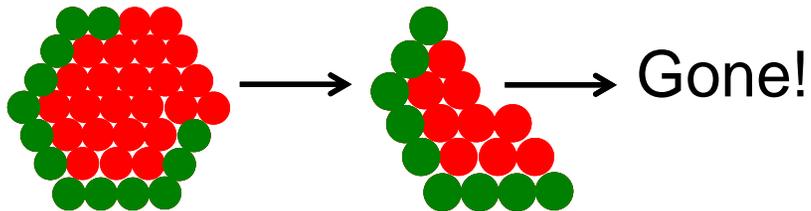
10 nm





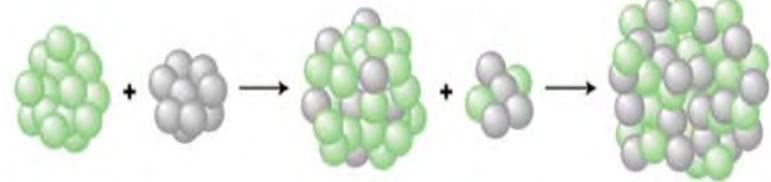
# Catalyst Degradation Mechanism

## 1. Dissolution

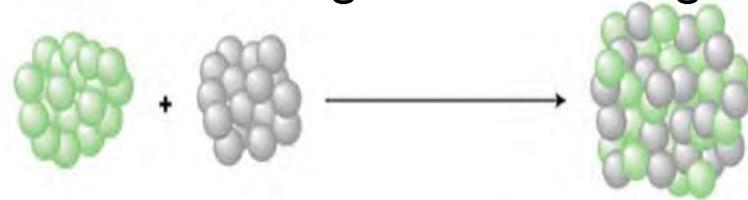


## 2. Coarsening

2a. Ostwald ripening: Dissolution + Redeposition



2b. Coalescence: Migration and Merge



Pt-Pd

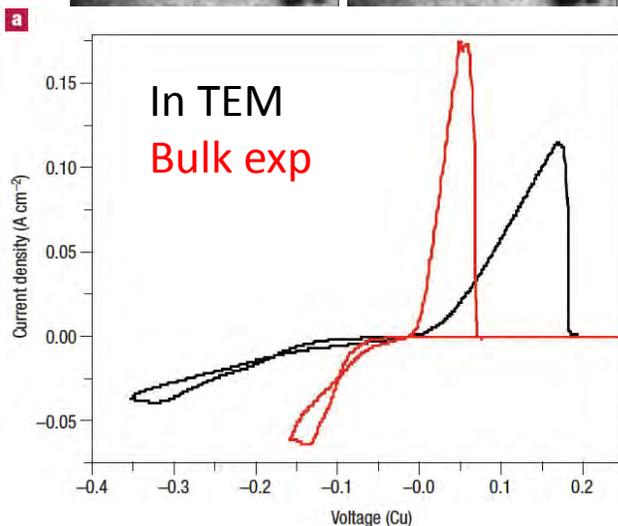
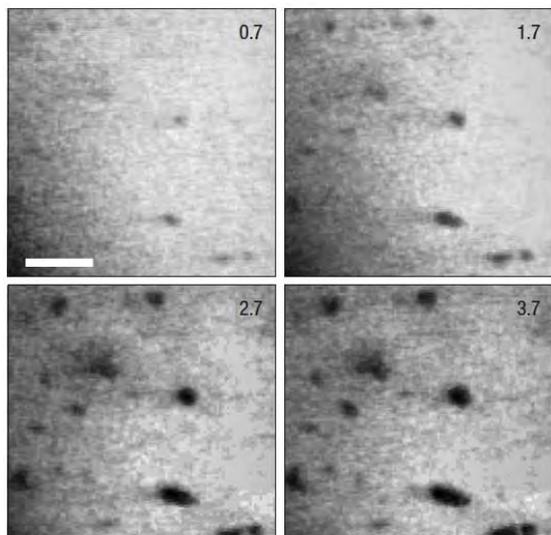
Pt-Co?

Is there any *in-situ* method to track the

same locations of nanoparticles?

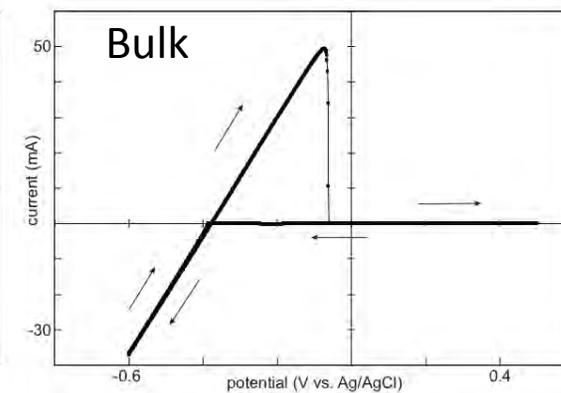
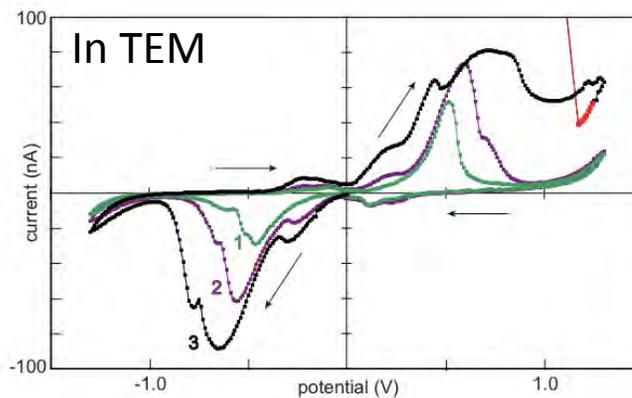
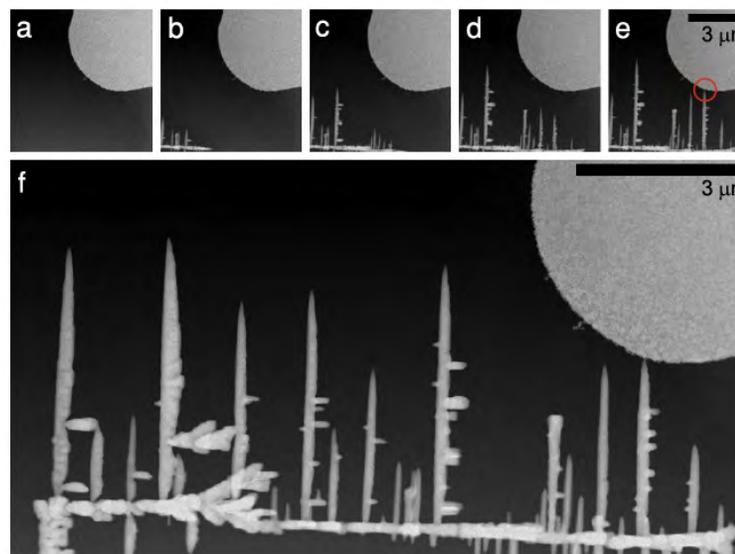
# Can we do electrochemistry in the TEM?

## Electrodeposition of Copper



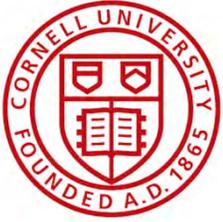
M.J. Willamson, Nat Mat. **2** (8), 532 (2003)

## Growth of Pb dendrites

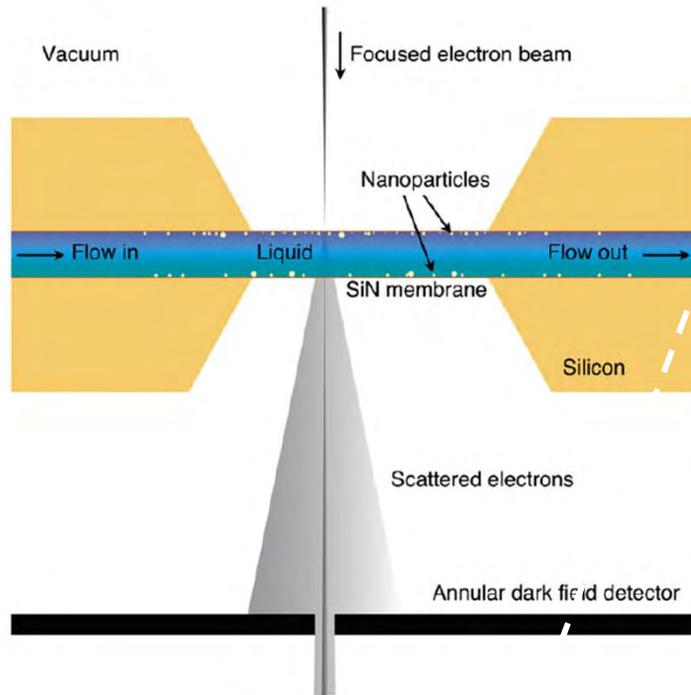


E.R. White, ACS Nano. **6** (7), 6308 (2012)

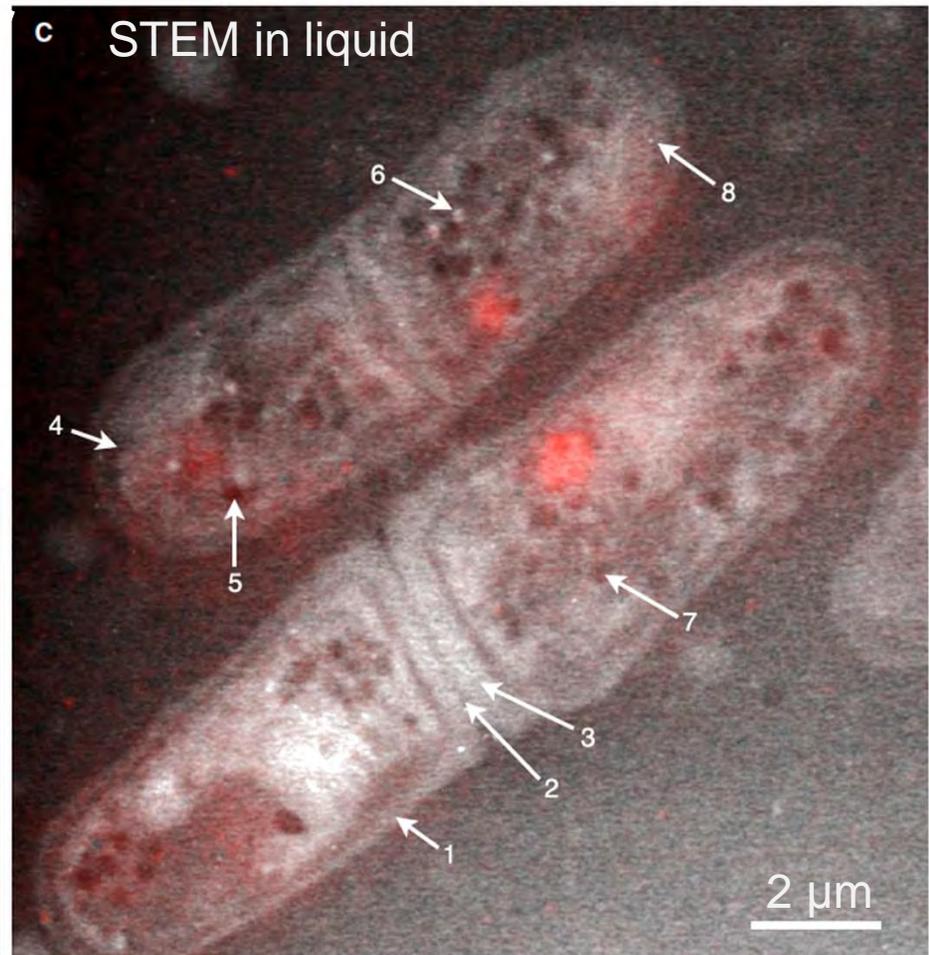
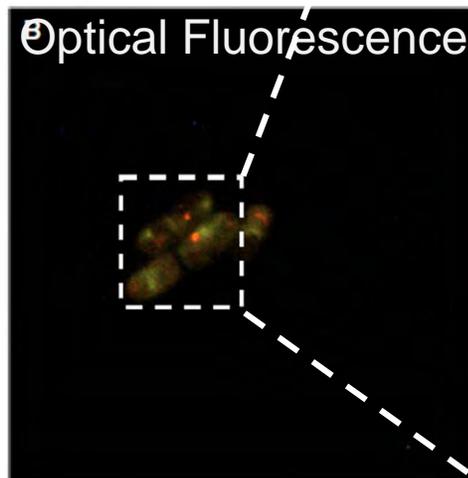
**Electrochemistry in a TEM is not trivial!**

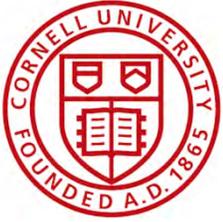


# TEM in Liquids



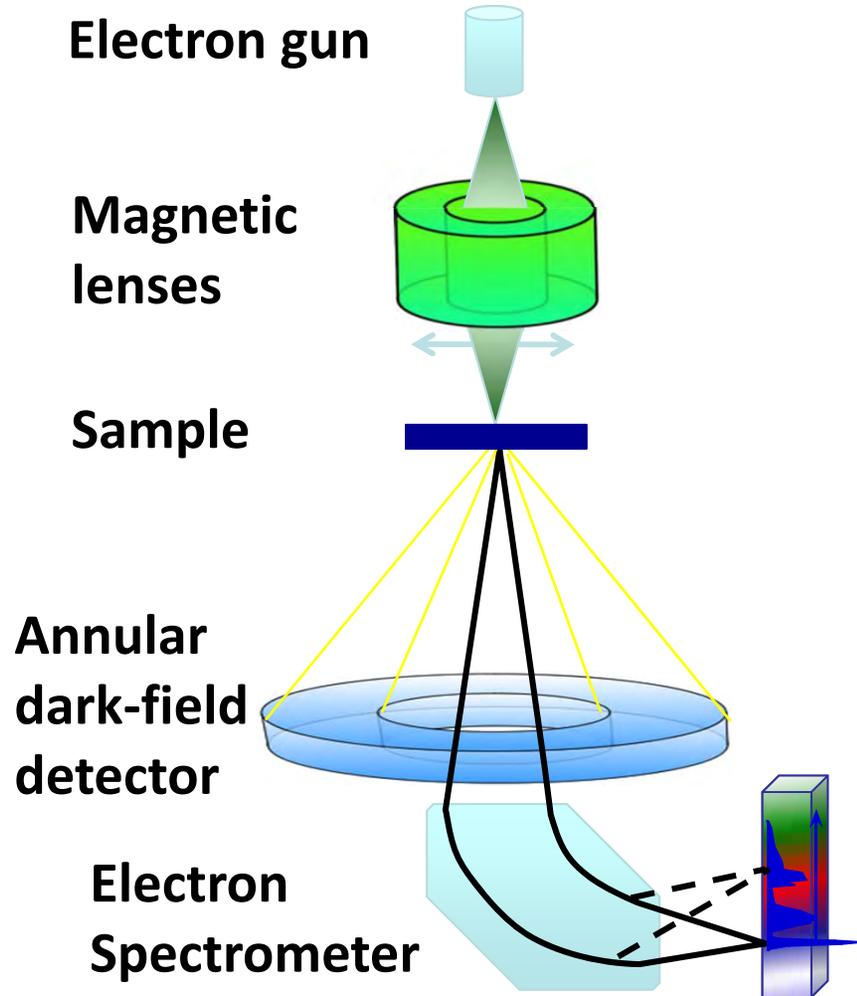
Peckys et al, "Fully Hydrated Yeast Cells Imaged with Electron Microscopy". *Biophysical Journal* **100**, 2522 (2011).



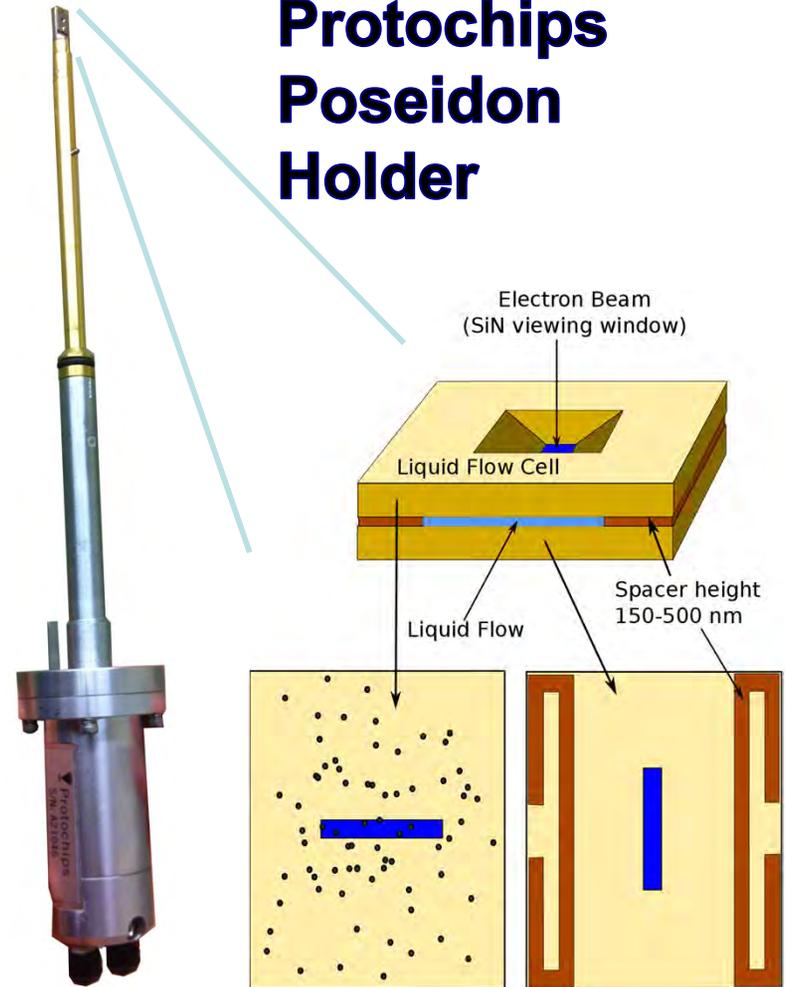


# Liquid cell *in situ* TEM

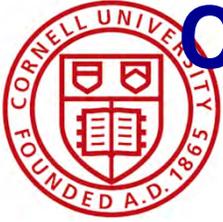
## Scanning Transmission Electron Microscope (STEM)



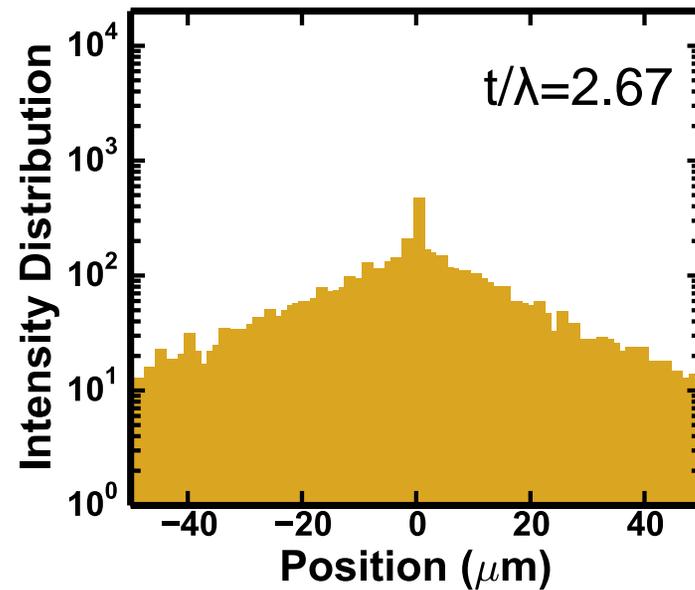
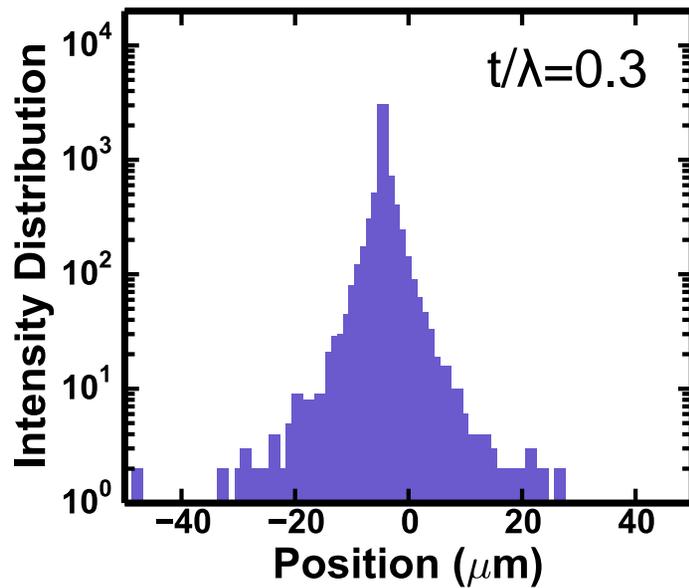
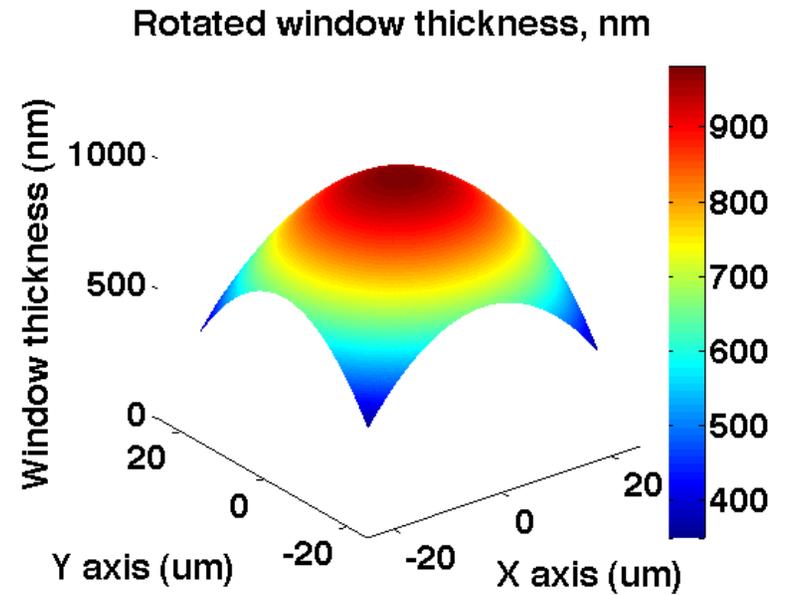
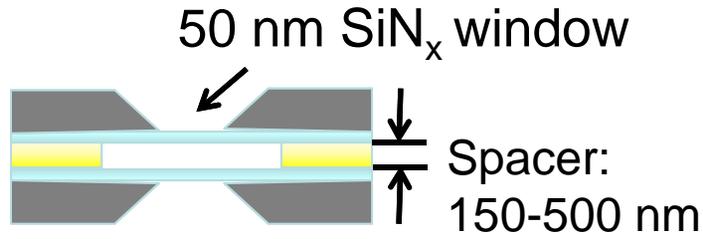
## Protochips Poseidon Holder

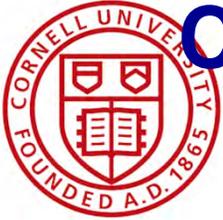


**STEM-Electron energy-loss spectroscopy**



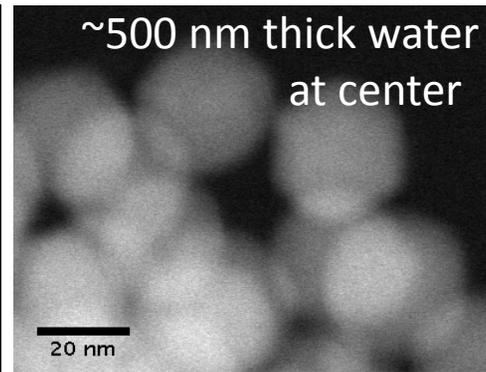
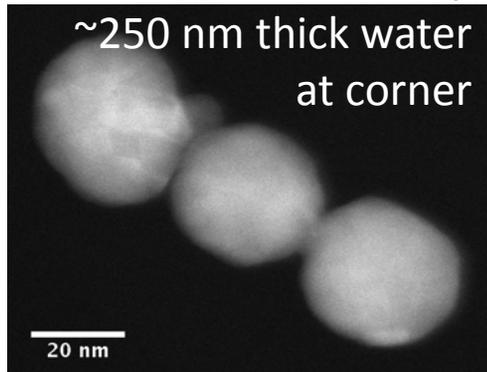
# Challenge #1: Thickness of the liquid layer





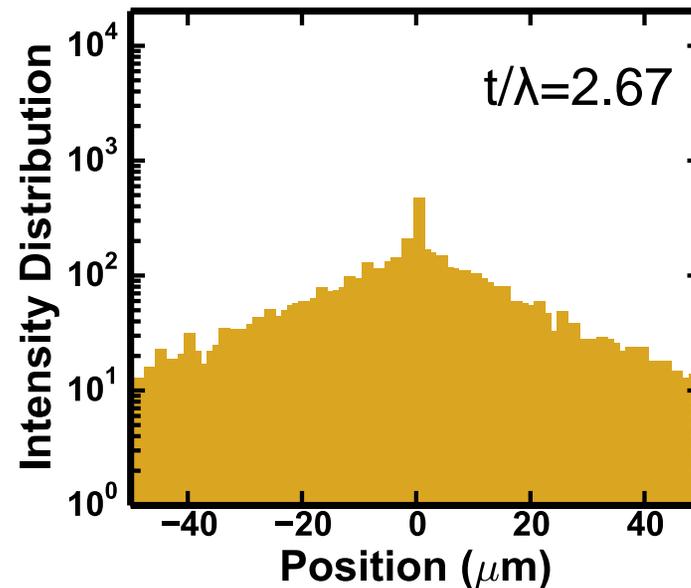
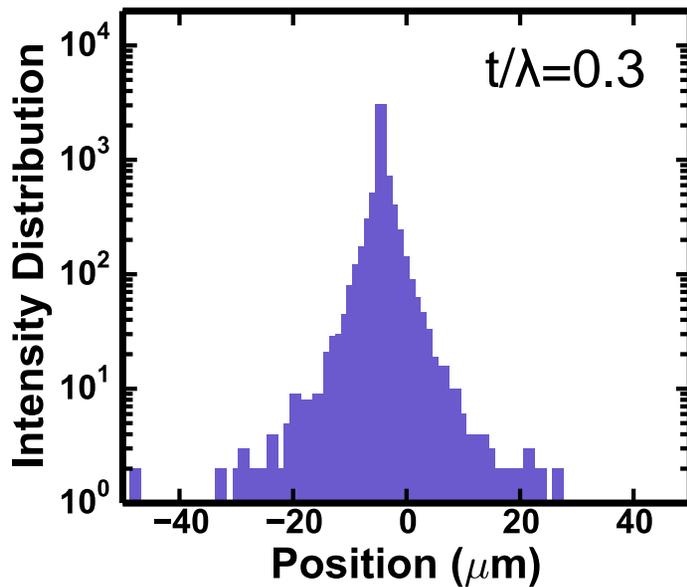
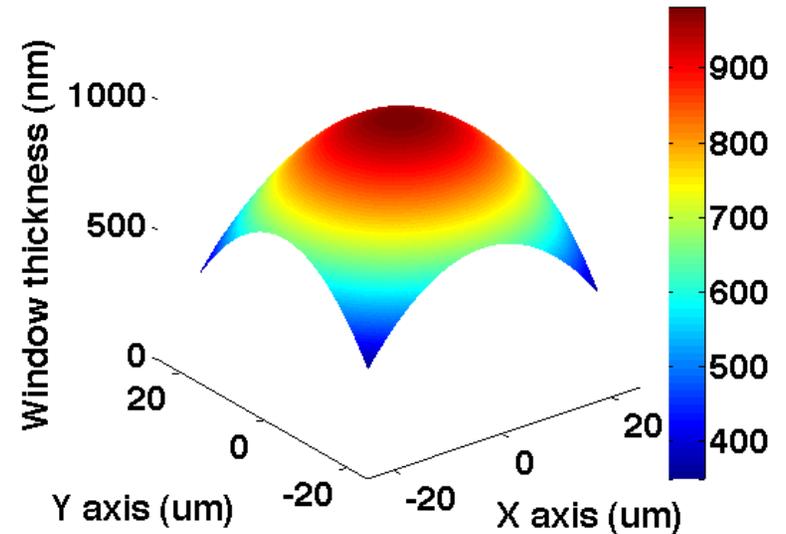
# Challenge #1: Thickness of the liquid layer

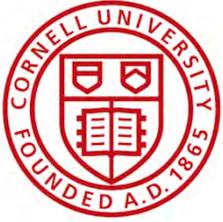
Gold nanoparticles in water



Reasonable to expect ~nm resolution in STEM

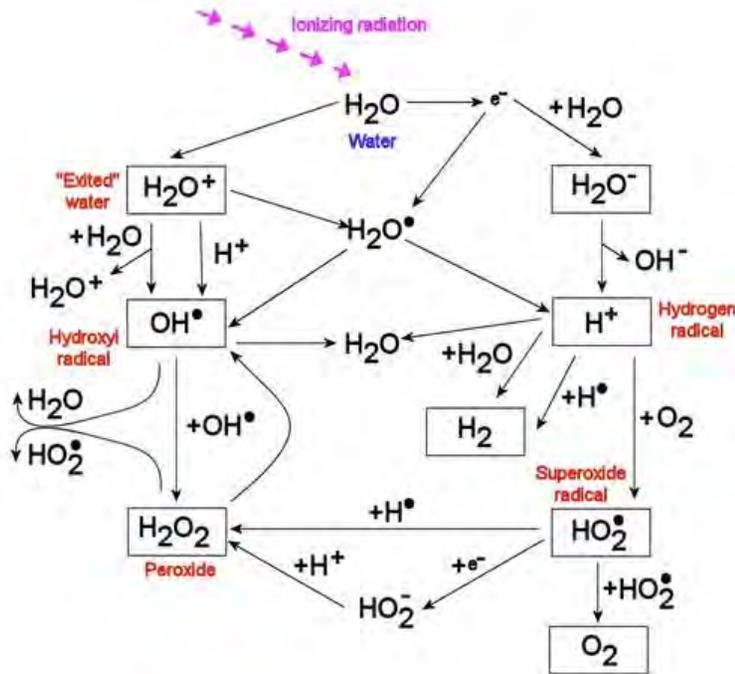
Rotated window thickness, nm





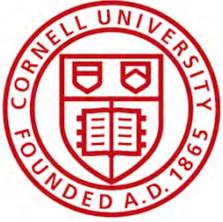
# Challenge # 2: Electron Beam Effects

High energy electron beam interacts with the solution.  
For water, this causes the production of radicals:



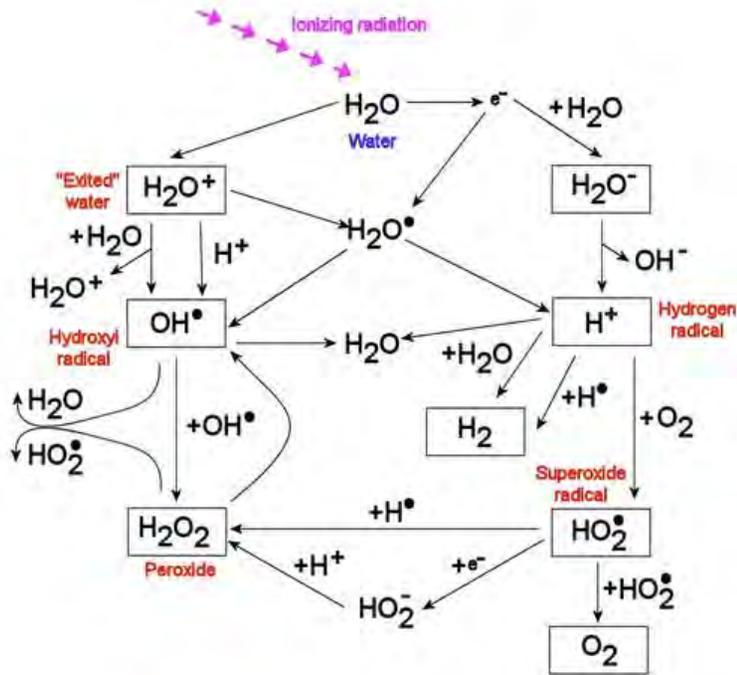
**Radiolysis** of intracellular water (H<sub>2</sub>O)

S.M. Carr, "Ionizing Radiation and Life." 2012



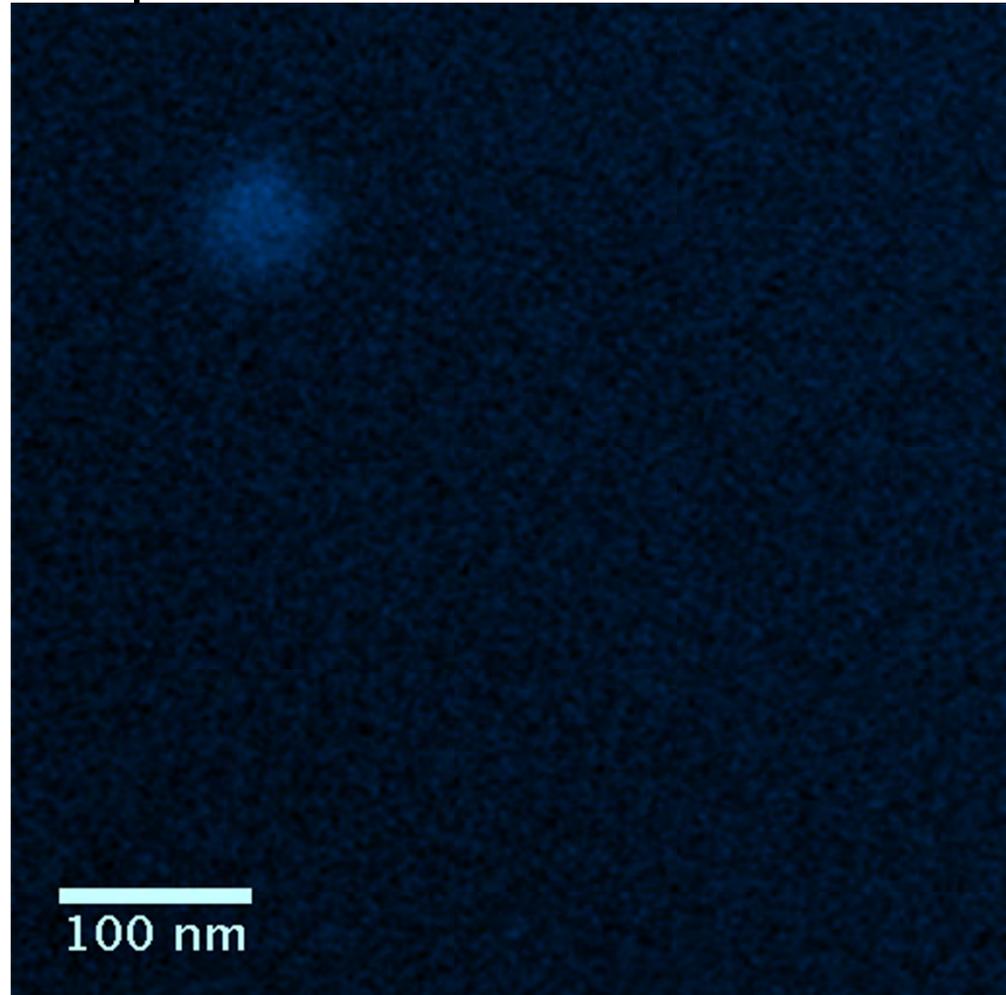
# Challenge # 2: Electron Beam Effects

High energy electron beam interacts with the solution.  
For water, this causes the production of radicals:



Radiolysis of intracellular water (H<sub>2</sub>O)

S.M. Carr, "Ionizing Radiation and Life." 2012

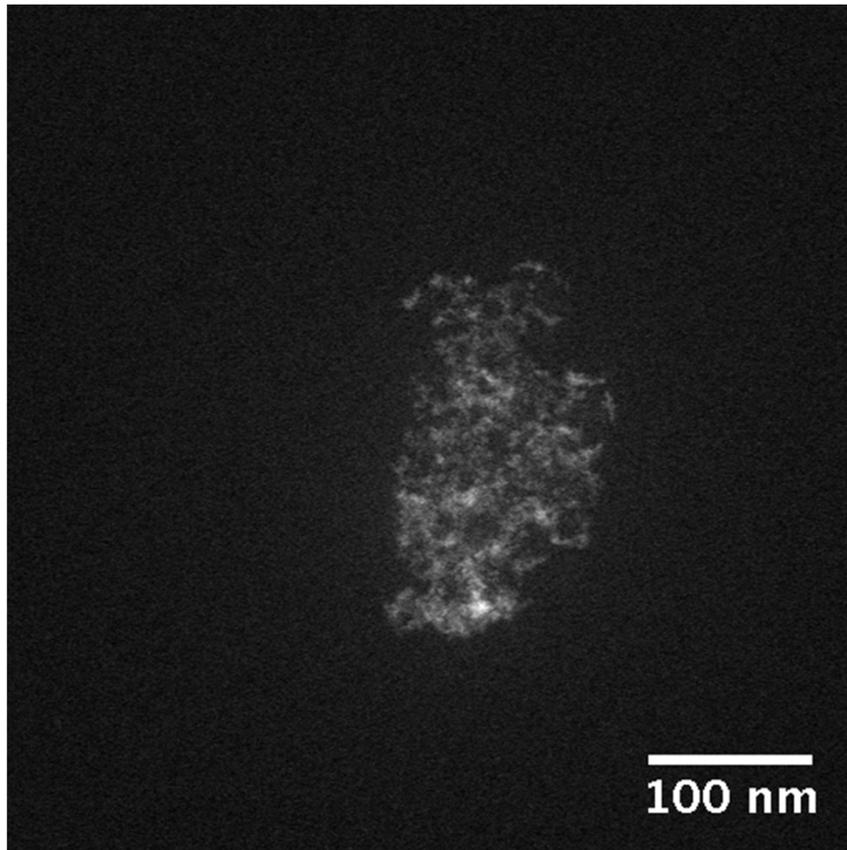


AgNO<sub>3</sub> precursor with excess sodium citrate surfactant

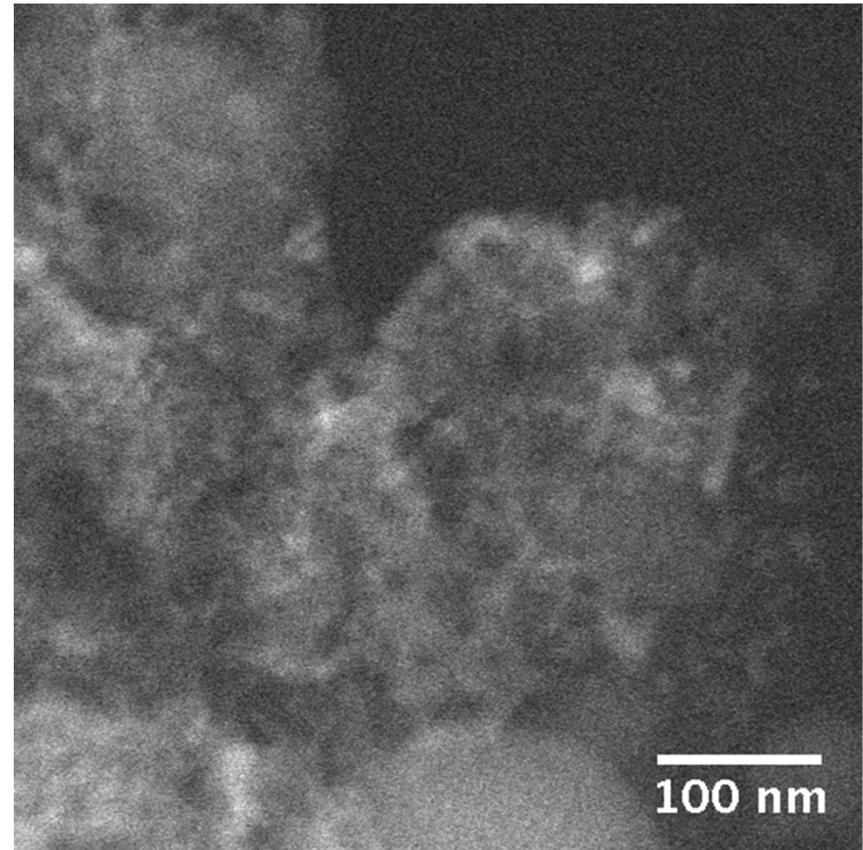


# Beam damage and pH

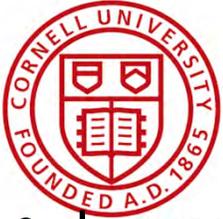
0.1 M  $\text{H}_2\text{SO}_4$



0.1 M NaOH (Pt in **AAEM**)



In base, and especially with an ionomer/membrane, we need low resolution studies!



# Overcoming beam interactions

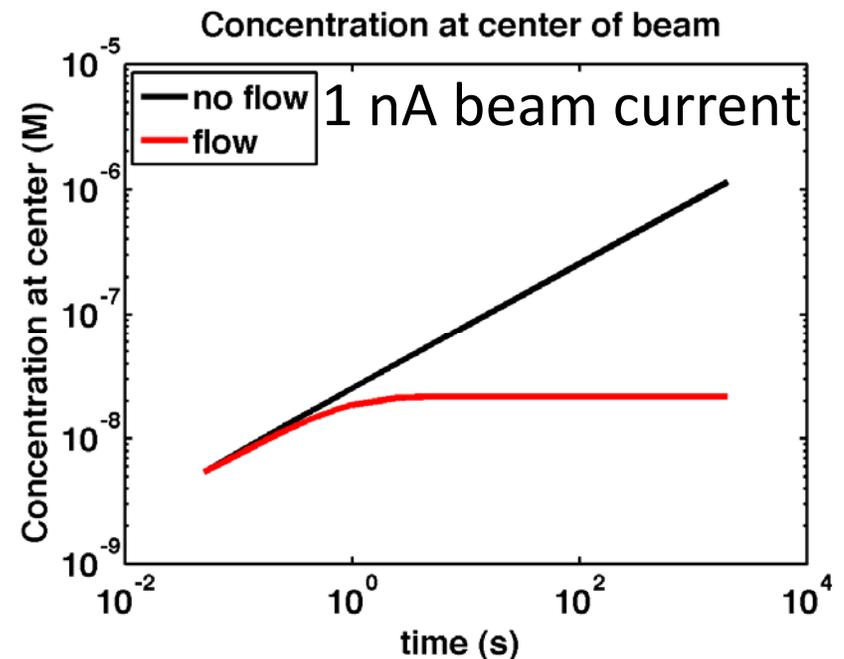
- Lower dose (at cost of signal)
- Short acquisition times or intermittent imaging
- “Low” resolution studies:  
dose  $\sim$  resolution<sup>-2</sup>  
If 100,000 e-/nm<sup>2</sup> for 1 Å,  
1000 e-/nm<sup>2</sup> for 1 nm  
10 e-/nm<sup>2</sup> for 10 nm
- **High** operating voltages:  
2 kinds of damage:
  - Knock on
  - Ionization/radiolysis dominating effect!

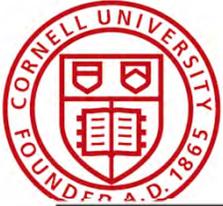
- Flow solution to prevent accumulation of free radicals

$$\frac{\partial C}{\partial t} = D \nabla^2 C$$

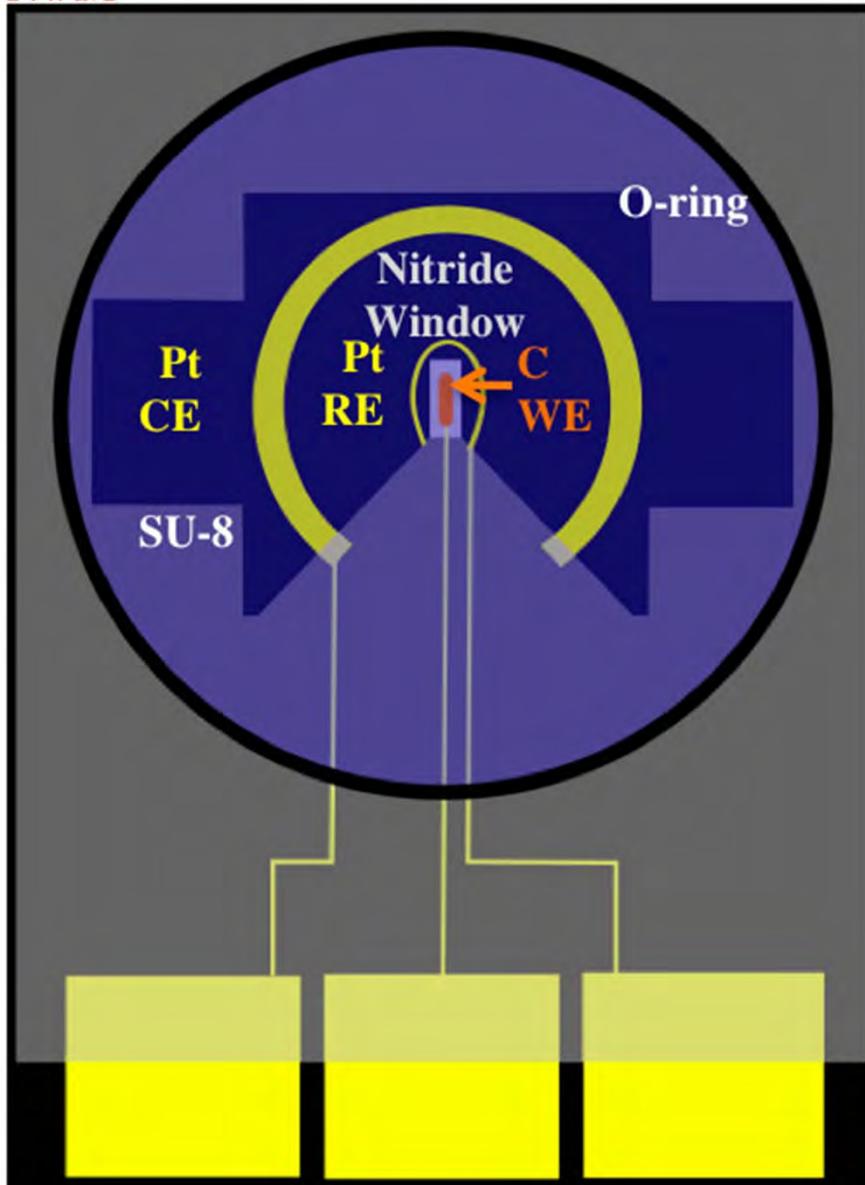
↑  
Diffusion

Change in concentration





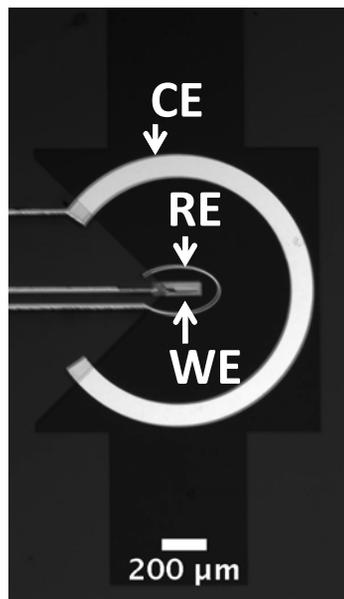
# Electrochemical Cell Chip Design



- electrochemistry from the material of interest, not the chip
  - Electrode only on viewing window
  - SU-8 covering contact leads
  - Electrode with minimal features (Carbon, not Au)
- low scattering from working electrode
- Quantitative electrochemistry with 3-electrode configuration
  - Floating potentiostat sensitive to pA current (Gamry)

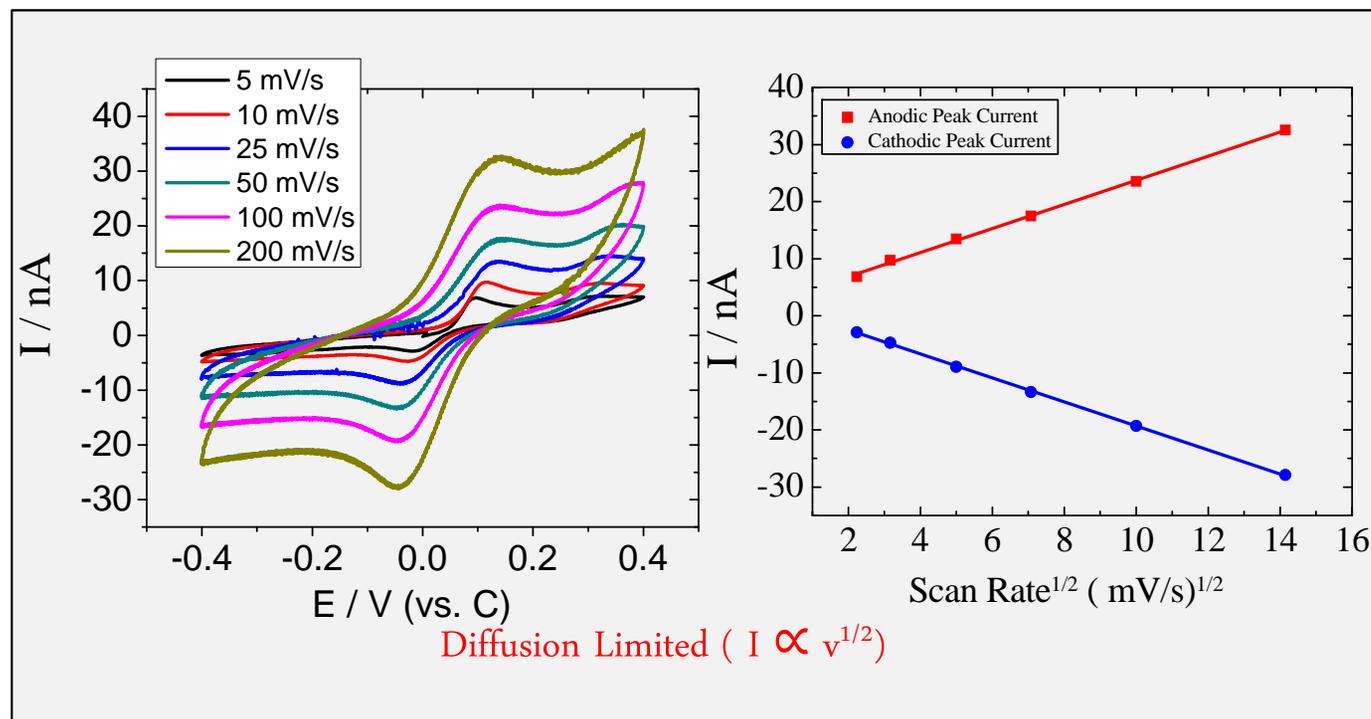
# CVs inside TEM with Electron Beam on

2 mM ferrocenemethanol in 0.2M Phosphate Buffer

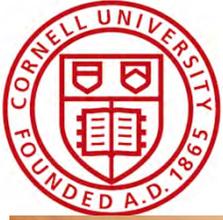


Use Pt Wire as WE

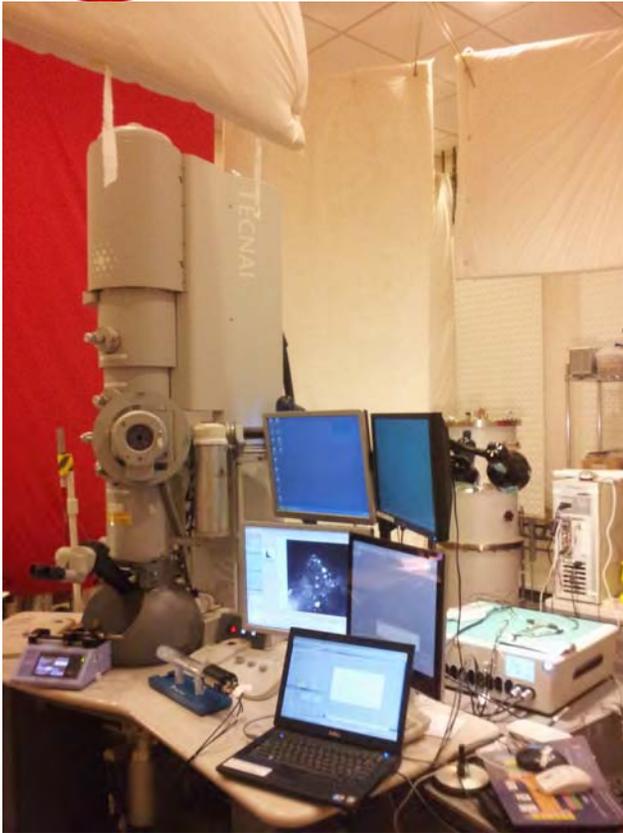
2 mM FcMeOH  
0.2 M PSB



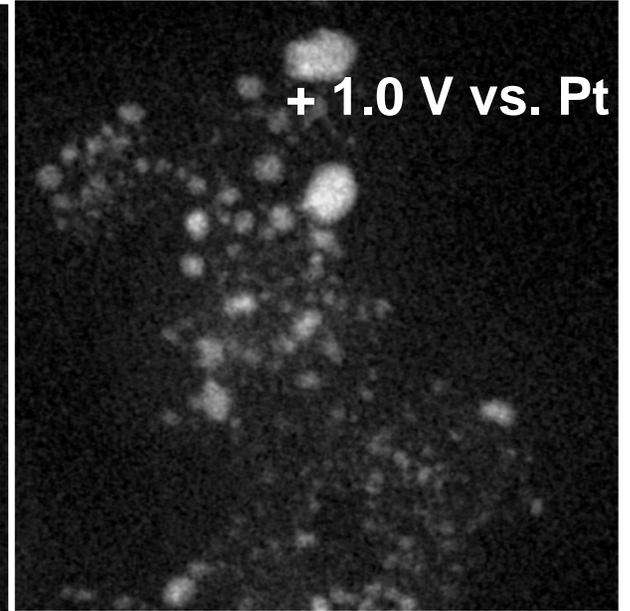
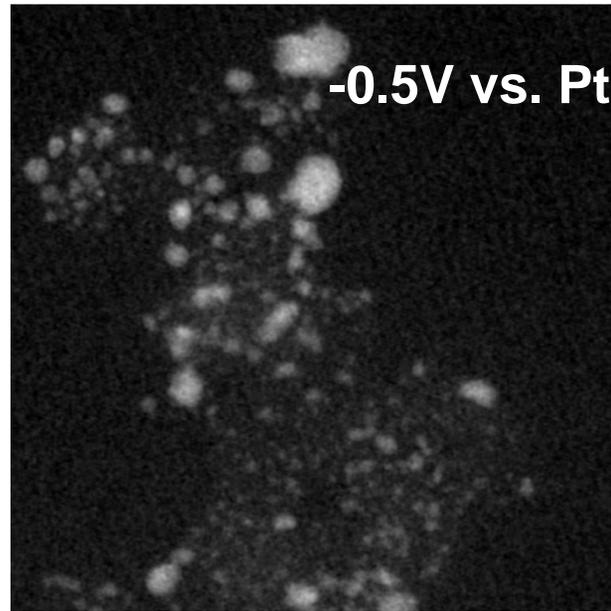
Similar electrochemical behavior to a standard microelectrode!



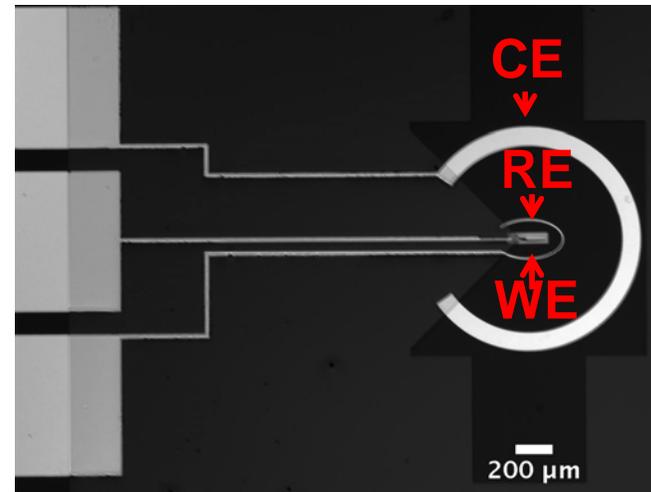
# *In-situ* Electrochemical Cycling of Fuel Cell Catalysts



*In-situ* Electrochemical Holder

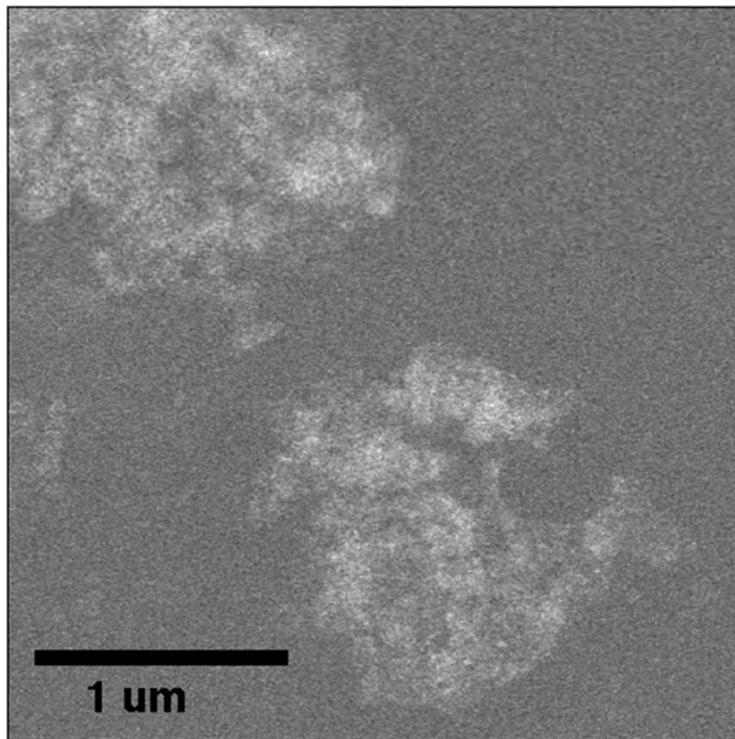
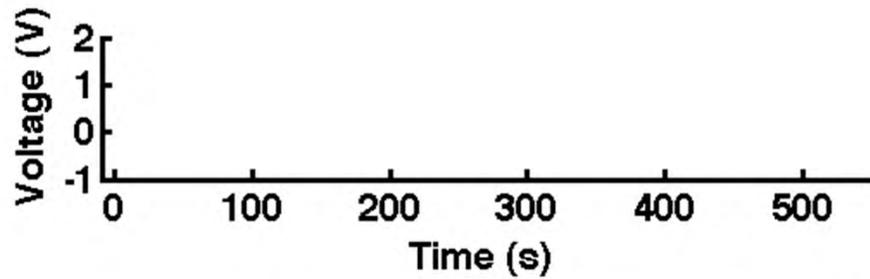


Pt<sub>3</sub>Co/C in 0.1 M Perchlorate Acid

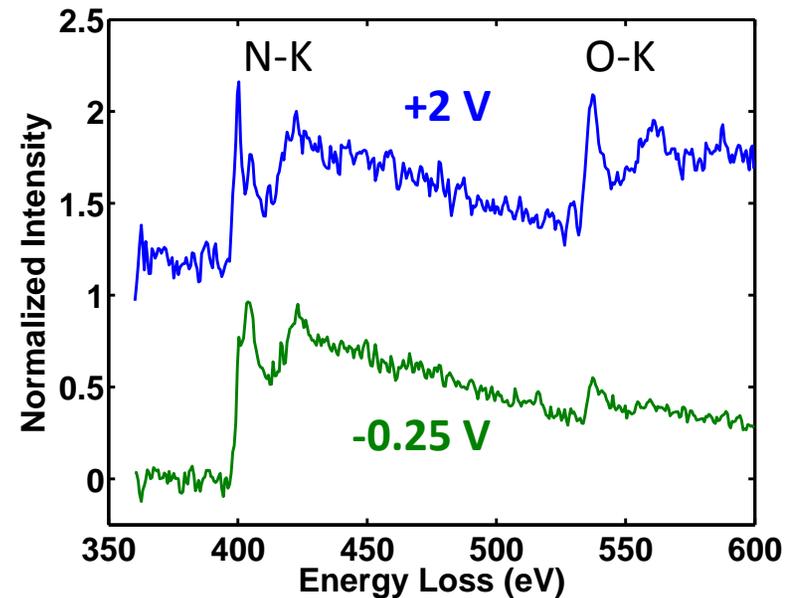
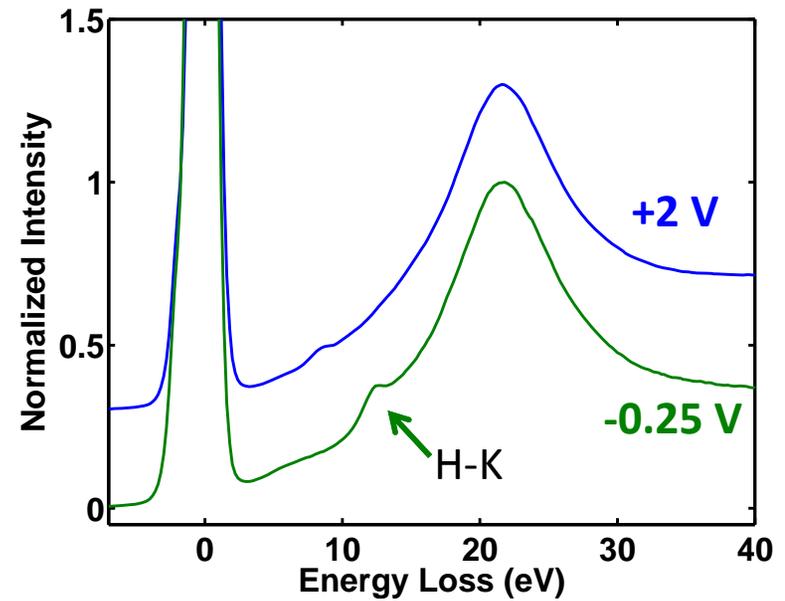




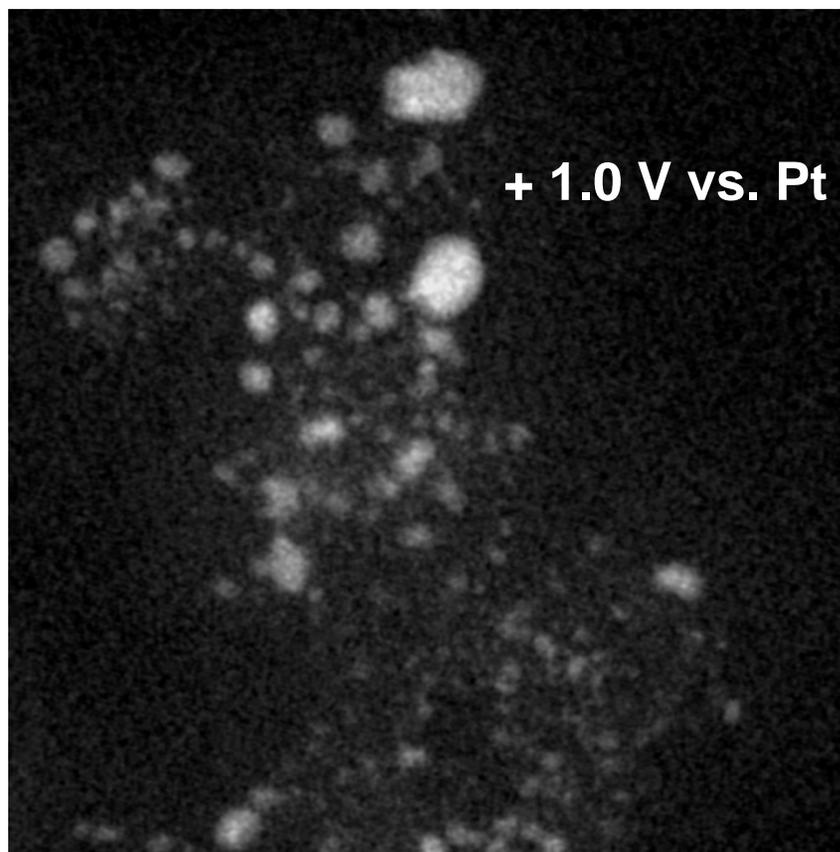
# Generating H<sub>2</sub> and O<sub>2</sub> bubbles in the TEM



Not driven by radiation damage!  
Driven by electrochemistry!

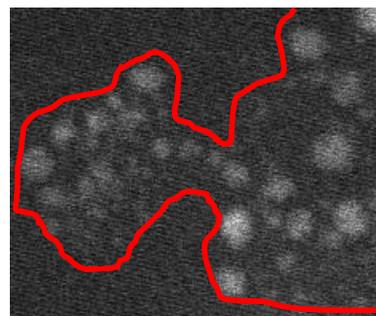


# Fuel Cell Electrochemistry in the TEM

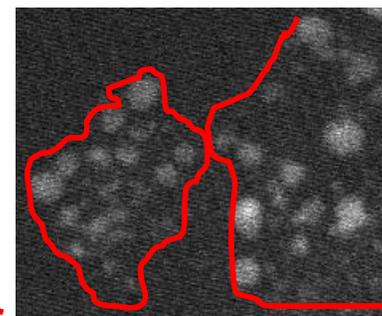


Pt<sub>3</sub>Co/C at +1 V in 0.1M HClO<sub>4</sub>

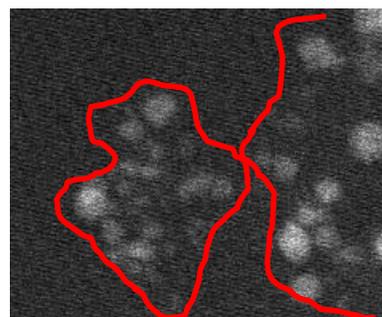
Initial



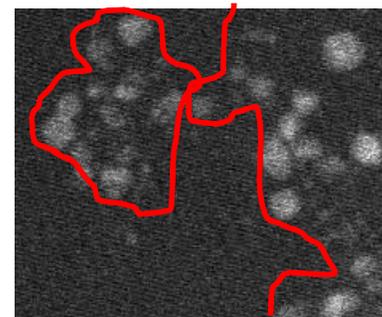
38s



86s



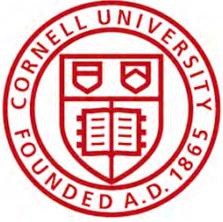
260s



50 nm



Cycling +0.5 V to +1.2 V at 500 mV/s for 220 cycles

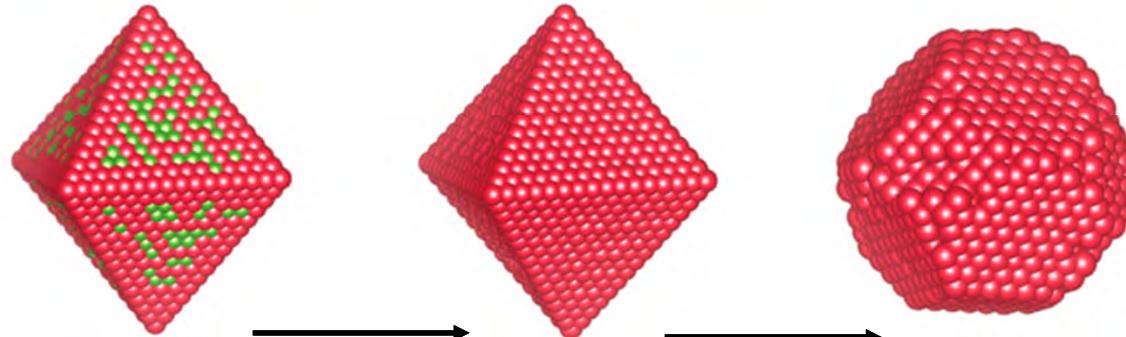


# Highly active fuel cell catalyst: PtNi

Activity of Pt-Ni surfaces: (111) >> (110) > (100) : want octahedral

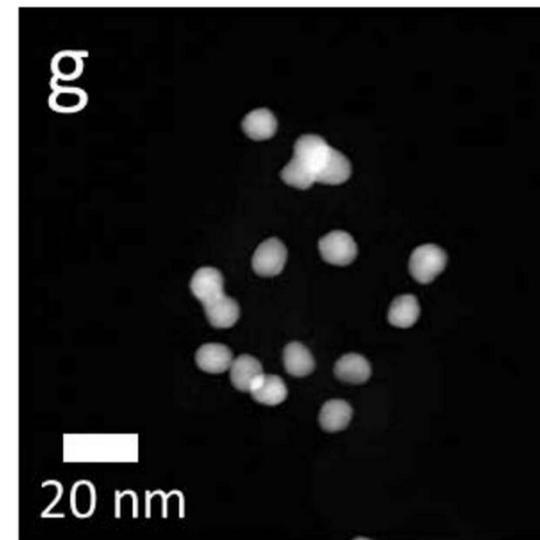
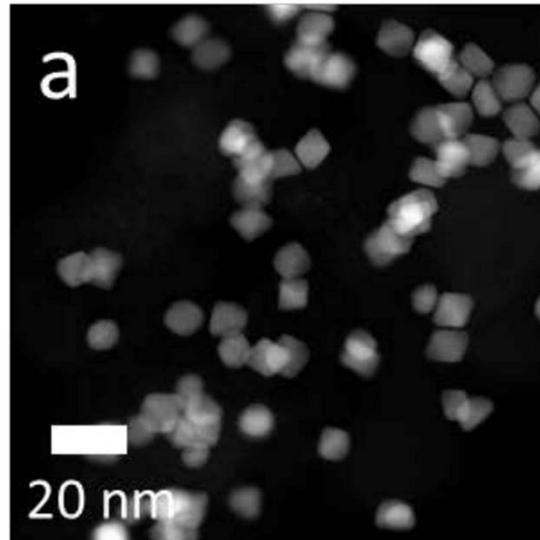
par versus

PtNi

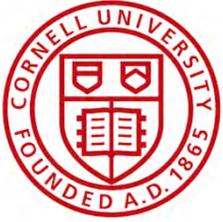


lose Ni on surface  
initial

lose octahedral shape  
after 8k cycles

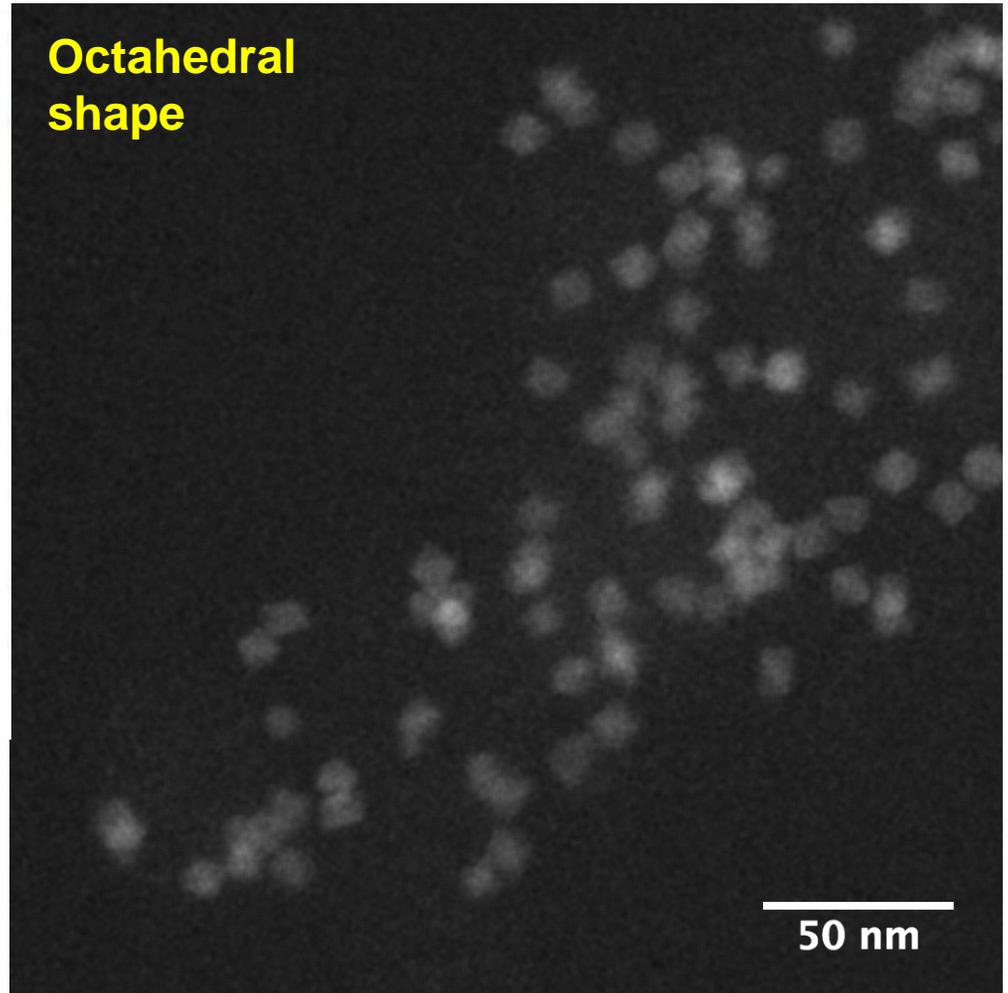
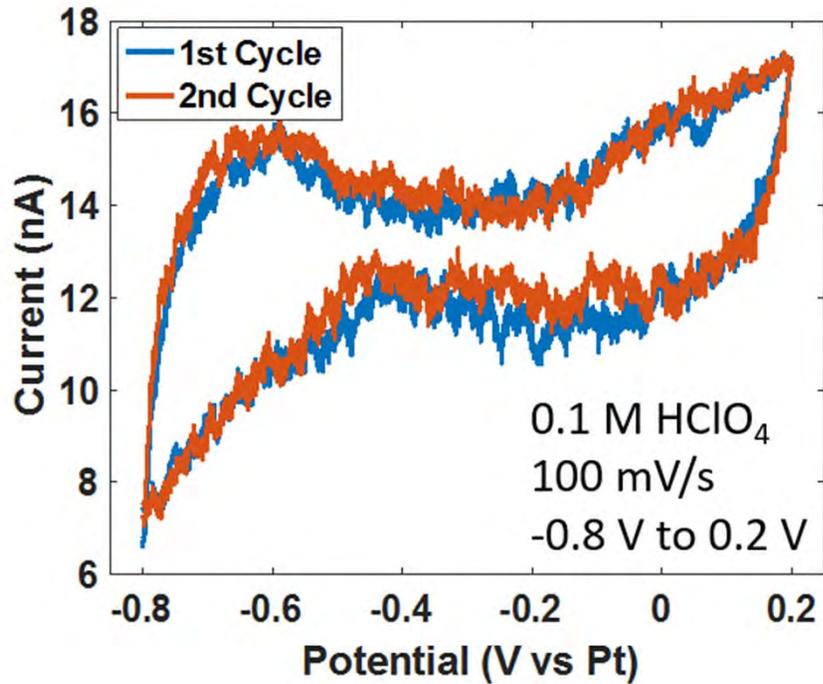


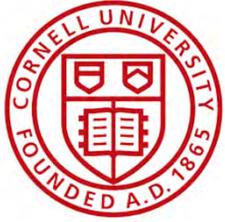
Vera Beermann, et al.  
NanoLetters, 2016,



# PtNi catalyst in electrochemical cell

PtNi octahedra in 250-320 nm 0.1M HClO<sub>4</sub>



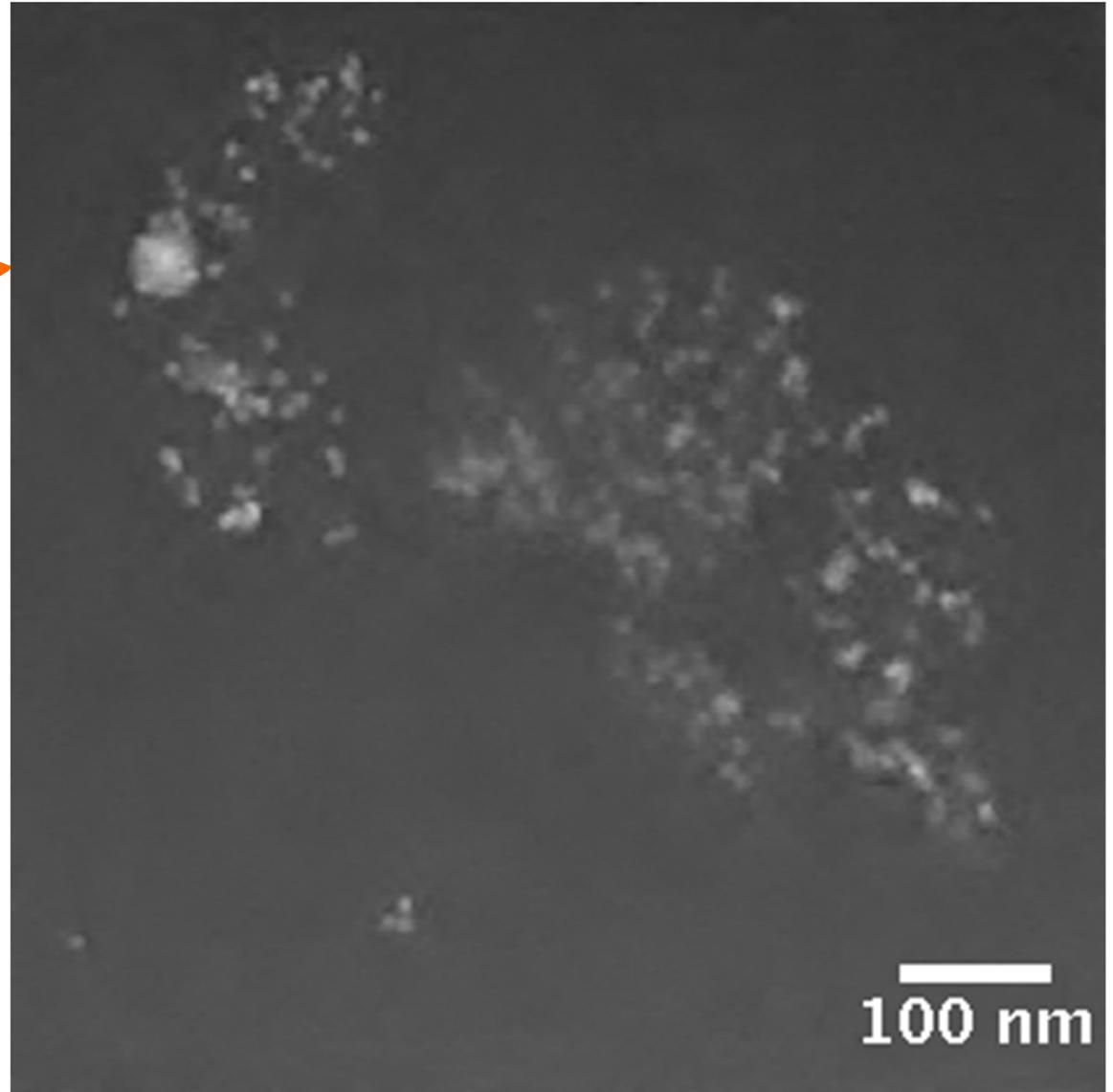


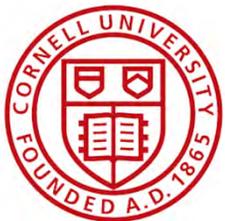
## First 20 CV up to 0.2 V vs Pt: Nickel particles disappear while octahedral shape remains

Ni-rich particles disappear during cycling, leaving behind a Pt-rich core.

Ni becomes spongy before disappearing.

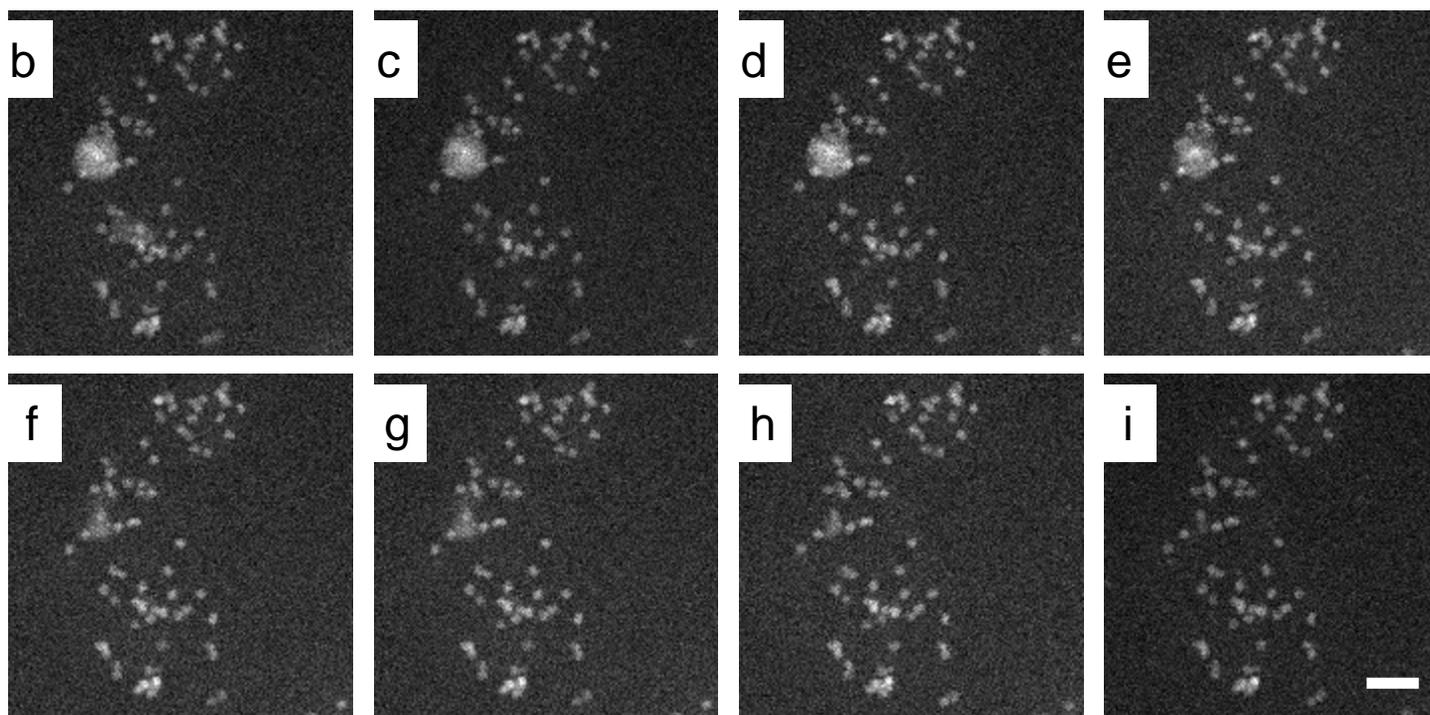
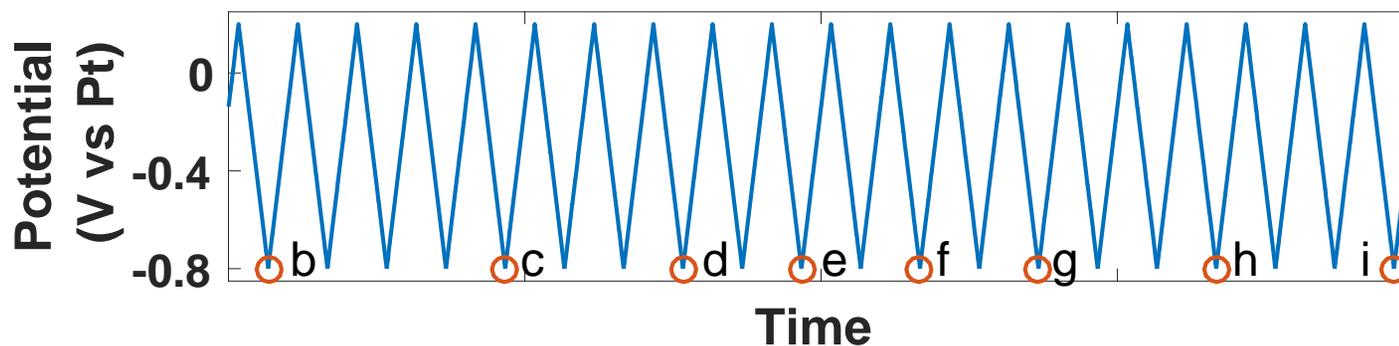
PtNi octahedra are stable during potential cycling from -0.8 V to 0.2 V vs Pt (0 to 1 V vs RHE)





# First 20 CV up to 0.2 V vs Pt: Nickel particles disappear while octahedral shape remains

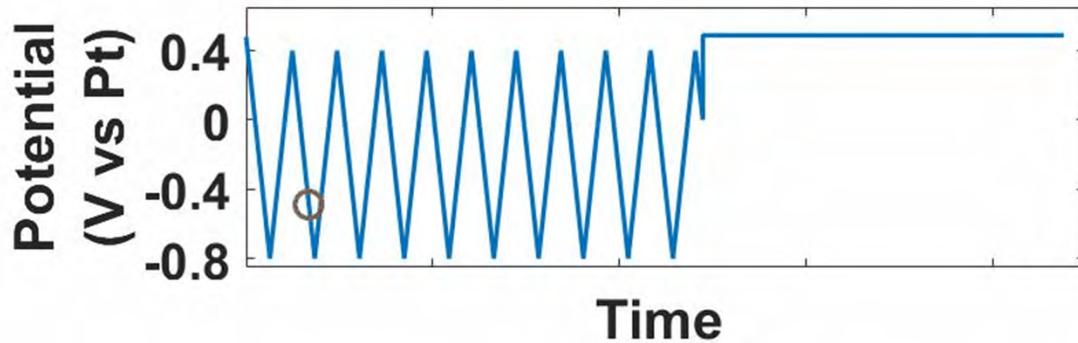
a



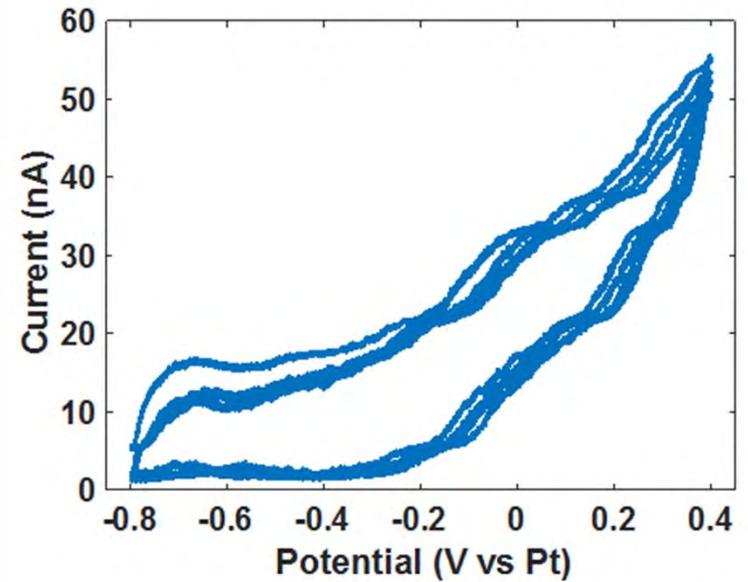
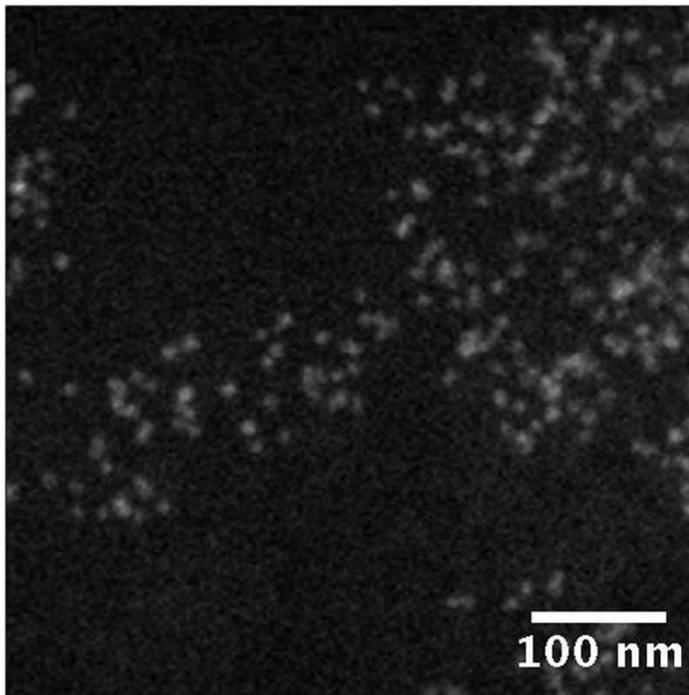
50 nm

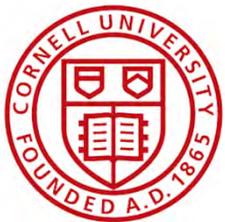


# 10 CV's from -0.8V to 0.4V vs Pt Then: high OCV potential (>~0.5V) coalesces PtNi

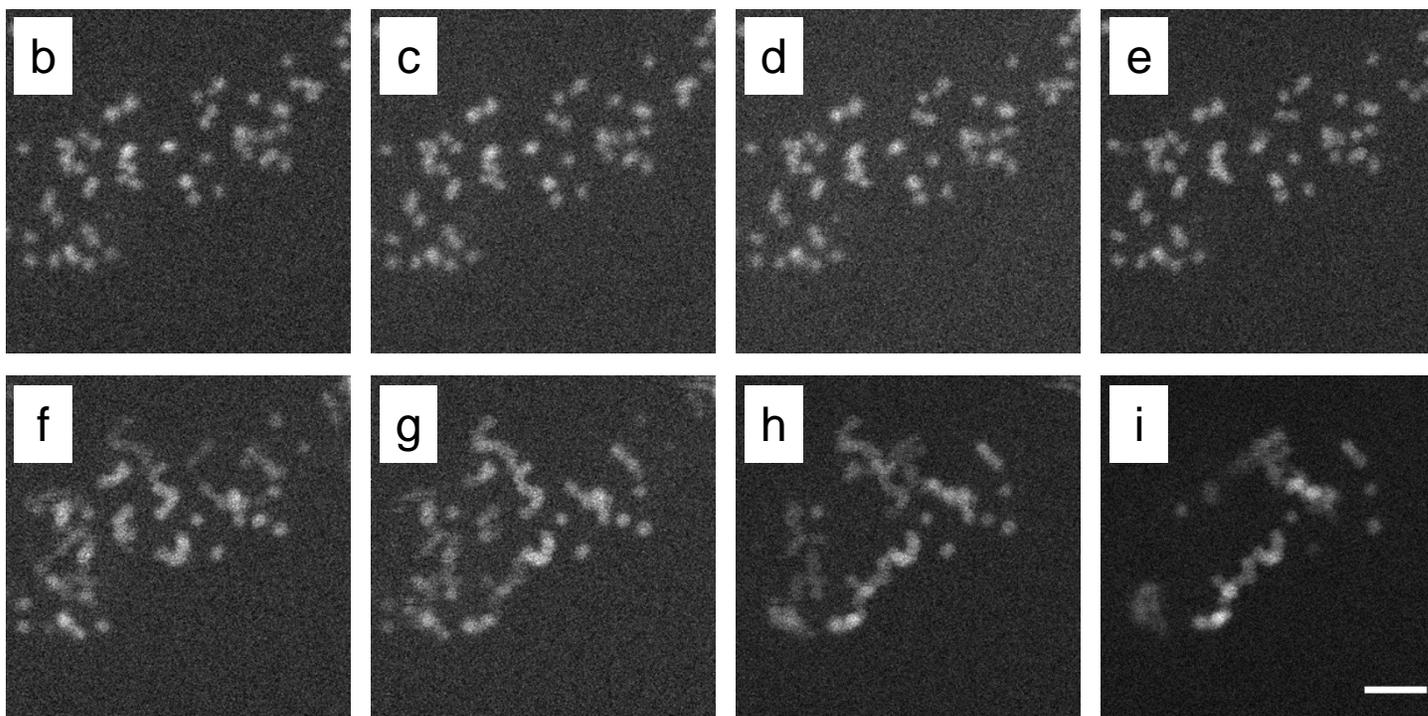
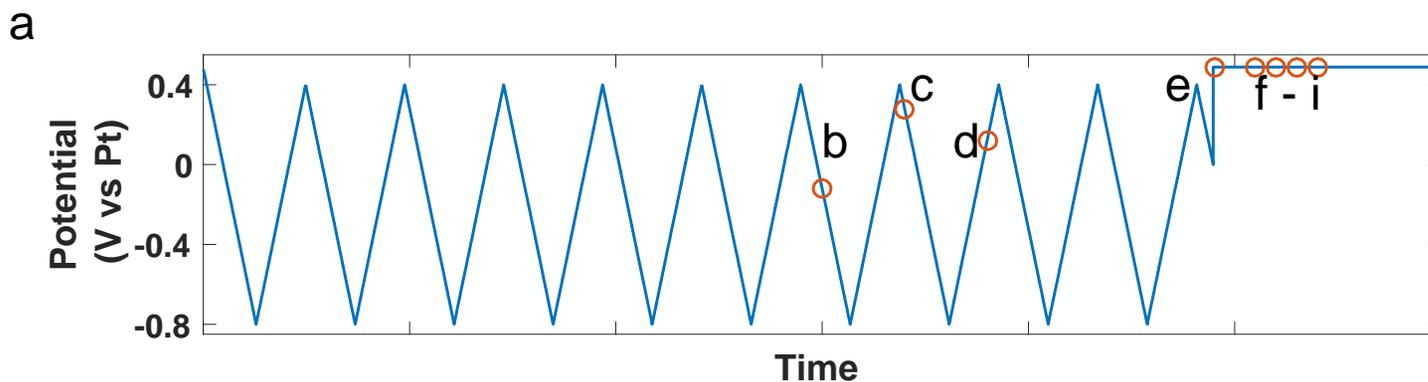


- Control your starting and shut down potentials!
- Cycling to 0.4 starts to soften the facets
- Holding at high potential coalesces particles

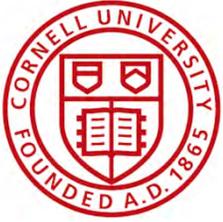




# 10 CV's from -0.8V to 0.4V vs Pt then: high OCV potential ( $>\sim 0.5V$ ) coalesces PtNi

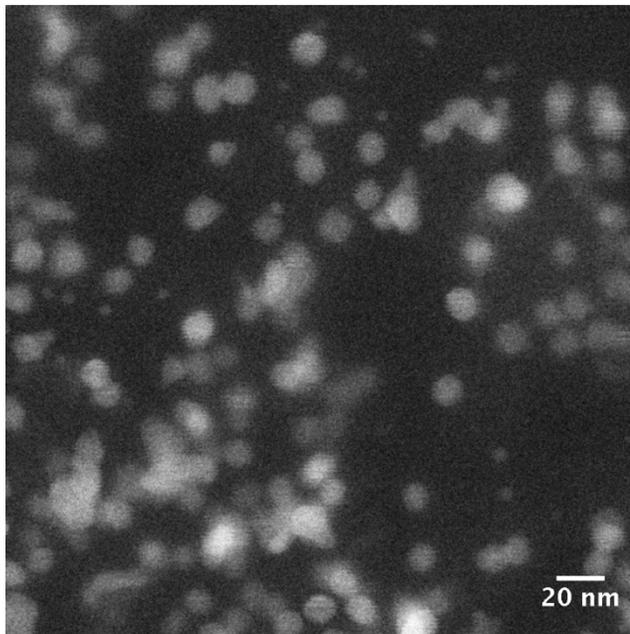


50 nm



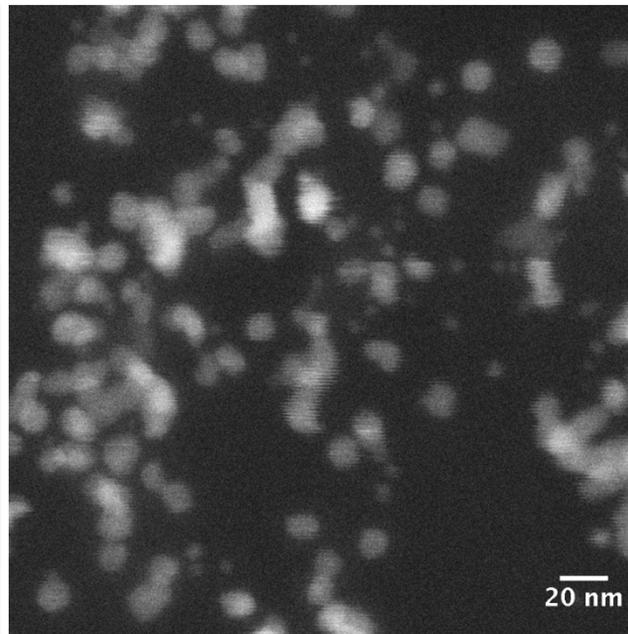
# After holding at high potential, particles on the electrode have lost their shape

elsewhere  
on the electrode



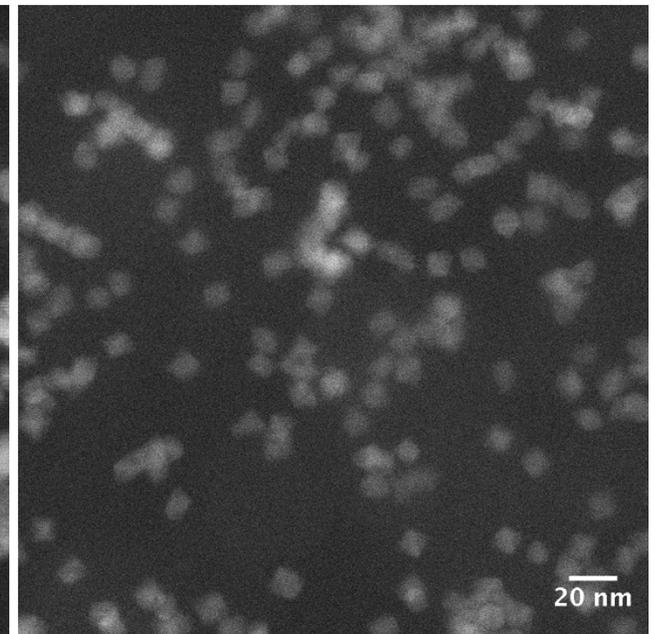
octahedral shape is no longer clear after cycling to higher potential

elsewhere  
on the electrode

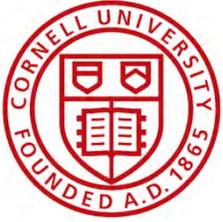


particles appear larger

off the electrode

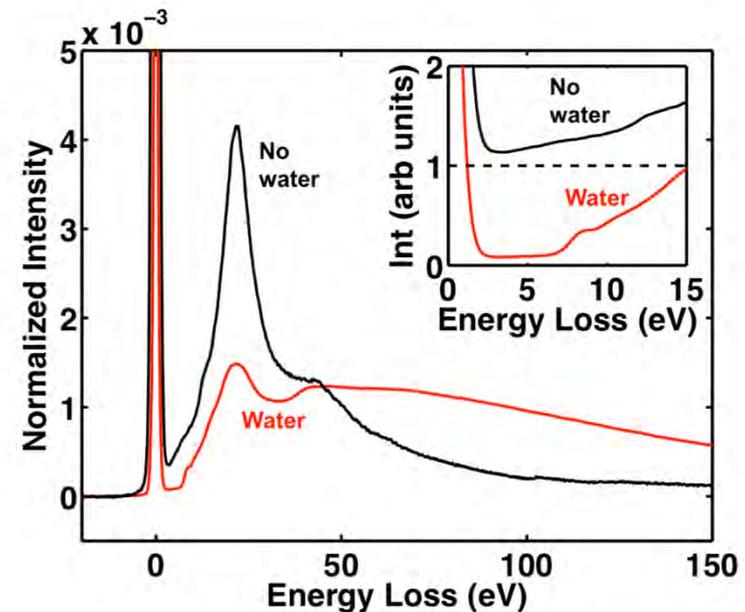
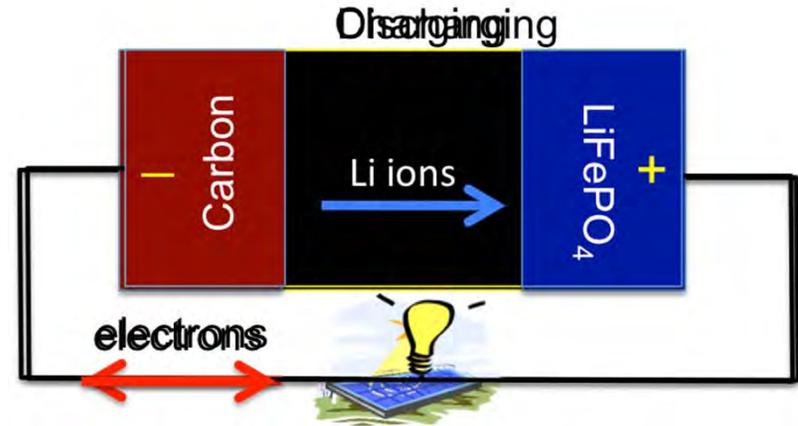


where there was no electrochemistry, the particles retain their octahedral shape



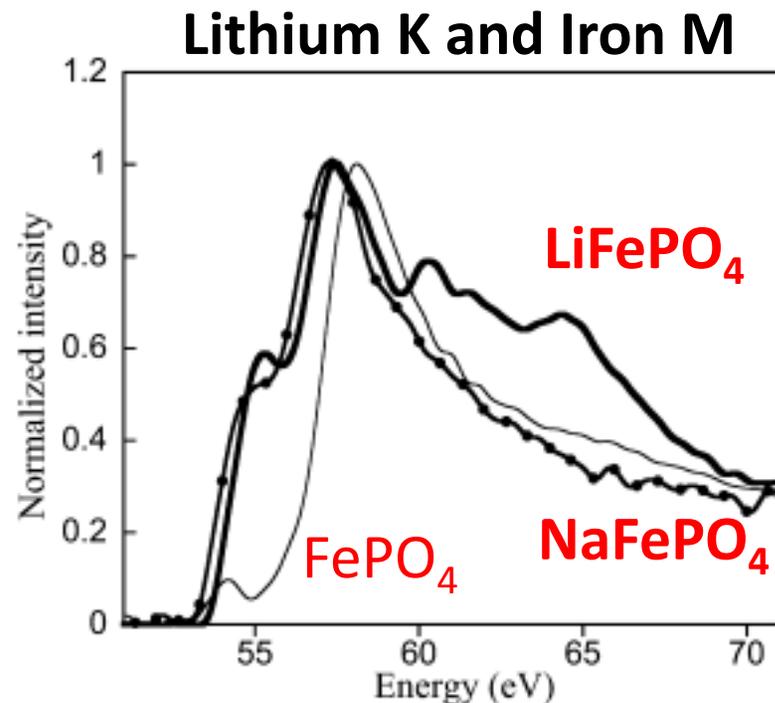
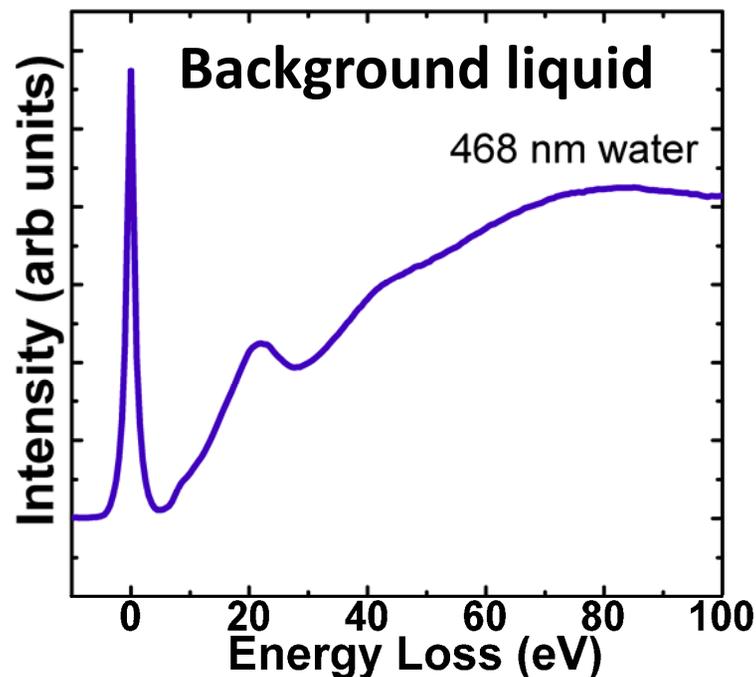
# Liquid cell TEM for Li-ion batteries

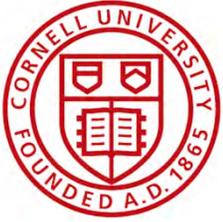
- Li-ion batteries
  - Complex electrochemical system where interfaces are critical
  - Lithiation mechanism often the limiting factor –  $\text{Li}^+$  transport in nanoscale materials
  - Inhomogeneous degradation
- Lithium is a challenging element to observe in the TEM:
  - Li K edge (54 eV) is overwhelmed by liquid plasmons



# How to track *Lithium*?

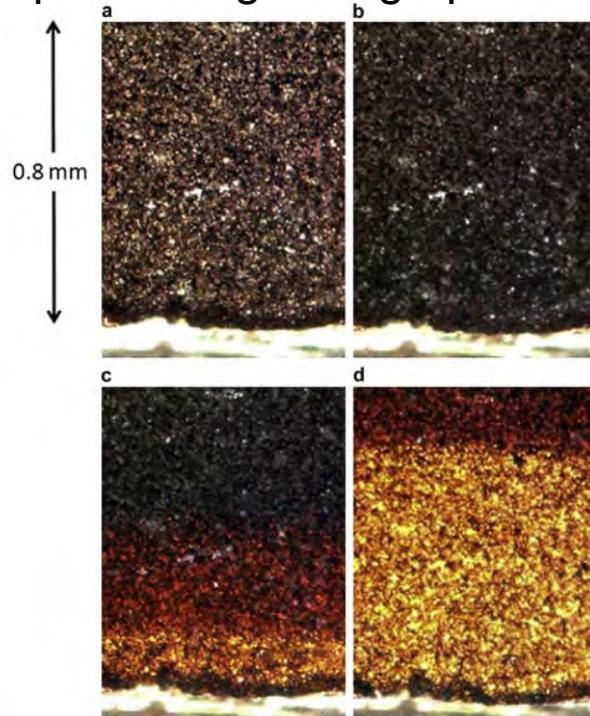
- Lithium is a challenging element to observe in the TEM
  - Small elastic cross section ( $\sim Z^2$ )
  - Li invisible by x-ray
  - Li K edge (54 eV) is overwhelmed by liquid plasmon and overlaps with transition metal M edges





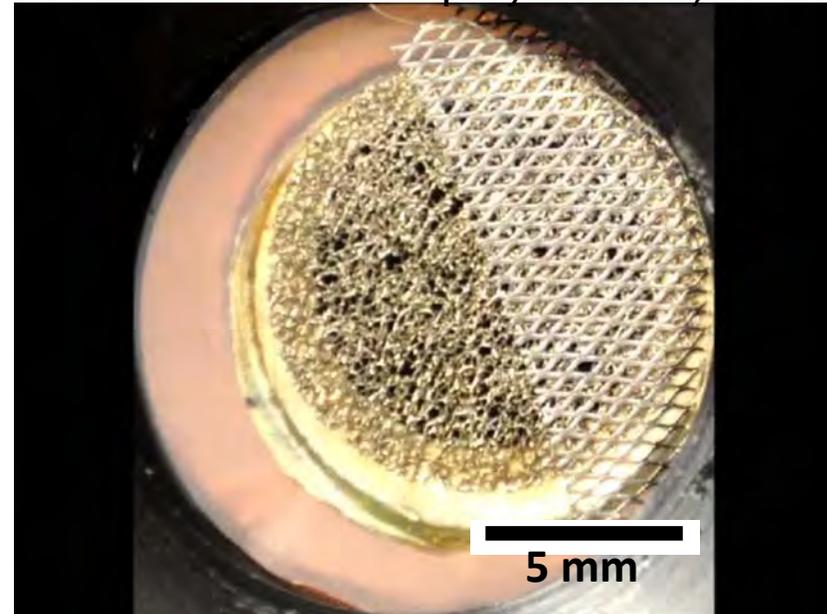
# Observing the Lithiation State in *In Operando* Batteries

Optical images of graphite lithiation

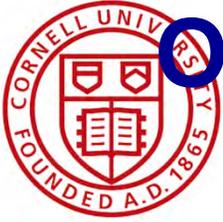


Harris, et al. Chem. Phys. Lett. (2010)

Color changes in Li-Sulfur  
(Red = Long-chain polysulfides  
Green = Shorter polysulfides)

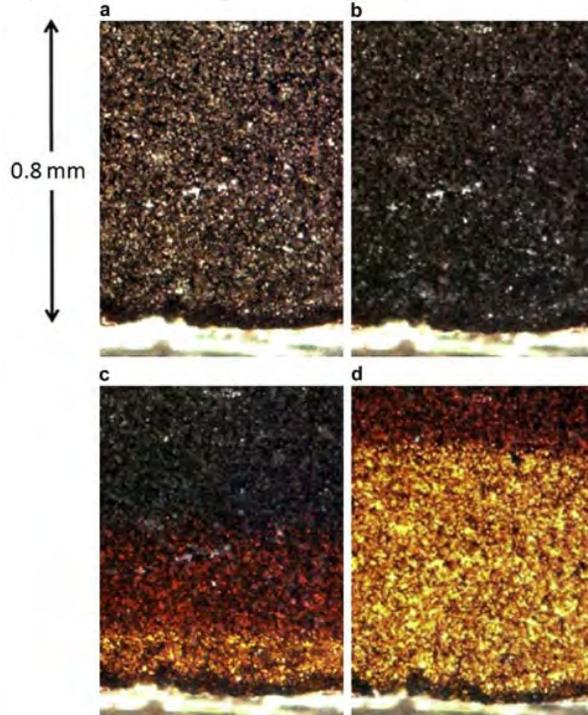


- Chemists rely on color changes: (dis)charging changes the electronic structure (few eV) = optical transition
- Valence EELS probes the unoccupied DOS = we can observe these electronic structure changes



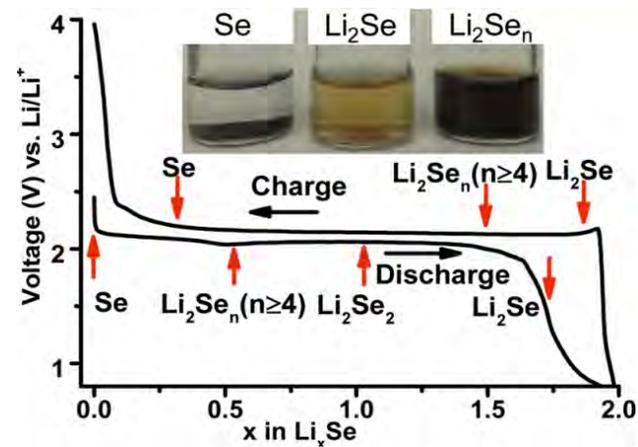
# Observing the Lithiation State in *In Operando* Batteries

Optical images of graphite lithiation



Steve J. Harris, et al. Chem. Phys. Lett. (2010)

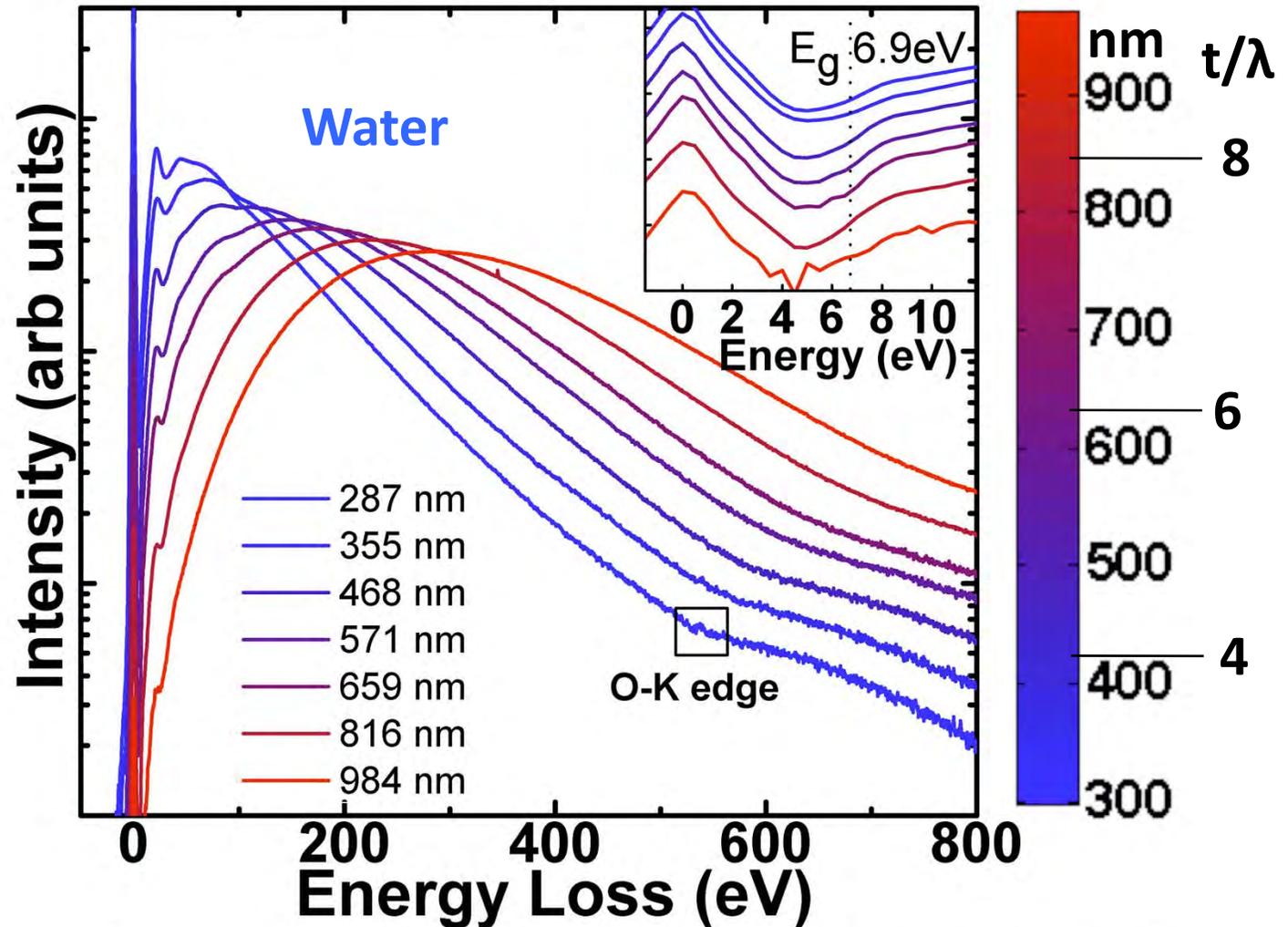
Optical images of Li/Se lithiation



Janjie Cui, et al. J. Am. Chem. Soc. 2013.  
DOI: 10.1021/ja402597g

- Electrodes change color: (dis)charging changes the electronic structure (few eV) = optical transition
- Valence EELS probes the unoccupied DOS = we can observe these electronic structure changes

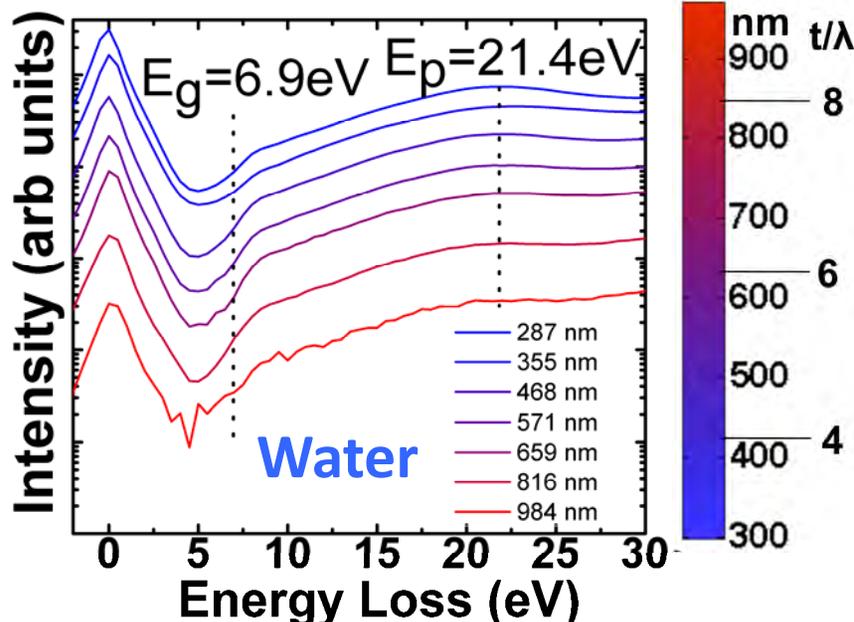
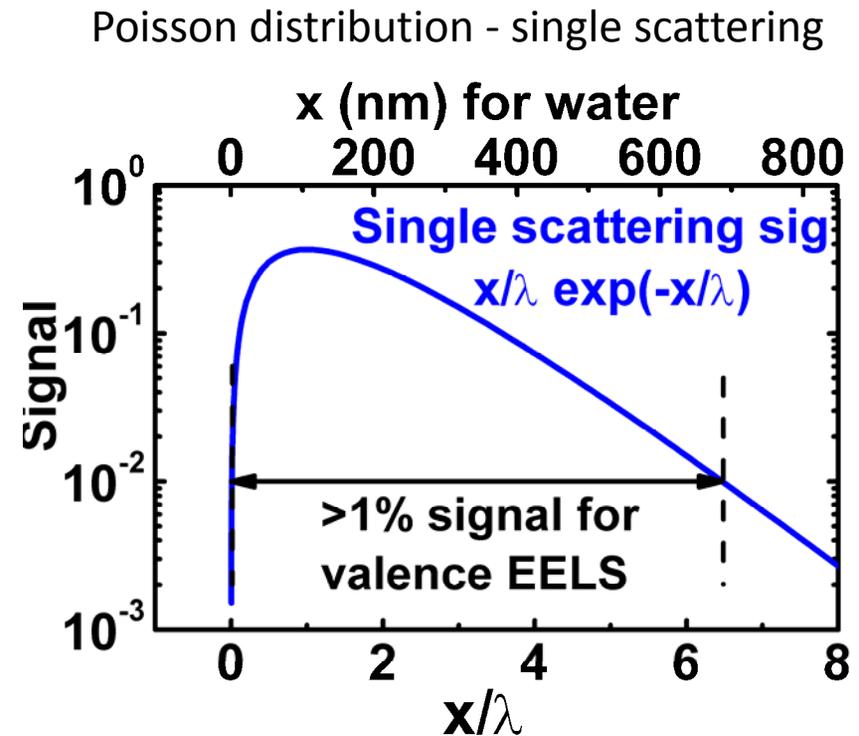
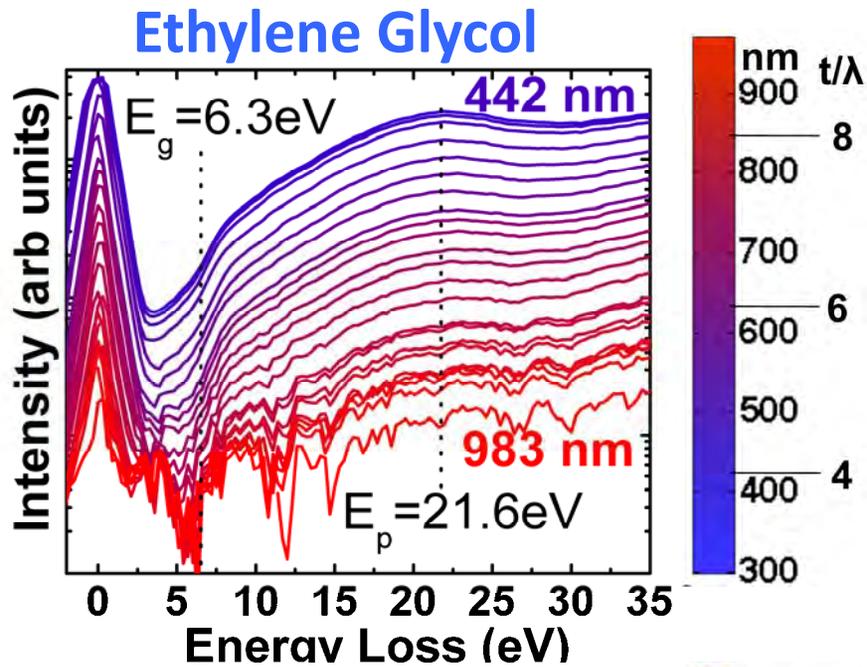
# Thickness limitations of core-loss EELS



**O-K core loss EELS – detectable  $< 3 \lambda$  (~300 nm)**

At higher thicknesses the bulk Plasmon overwhelms the signal.

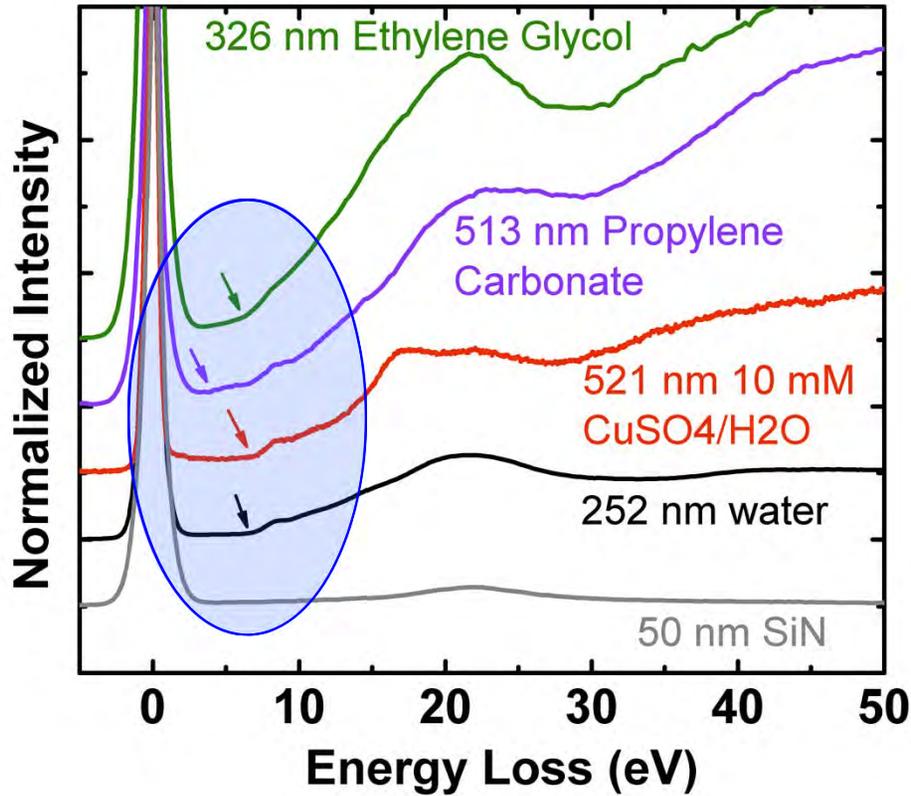
# Thickness limitations of valence EELS



**Valence EELS –  
resolvable  $< 6.5 \lambda$**

There is little lower-energy plural scattering to obscure the signal.

# Valence EELS



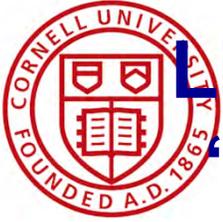
Gives information about:

- Optical gap
- Local electron density
- Thickness
- Can see “optical transitions”

Resides at lower energies

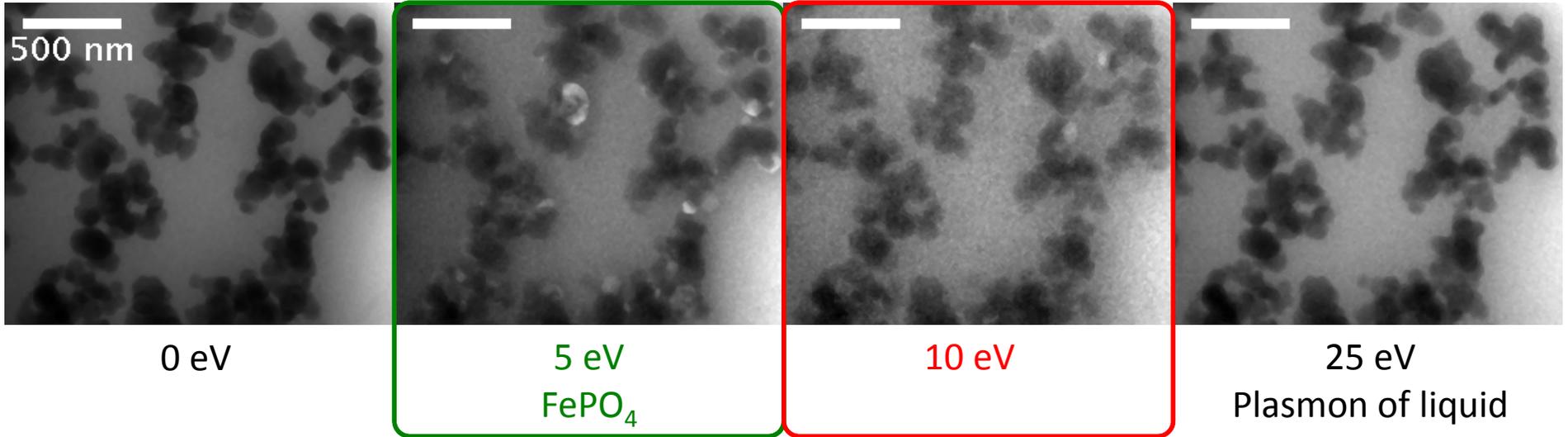
➔ less affected by multiple scattering

Liquid	Optical Gap by EELS (eV)	Optical Gap by UV-VIS (eV)	
Water	6.9	6.5*	*from CRC
10mM CuSO4/Water	5.8	4.1	
Propylene Carbonate	4.2	5.5	
Ethylene Glycol	6.3	6.3	

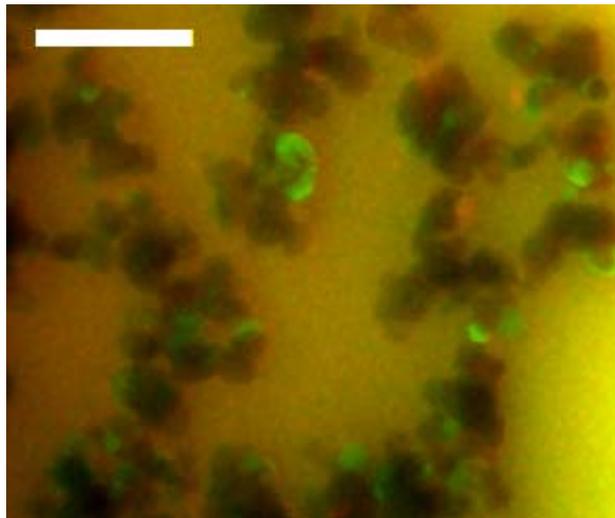
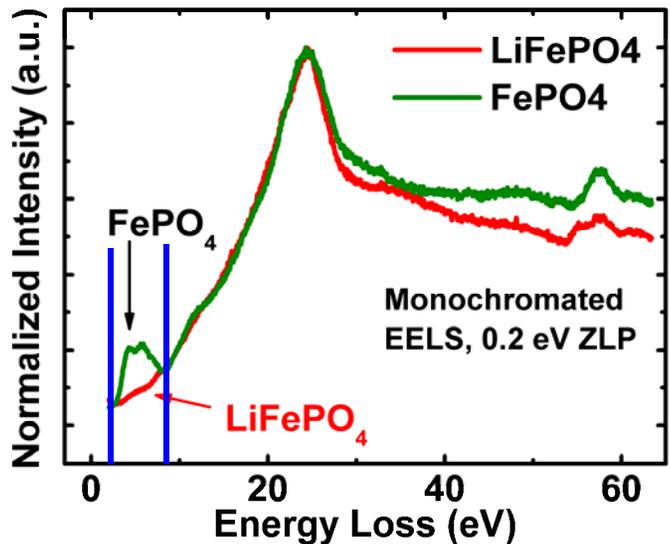


# LiFePO<sub>4</sub> in Liquid by Energy Filtered TEM 'Optical' Spectroscopy at nm resolution

Chemical (not diffraction) contrast!



Monochromated EELS, 0.2 eV resolution (DRY)



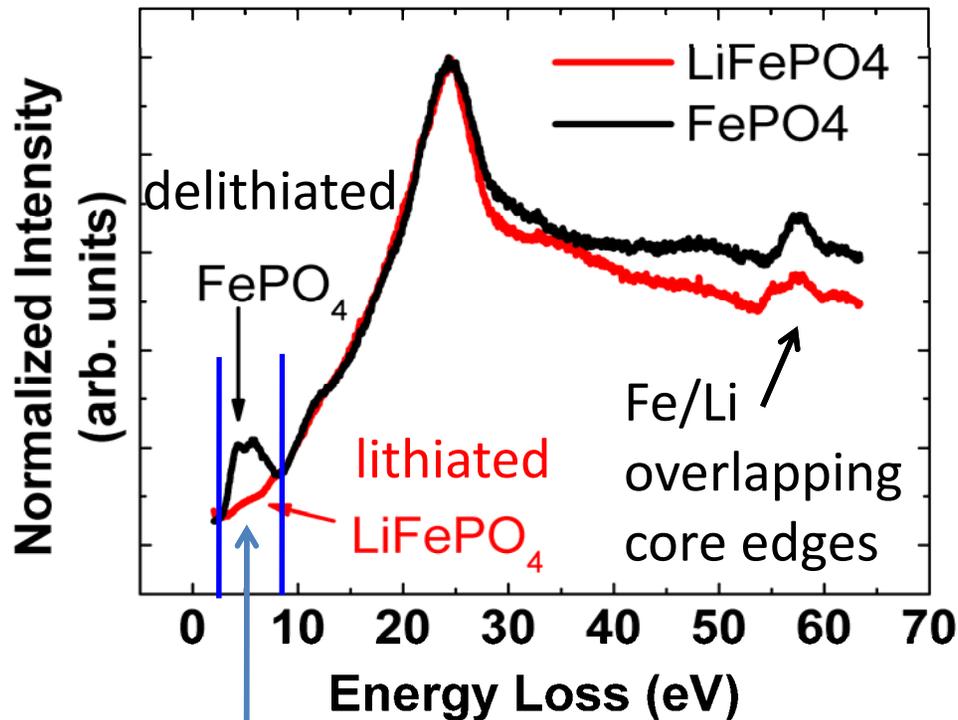
FePO<sub>4</sub> region  
at 5 eV

Interband  
transition at  
10 eV

Holtz et. al. Microsc.  
Microanal. 19, 1-9, 2013.

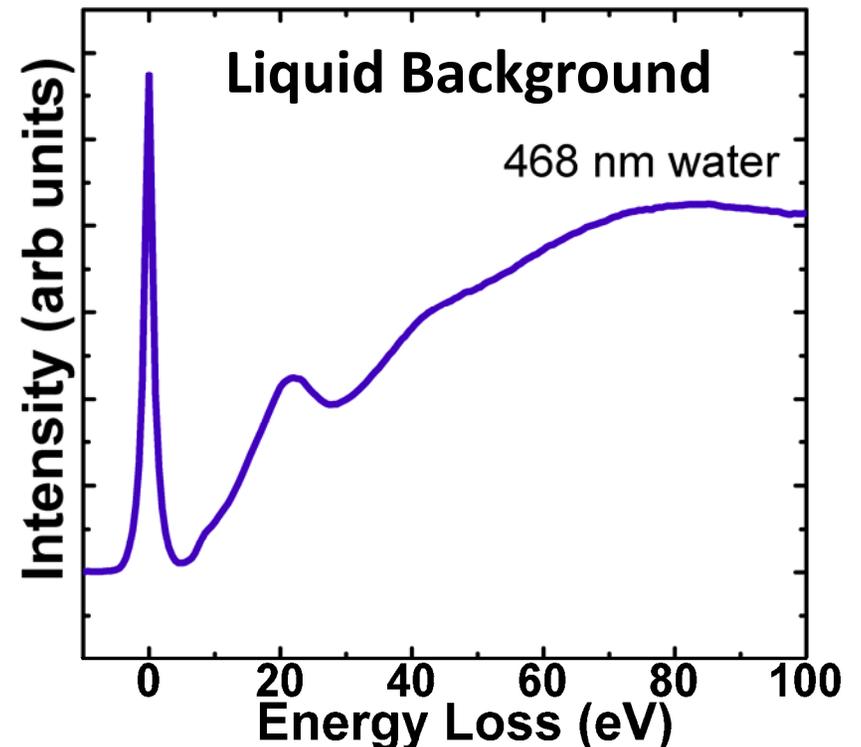
# Valence EELS of $\text{LiFePO}_4$ : a battery cathode

Monochromated EELS,  
0.2 eV resolution



Interband transition, when lithiated, the Fe bands are filled & peak disappears

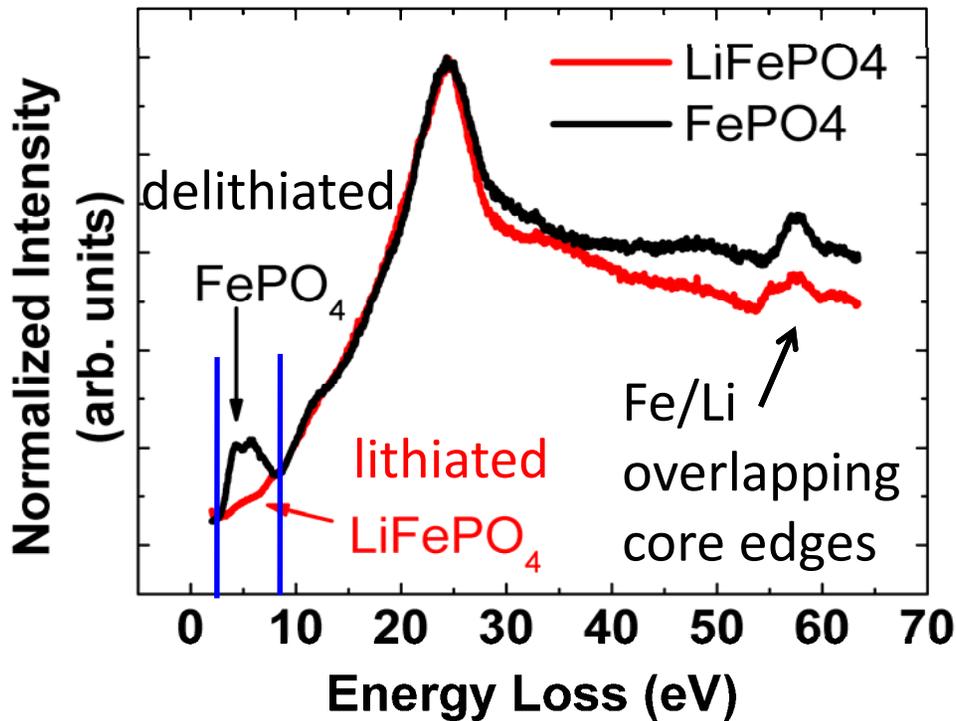
EELS of water



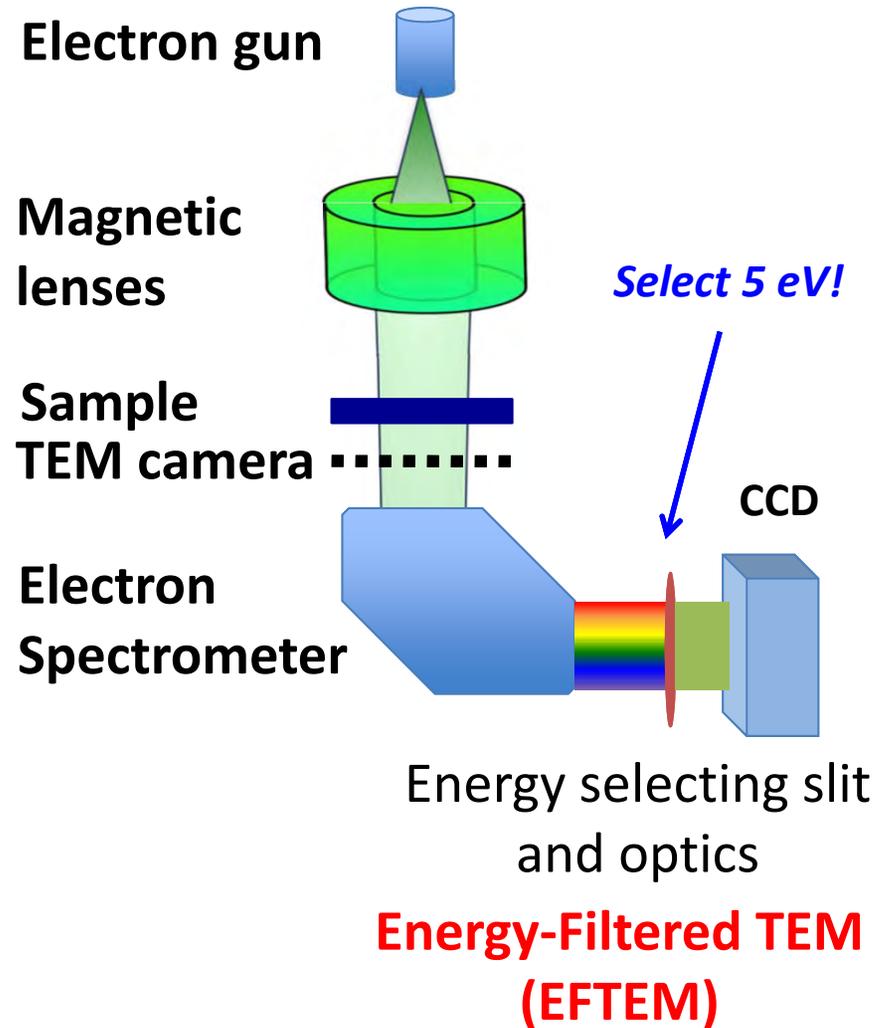
- Best possible signal to background in liquid
- Large cross section for scattering = strong signal

# Valence EELS for Batteries

Monochromated EELS,  
0.2 eV resolution

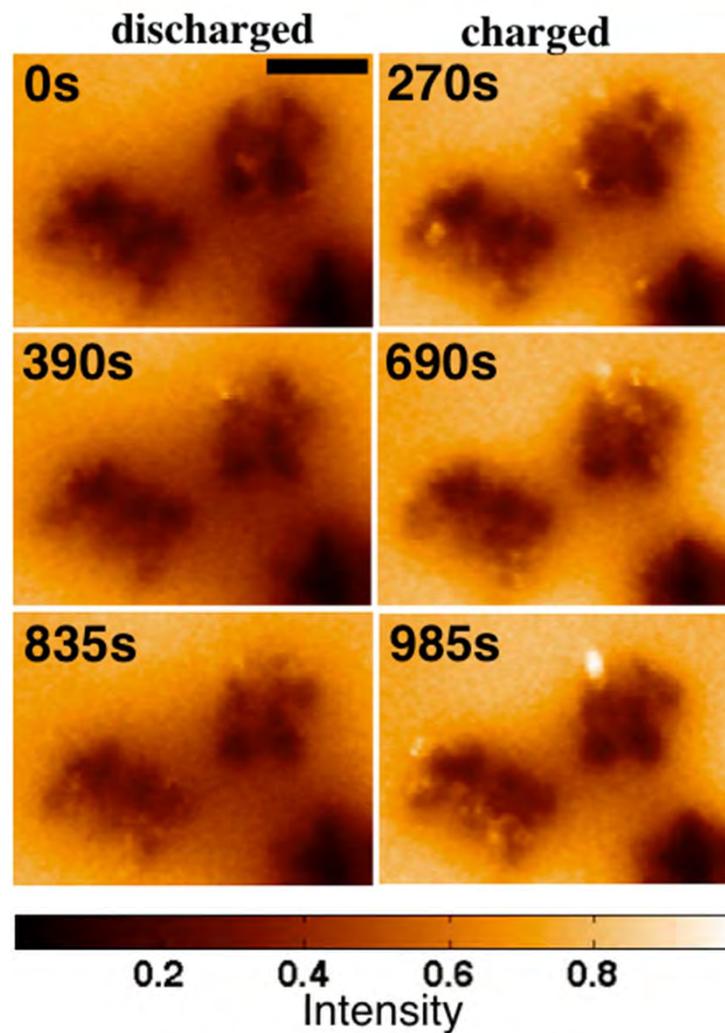
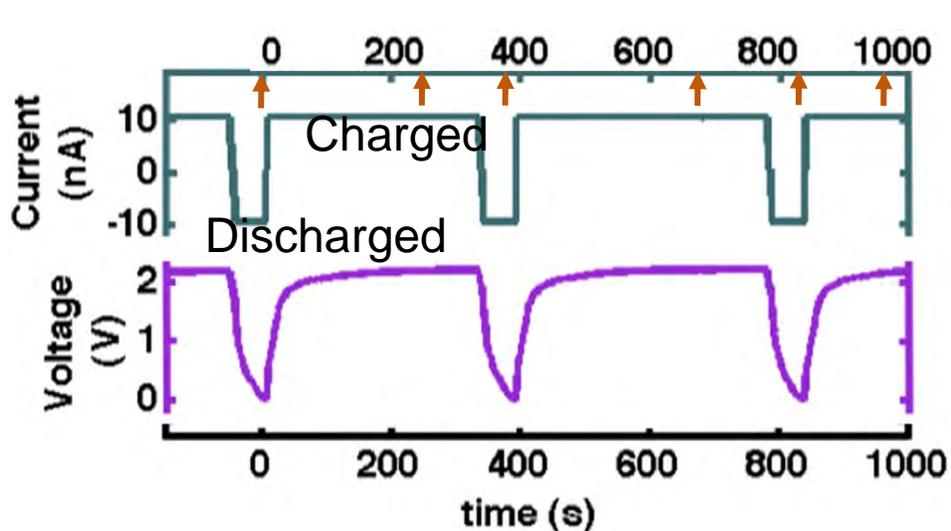


Transmission  
Electron Microscope (TEM)

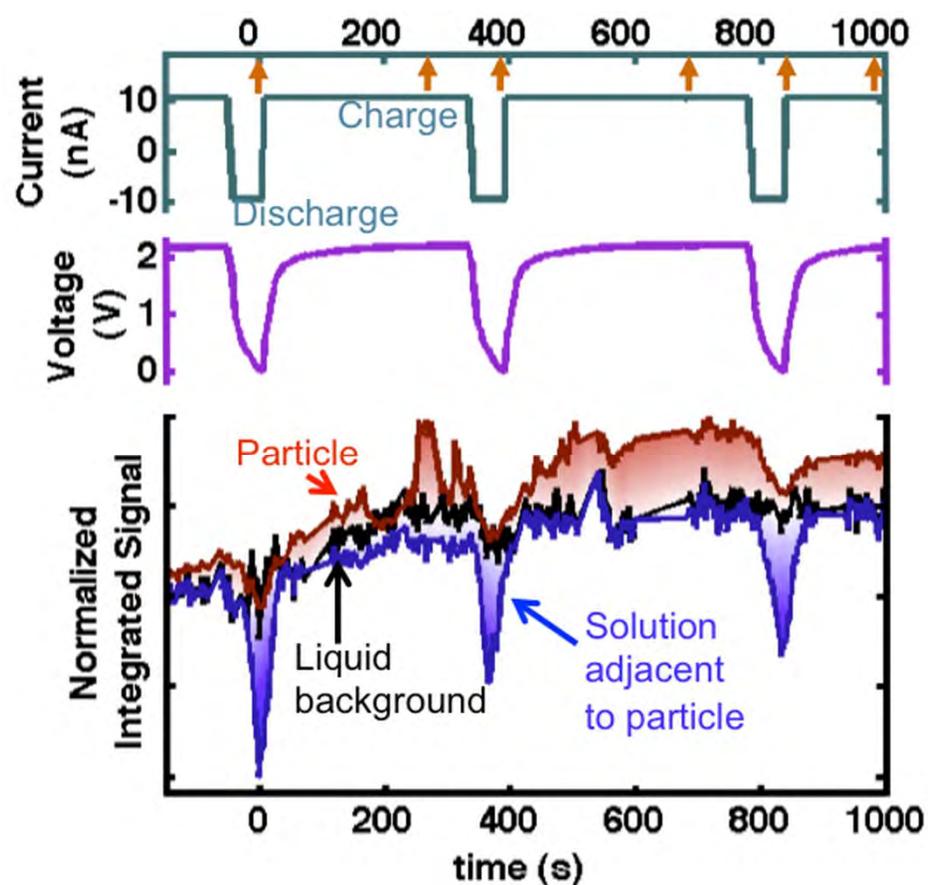


Want *rapid spectroscopic*  
*images of the 5 eV signal!*  
**EFTEM imaging!**

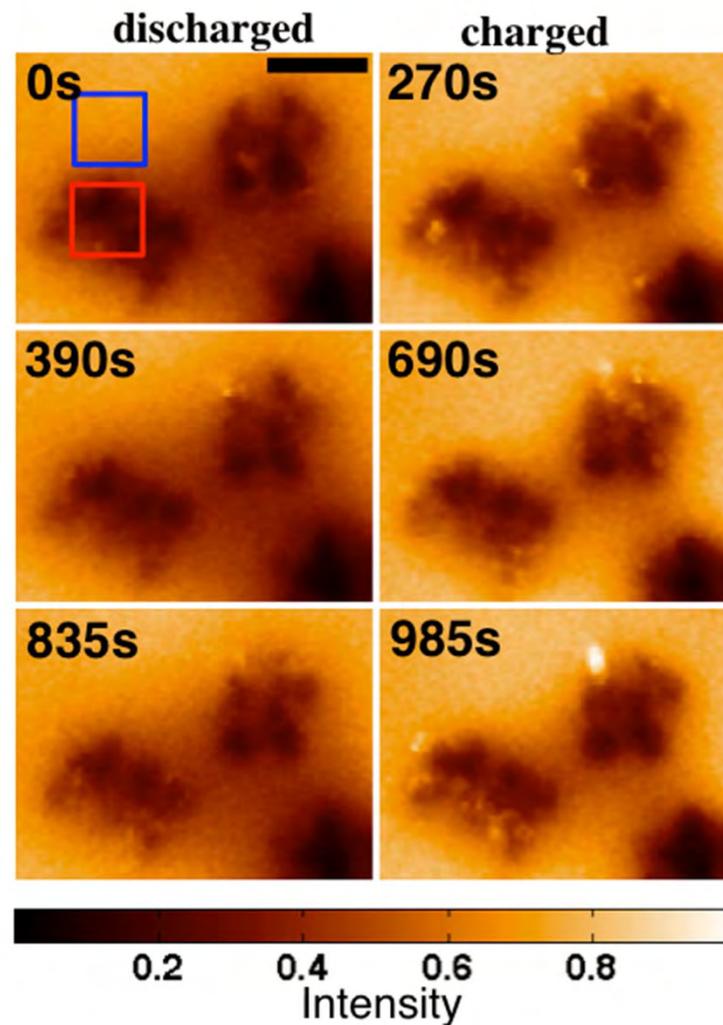
# 5 eV Images: Bright = FePO<sub>4</sub> Li Concentration Change during Cycling



# 5 eV Images: Bright = FePO<sub>4</sub> Li Concentration Change during Cycling

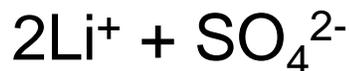
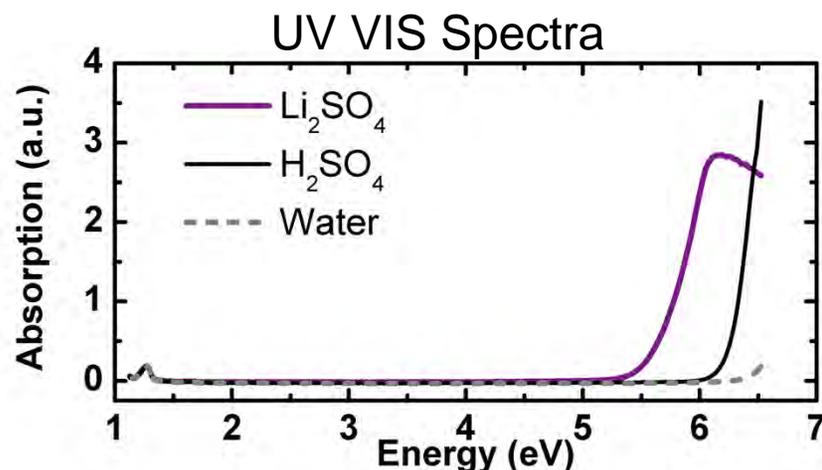
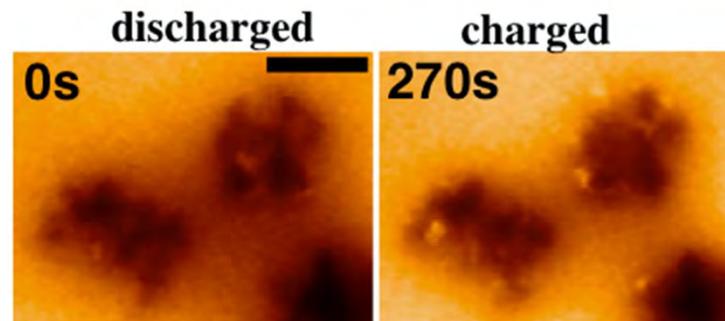
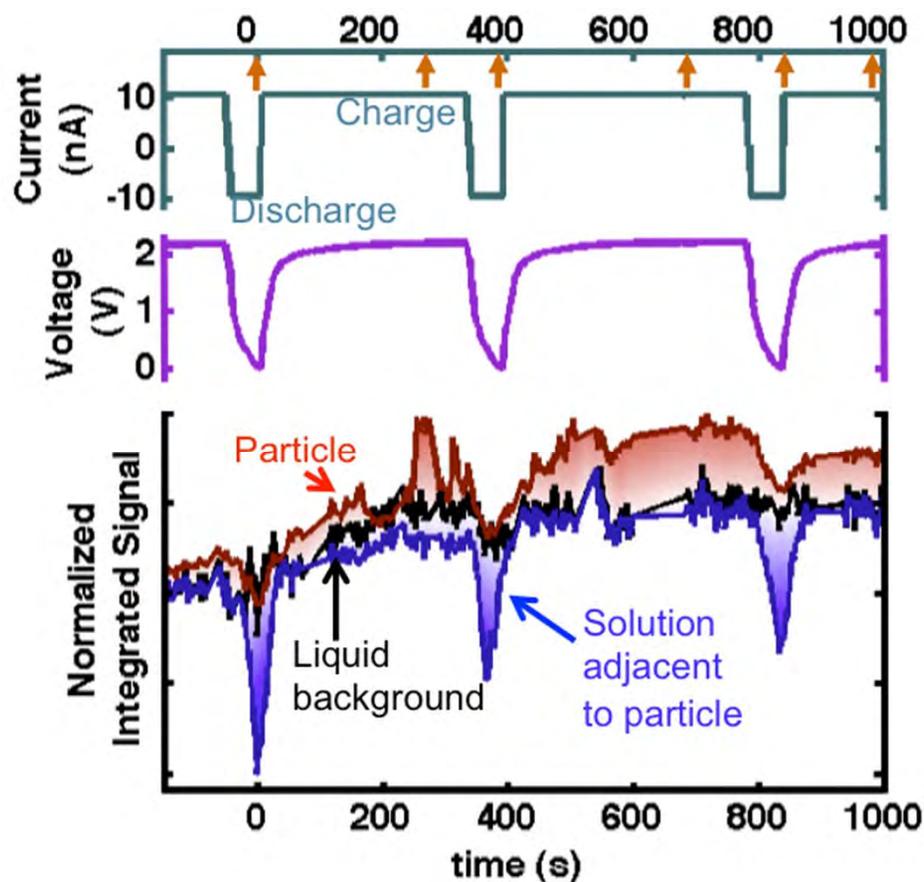


**Intensity from FePO<sub>4</sub> tracks charging!**



# 5 eV Images: Bright = FePO<sub>4</sub>

## Li Concentration Change during Cycling

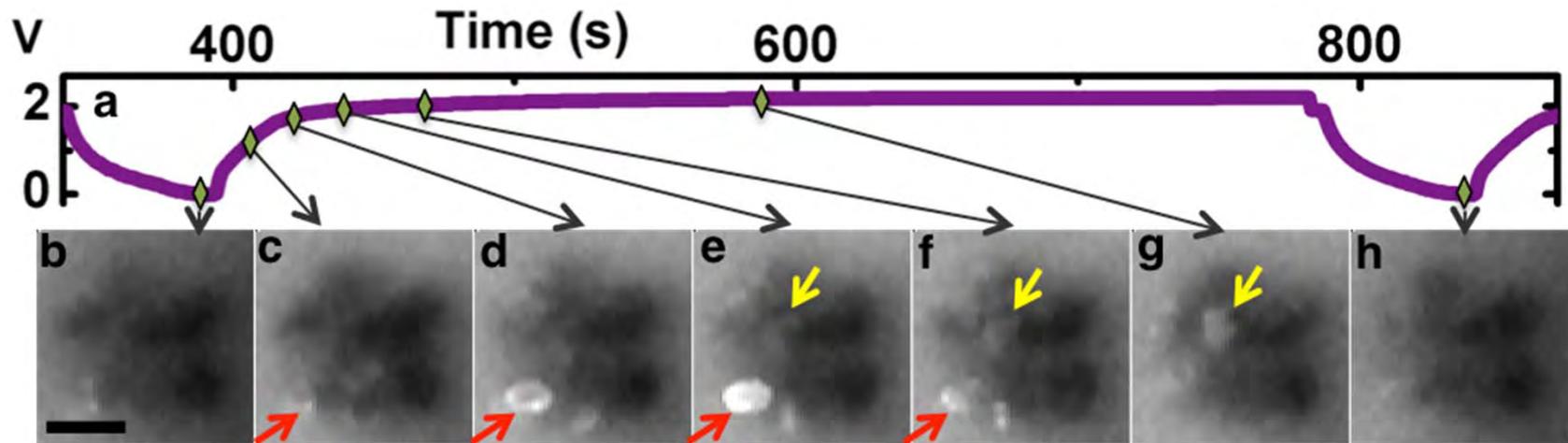


JDFT shows  $\text{LiSO}_4^-$  is the cause of bright 6 eV signal!

**Intensity from FePO<sub>4</sub> tracks charging!**

**Intensity in solution tracks  $\text{LiSO}_4^-$  follows discharging!**

# Mechanism of Lithiation



2 dominant modes of lithiation for this experiment:

We see

(red) core-shell type structures during growth of the FePO<sub>4</sub>

(yellow) Propagating from Left to Right

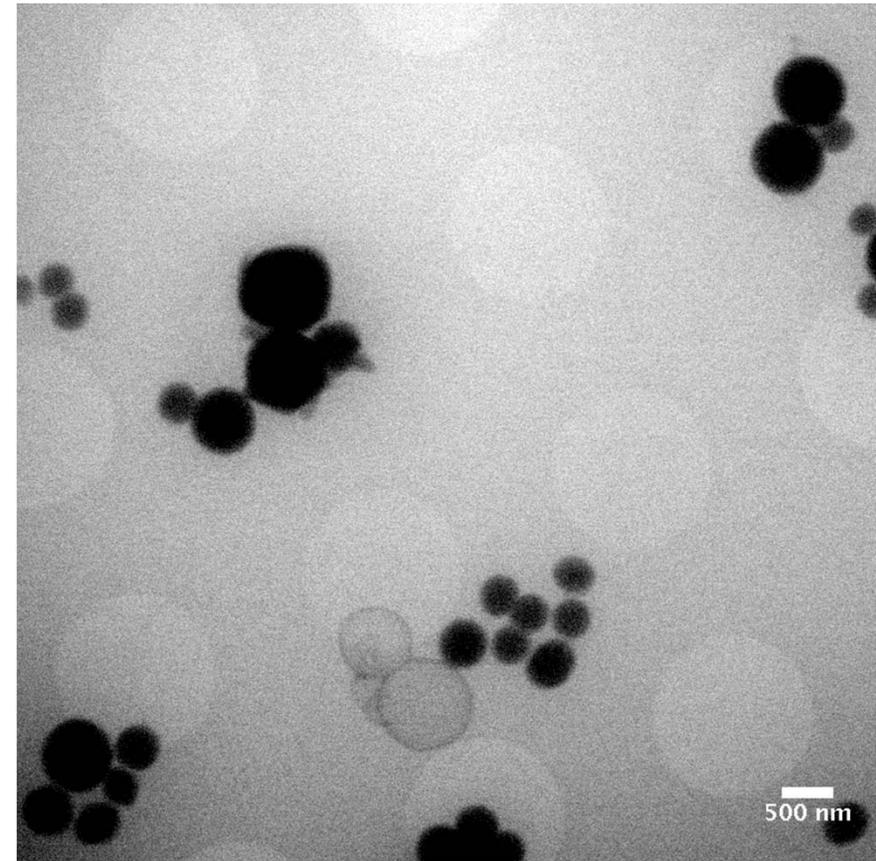
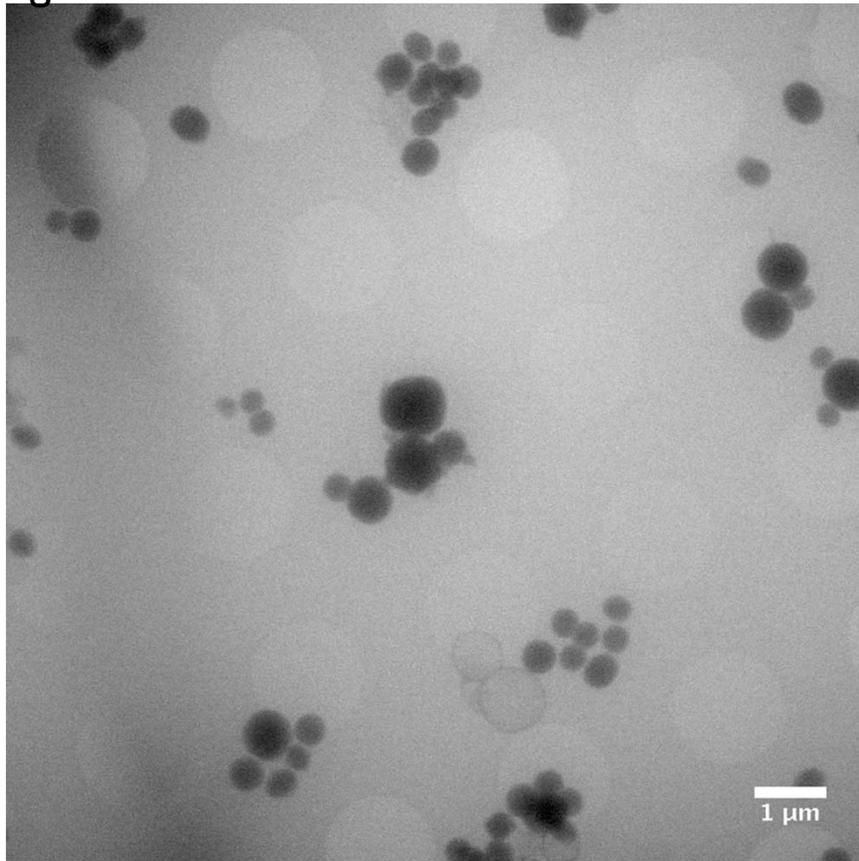
Highly dependent on particle size and experimental conditions



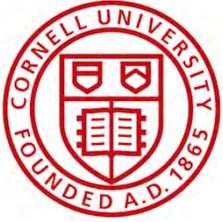
# AirSTEM Images of Sulfur in PAN

**Another way to image the encapsulated sulfur effectively is to make sure that there is no sublimation hence we image our samples in air instead of vacuum.**

Therefore, sample was then taken out of the F20 (Vacuum STEM) and imaged in the AirSEM using our homebuilt Air-STEM detector

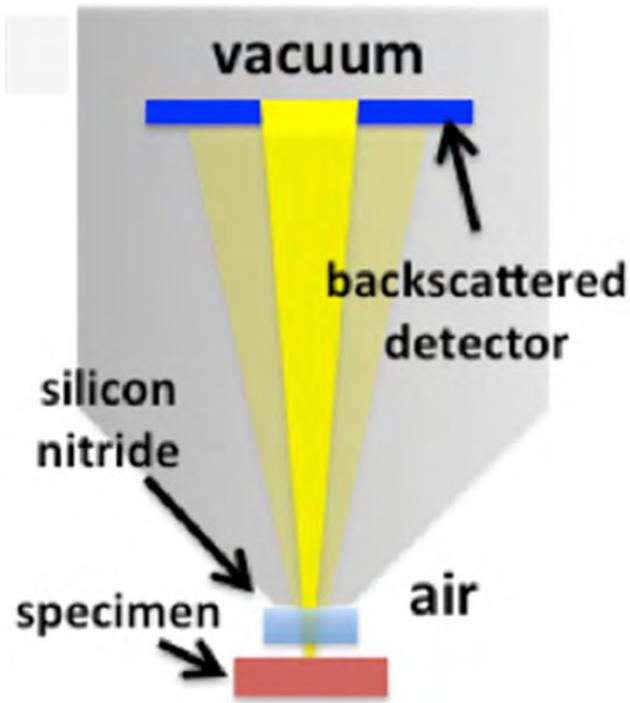


**As you can the resolution of the F20 and Air-STEM are comparable!**

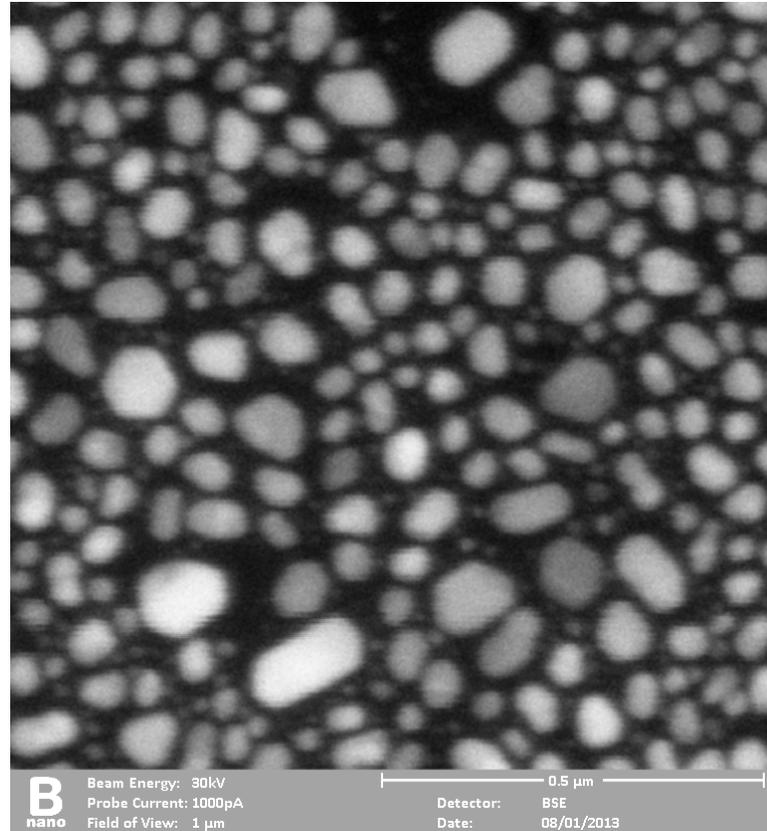


# airSEM: Scanning Electron Microscopy without a Vacuum Chamber

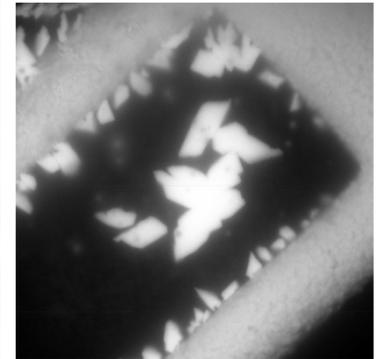
- Sample is in air, can be wet or insulating



A thin, electron transparent membrane separates the column from air

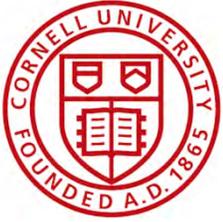


~ 8nm resolution on Au nanoparticles

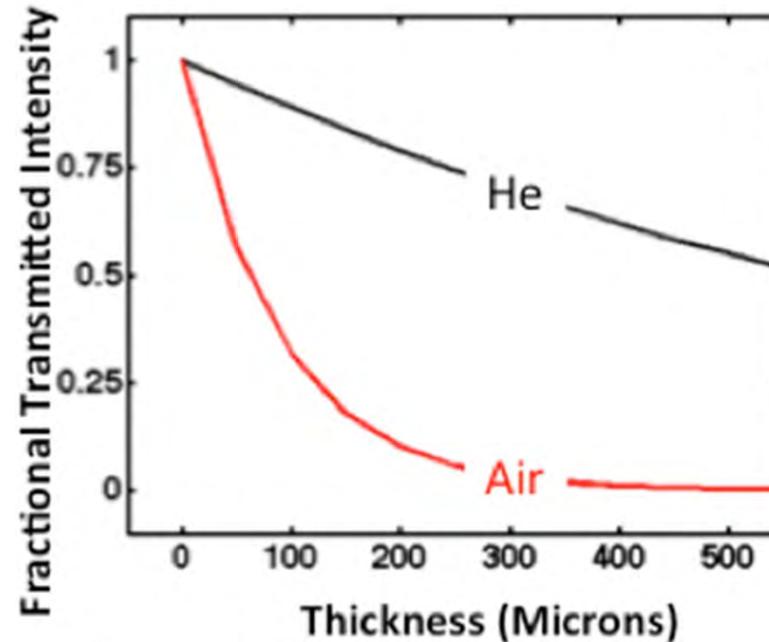


Watching Pb dendrites Grow in solution

Can image through 50-100 microns of air



## How Much Unscattered Beam is left?

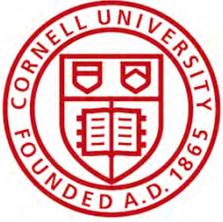


As long as some unscattered beam is left, we can form a high-resolution image

Mean free path in air is 150  $\mu\text{m}$  at 30 keV  $\Rightarrow$  30% loss at 50  $\mu\text{m}$

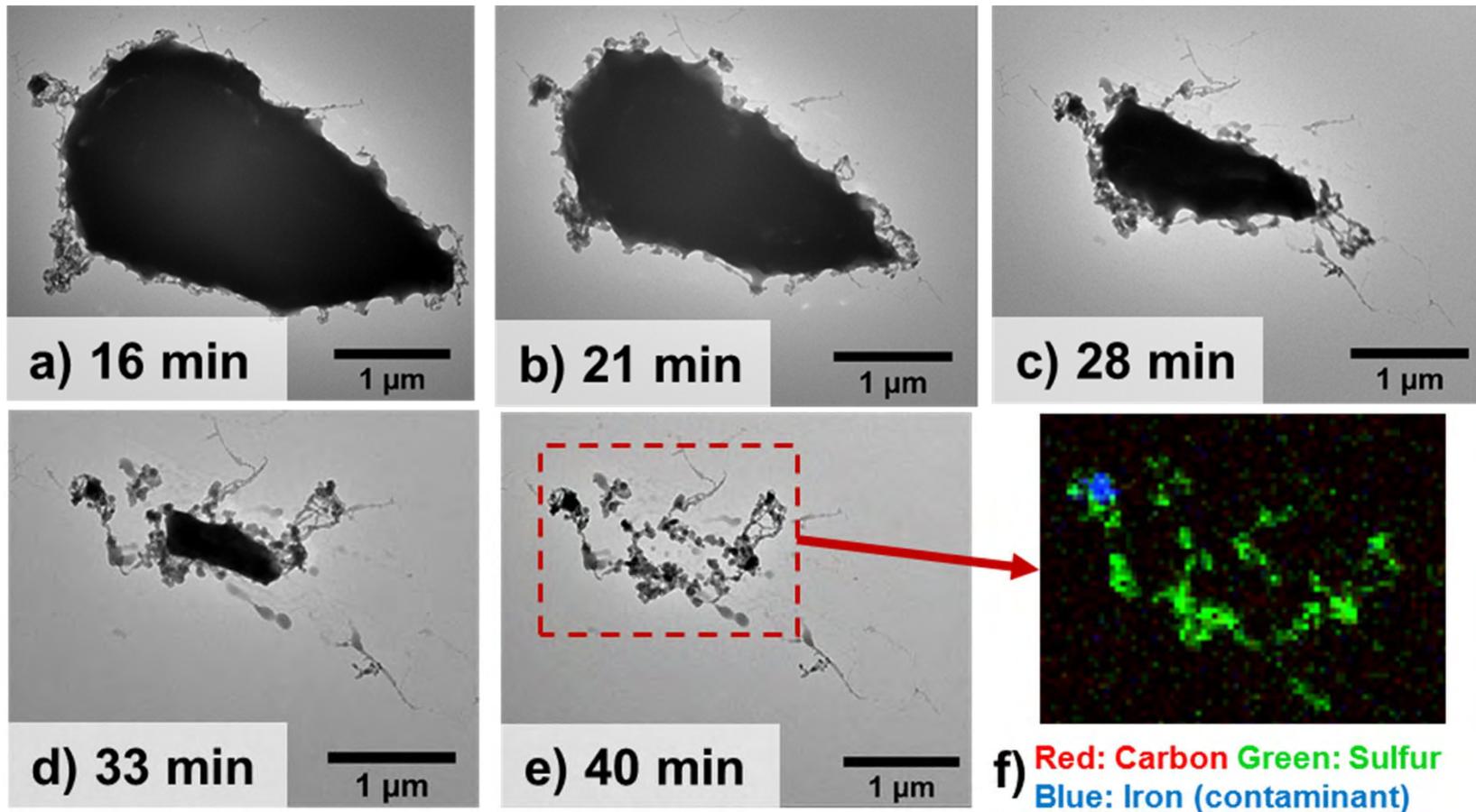
But mean free path in silicon nitride is 30 nm! (our membrane is 20 nm)  $\Rightarrow$  50% loss

**Solution: Graphene window - even 6 ML would give only a 5% loss**



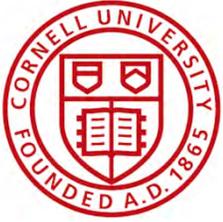
# Sulfur Sublimates at Room Temperature in the TEM

Sublimation rates is  $\sim 1$  ML/second (same as bulk)



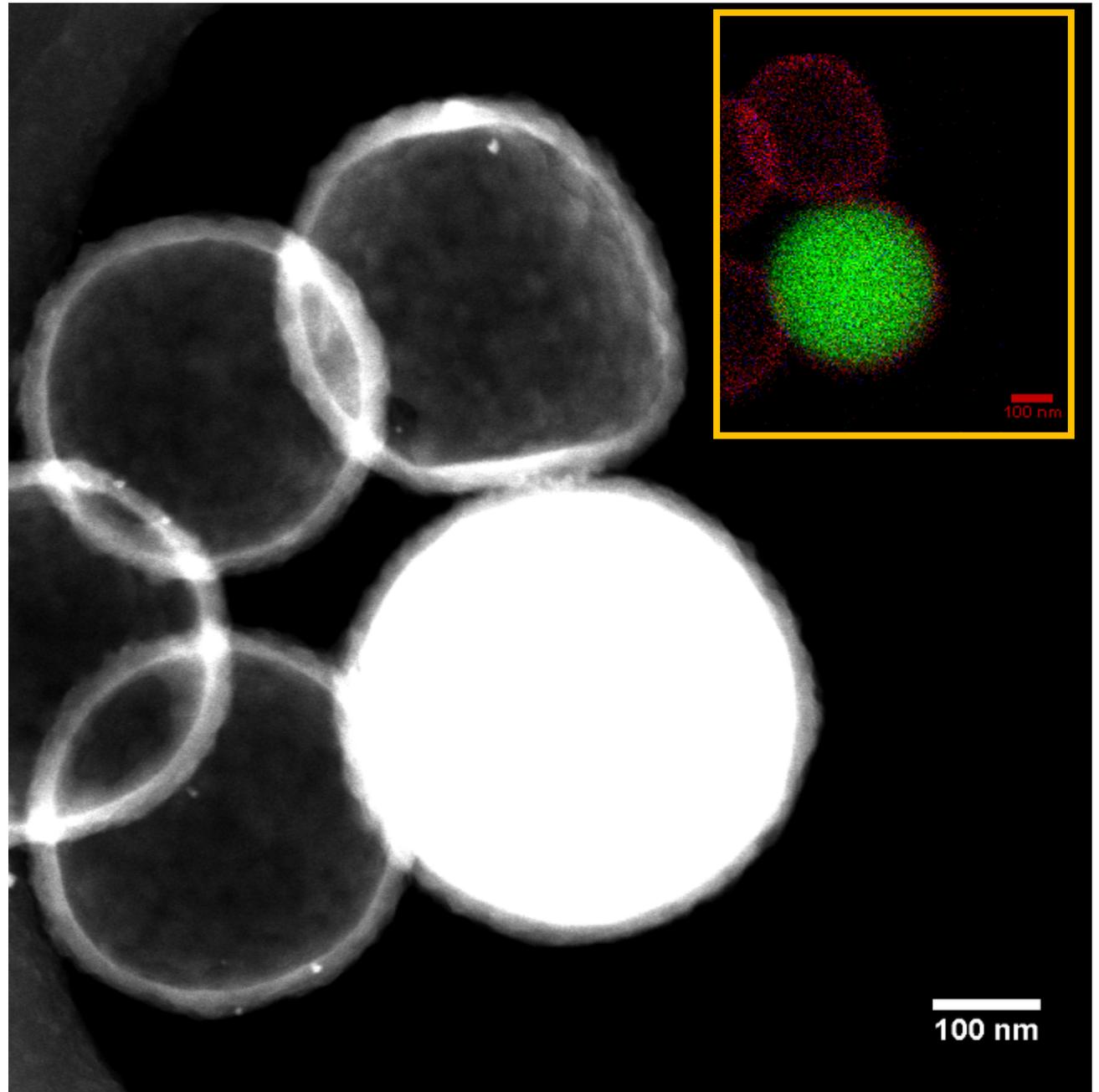
Remnant is “white sulfur” - amorphous & polymeric

T=18°C, P=  $8.8 \times 10^{-8}$  Torr, Beam off between images

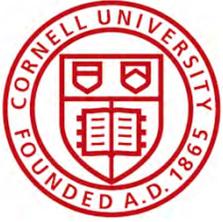


# Sulfur in PAN

Red = Carbon  
Green = Sulfur

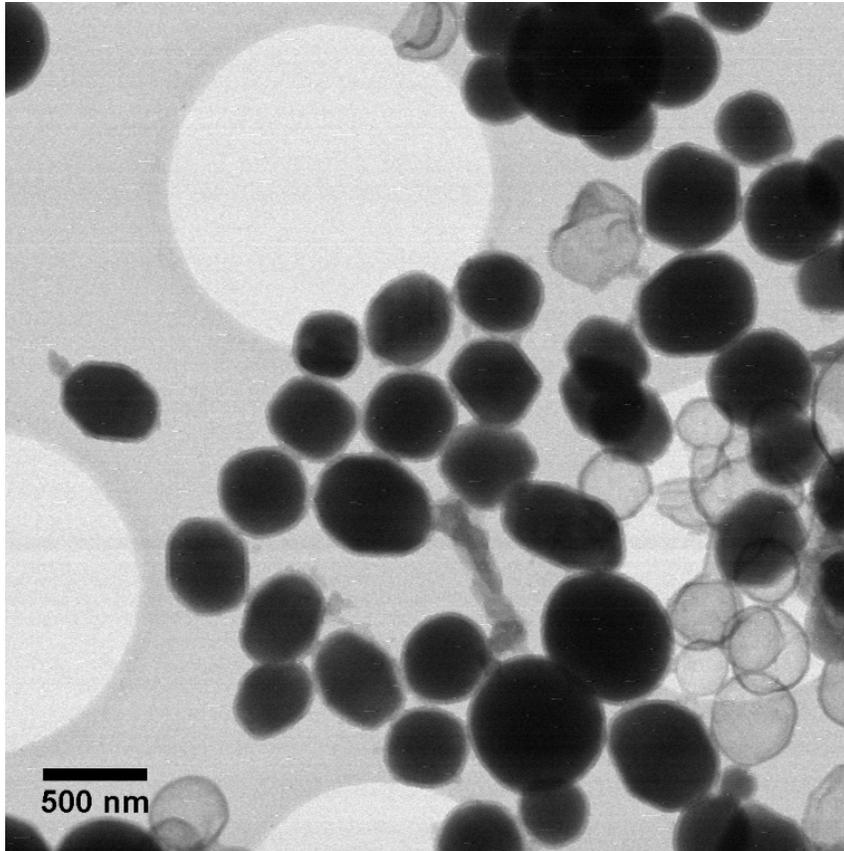


Images taken by B. Levin

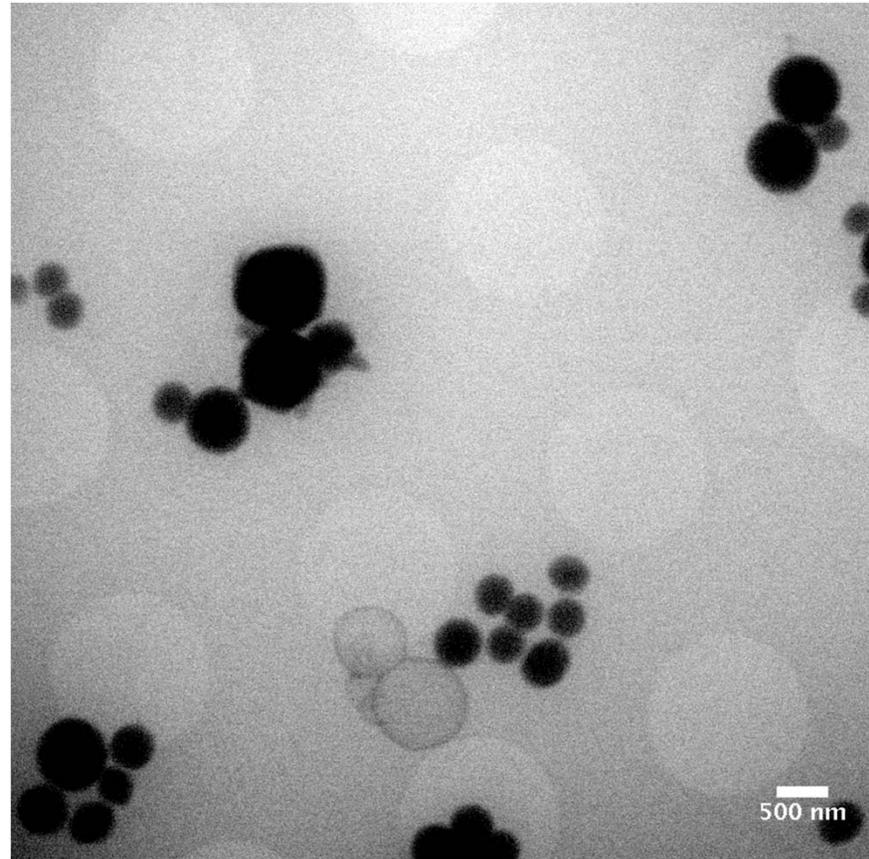


# Bright-Field STEM Images of Sulfur-filled PAN Shells

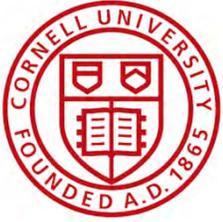
200 keV Tecnai F20



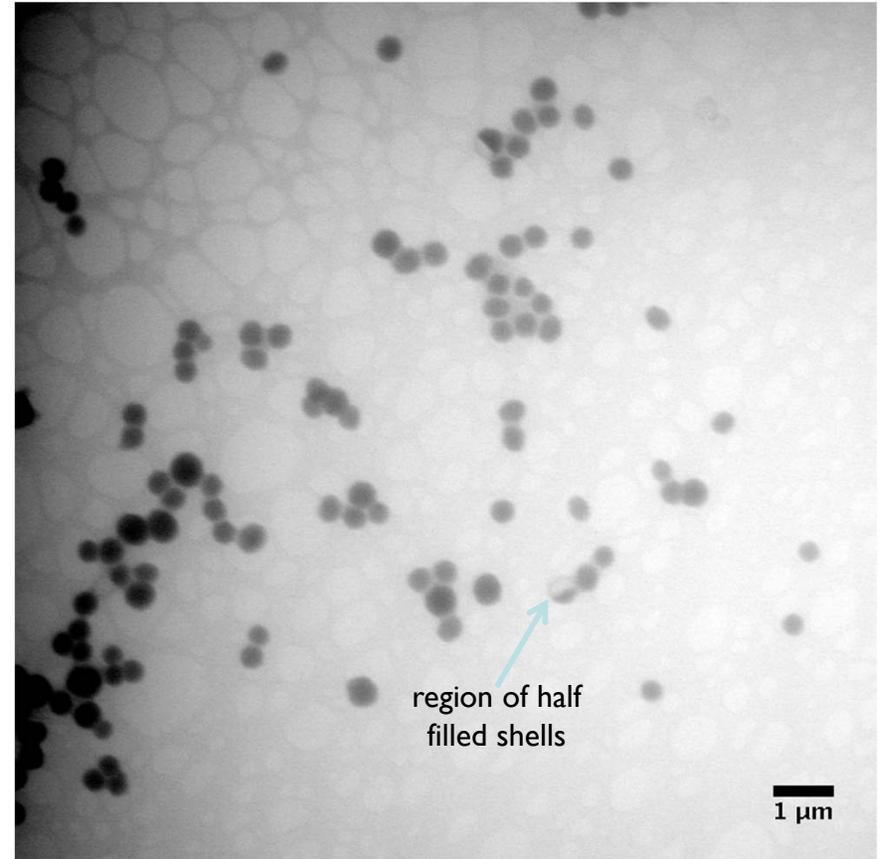
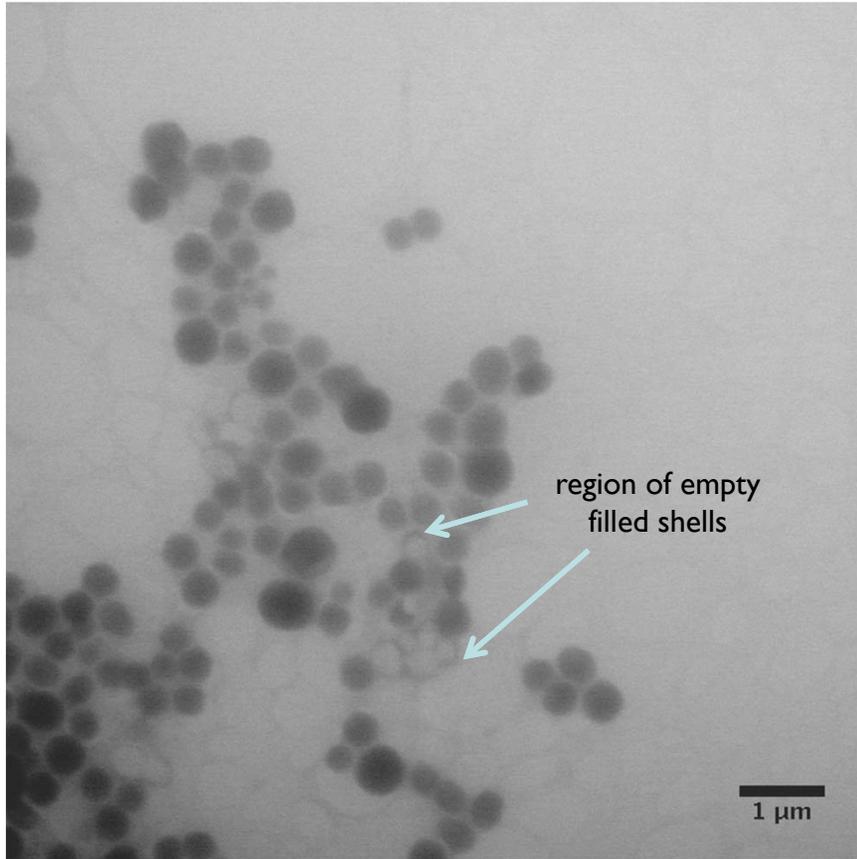
30 keV bNano AirSTEM

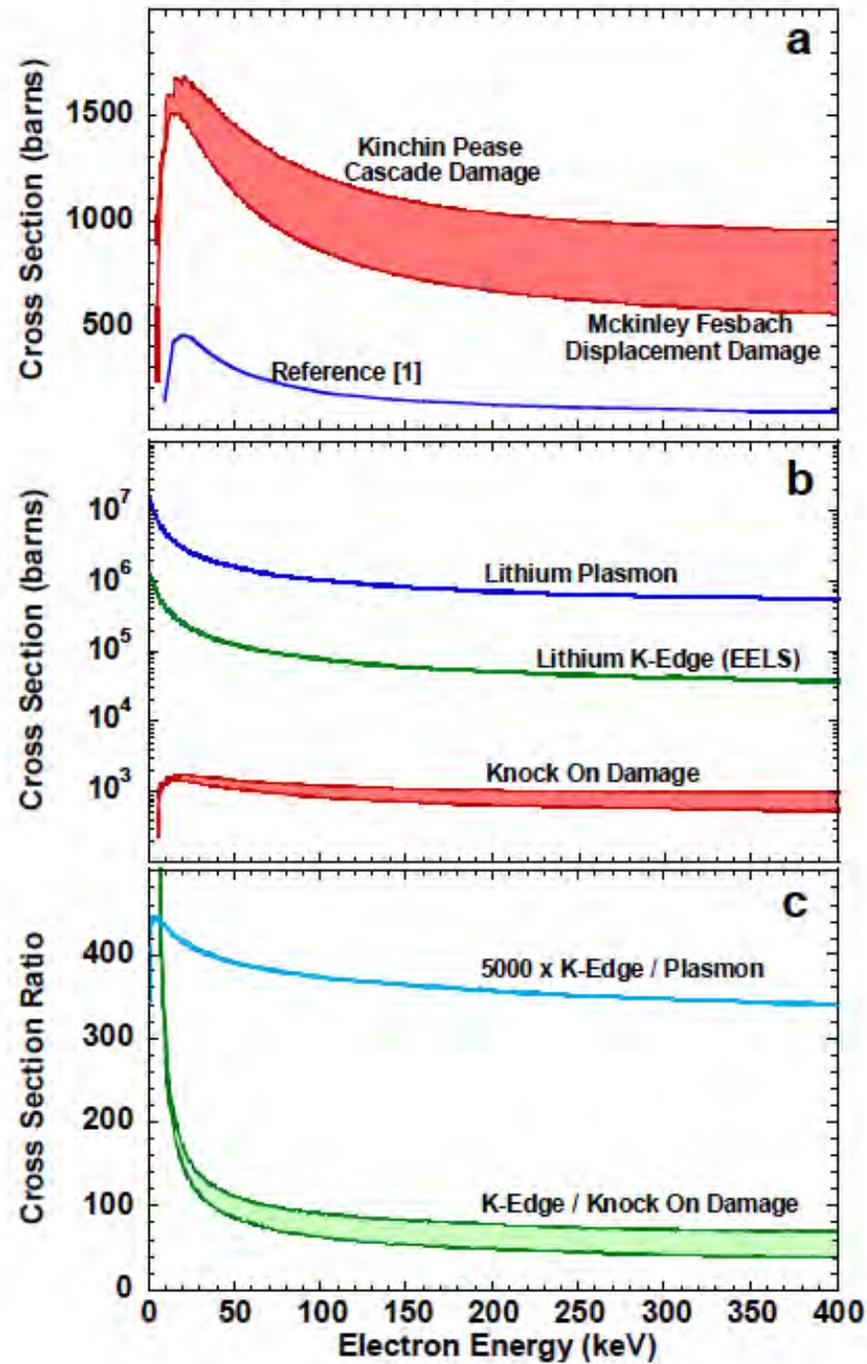
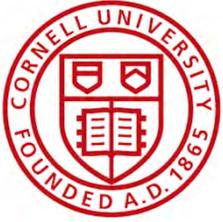


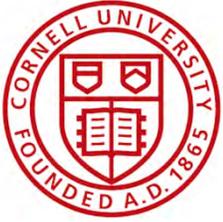
Sulfur sublimates in vacuum – leaving empty shells if there were large pores



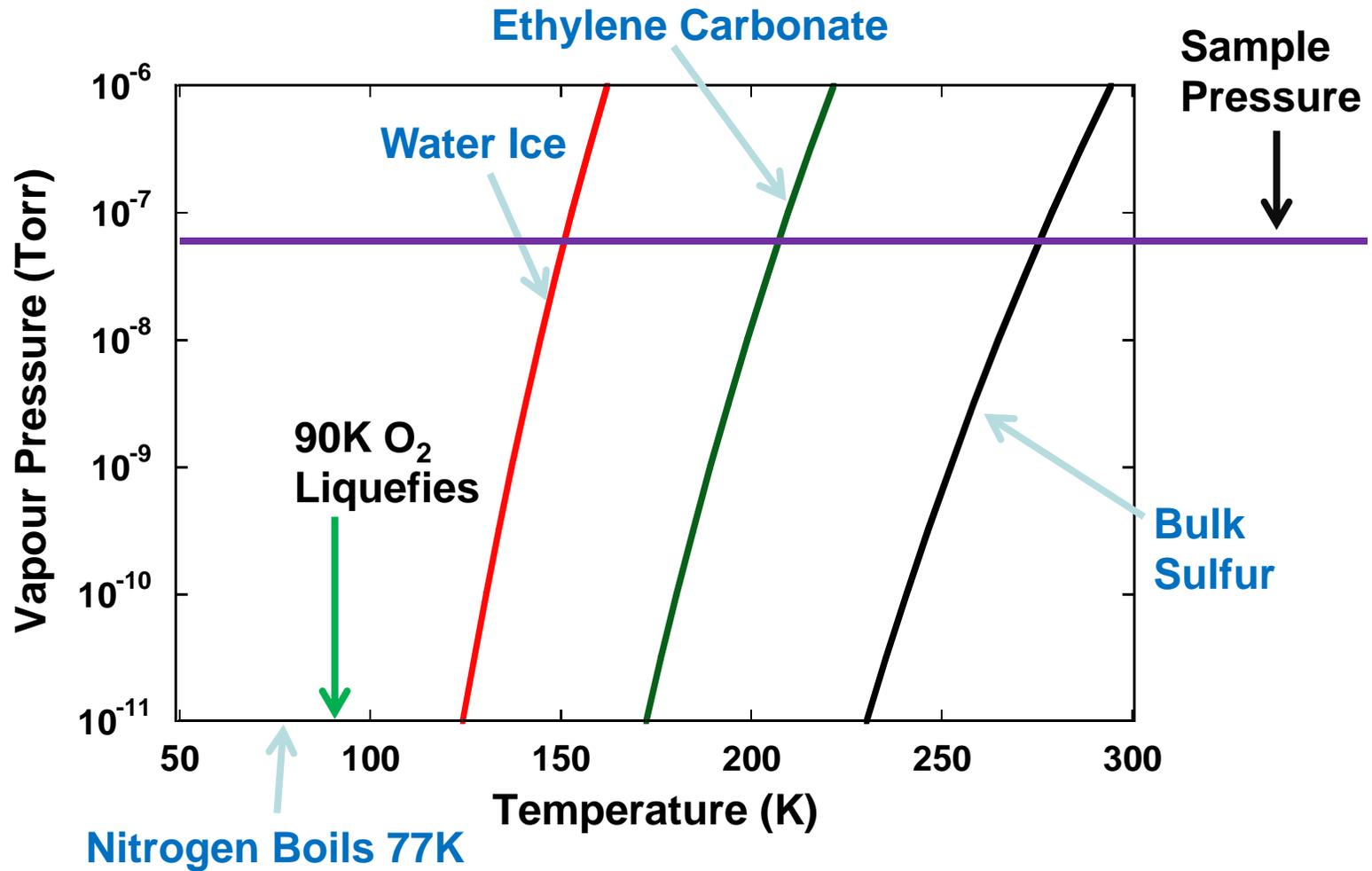
# Fresh sample: AirSTEM: Most shells filled, But a few empty shells and (new!) half-filled shells

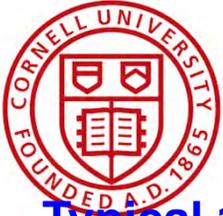






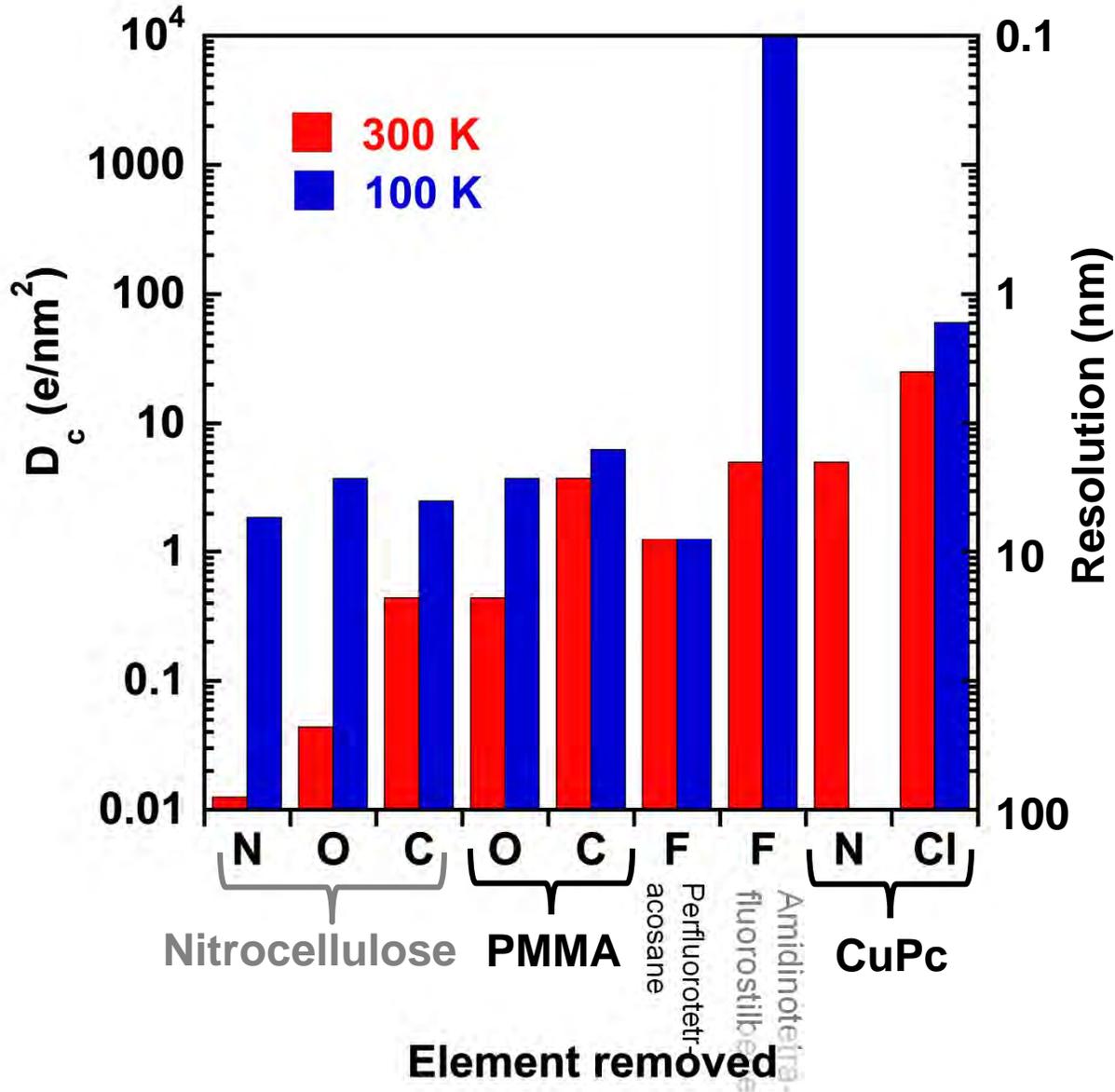
# Why do we want a cryo-holder?

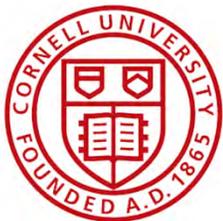




# Critical Dose for Mass Loss

Typical 10-50 x decrease in mass loss on cooling -> 3-7x resolution improvement

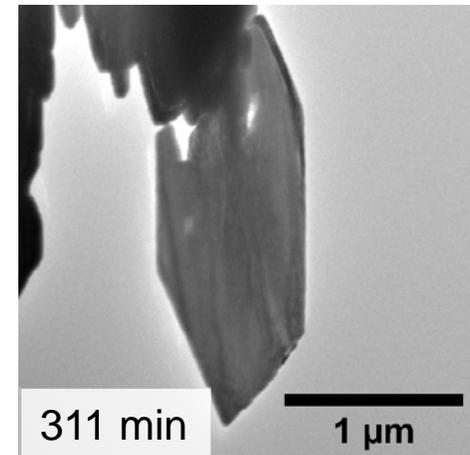
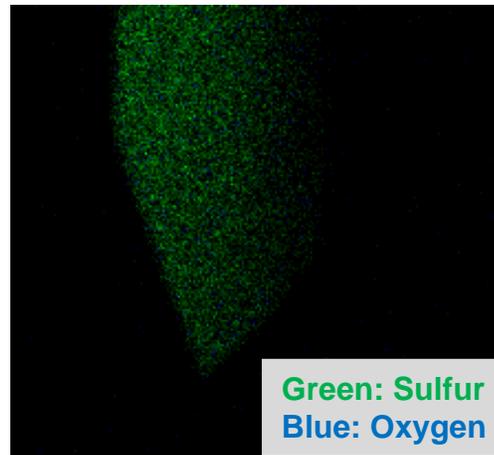
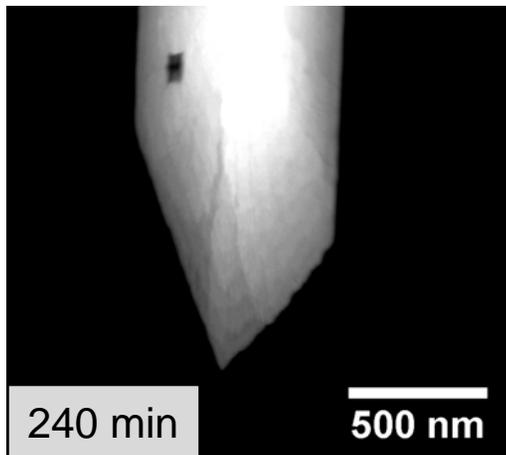
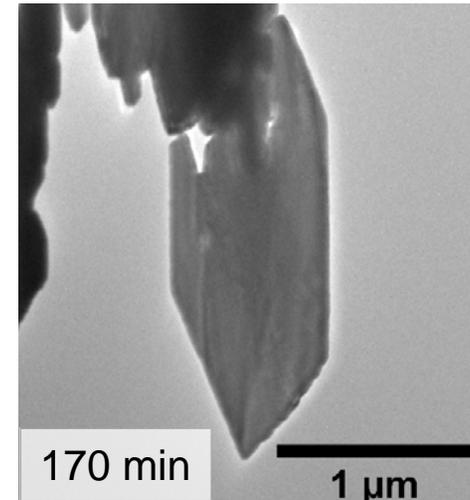
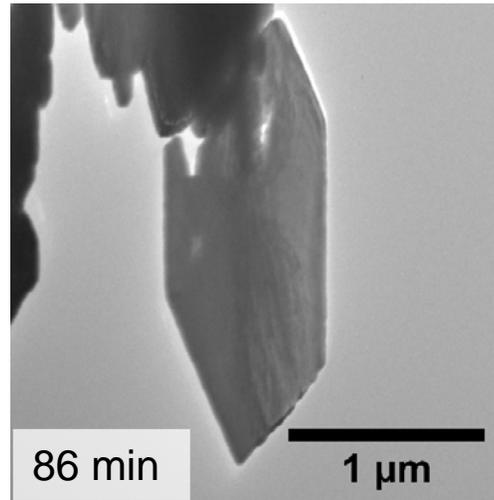
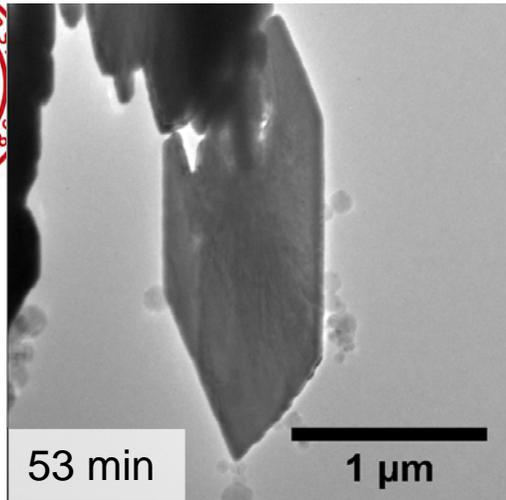




# Sensitivity to Radiation Damage

**G = # of Reactions/100 eV of *deposited* energy**

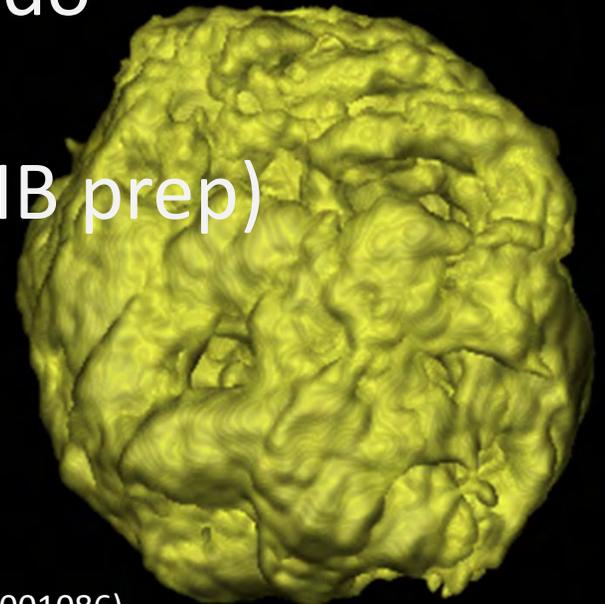
Compounds and sites of attack	$G_{-M}$ (loss)	$G_{H_2}$	$G_{CO_2}$	$G_{CO}$	$G_{H_2O}$	$G_{CH_4}$
<b>Hydrocarbons,</b>						
saturated						
C-H, C-C	6-9	3.8-5.6				0.2-0.7
unsaturated						
C-H, C-C	11-10	0.8-1.2				0.13
aromatic						
C-H, side chain C-C	0.2-1	0.01-0.18				
<b>Alcohols</b>						
HC-OH, C-COH	3-6	3.5-4.5		0.04-0.23	0.3-0.9	
<b>Ethers</b>						
C-H, C-OR	7	2.0-3.6		0.06-0.13		
<b>Aldehydes and ketones</b>						
C-H, C-C=O	7	0.8-1.2		0.6-1.6		0.1-2.6
<b>Esters</b>						
C-H, O=C-OR	4	0.5-0.9	0.3-1.6	0.15-1.6		0.4-2.0
<b>Carboxylic acids</b>						
C-H, C-COOH	5	0.5-2.3	0.5-4.0	0.1-0.5	0.1-2.2	0.5-1.4
<b>Amino acids</b>						
	$G_{-M}$	$G_{H_2}$	$G_{CO_2}$	$G_{NH_3}$		
<b>Leucine</b>						
C-H, C-NH <sub>2</sub> , C-COOH	14	0.5	2.8	5.1		
<b>Valine</b>						
C-H, C-NH <sub>2</sub> , C-COOH	8	0.2	0.6	4.1		

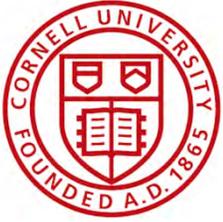


Time series of images on Sulfur particle including HAADF and EDX to identify Sulfur. Temperature  $-175^{\circ}\text{C}$ , pressure  $8.83 \times 10^{-8}$  Torr. Particle is clearly unchanged by time in vacuum, in contrast to Sulfur at room temperature. Ice contamination in the first image burned away under the beam within  $\sim 2$  min. Very little ice seen in EDX.

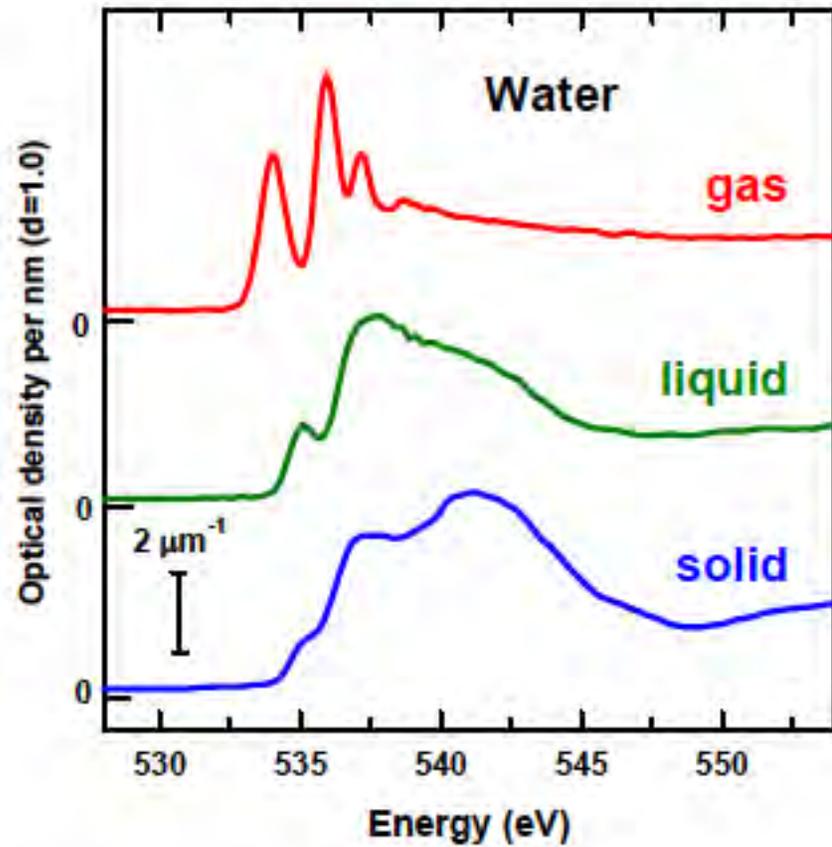
# Capabilities

- Atomic-scale composition and bonding
- 3D Imaging at the nanoscale
- *In-situ* liquid flow, electrochemical cycling to observe reactions at TV rate
- Air-SEM for larger-scale in-operando experiments
- Cryo-electron microscopy (+cryoFIB prep)
  - Organic/inorganic interfaces
  - Liquid/Solid interfaces

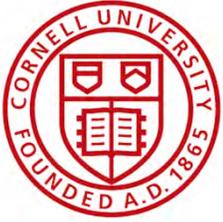




# Oxygen K-Edge of $H_2O$



XAS Data from Adam Hitchcock



# New Detectors in Cryo-EM:



## THE REVOLUTION WILL NOT BE CRYSTALLIZED

**MOVE OVER X-RAY CRYSTALLOGRAPHY.  
CRYO-ELECTRON MICROSCOPY IS  
KICKING UP A STORM IN STRUCTURAL**

In a basement room, deep in the bowels of a steel-clad building in Cambridge, a major insurgency is under way. A hulking metal box, some three metres tall, is quietly beaming terabytes' worth of data through thick orange cables that disappear off through the ceiling. It is one of the world's most advanced cryo-electron microscopes: a device that uses electron beams to photograph frozen biological molecules and lay bare their molecular shapes. The microscope is so sensitive that a shout can ruin an experiment, says

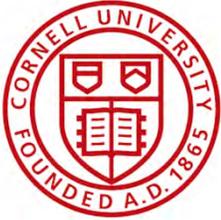
NATURE **525** (2015) 172

'ribosomania'

"X-ray crystallography has largely fallen by the wayside in the laboratory of Venki Ramakrishnan, who shared the 2009 Nobel. For large molecules, "it's safe to predict that cryo-EM will largely supersede crystallography", he says."

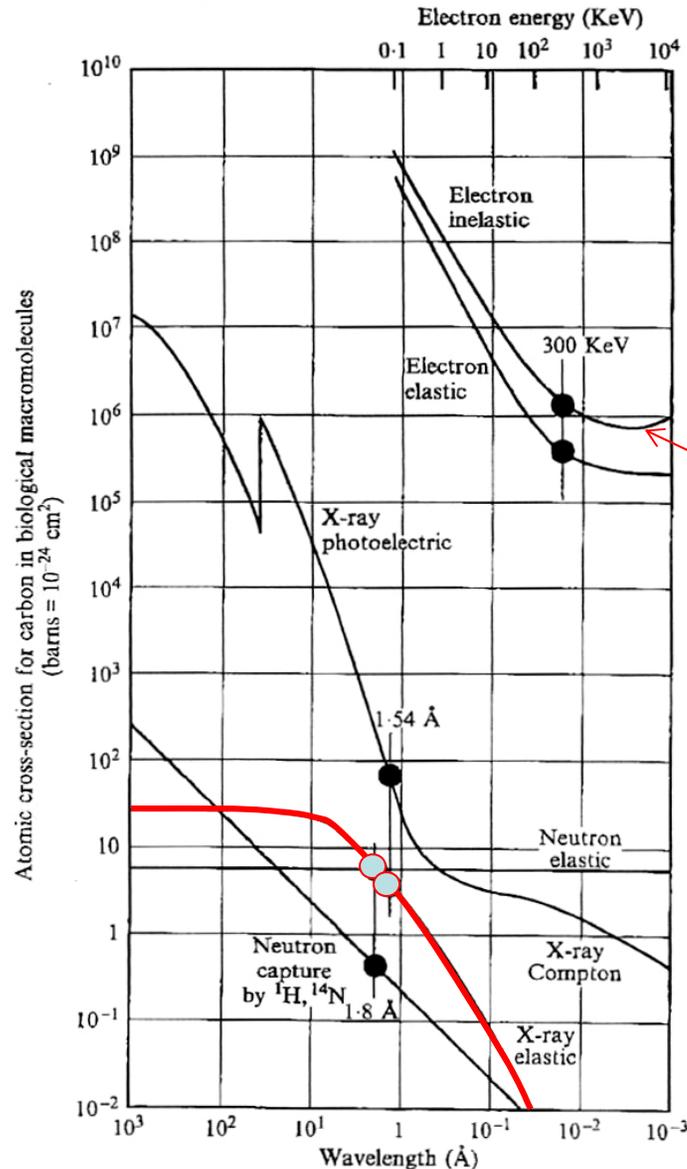
The Nobel Prize in Chemistry 2017 was awarded jointly to Jacques Dubochet, Joachim Frank and Richard Henderson

"for developing cryo-electron microscopy for the high-resolution structure determination of biomolecules in solution."



# How Bad is Radiation Damage?

R. Henderson, Quarterly Reviews of Biophysics 28 (1995) 171-193.



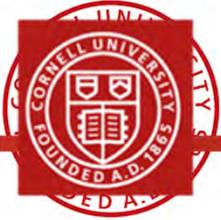
It's not the cross-section, but

How many damaging events per useful imaging event?

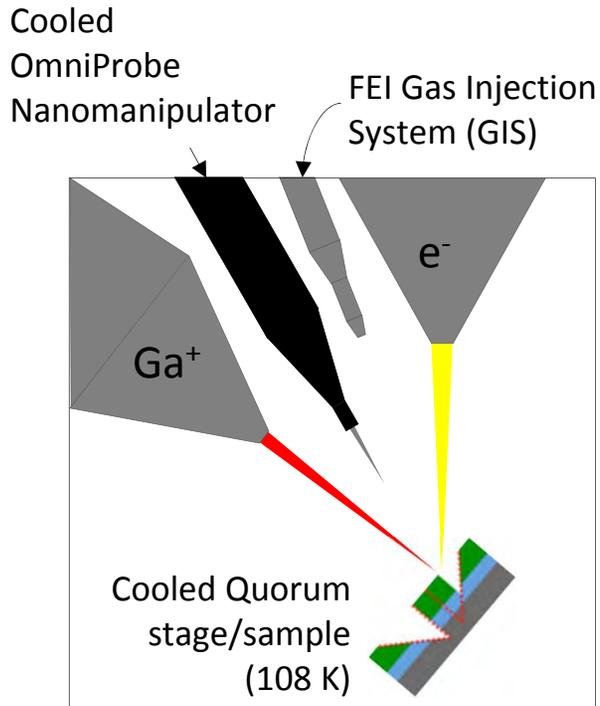
Least Damage:

Elastic imaging - Electrons wins

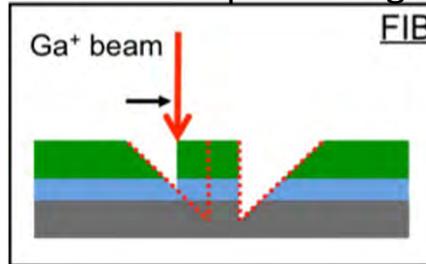
Inelastic imaging - Soft X-rays win



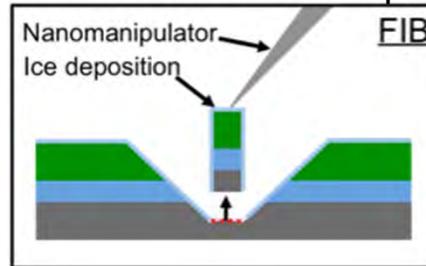
# Cryo-FIB lift-out



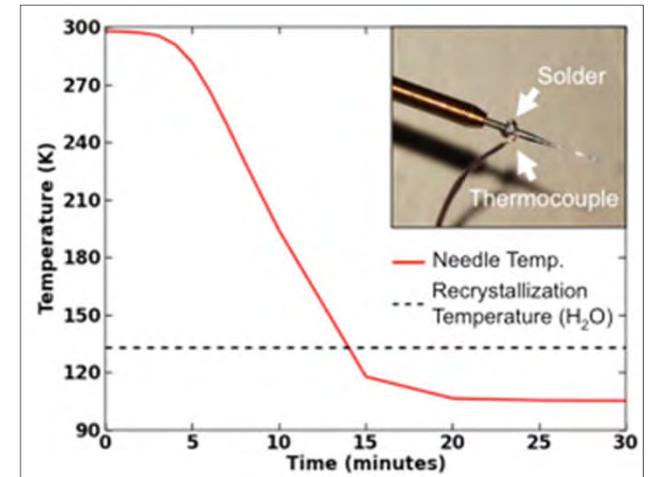
## Frozen Sample Milling:



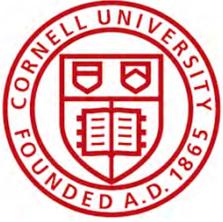
## Lift-Out with Water Vapor:



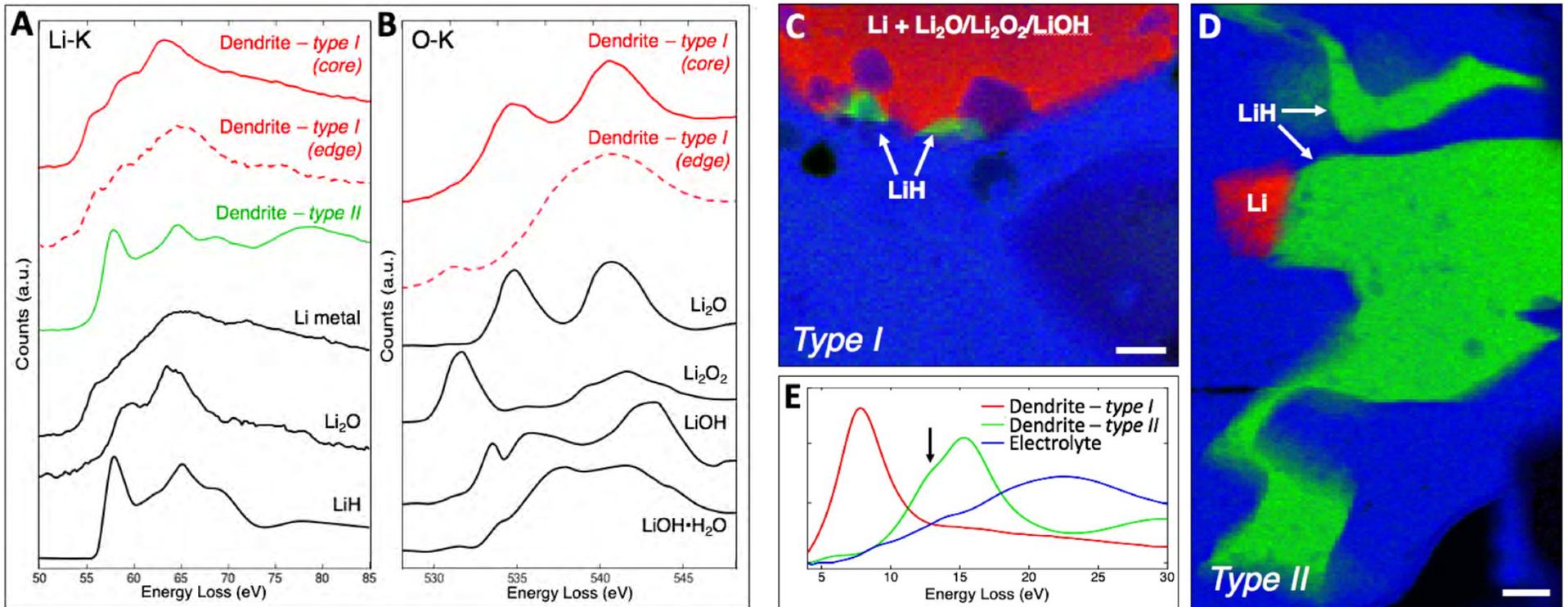
## Needle Cooling:



Use water vapor instead of Pt for attachment.  
Cooled nanomanipulator maintains the sample temperature during lift-out.



# Two structurally and chemically distinct dendrite structures

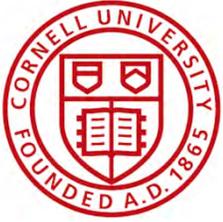


## Type I dendrite:

- lithium metal, partially oxidized at its core
- transitioning to peroxide and hydroxides near the surface

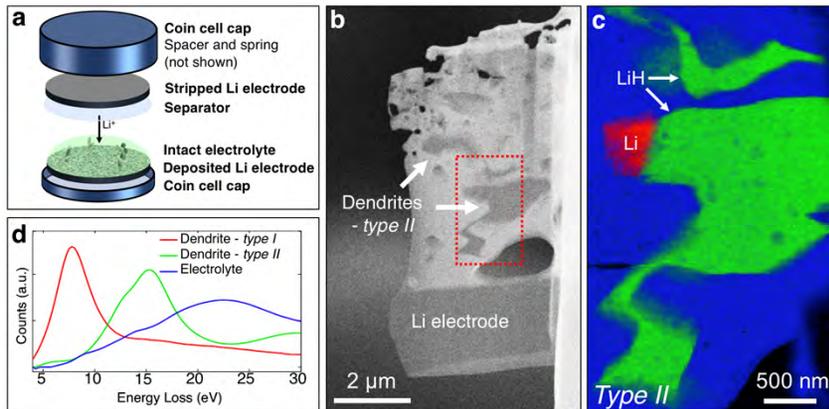
## Type II dendrite:

- uniform lithium hydride

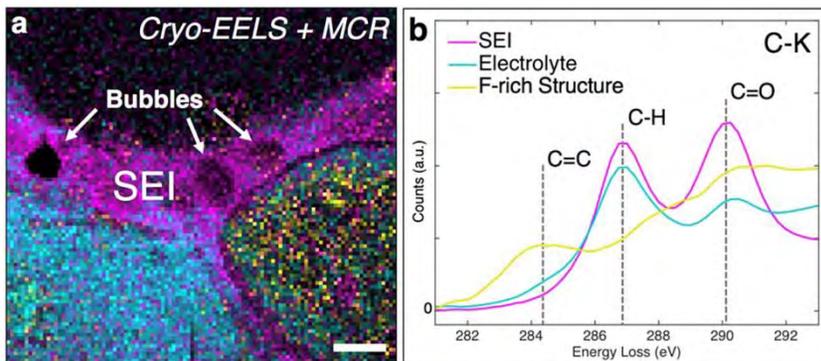


# Cryo-STEM for characterization of liquid/solid interfaces

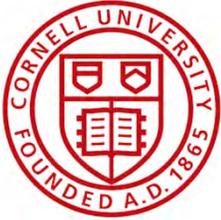
## Nanoscale mapping of dendrites in Li metal coin cell battery



## Spatial variations of carbon bonding

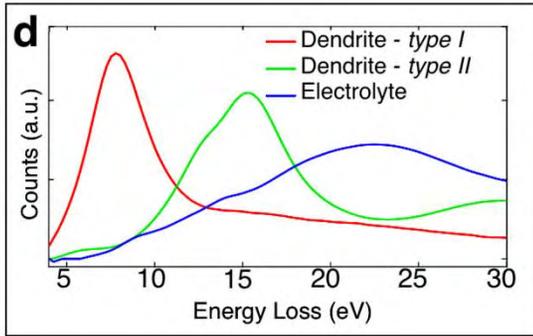
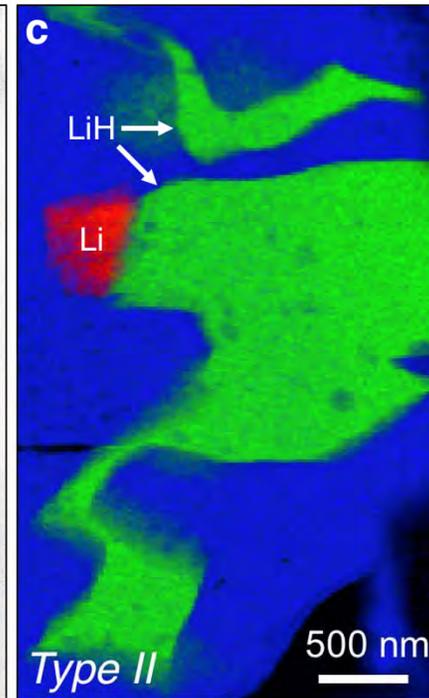
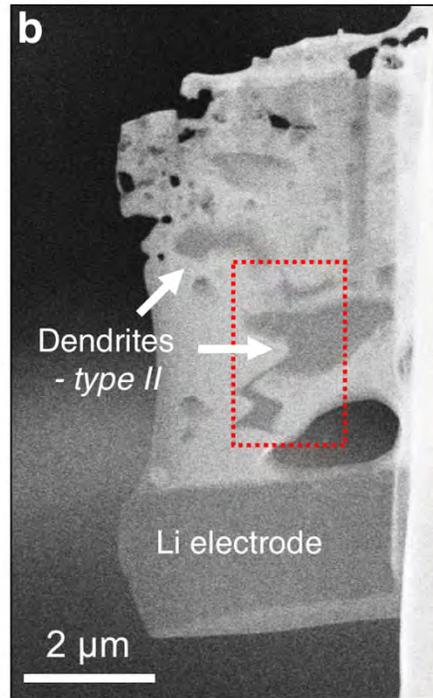
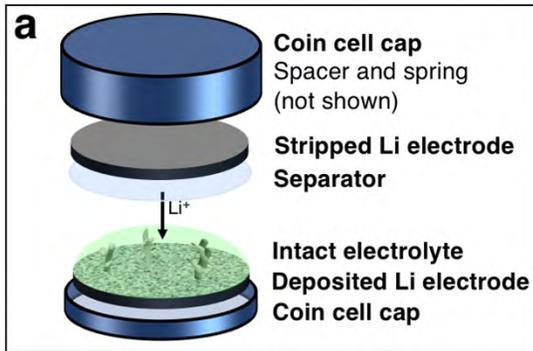


- High-resolution characterization of structure, chemistry and bonding of materials and devices by electron microscopy and spectroscopy at both room and cryogenic temperatures
- Demonstrated an approach to access buried liquid-solid interfaces by cryo-FIB and cryo-STEM/EELS
- Application of these techniques to image the structure and chemistry of reactive or beam sensitive materials and of processes at liquid/solid interfaces such as those in Li-metal coin cell batteries

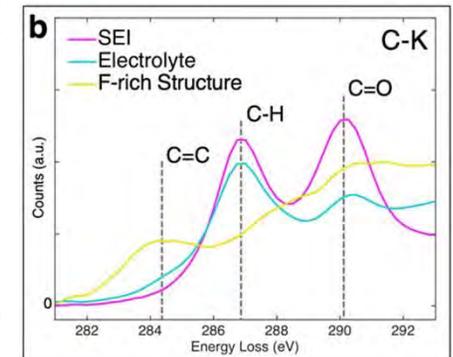
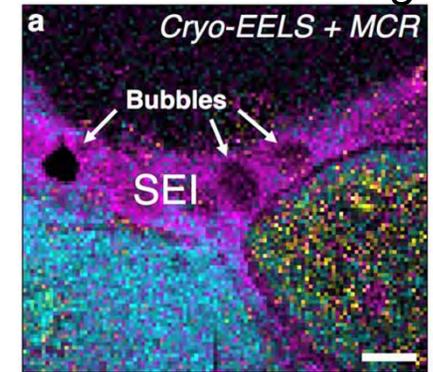


# Cryo-STEM for characterization of liquid/solid interfaces

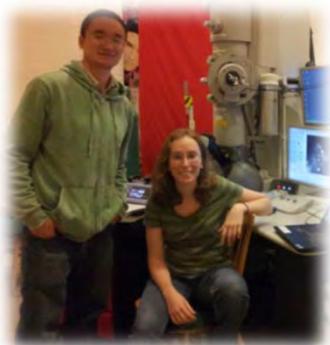
## dendrites in Li metal coin cell battery



## Spatial variation of carbon bonding



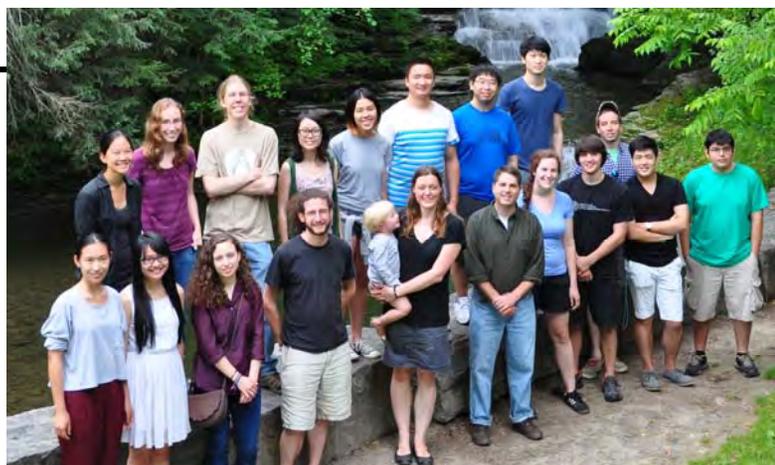
# Acknowledgements



*Abruña Group*

**Yingchao Yu**

Jie Gao  
Deli Wang  
Nikky Ritzert  
Jimmy John  
Eric Rus



*Muller Group*

**Megan Holtz**

Kayla Nguyen  
Barnaby Levin  
Elliot Padgett  
Pinshane Huang  
Paul Cueva  
Qingyun Mao  
Nina Andrejevic  
Celesta Chang

Huolin Xin\*  
Robert Hovden\*  
Julia Mundy\*

*Kourkoutis Group*

Michael Zachman  
Katy Spoth



*General Motors*

Z. Vic Liu  
J. Zhang  
Nalini Subramanian  
Rohit Makharia  
Fred T. Wagner

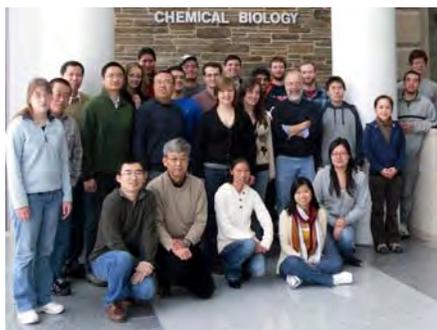


*Arias Group*

**Deniz Gunceler**  
**Shankar Sundararaman**  
Katie Schwartz  
Kendra Letchworth-Weave

*Corning*

Weirong Jiang



*Microscopes*

John Grazul  
Mick Thomas  
Earl Kirkland  
**NSF MRSEC**



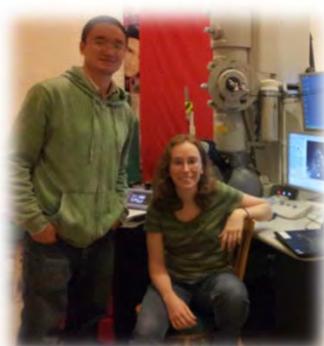


## David Muller

**Megan Holtz**

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Paul Cueva  
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## Kourkoutis Group

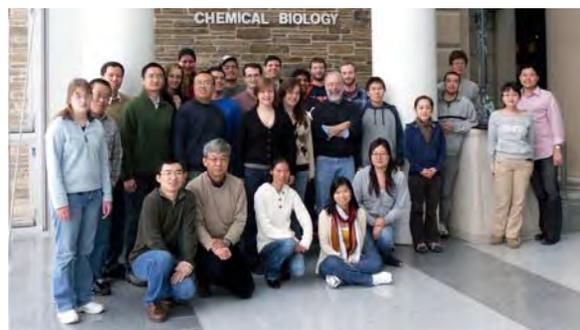
Michael Zachman  
Katy Spoth

## Microscopes

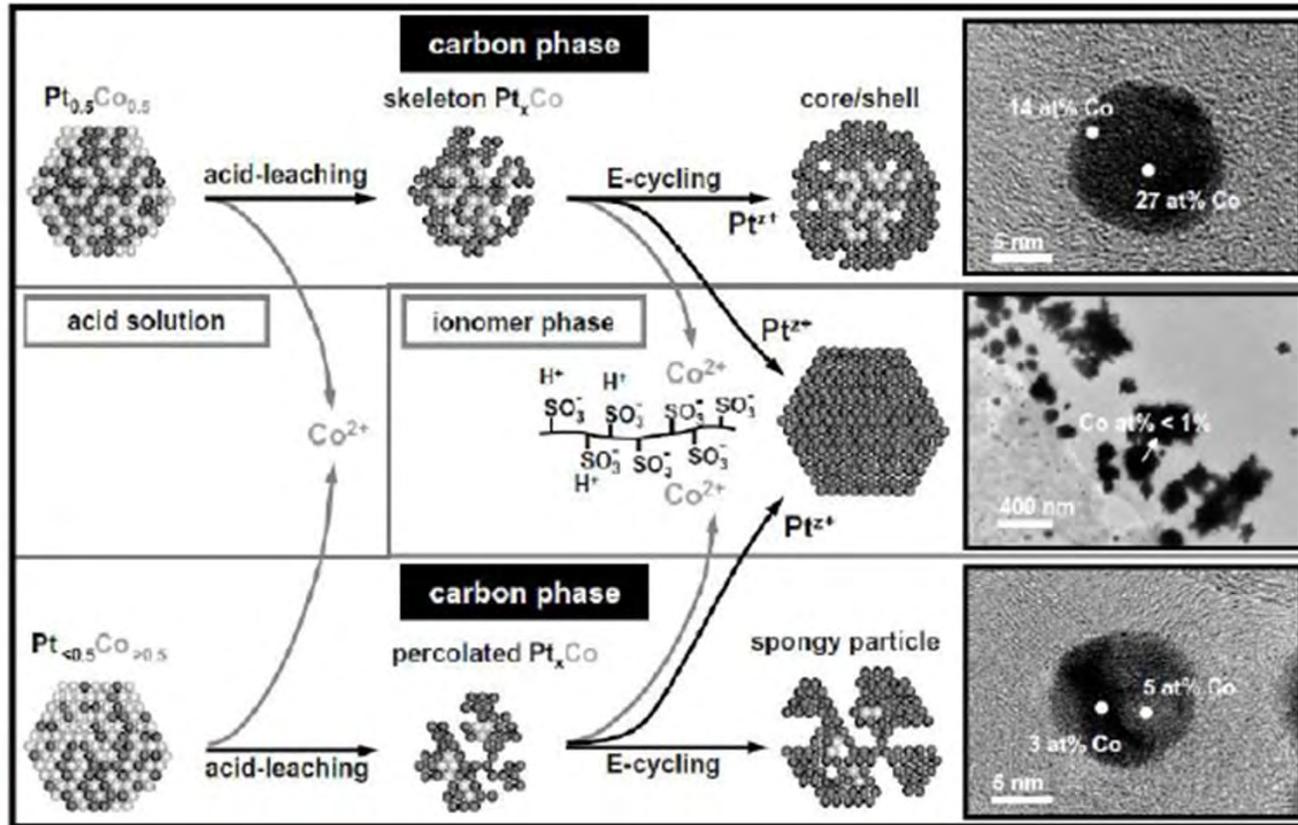
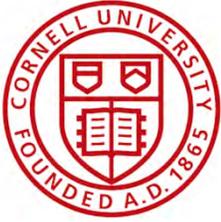
John Grazul  
Mick Thomas  
Earl Kirkland

## Protochips (John Damiano)

**Gamry (Burak Ulgut)**



Work supported by the Energy Materials Center at Cornell, DOE EFRC BES (DE-SC0001086). EM Facility support from the NSF MRSEC program (DMR 1120296).



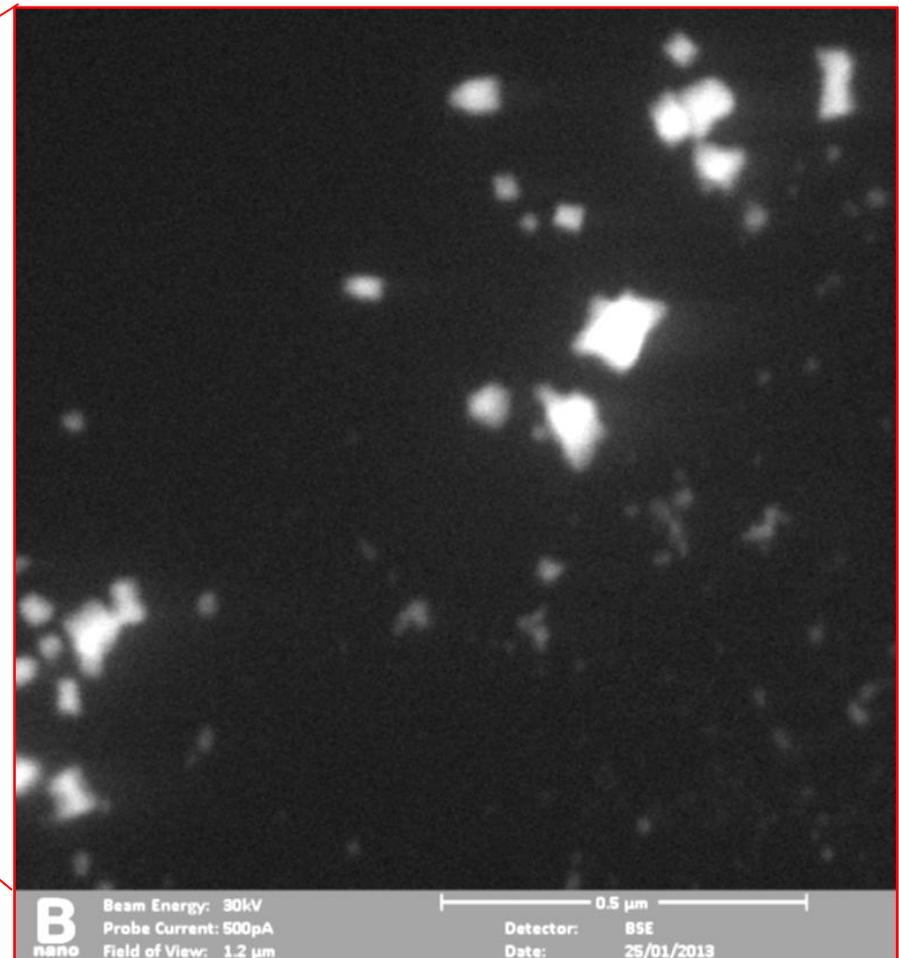
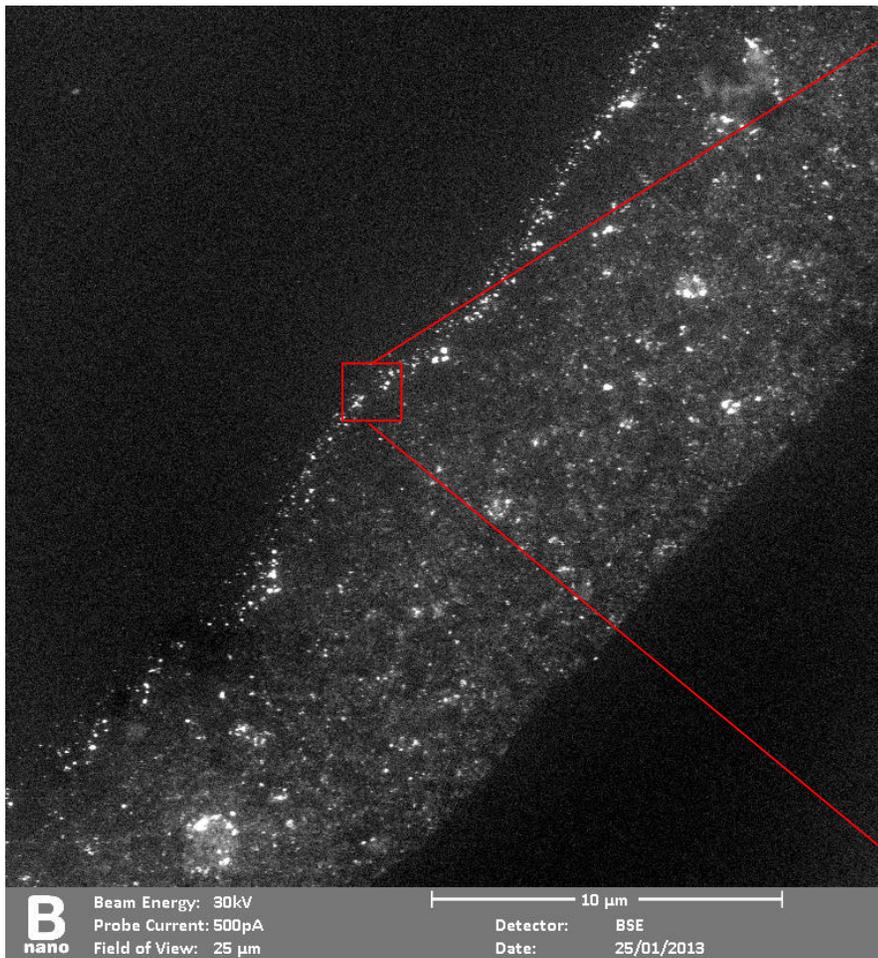
**Core-shell structure**  
 arose from acid leaching of relatively Pt-rich particles (skeleton Pt surface) followed by electrochemical cycling to smooth out the surface Pt layer.

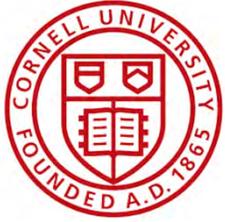
(small to intermediate particles)

**Spongy (“percolated”) structure**  
 arose from acid leaching of Co-rich particles. (large particles)

S. Chen, H. A. Gasteiger *et al.*, J ECS, 157, A82 (2010)

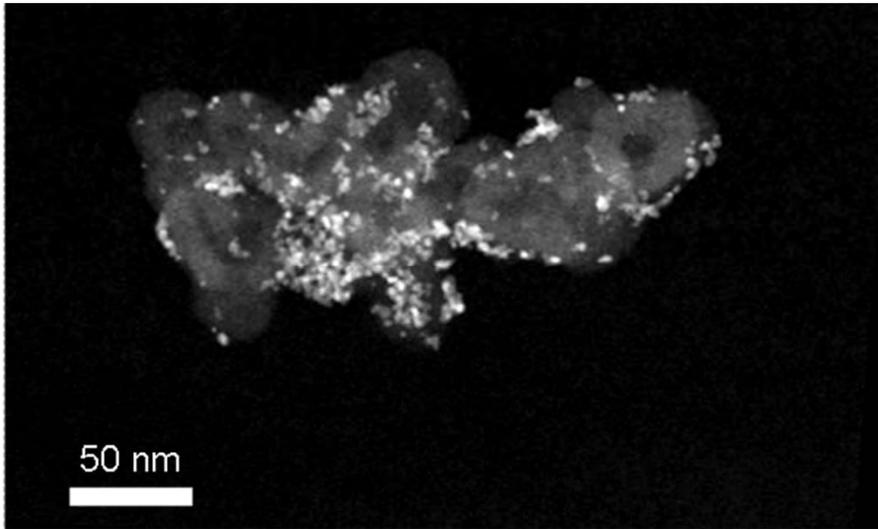
# *Air-SEM Backscatter images of Pt<sub>3</sub>Co Nanoparticles in a Fuel Cell Cathode after 30k Cycles*



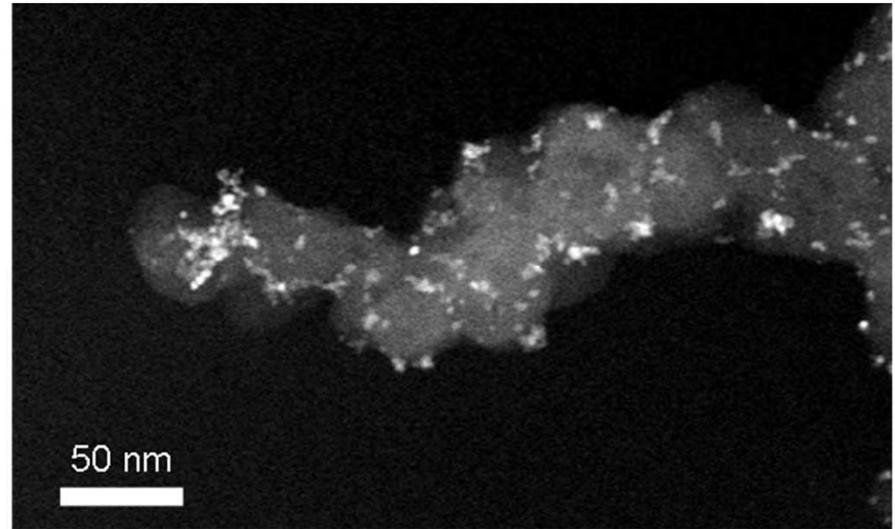


## *Effects of Corrosion in 2D*

**Corroded Carbon Support**



**Non-corroded Carbon Support**



- Corroded carbon shows holes - pore? Voids? Connectivity?
- Does the Pt catalyst get inside the holes?

## Microscopy

*Yingchao Yu*  
*Megan Holtz*  
*Huolin L. Xin*  
*Robert Hovden*  
*Julia A. Mundy,*  
*Lena Kourkoutis*  
*Randy Cabezas*  
*Peter Ercius*  
*Barnaby Levin*  
*Kayla Nguyen*

## Synthesis and E-Chem

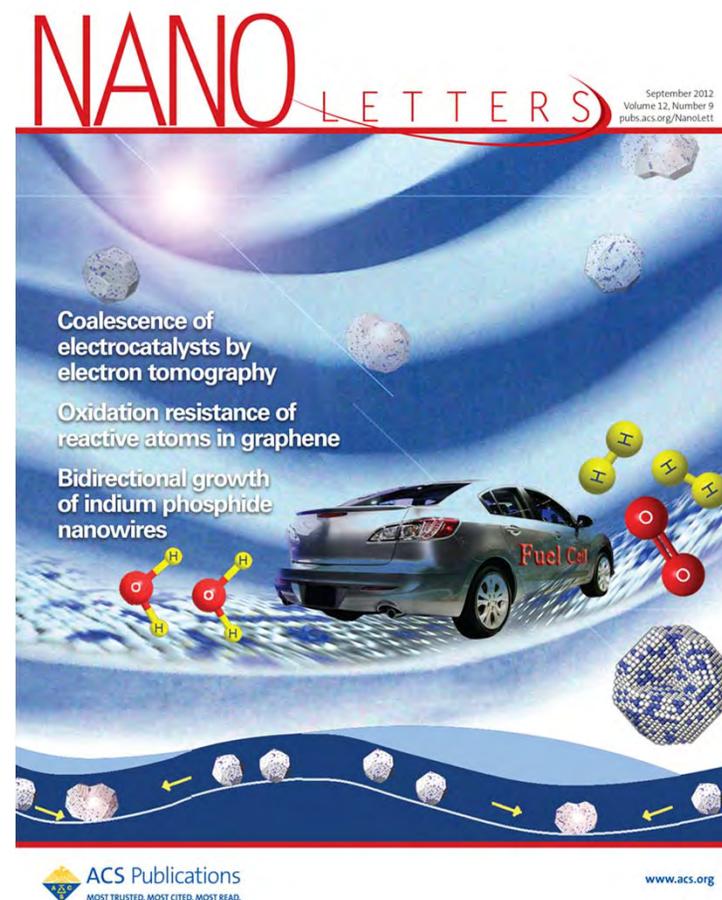
*Deli Wang*  
*Eric Rus*  
*Hao Chen*  
*Frank DiSalvo*  
*Hector Abruña*

## General Motors

*Z. Vic Liu*  
*J. Zhang*  
*Nalini Subramanian*  
*Rohit Makharia*  
*Fred T. Wagner*

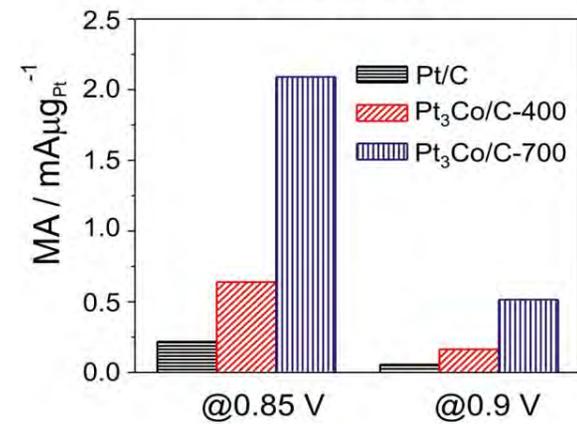
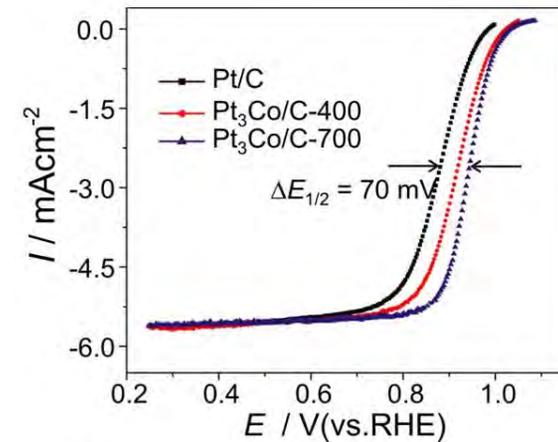
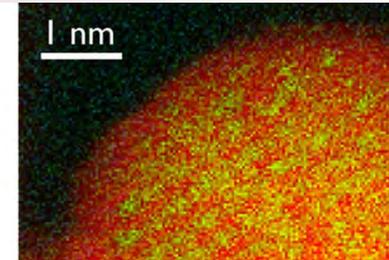
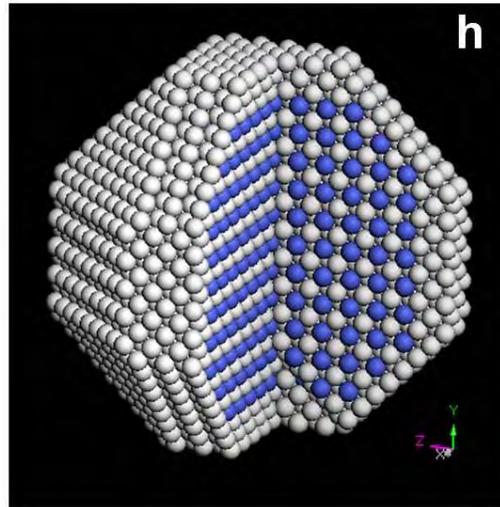
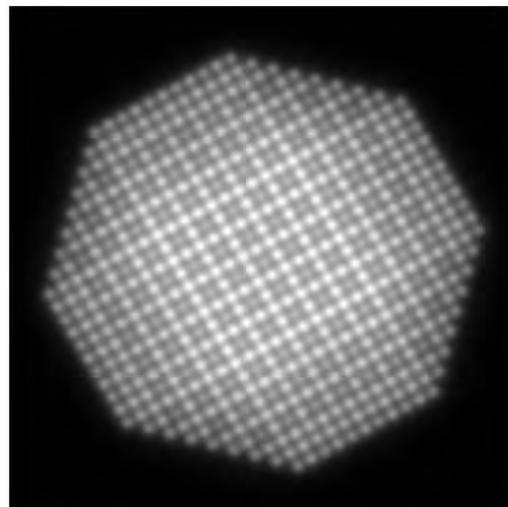
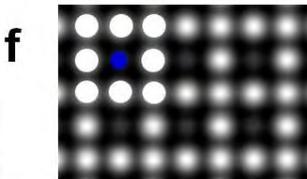
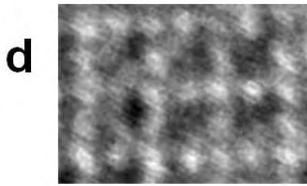
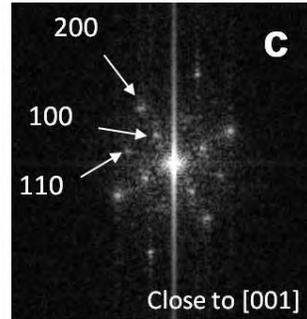
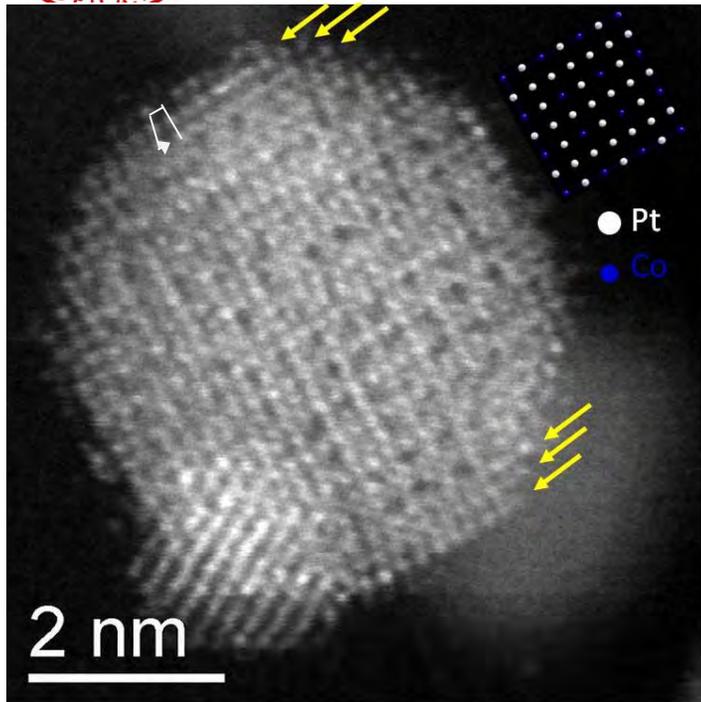
## Protochips

*David Nakahashi*  
*John Damiano*

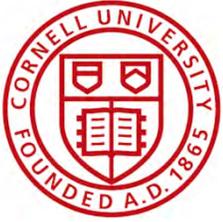




# Ordered Intermetallic vs. Disordered Alloy

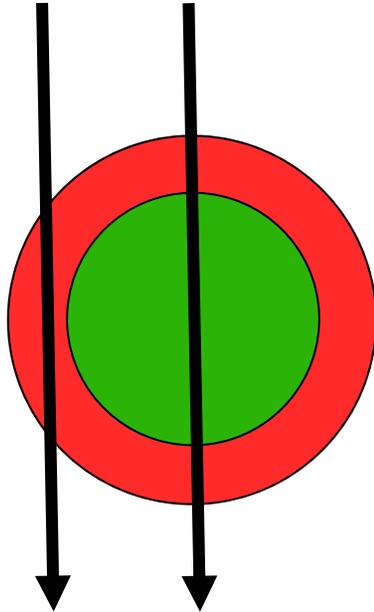


Wang, et al, Nature Materials, 2012, in pres.

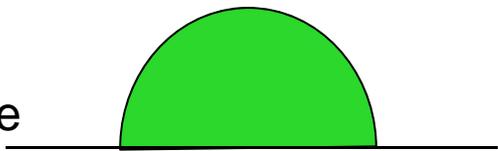


# Imaging in Projection

Core-Shell



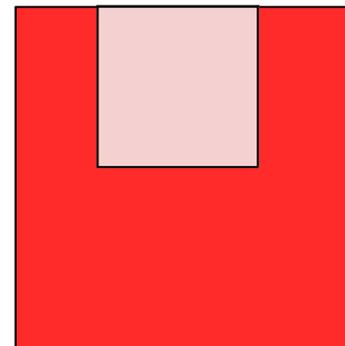
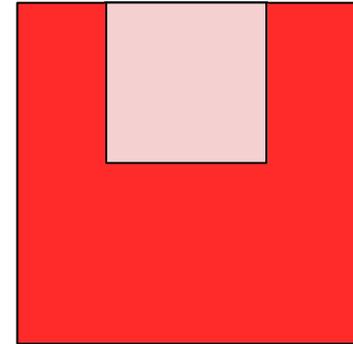
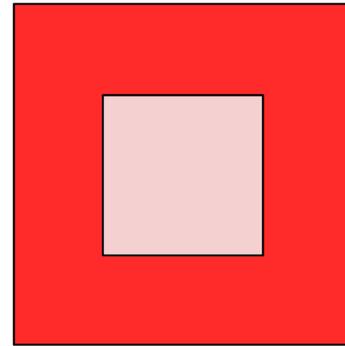
Core

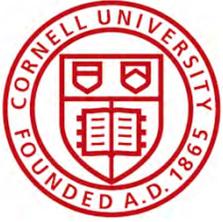


Shell

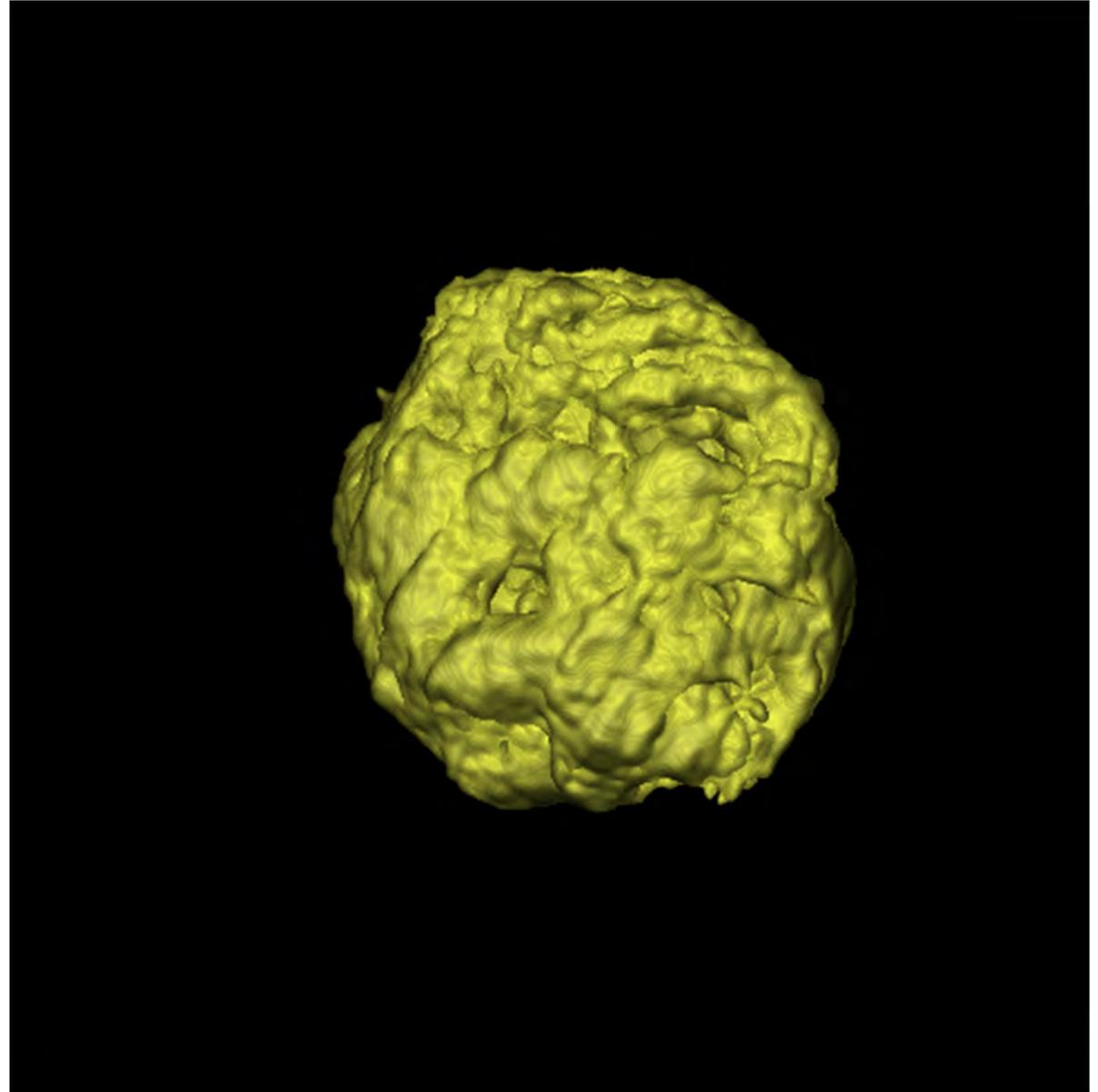
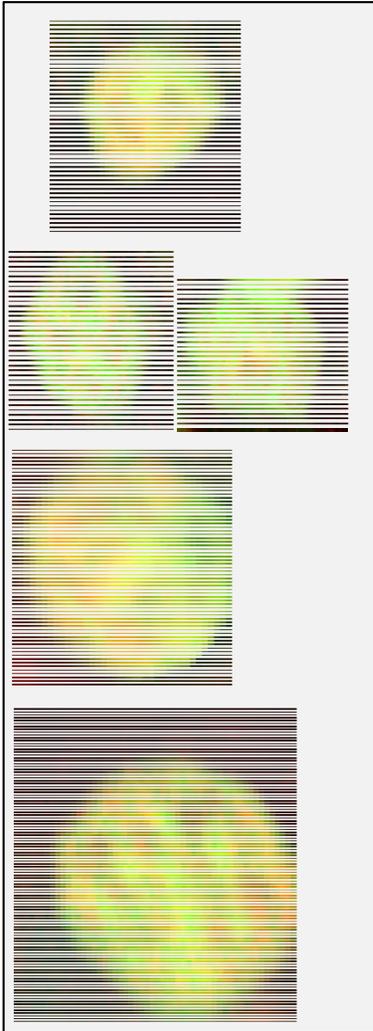


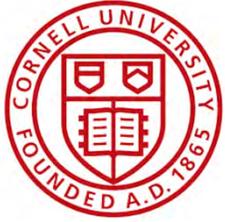
Inside or Outside?





# Porous, acid-leached $\text{PtCu}_3$

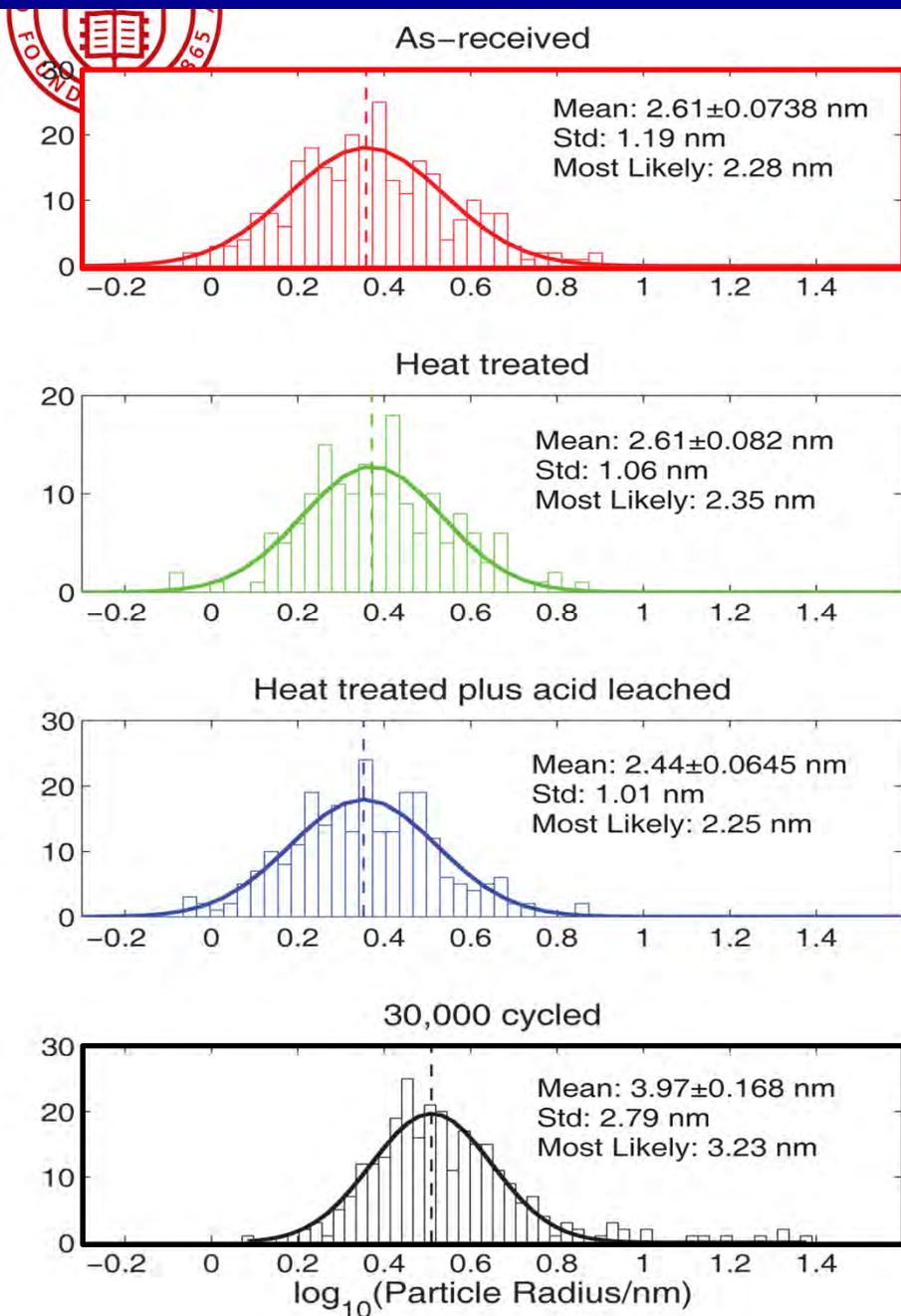




*EST Reconstructed PtCu<sub>3</sub> Nanoparticle*  
*(Equal-Slope Tomography, John Miao, Mary Scott, UCLA)*



# Particle Size Distributions (radius not diameter)



## Reactivity loss

**60%**

Can this number be explained by the loss of surface area?

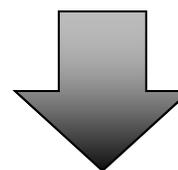
## Surface area loss

Assume no loss of cobalt

Surface loss: **15.8%**

Assume all Co lost (volume loss of ~25%)

Surface loss: **36.8%**

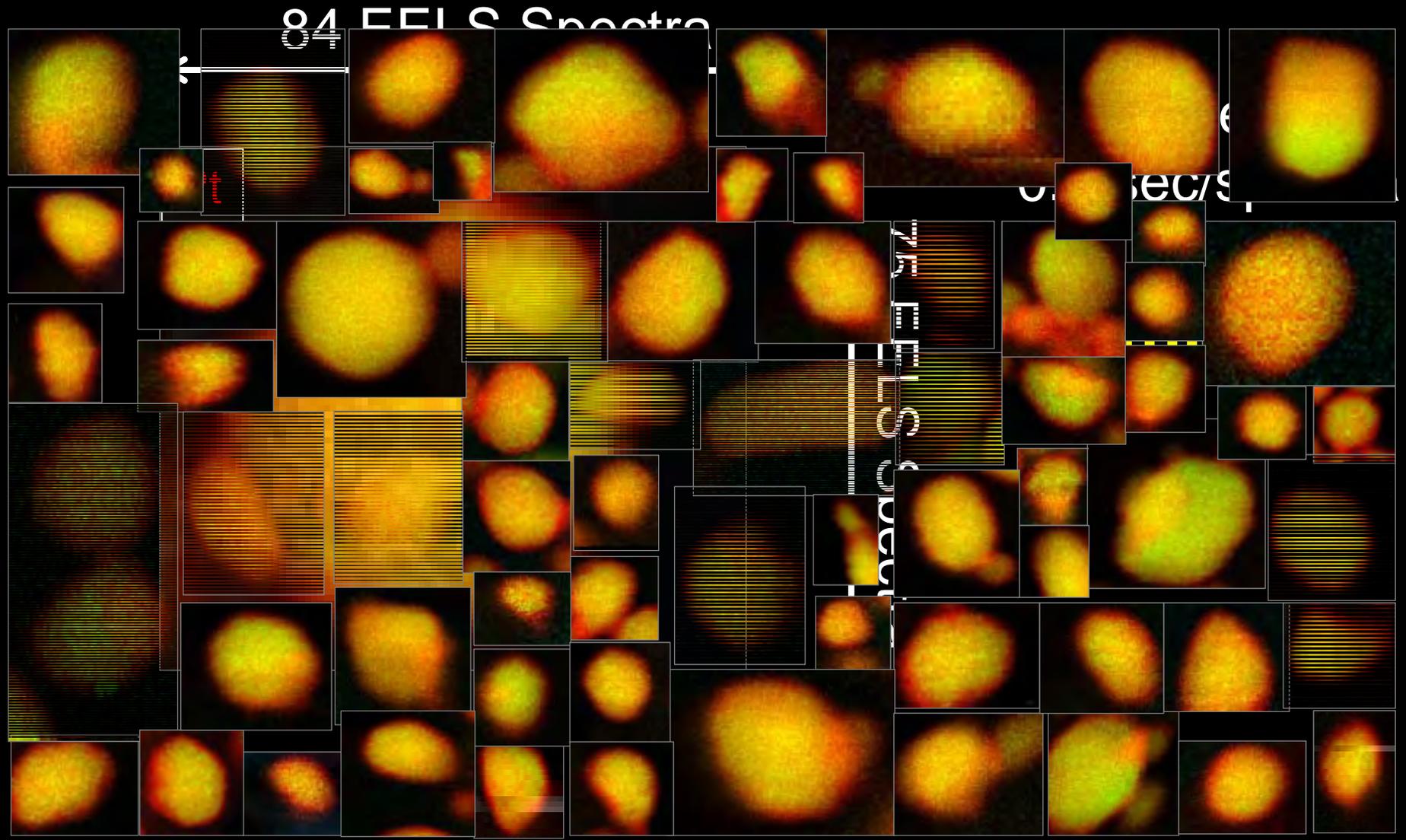


Bigger particles have lower reactivity per unit area. Why?

# Pre-voltage Cycling Particles: Pt Rich Shell

Pt  
Co

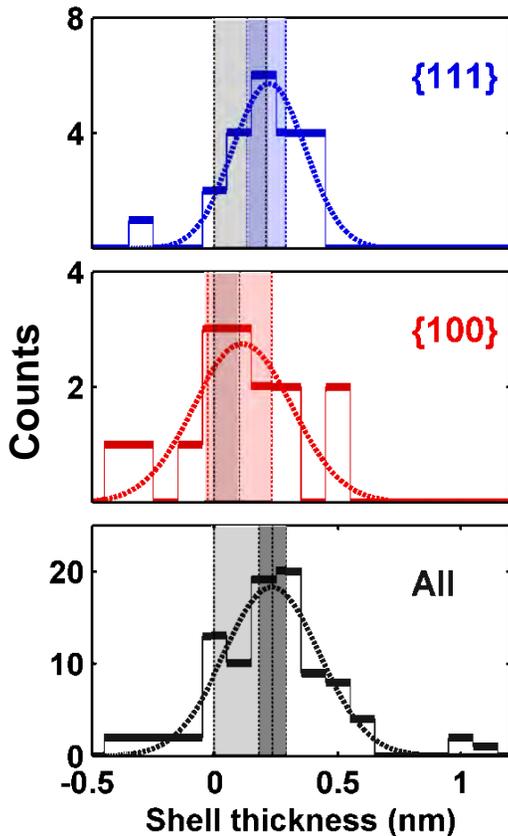
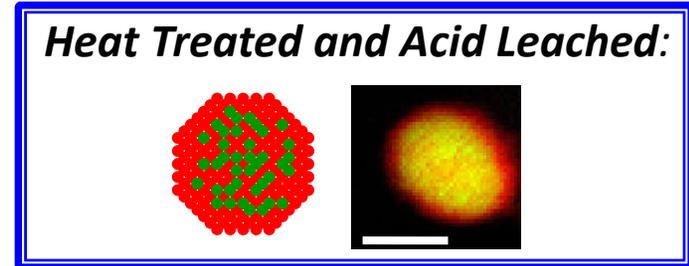
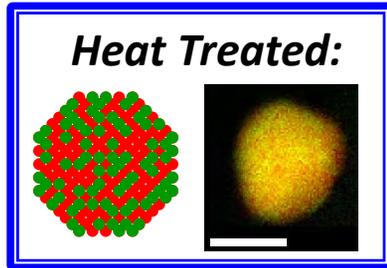
10 nm



# Heat Treated and Acid Leached: From Pt-rich {111} surface → 3 ML Pt shell on all surfaces



Scale bar:  
5 nm



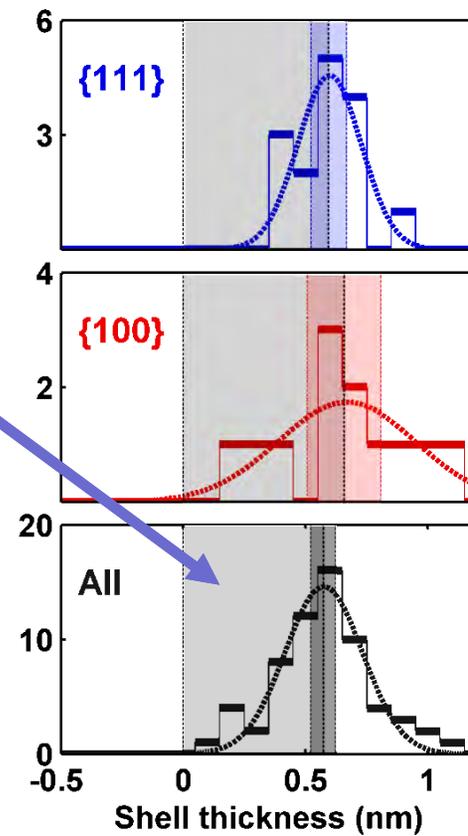
Pt segregation on {111} only

3 ML Pt-shell:

- No Co neighbors
- No Co 2<sup>nd</sup> neighbors
- No Co-Pt bonding

Yet:

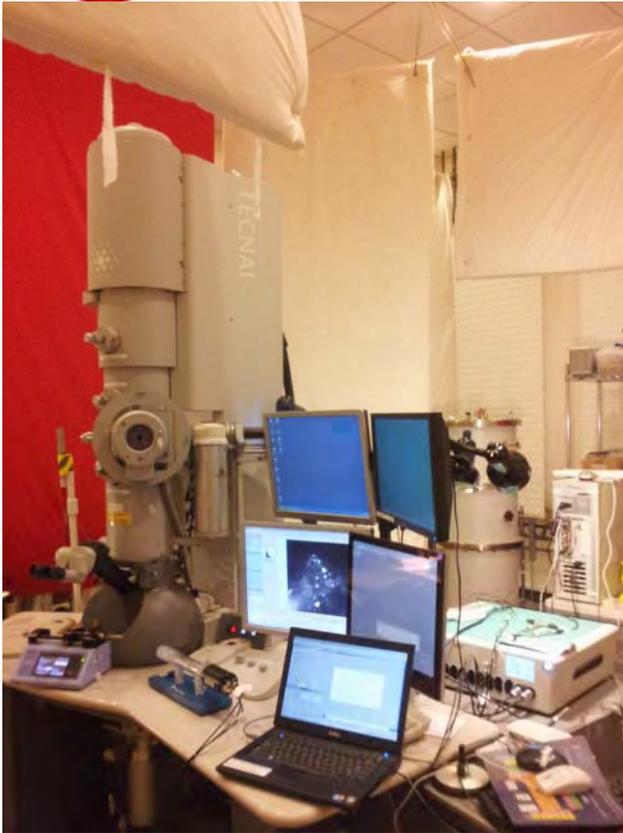
- Enhanced activity over pure Pt.



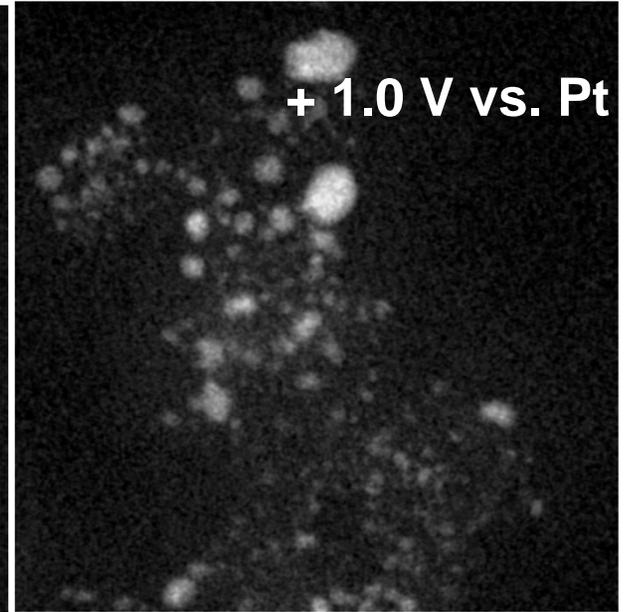
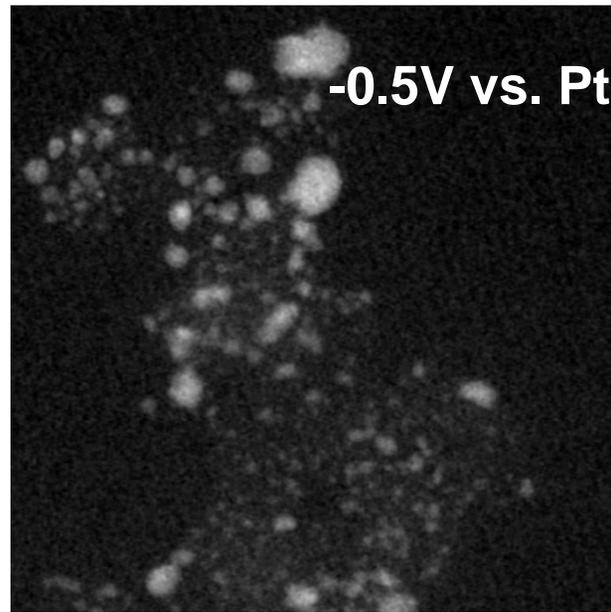
No statistical correlation between facet and shell thickness



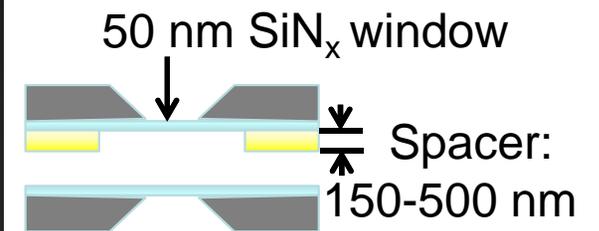
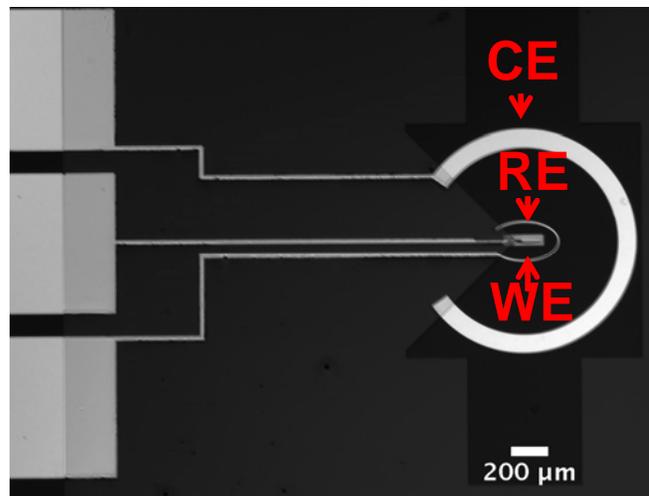
# *In-situ* Electrochemical Cycling of Fuel Cell Catalysts



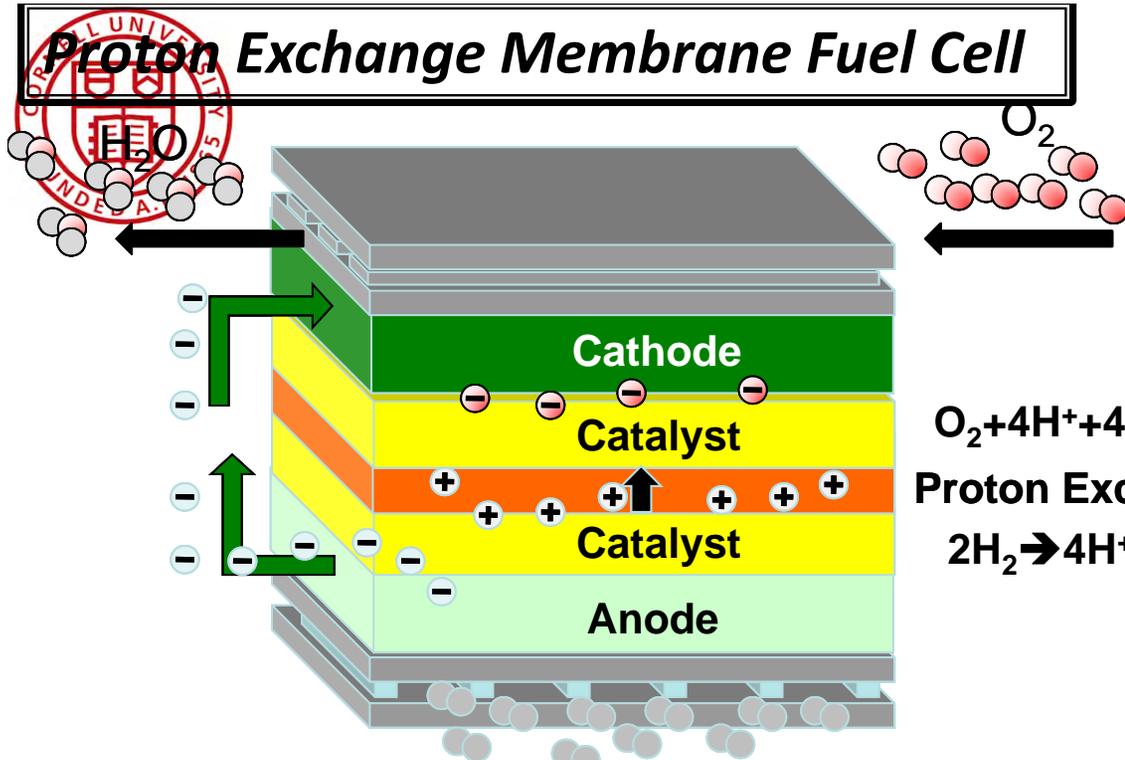
Protochips  
Electrochemical Holder



Pt<sub>3</sub>Co/C in 0.1 M Perchlorate Acid



# Proton Exchange Membrane Fuel Cell



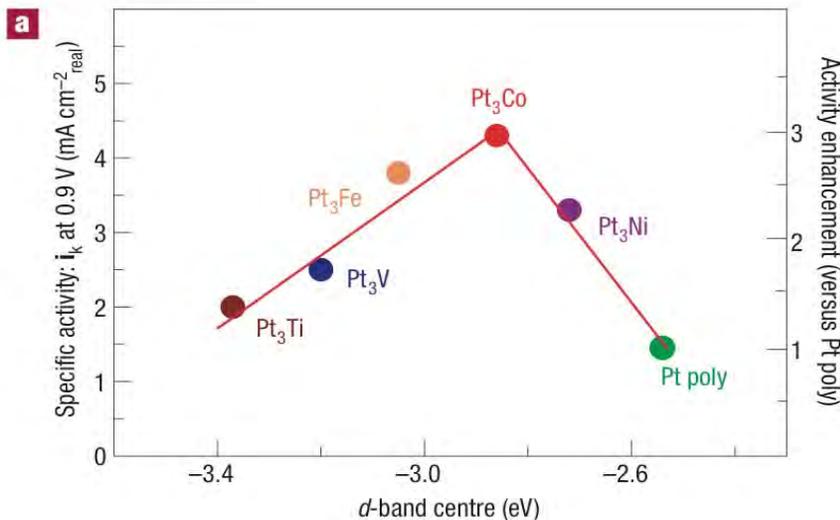
**Limiting Step:  
Oxygen Reduction  
Reaction in Cathode**



Proton Exchange Membrane



## Improved Catalytic Activity v. Pure Pt Metal

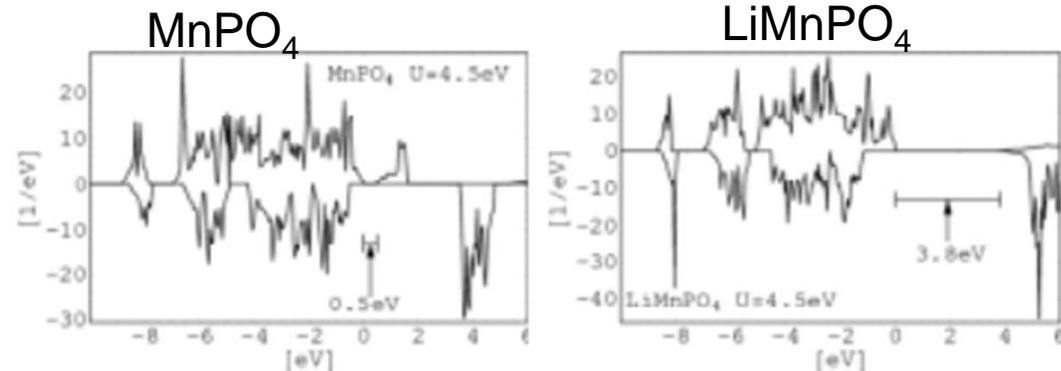


V. R. Stamenkovic et al. *Nature Mater.* **6**, 241 (2007).

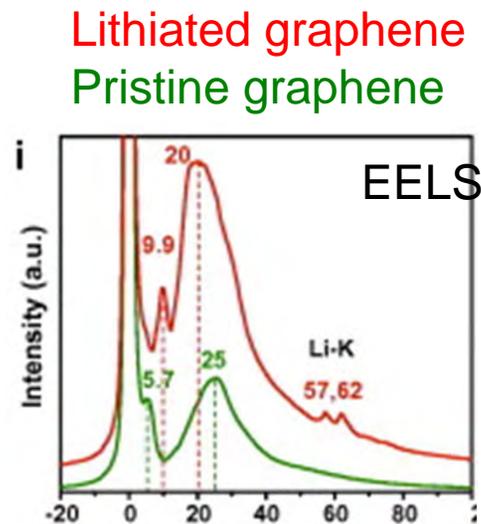
## *Pt<sub>3</sub>Co* catalyst particles:

- What is the atomic distribution of Pt and Co in these particles?
- How durable are these alloys?

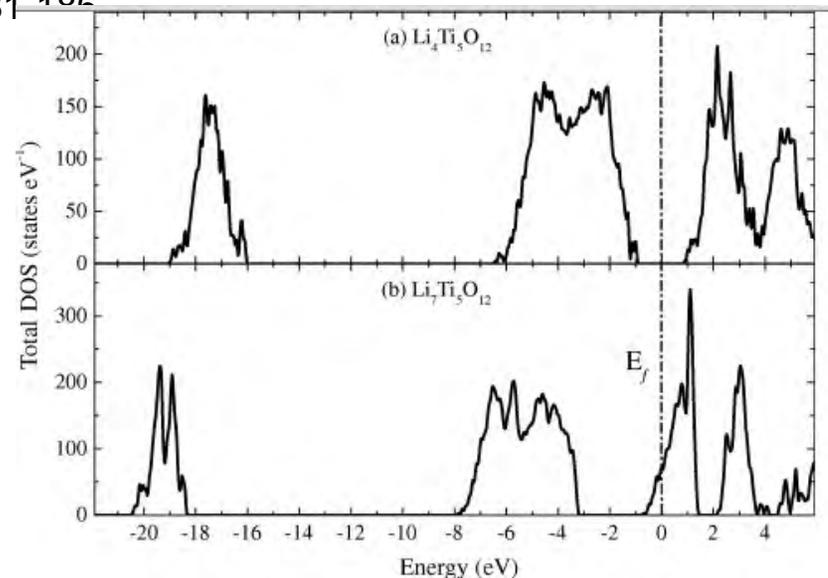
# Liquid cell EFTEM technique with new battery materials



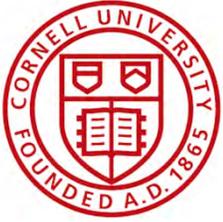
Fei Zhou, et. al. Solid State Communications 132.3 (2004): 181-186



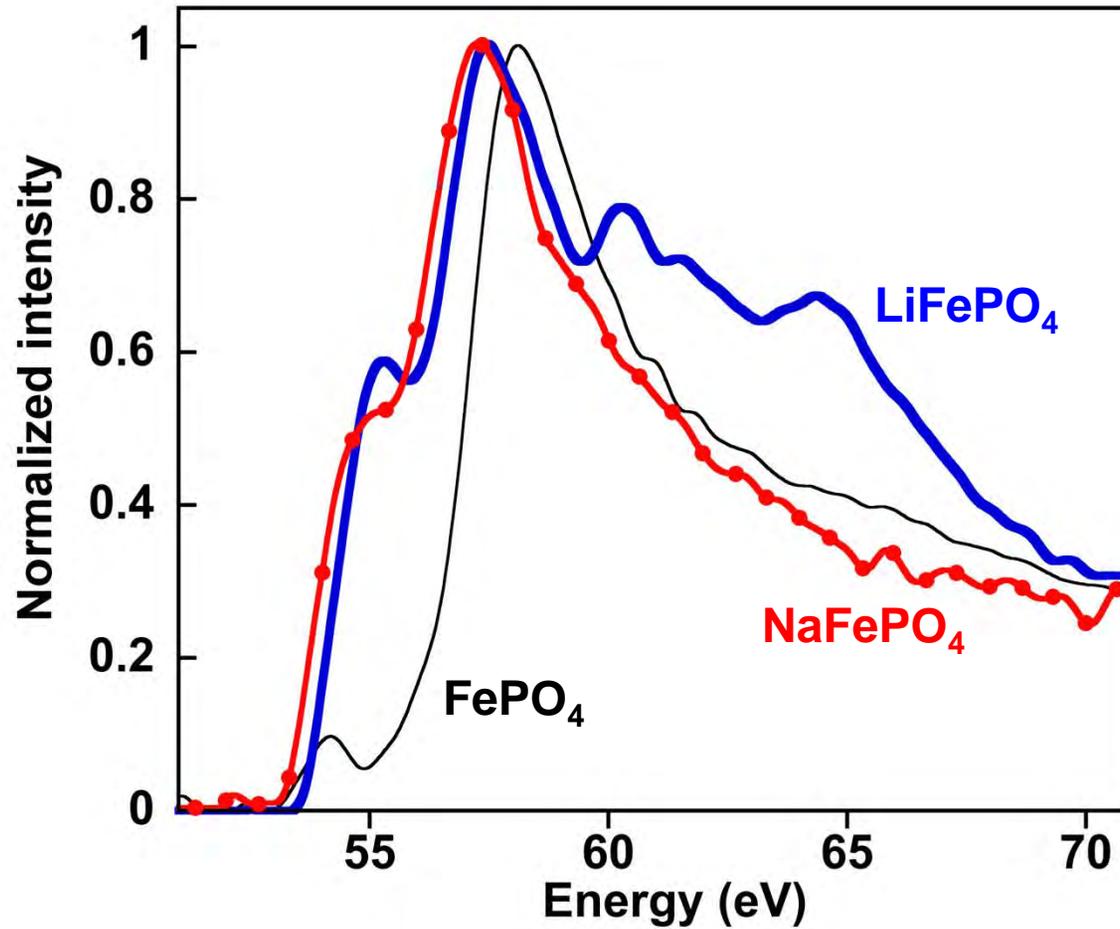
Xiao Hua Liu, et al. Carbon, **50** 10, (2012) 3836–3844



Ouyang C.Y, et. al. Echem Comm. **9** 5 (2007): 1107-1112

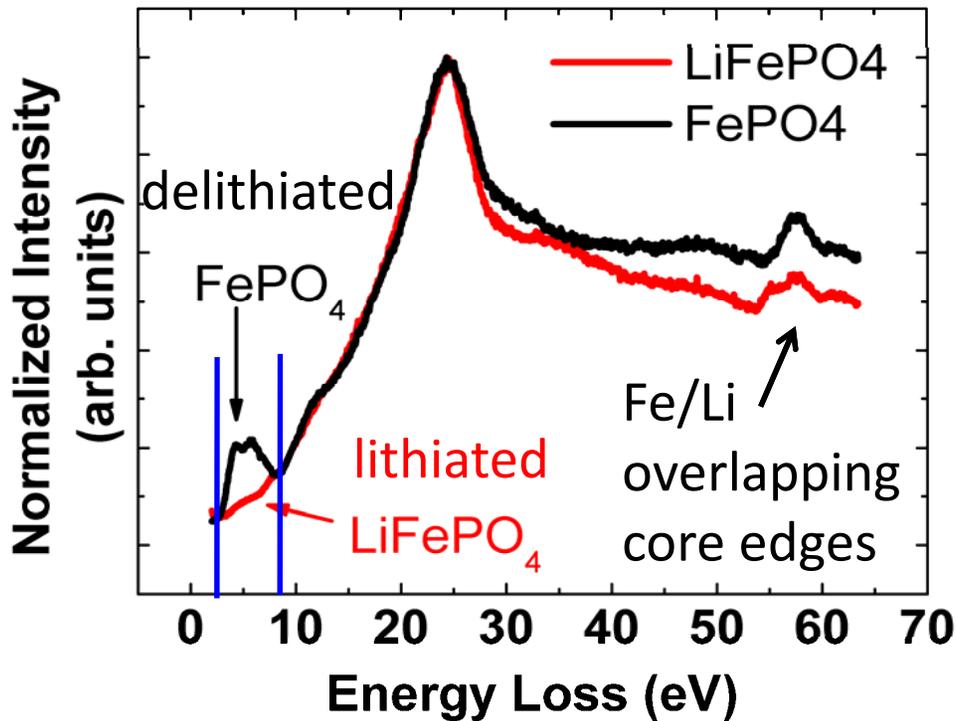


# Li-K Edge Problems II

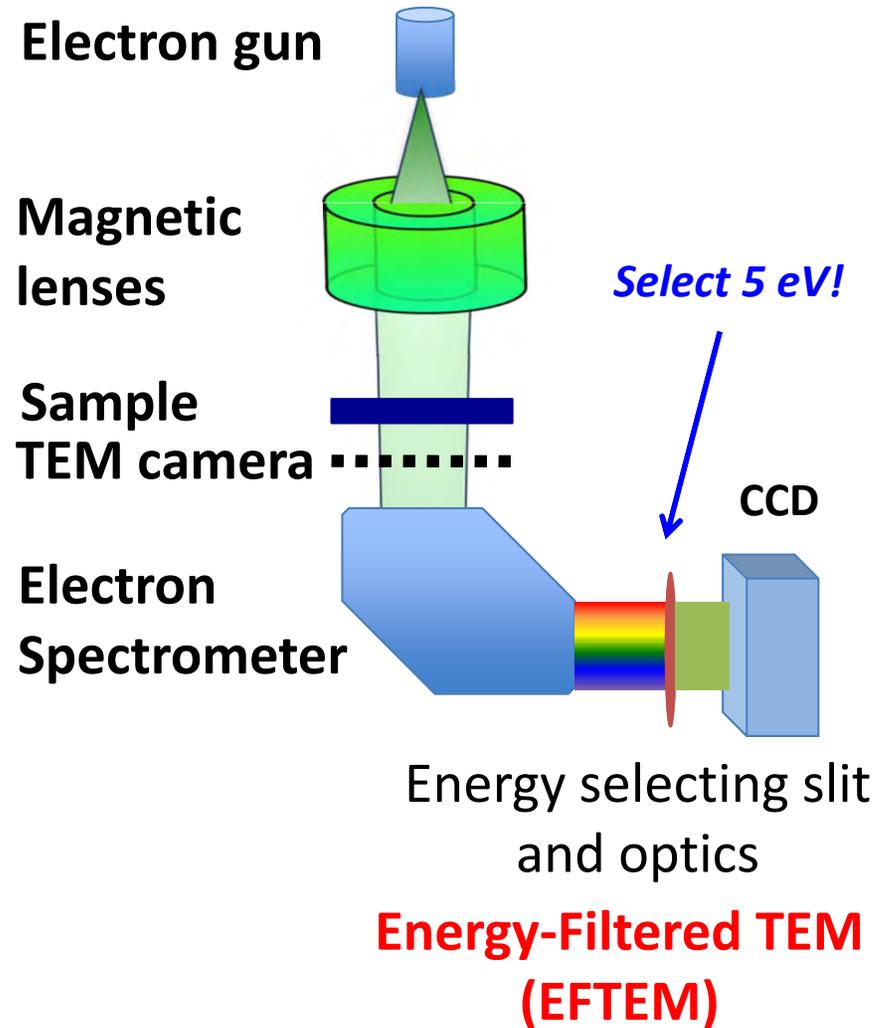


# Valence EELS for Batteries

Monochromated EELS,  
0.2 eV resolution

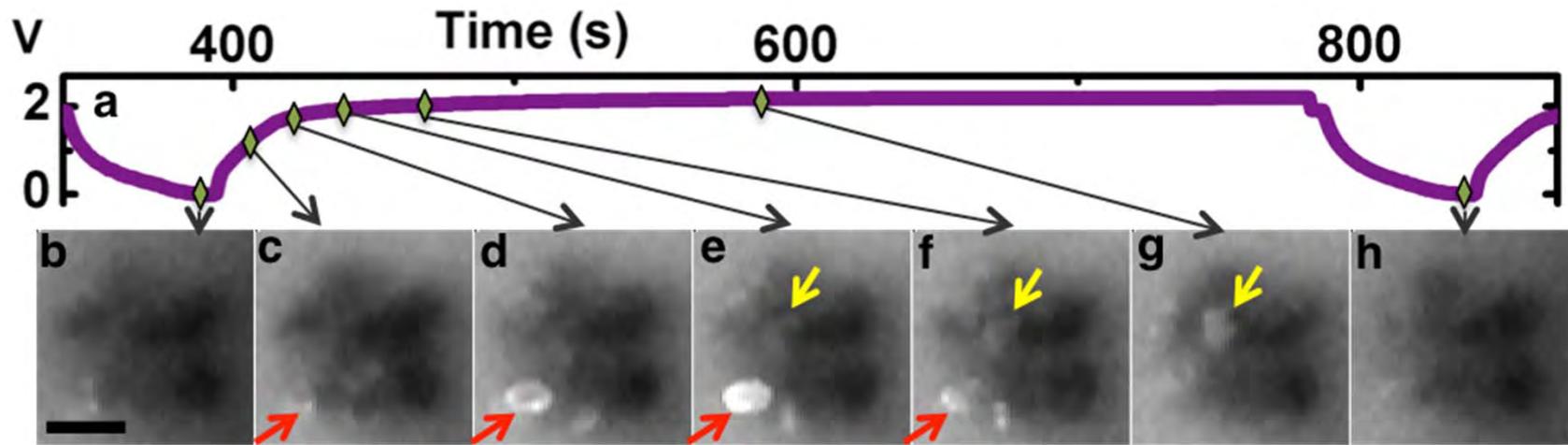


Transmission  
Electron Microscope (TEM)



Want **rapid spectroscopic**  
**images of the 5 eV signal!**  
**EFTEM imaging!**

# Mechanism of Lithiation

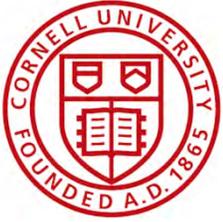


2 dominant modes of lithiation for this experiment:

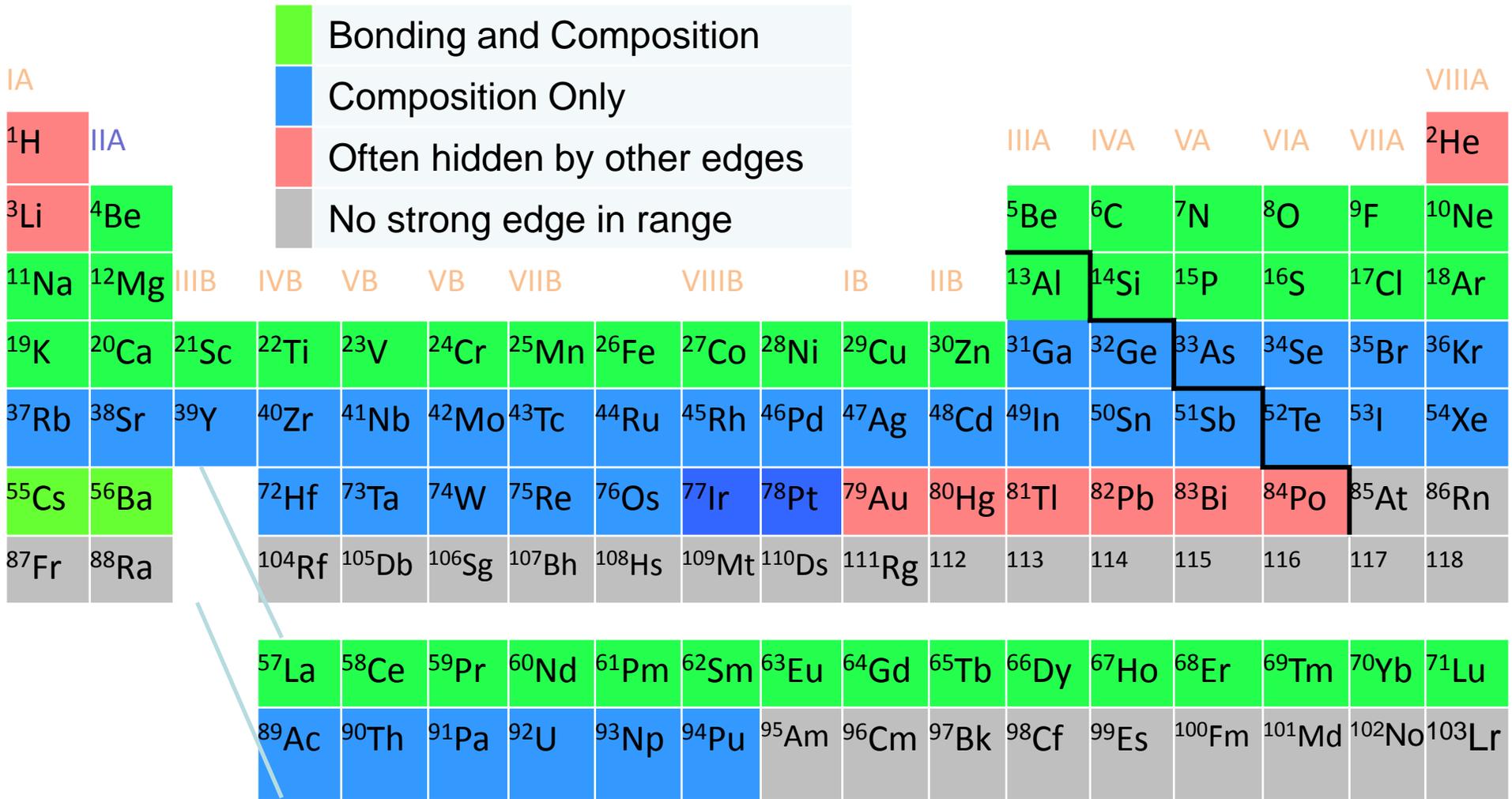
We see core-shell type structures during growth of the  $\text{FePO}_4$

Slow nucleation for domino cascade

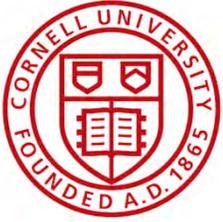
Highly dependent on particle size and experimental conditions: further investigation necessary



# Elements Accessible by EELS

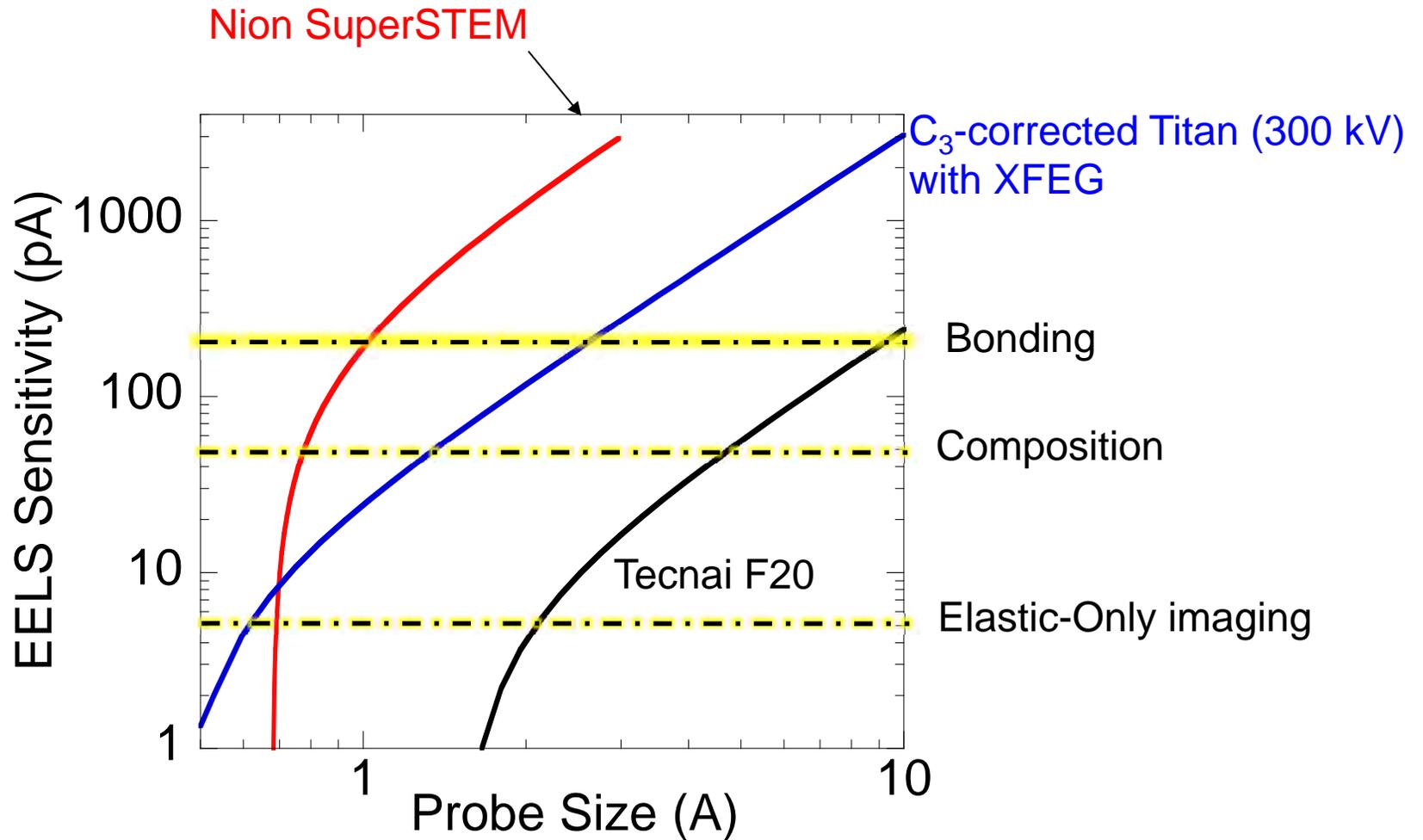


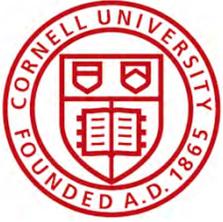
Imaging Pt: P. Cueva et al, *Microscopy and Microanalysis* **18**, 667 (2012).



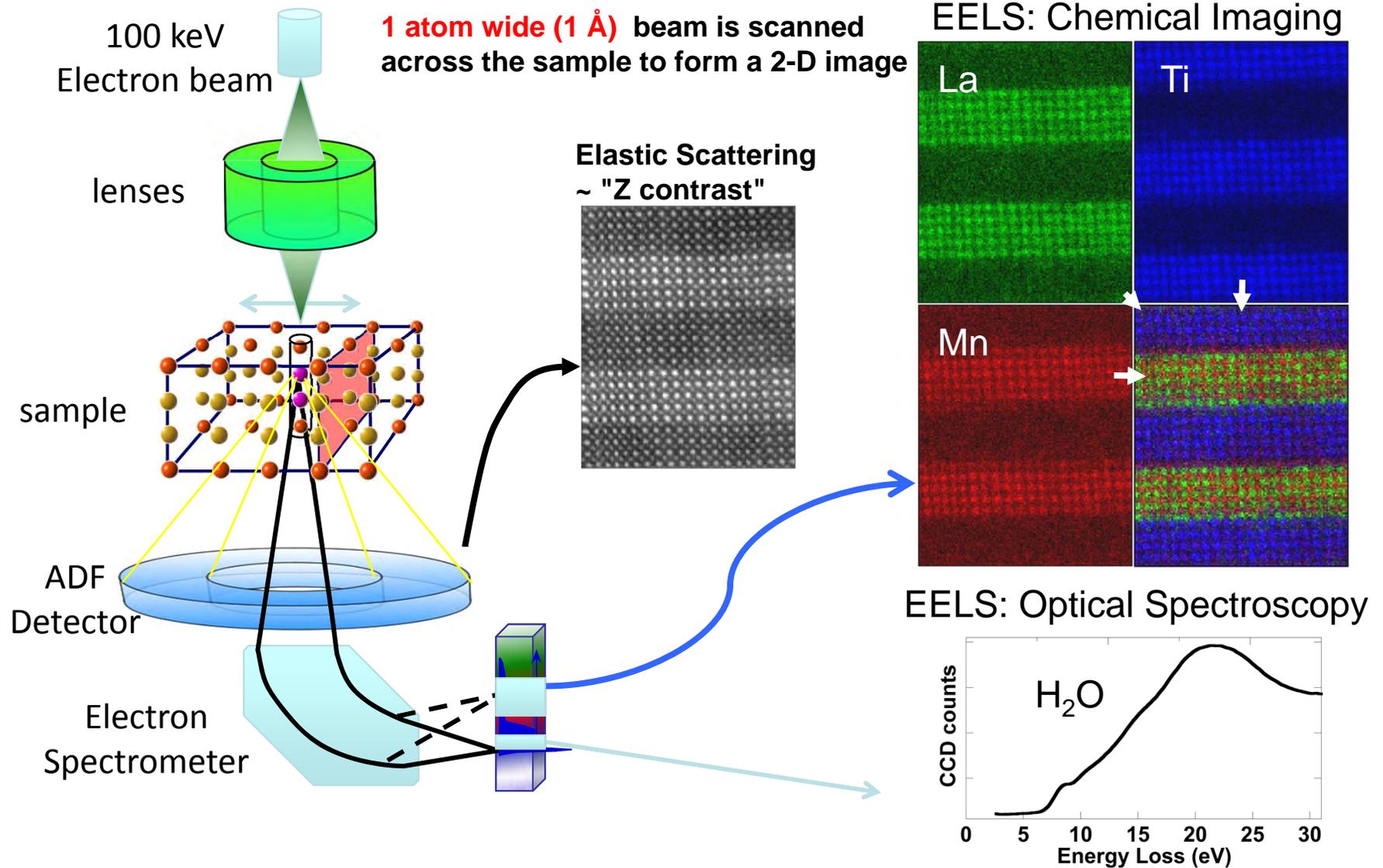
# Tradeoff: Probe Size for Beam Current

$$I_{coll} = \beta \times (\text{Probe Area}) \times (\text{Probe Solid Angle}) \times (\text{Collection Efficiency})$$





# Scanning Transmission Electron Microscopy

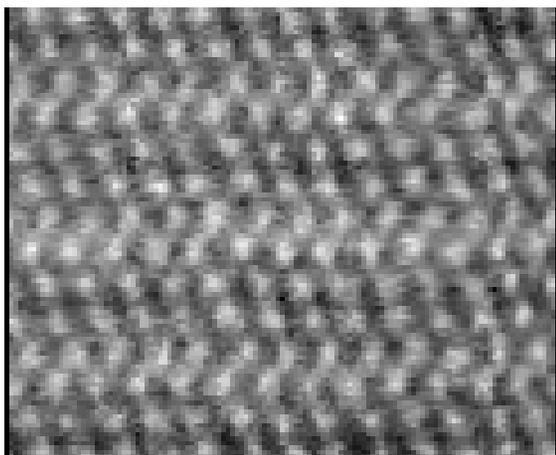


Muller, Kourkoutis, Murfitt, Song, Hwang, Silcox, Dellby, Krivanek, *Science* **319**, 1073 (2008).

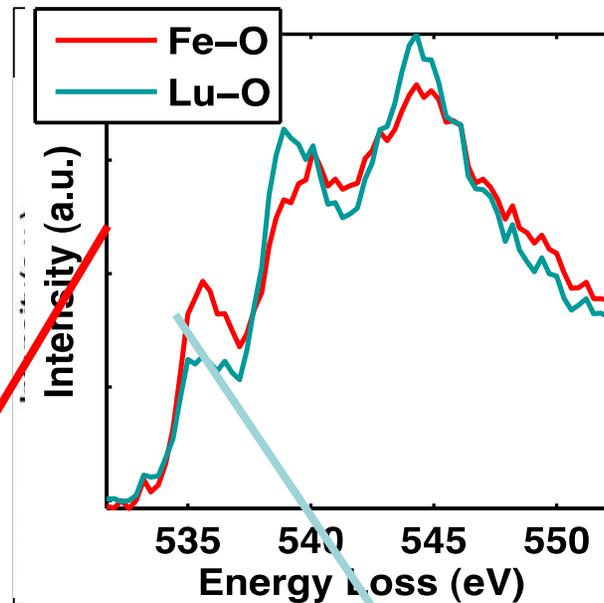
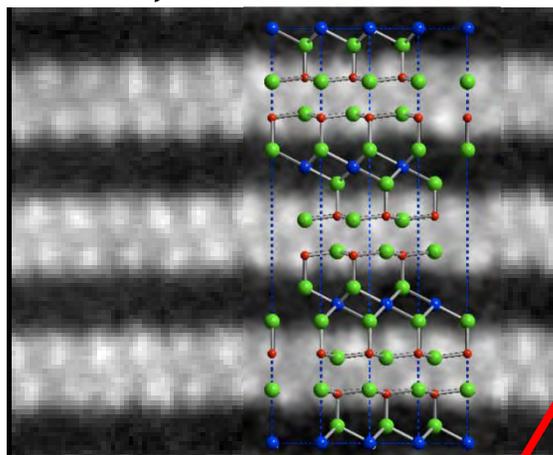
# Mapping Chemical Environment of O atoms in $\text{LuFe}_2\text{O}_4$



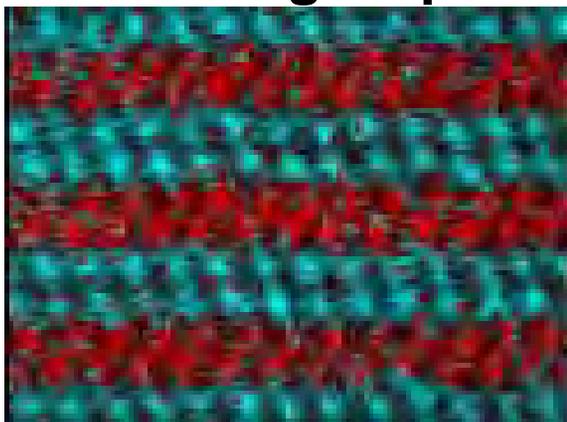
O K-edge



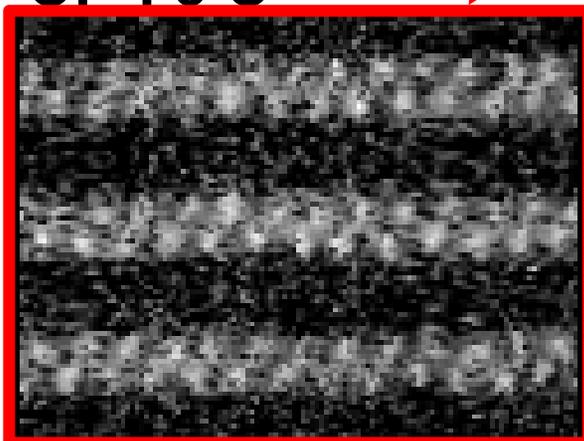
Fe  $L_{2,3}$  - edge



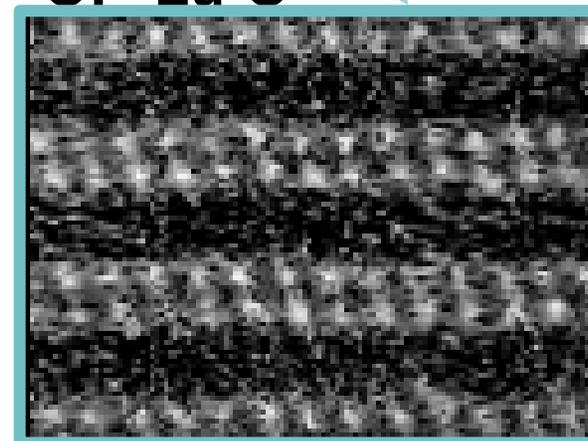
O Bonding Map

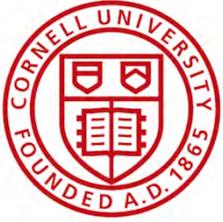


O: "Fe-O"

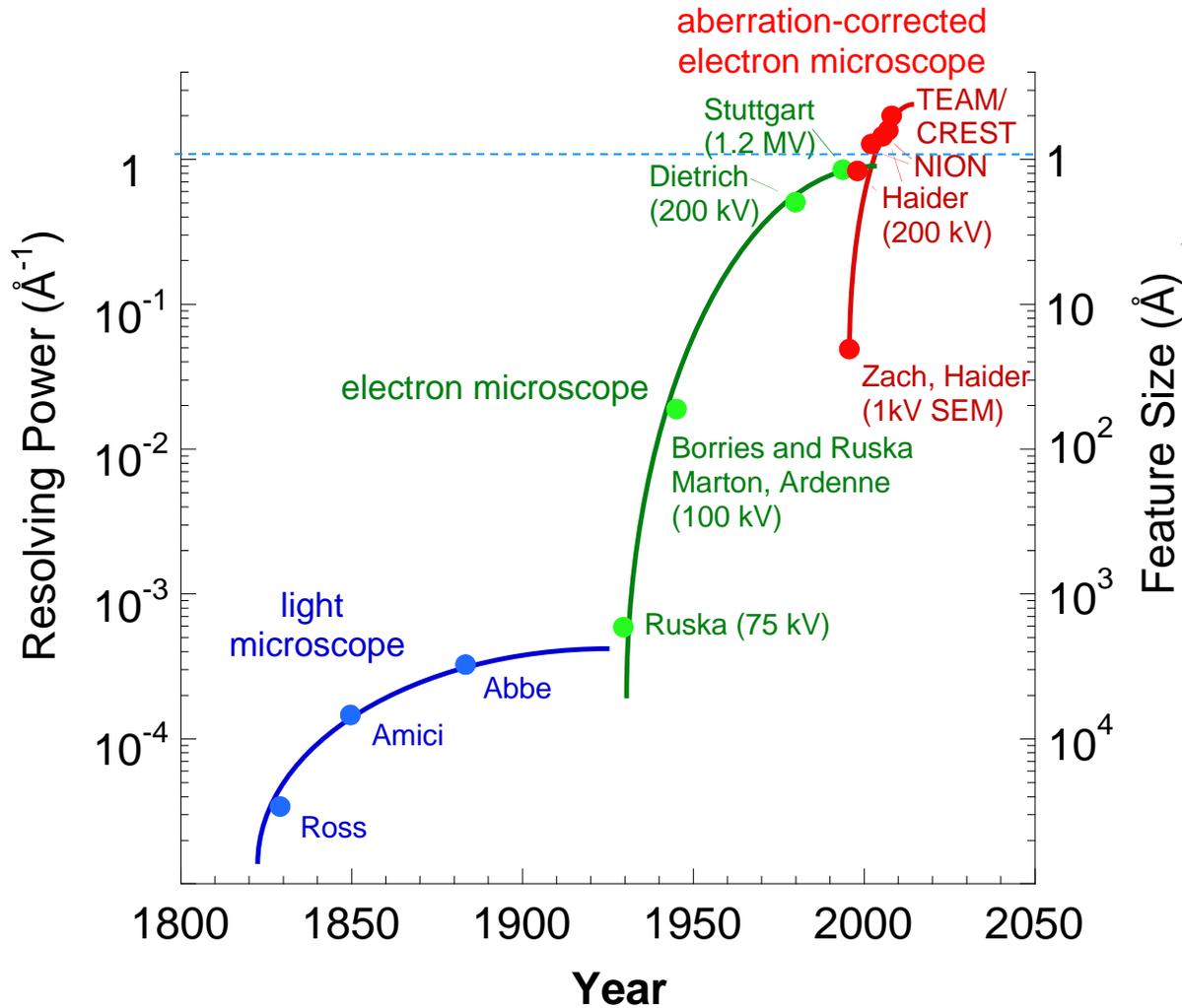


O: "Lu-O"



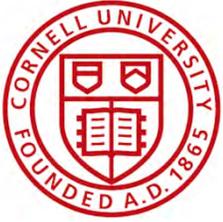


# Hardware Advances in Microscopy

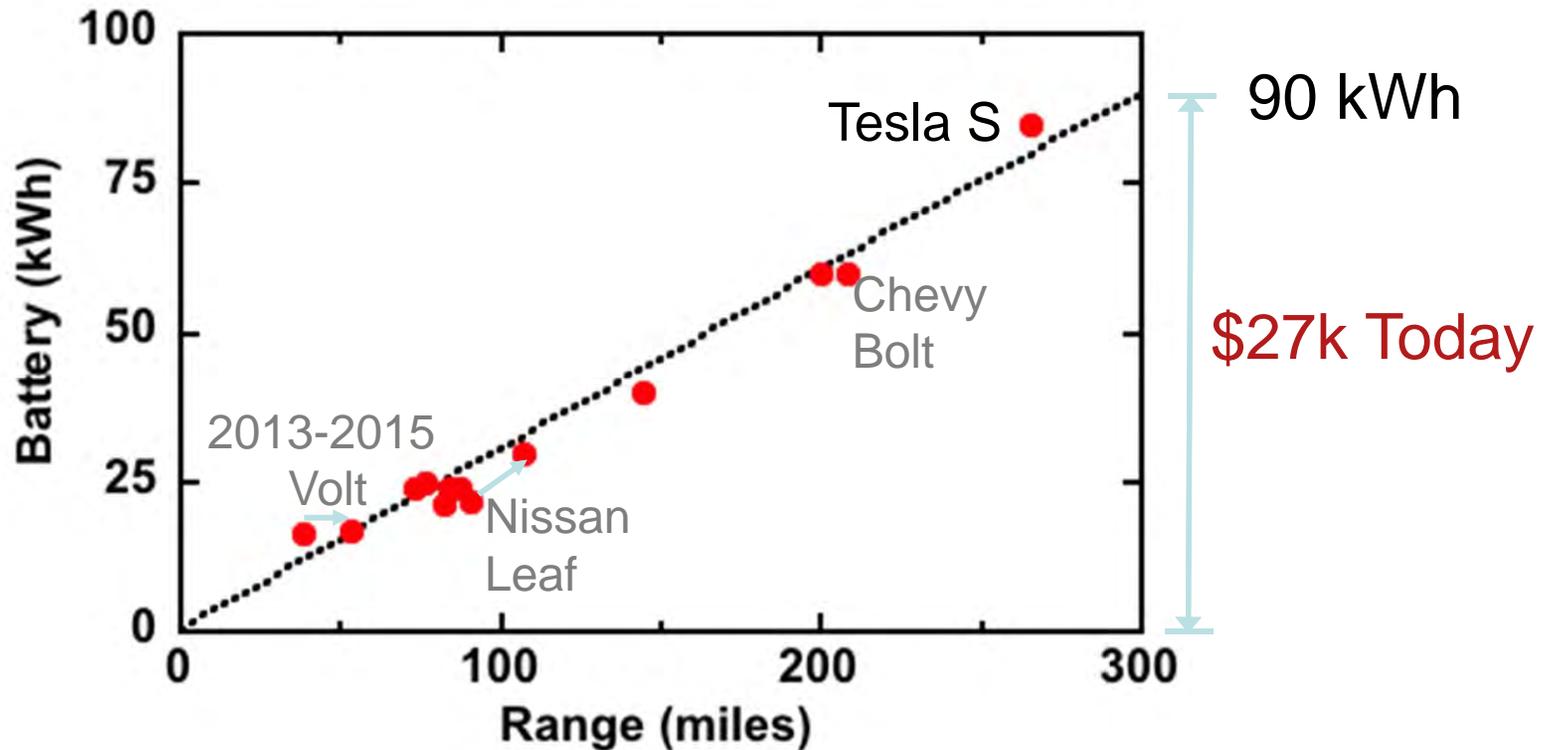


Corrected optics have enabled practical Sub-Angstrom resolution

Muller, Nature Materials, (2009), Adapted from Rose (2009)

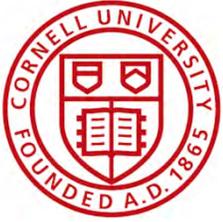


# Electric Vehicle Range is Set by Size of Battery

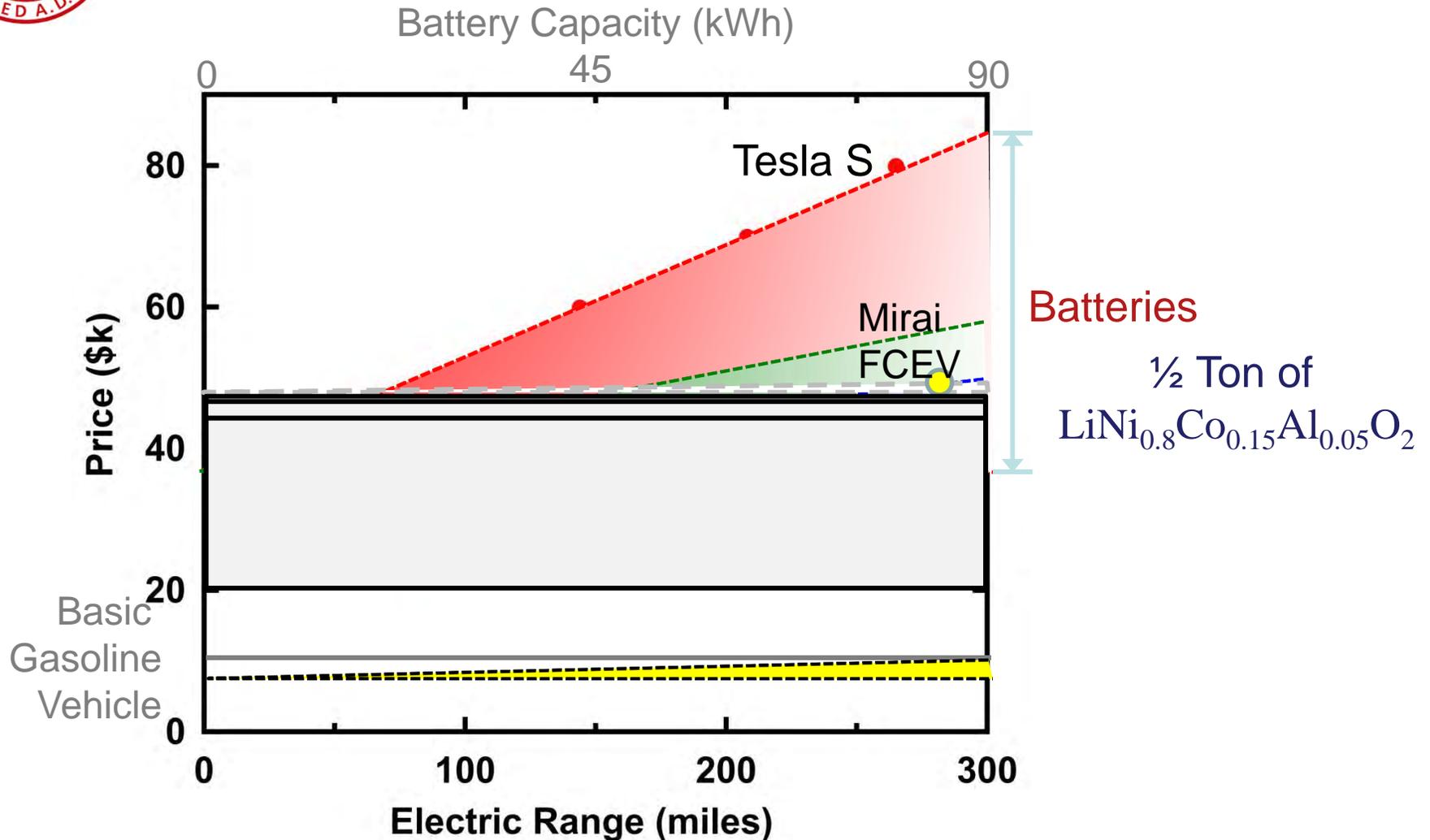


How much does 90 kWh cost?

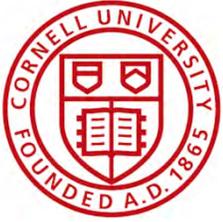
- Tesla : \$27,000 (inc.  $\frac{1}{2}$  ton of  $\text{LiNi}_{0.8}\text{Co}_{0.15}\text{Al}_{0.05}\text{O}_2$ )
- Future Li-ion : \$13,500 - \$20,000
- **To beat Gas** : \$1,000-\$3,000  **New Materials (<33 \$/kWh)**



# Electric Vehicle Price & Range Dominated by Battery

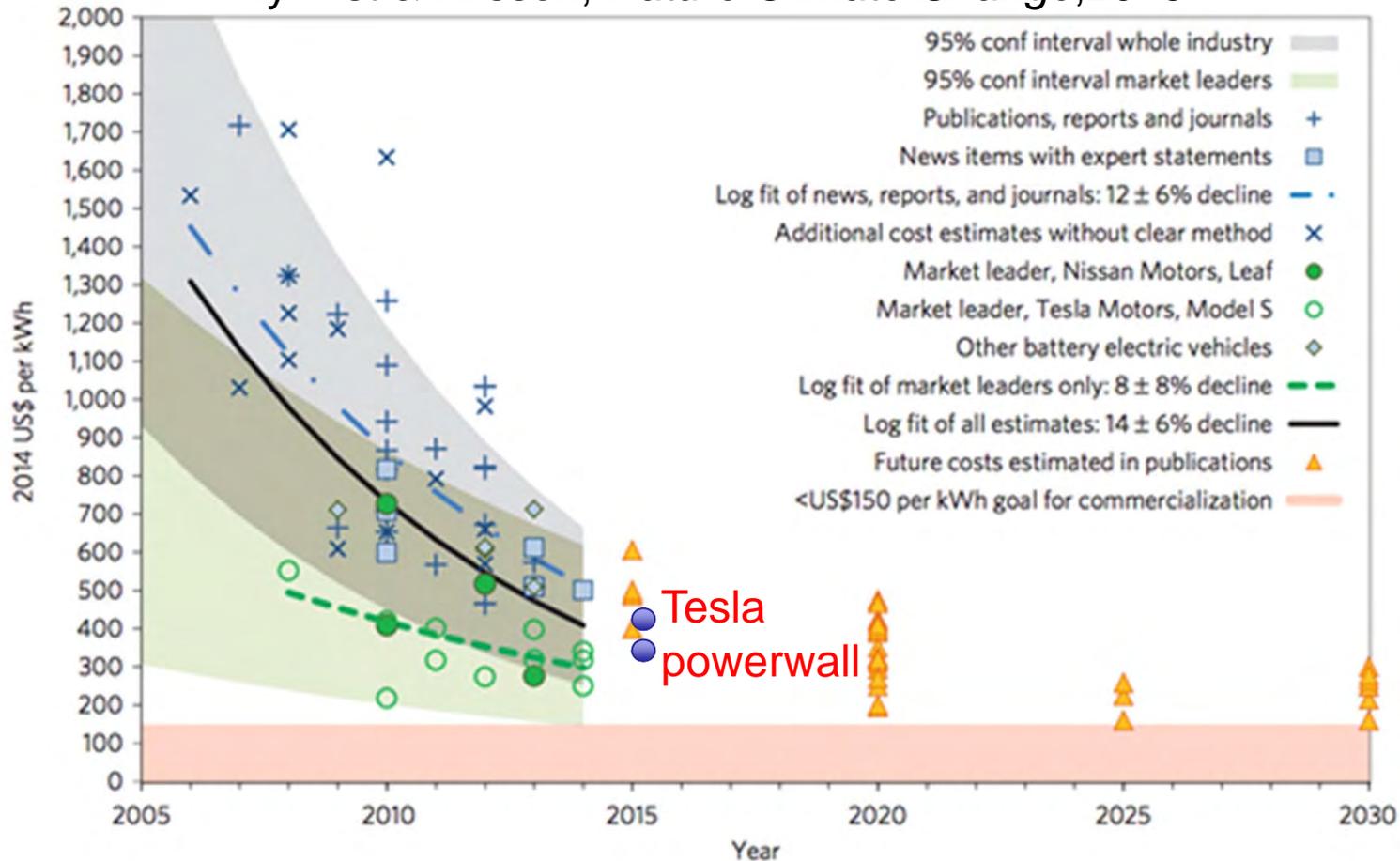


**To beat Gasoline: Need 90 kWh Battery for < \$3000  
(or Fuel cell)**



# Battery Cost is Reducing Exponentially

Nykvist & Nilsson, *Nature Climate Change*, 2015



$$\text{\$150/kWh} * 90 \text{ kWh} = \text{\$13,500}$$