

Workshop on Predictive Theoretical and Computational Approaches
for Additive Manufacturing, October 7-9 2015, Washington D.C.



LA-UR-15-27758

Towards Modeling and Simulation of Additive Manufacturing of Metals at LANL

Session 1: Theoretical Understanding of Materials Science and Mechanics

Marianne M. Francois

mmfran@lanl.gov

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Operated by Los Alamos National Security, LLC for the U.S. Department of Energy's NNSA

With contribution from

- C. Bronkhorst
- N. Carlson
- C. Newman
- V. Livescu
- S. Vander Wiel
- T. Haut
- S. Runnels
- J. Bakosi
- J. Gibbs
- J. Mayeur
- A. Trainer
- L. Parietti
- D. Teter
- J. Carpenter
- G. Gray
- T. Lienert
- T. Holesinger
- A. Clarke
- D. Turret
- C. Knapp
- J. Shlachter
- M. Schraad
- B. Archer
- K. Lam

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Questions to be addressed

- Fundamental scientific (AM) issues ?
- Fundamental theoretical and computational approaches needs to fully understand AM ?
- Most important questions in materials and mechanics, engineering and mathematics ? Technical challenges for predictive theoretical and computational approaches in order to enable widespread adoption of AM ?
- Opportunities in R&D for theoretical and computational materials science, mechanics and multiscale computations
- **To address these questions, I will use LANL nascent effort in modeling and simulation of additive manufacturing of metals as example**

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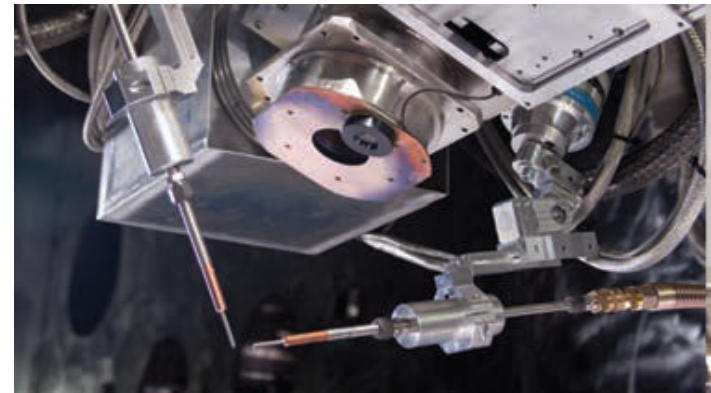
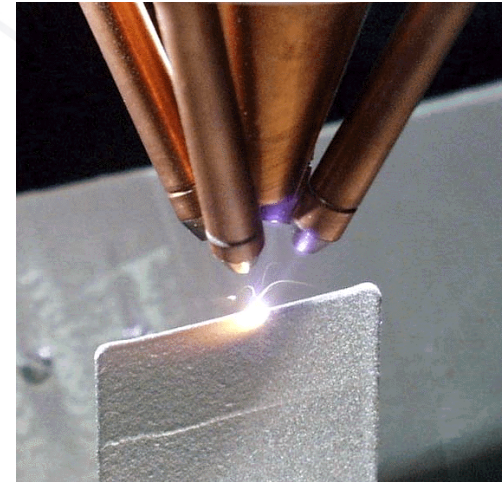
Fundamental AM issues

- How do we move toward a science-based qualification of AM metals ?
- How do we shorten qualification cycle ? (reduce cost)
- Currently, there is no Additive Manufacturing (AM) certified process, no standard, and there is a lot of variability
 - Multiple AM processes (powder-bed, directed energy deposition,...), various operating conditions and control parameters (power, scanning patterns,...), feedstock quality, post-processing (heat treatment)
- **→ Need for fundamental understanding through scientific methodology integrating experiments with theoretical modeling and simulation.**

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LANL Additive Manufacturing of Metals

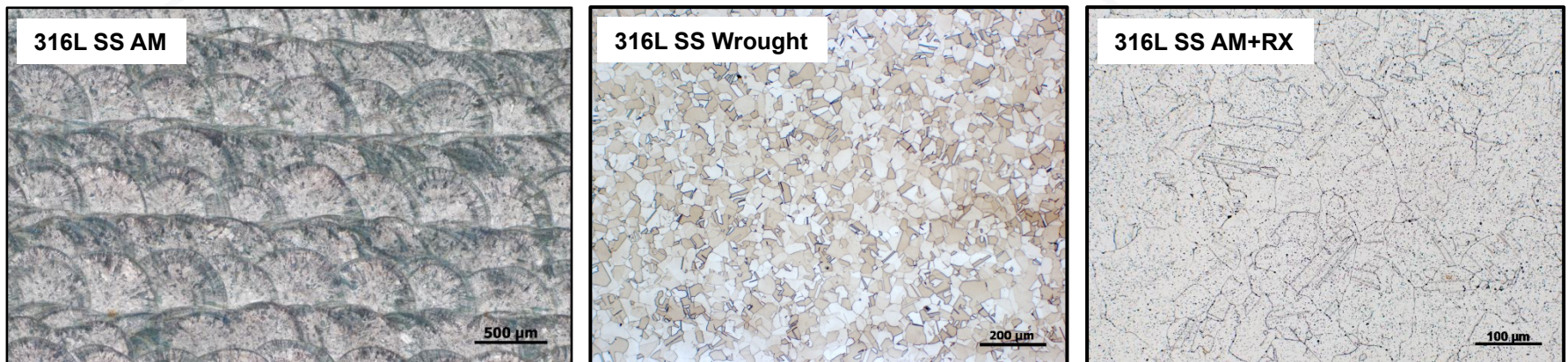
- Part of multi-lab effort (LLNL, SNL, LANL and NSC)
- LANL focuses on directed energy deposition system
 - Ebeam or laser as heat source
 - Powder or wire as feedstock
 - Optomec (laser/powder), Sciaky (e-beam/wire)



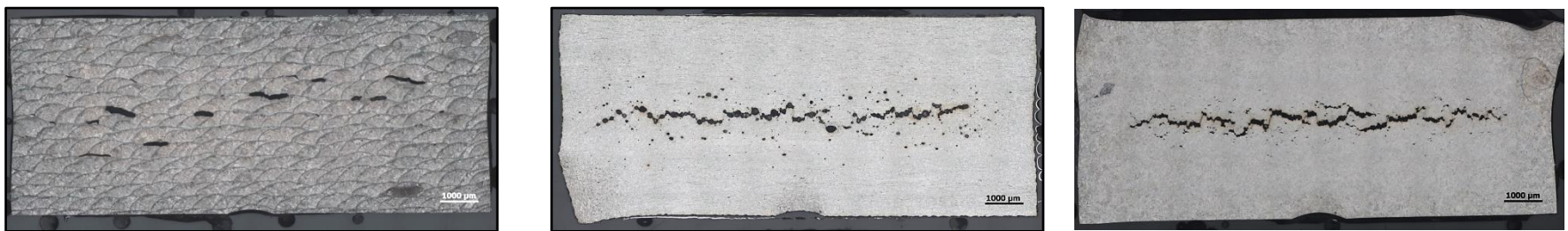
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Fundamental scientific issues: Same 316L SS material but different microstructures

- Difference in processing results in microstructural differences



- Difference in microstructure results in performance differences



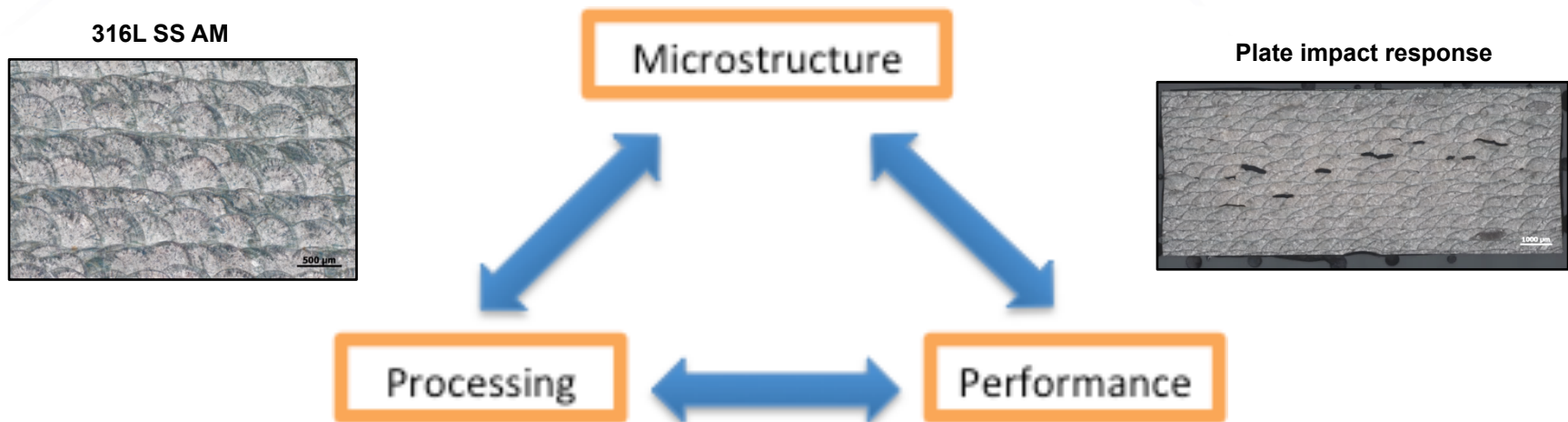
257m/s Flyer Impact Triple Shot

Ref: Gray, LANL MST-8

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Theoretical and Computational Approach Needs

- Integration of processing and performance through microstructure prediction



- Need to develop **advanced modeling and simulation capabilities** for AM processes along with **experimental testing** as part of our methodology **towards prediction and control**

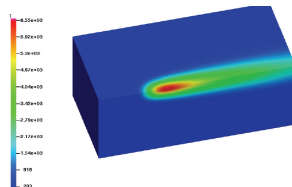
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Long term vision: microstructure-aware modeling from materials processing to mechanical performance

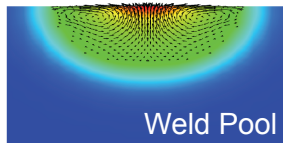
Process Modeling ↔ **Microstructure Modeling** ↔ **Properties Modeling** ↔ **Performance Modeling**

Liquid/solid phase change

3D multi-physics
microstructure-aware
solidification
capability

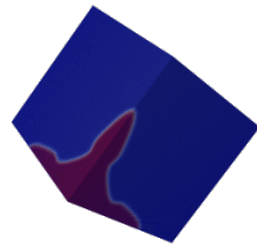


Moving heat source

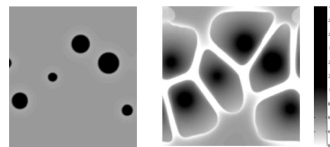


Weld Pool

Direct numerical
simulation of grain growth



3D single grain growth

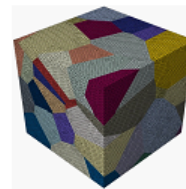


Initial grain
distribution
(nucleation
site)

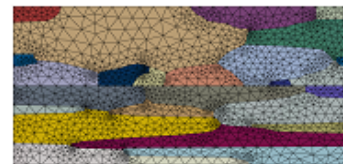
Final grain
shape and
composition

Solid/solid phase transformation

Polycrystal models to
determine elastic, plastic,
and damage properties

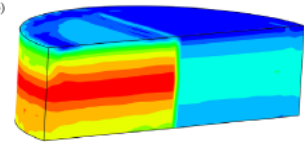
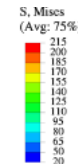


Polycrystal and grain
boundary properties



AM specific interface properties

Thermal-mechanical
models to predict
elastic, plastic, damage,
and failure processes

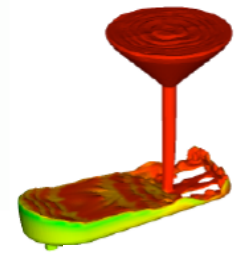


Mesoscale to macroscale
prediction of performance

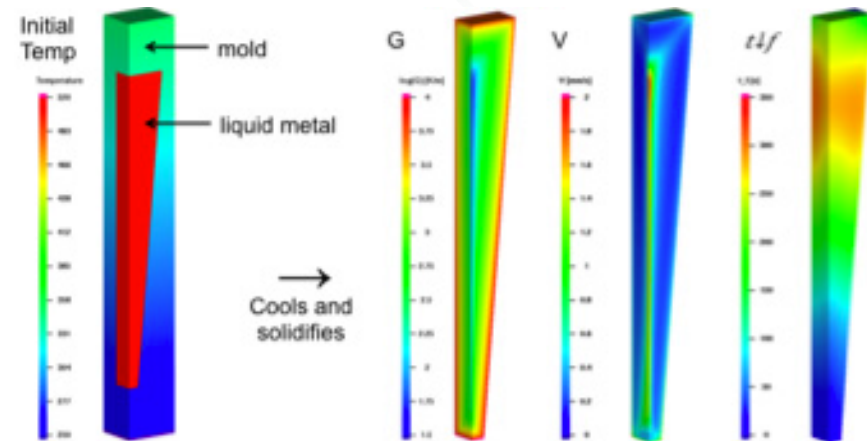
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TRUCHAS a computational tool for modeling material processing

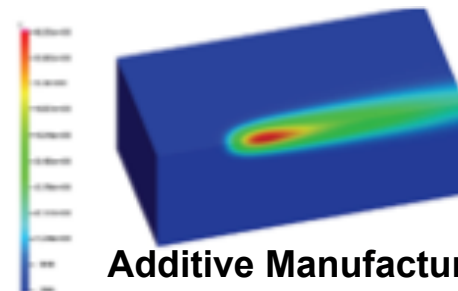
- A 3D multiphysics package: fluid flow with interface tracking and surface tension, heat transfer with phase change, species diffusion, chemical reaction and solid mechanics
- Complex geometries
- Initially developed to model casting processes, extended to model welding in 2006
- Currently being extending to model direct energy deposition AM processes



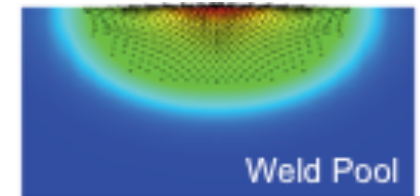
Mold Filling



Nuclear Fuel Rod Casting



Additive Manufacturing



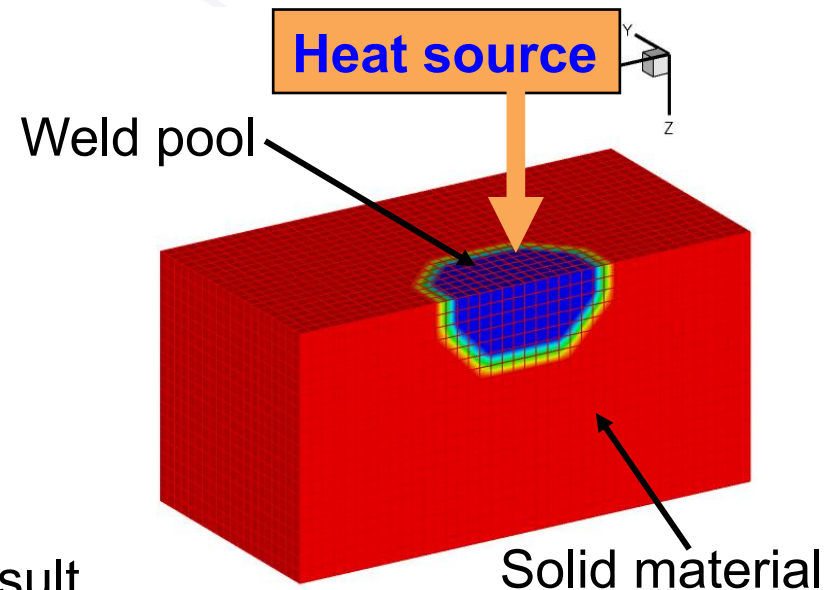
Welding

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3D Simulations of Laser Spot Welding of 304 Stainless Steel – a Validation Study

Ref: Parietti, Lam, LA-UR-06-7622

- Domain $[-0.5, 0.5] \times [0, 0.5] \times [0, 0.5]$ mm
- Laser power $Q = 530$ W
- Properties as in He et al. (2003)
 - Solidus temperature 1697 K
 - Liquidus temperature 1727 K
 - $d\sigma/dT = -4 \times 10^{-4}$ N/m
- Heat Transfer + Phase change + Flow
- Heat source applied from $t=0$ to 4 ms result in melting of the solid material
- At $t=4$ ms, solidification starts
- Convection in melt driven mainly by thermocapillary force



Surface Heat Flux with Gaussian distribution

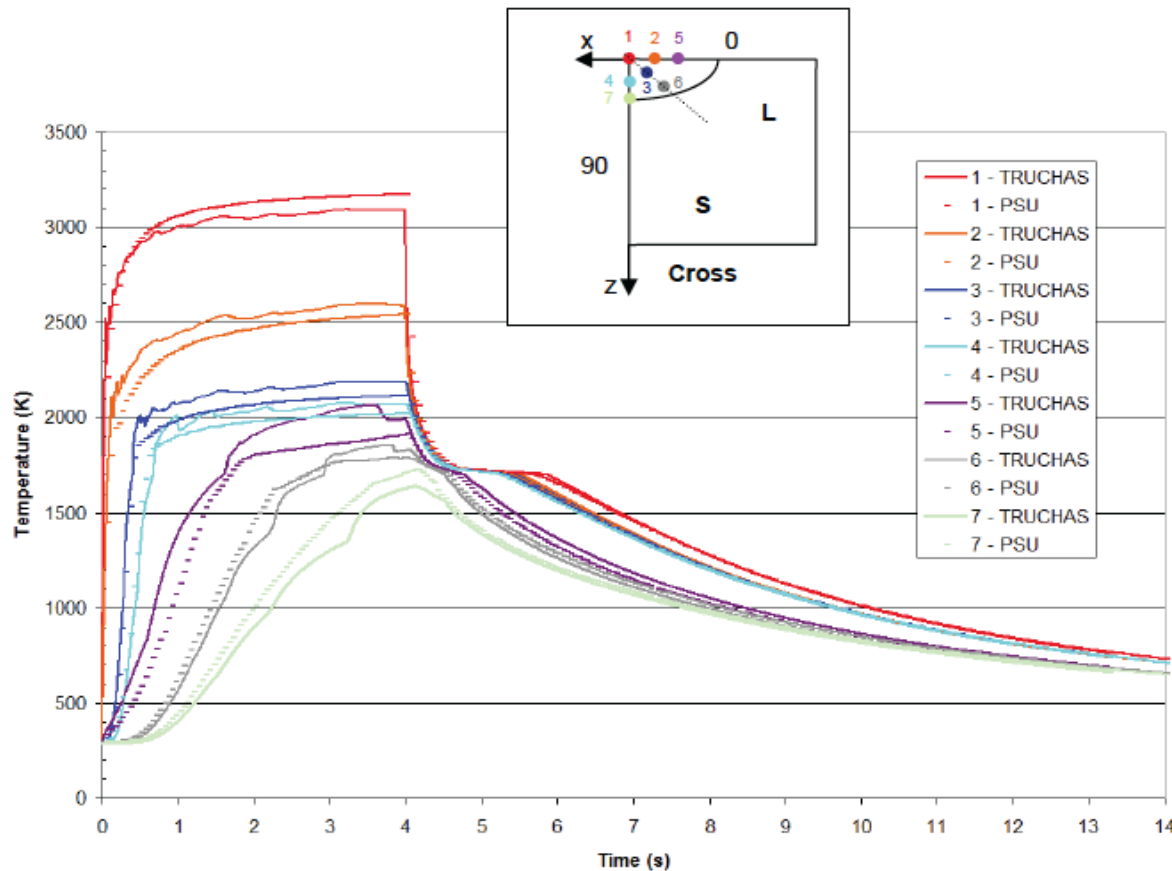
$$\Phi(x, y) = \frac{3Q\eta}{\pi r_b^2} e^{-\frac{3(x^2 + y^2)}{r_b^2}}$$

η absorption coefficient
 r_b beam radius

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Laser Spot Welding of 304 Stainless Steel - Validation Temperature Histories Cross-section

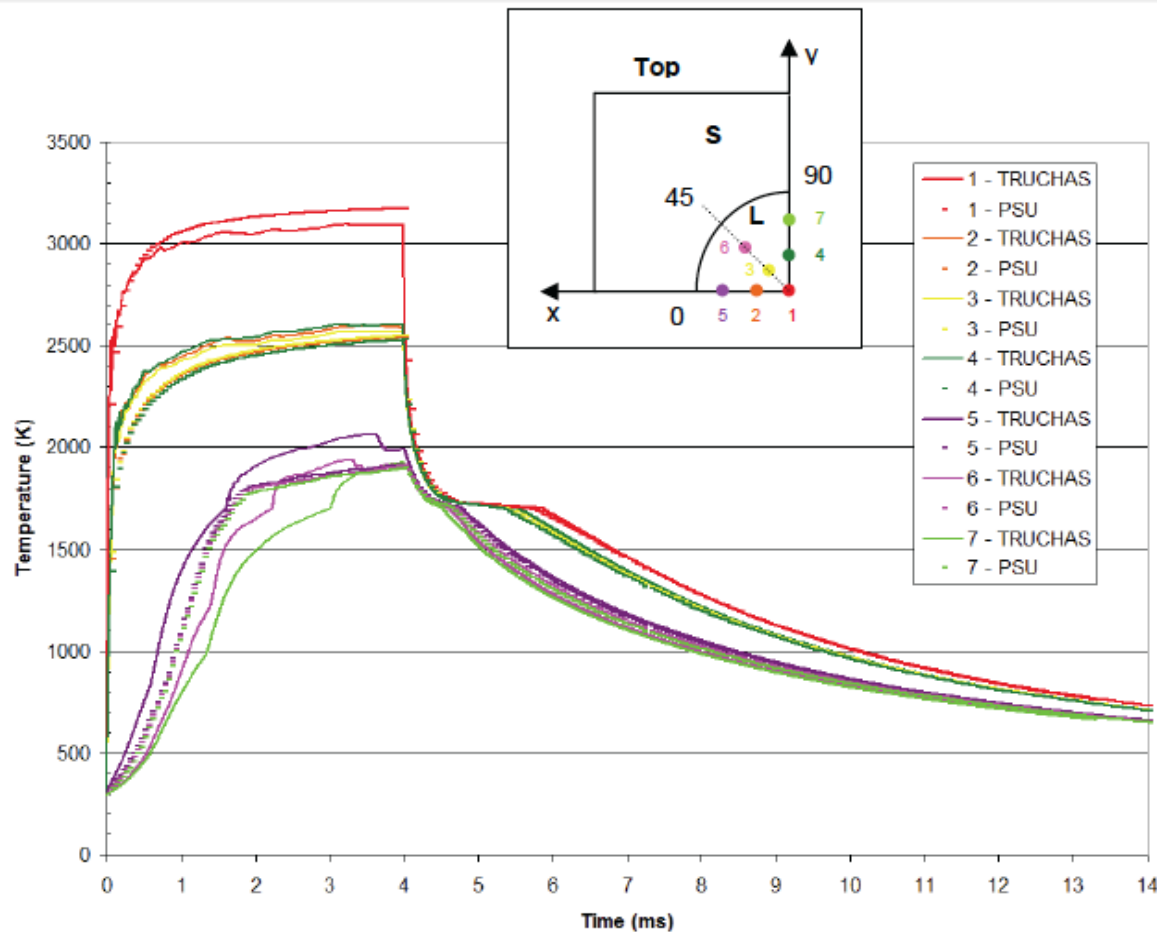
Ref: Parietti, Lam, LA-UR-06-7622



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Laser Spot Welding of 304 Stainless Steel - Validation Temperature Histories Top Surface

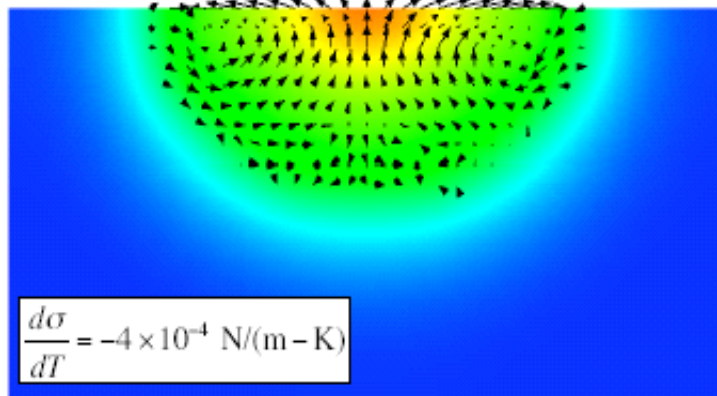
Ref: Parietti, Lam, LA-UR-06-7622



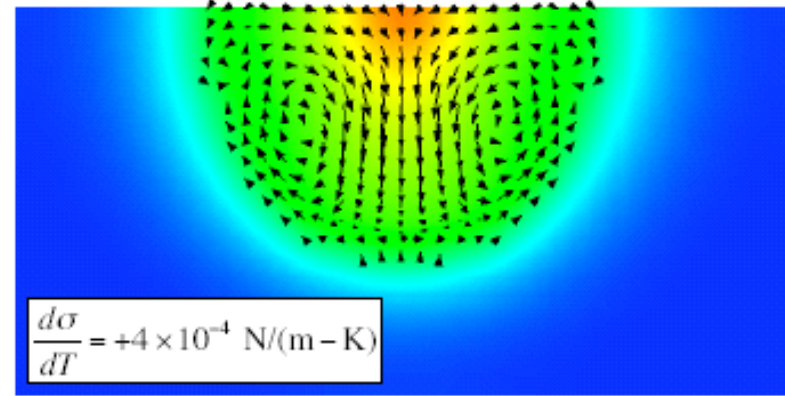
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Marangoni convection determines weld pool shape

Laser Spot Welding of 304 Stainless Steel
Temperature contours and velocity vectors



Negative gradient
→ outward flow
→ shallower weld pool



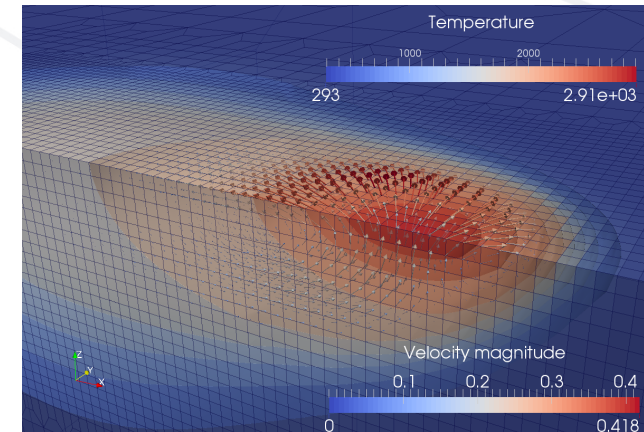
Positive gradient
→ inward flow
→ deeper weld pool

- Surface tension properties vary with composition and temperature
- Any impurity will change melt pool shape and size
- In AM: powder particles effects (fully or partially melted), melting and re-solidification cycle (scanning pattern effect), chemical composition, surface instabilities, ...

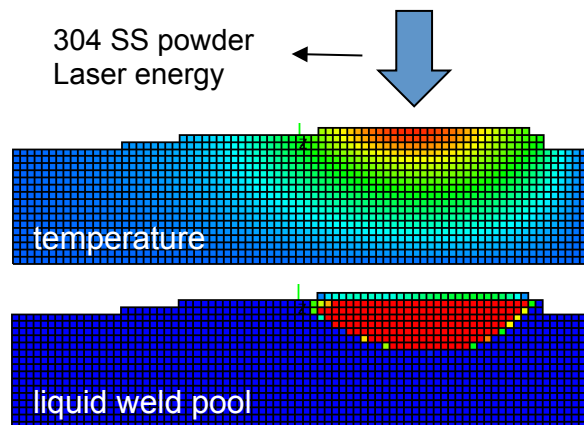
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ASC Truchas code extended to model AM

- Open source Truchas metal casting code is being extended to AM process modeling, esp. directed energy processes (e.g. LENS)
- Capabilities assessed via testing on AM process problems involving heat transfer/phase change, weld pool fluid flow (Marangoni effect), and residual stress/distortion.



Marangoni flow in weld pool during single weld bead pass in 304 SS.

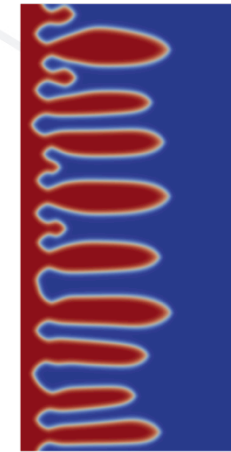


- Short-term: Implemented preliminary heat/mass deposition models:
 - moving heat flux boundary conditions for simpler models;
 - fully moving powder/laser energy deposition at evolving material surface embedded within the computational domain (volume-of-fluid).
- Next: more physics models and V&V
- Microstructure-aware solidification models

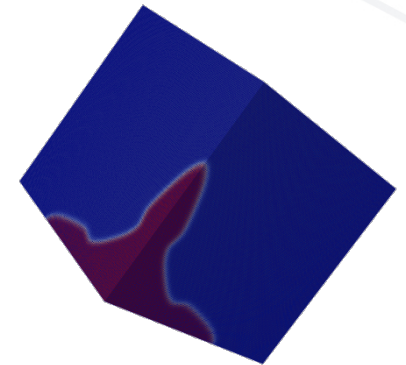
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Microstructure capability development

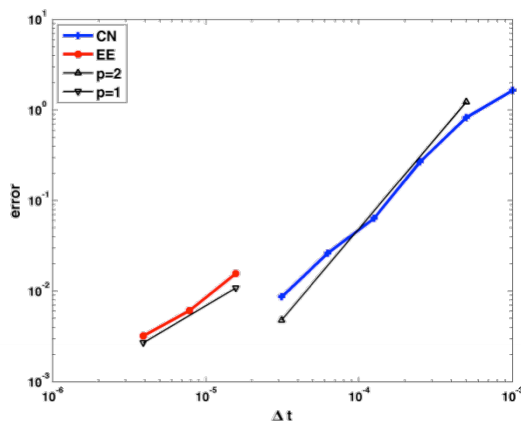
- Phase field approach to microstructure solidification simulation utilizing modern algorithms and software.
- *Finite element* allows for high-order spatial discretization on unstructured 2D/3D meshes.
- *Implicit time integration* allows for stable, second order with large time steps.
- Second-order implicit shows better efficiency over first-order explicit.
- Unstructured mesh allows for irregular (curved) geometries.



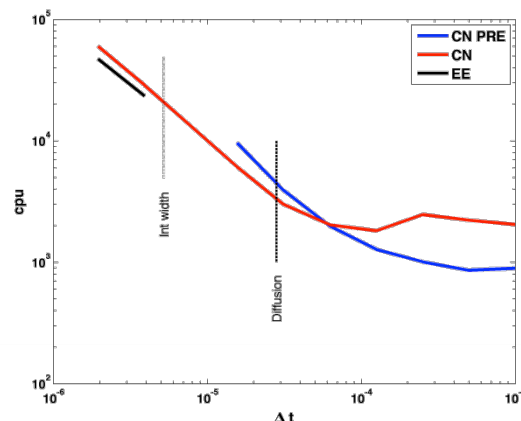
Multiple dendrite growth along flat boundary. (Preliminary result)



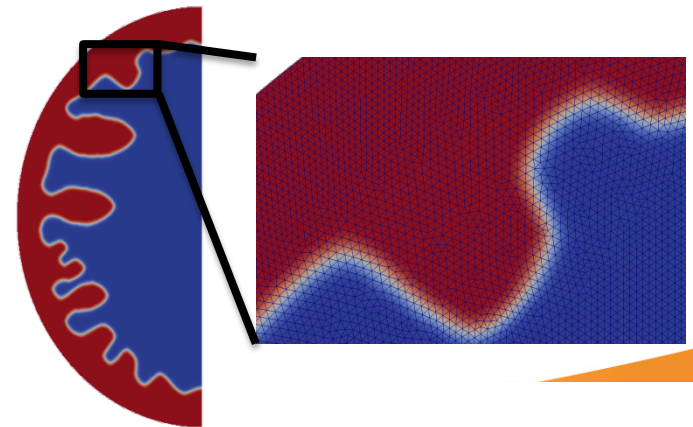
3D single dendrite growth.



Accuracy as a function of timestep size for implicit (--) vs explicit (--).



Efficiency as a function of timestep size for implicit (--, --) vs explicit (--).

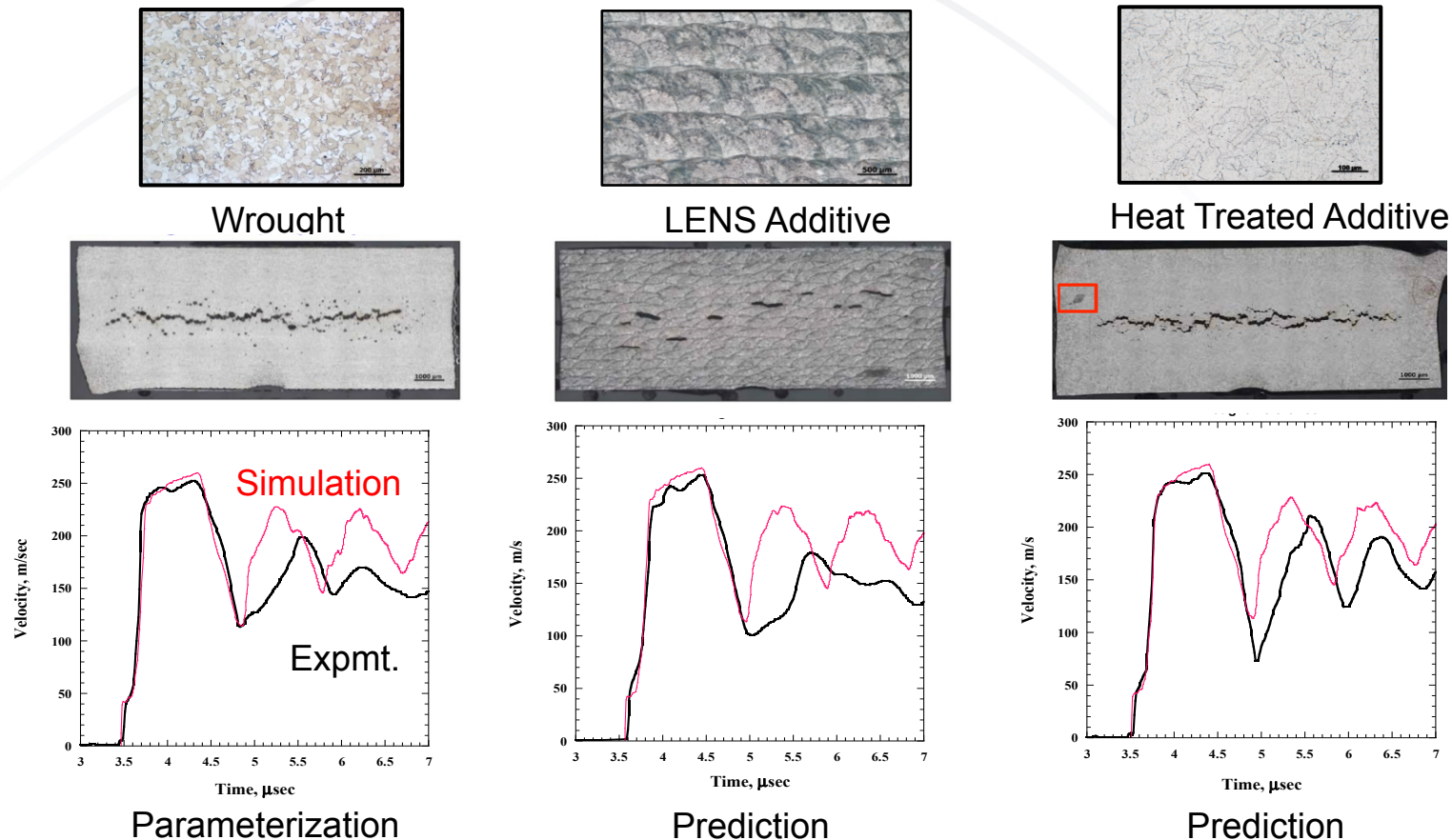


Multiple dendrite growth along curved boundary, unstructured mesh. (Preliminary result)

Additively Manufactured Material – 316L Performance

Modeling of Plate Impact Response

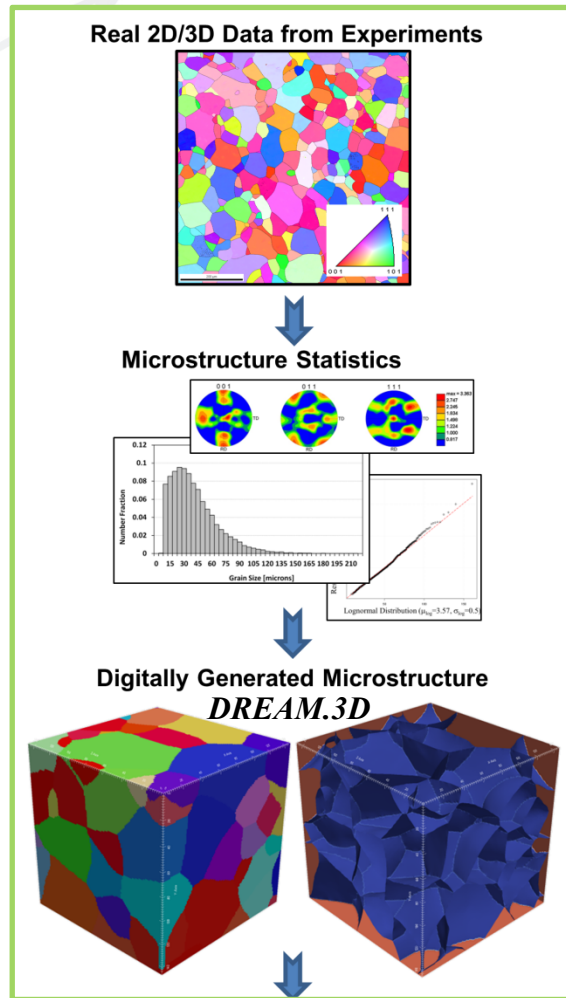
AMIT Materials & Expts., G. Gray et al.



- Influence of microstructure on damage performance is significant.
- Not yet possible to represent the additive microstructure adequately to successfully predict dynamic damage with simple macro-scale models.

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316L Metallographic Characterization

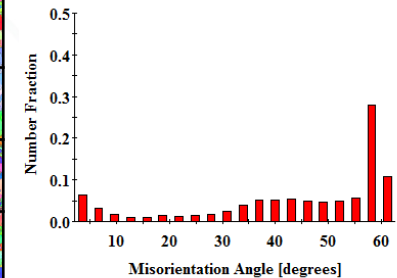
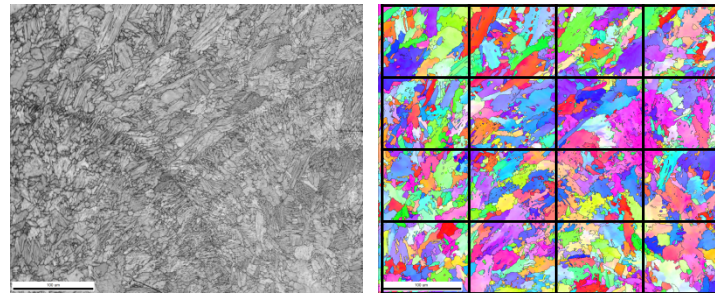


PATRAN FEM Software with
Advanced Surface Meshers

ABAQUS + Dynamic Damage Model

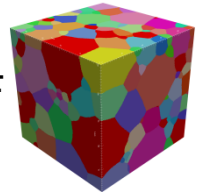
AM material

- EBSD data acquired
- Working on texture and Euler angles data export

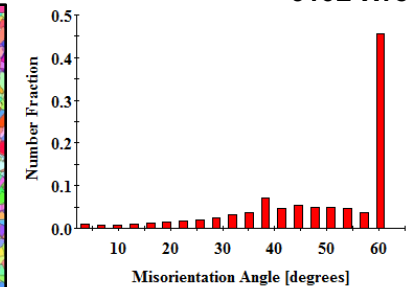
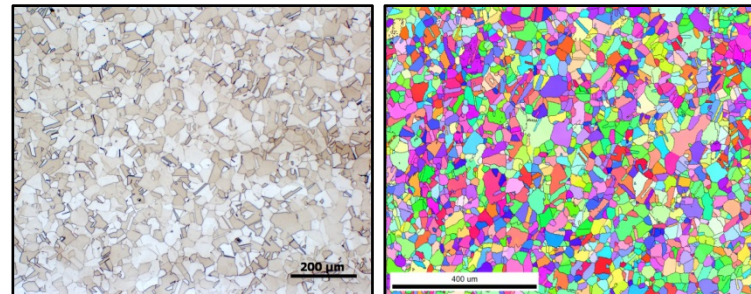


Wrought material

- EBSD data collected
- Found statistical distribution of grain sizes and ODF
- Working on twin insertion within Dream3D

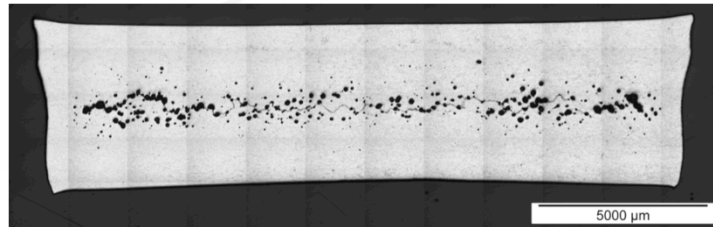


316L Wrought

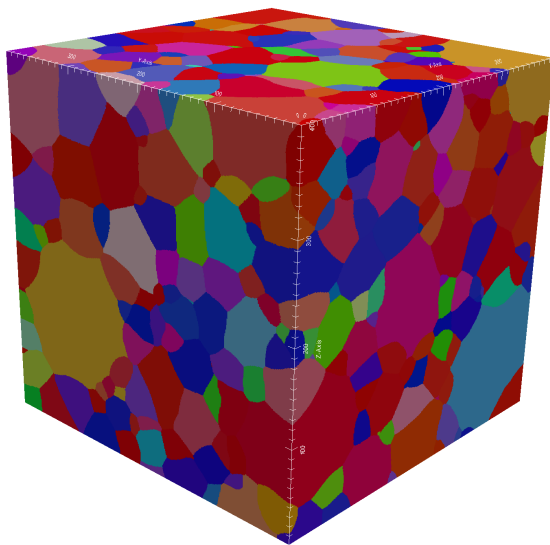


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Macroscale Performance Linkages To Microstructural Modeling

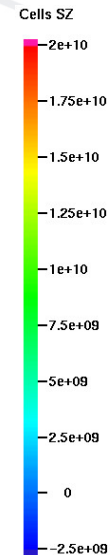


Tantalum on Tantalum Experiment

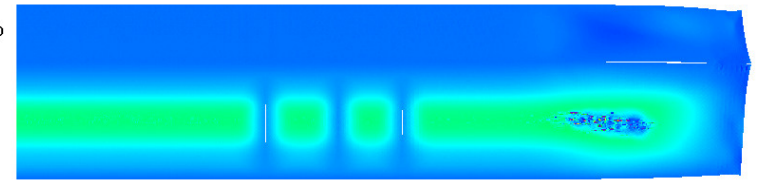


Virtual Equivalent Microstructure

Prediction of macroscale performance will require microstructural representation

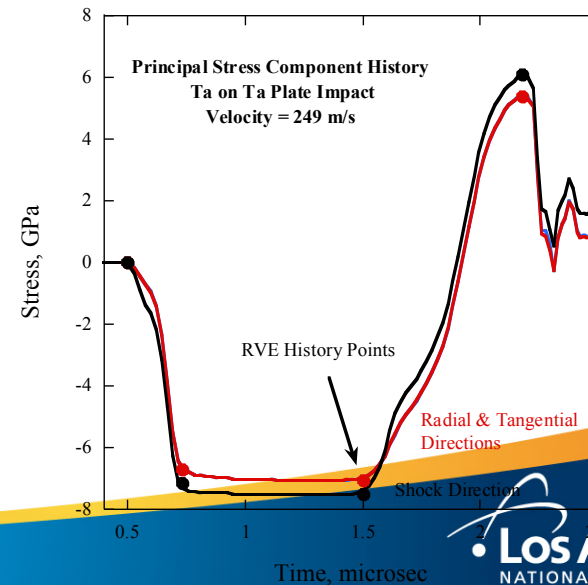


Macroscale Calculation of Experiment



S_{zz} Contour
Time = 2.166 μ s

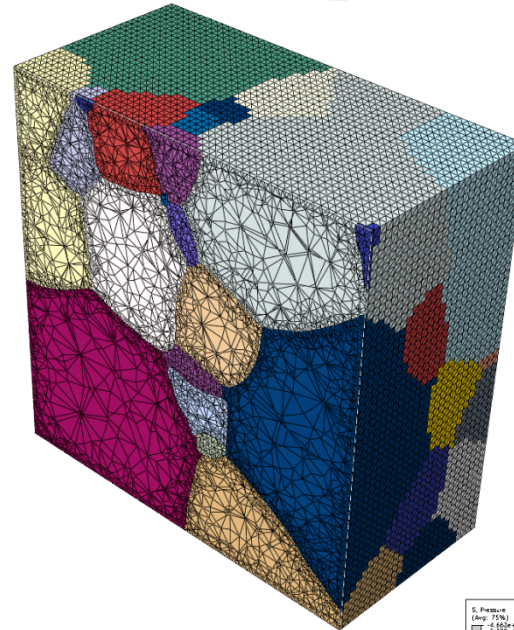
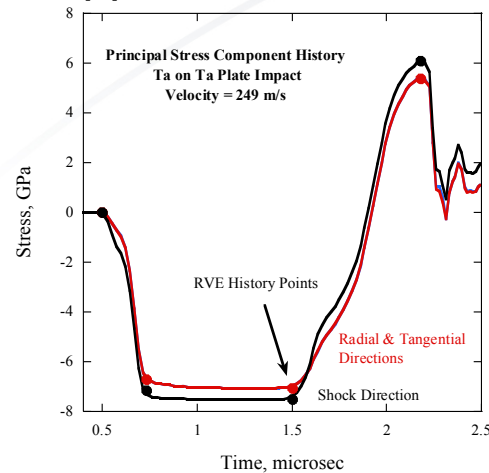
Calculated Principal Stress History



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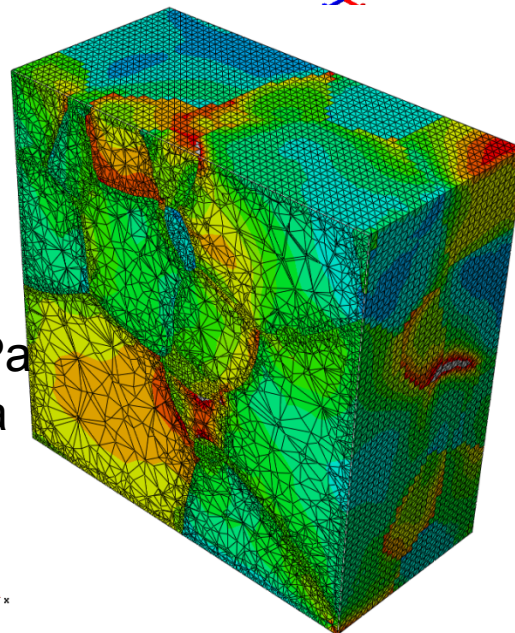
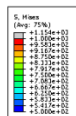
Dynamic Loading Conditions Applied to Statistically Equivalent Tantalum Numerical Microstructures

Applied Stress History

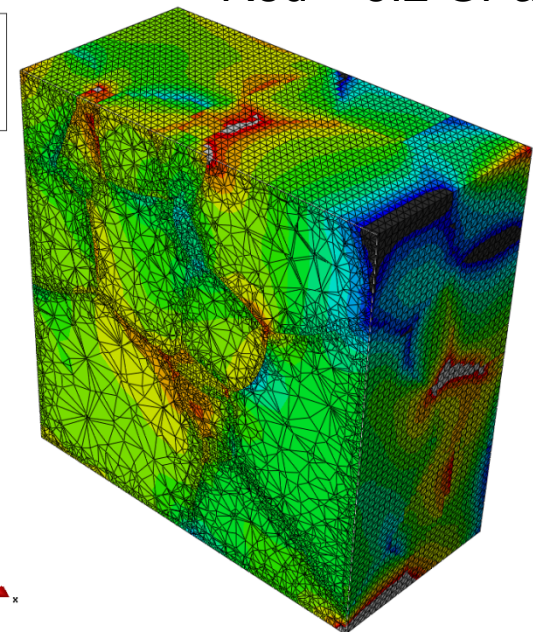
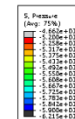


70 grain Realization

Pressure
Blue = 5.9 GPa
Red = 5.2 GPa



Mises Stress
Blue = 500 MPa
Red = 1.0 GPa



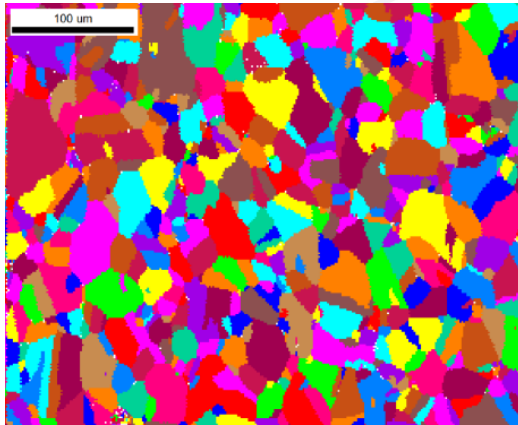
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LLC for the U.S. Department of

Additively Manufactured Material – 316L

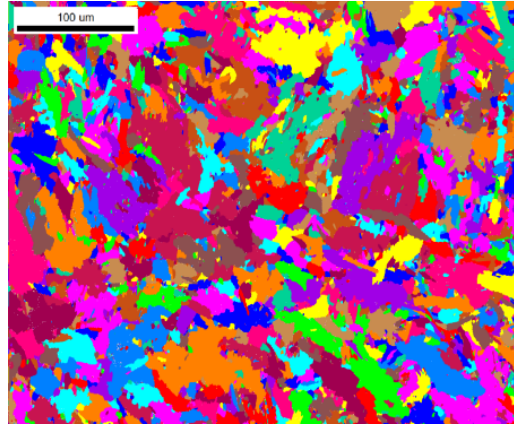
Performance Modeling of Strength Difference

Different colors represent different crystallographic orientations



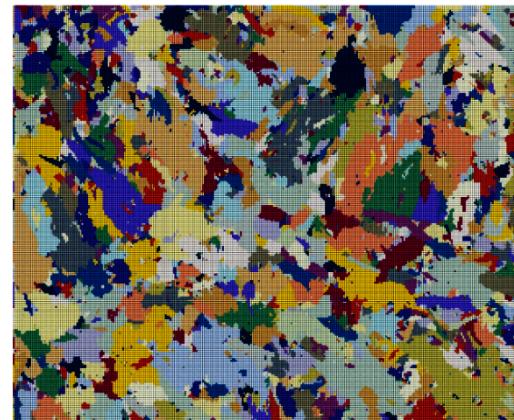
Wrought

Mean Grain Size = 15.9 μm



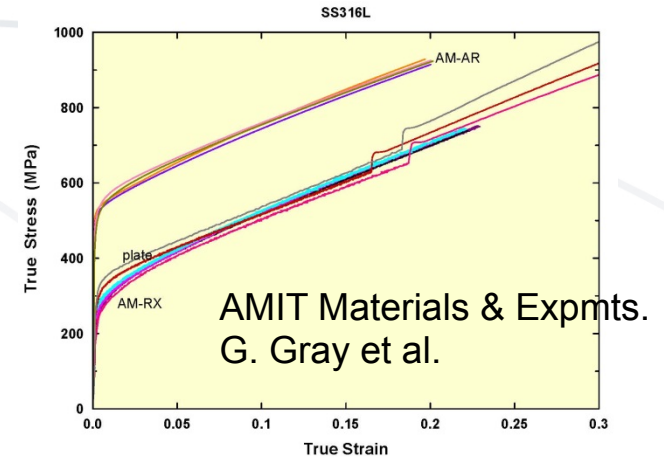
LENS Additive

Mean Grain Size = 4.4 μm



Numerical Polycrystal Models

Crystallographic orientation is translated to the numerical models but not color represented.



Greater strength in additive material

- Smaller grains generally increase strength in metals.
- Hypothesis: Grain size difference in part responsible for observed strength difference.

Hall-Petch relationship between grain size and plastic flow resistance being applied at grain level in polycrystal models.

Long-term Objectives for Predictive Methods in AM

- Integration of processing and performance modeling through microstructure prediction
- Validation with experimental testing (in-situ) as part of our methodology towards prediction and control
- Multiscale process modeling: microstructure-aware
- AM materials modeling and multiscale mechanical response (performance) modeling
 - Processing phase change and microstructural evolution
 - Cooling internal stress development linked to microstructure
 - Plasticity and structural feature damage prediction

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Opportunities for Theoretical and Computational Predictive Methods in AM

■ Material Processing

- Melting/solidification and re-melting/solidification cycles, melt pools, microstructure morphology evolution, composition distribution (alloy), liquid-solid phase change models
- Linkage of microstructure information to macroscale model (thermal gradient and cooling rate maps)
- Residual stresses

■ Mechanics of Materials

- AM materials models (properties,...), solid/solid phase transformation
- Plasticity/damage modeling
- Linkage of microstructure information to macroscale model

■ Faster computational methods

- Reduced order models, fast emulators for process control
- Robust, efficient and accurate numerical methods (implicit methods) for high-fidelity physics-based simulation

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